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AD-A015 907

DEVELOPMENT OF ENCINEERING DATA ON THE MECHANICAL AND PHYSICAL PROPERTIES OF ADVANCED COMPOSITES MATERIALS

IIT RESEARCH INSTITUTE

PREPARED FOR
AIR FORCE MATERIALS LABORATORY

FEBRUARY 1974

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REPORT DOCUMENTATION PAGE	READ DISTRUCTIONS DEPORE COMPLETING FORM
	1. RECIPIENT'S CAVALOG NUMBER
APML-TR-72-205 Part II	
e. TITLE (and Section) Development of Engineering	TYPE OF REPORT & PERIOD COVERED
Data On The Mechanical And Physical Prop	Final Report
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7. Au THORra:	- D6063
Hofer, Jr., K.E.; Rao, N. and Larsen, D.	
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S DESCRIPTION OF AND	M. PROGRAM EL EMENT PROJECT TASK
5. Performing ORGANIZATION NAME AND ADDRESS IIT Research Institute	16. PROGRAM ELEMENT, PROJECT, TASK AREA & WORK UNIT NUMBERS
10 W. 35th Street	
Chicago, Illinois 60616	
II. CONTROLLING OFFICE NAME AND ADDRESS	12. REPORT DATE
	Pehrnary 1974
	499
14 MONITORING AGENCY NAME & ADDRESS(st different from Controlling Office)	18. SECURITY CLASS. (of Mio report)
	Unclassified
	ISO. DECLASSIFICATION DOWNGRADING
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thermo-humidity cycling, interlaminar shear, flex tests, boron/
epoxy composites, graphite/epoxy, composites, aluminum/boron composites, titanium/Borsic composites, laminate data, laminate fabrication, moisture weight gain

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were 6061 Aluminum/Boron and 6Al-4V - Titanium/BorSic Composites. The resin matrix systems were procued in the form of prepreg tapes and specimens were fabricated at IITRI materials laboratory. The metal matrix composites were fabricated by vendors in laminate form and supplied to IITRI for specimen fabrication and testing. The environments included steady state and cyclic thermal, thermo-humidity, and humidity conditioning.

Part II

DEVELOPMENT OF ENGINEERING DATA ON THE MECHANICAL AND PHYSICAL PROPERTIES OF ADVANCED COMPOSITES MATERIALS

K. E. Hofer, Jr. N. Rao D. Larsen

IIT Research Institute

Technical Report AFML-TR-72-205, Part II
February, 1974

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Air Force Materials Laboratory
Air Force Systems Command
Wright-Patterson Air Force Base, Ohio

FOREWORD

This technical report summarizes the work accomplished during Contract No. F33615-71-C-1713, "Development of Engineering Data on the Mechanical and Physical Properties of Advanced Composite Materials;" it was prepared by the Mechanics Research Division of the IIT Research Institute. The work reported herein was accomplished under the joint sponsorship of two divisions of the Air Force Materials Laboratory the Systems Support and the Advanced Development Division, under Project No. 7381, "Material Application, Task No. 738106, Design Data Development and Advanced Composite ADP." Messrs. M. Knight, AFML/MXE, of Systems Support Division and Capt. 1. Woodrum and R. Neff, of the Advance i Development Division three the Air Force Project Engineers.

An advanced composite team under the direction of the Materials Engineering Section of the Mechanics Research Division performed the work described herein. IITRI personnel associated with this program and their respective responsibilities are delineated below:

- K. E. Hofer, Program Manager
- N. Rao, Overall Engineering and Scheduling
- V. Humphrevs, Analytical Methods and Reporting
- D. Larsen, Thermophysical Testing
- R. Labedz, Fabrication
- H. Lane, Static Test Engineer
- L. C. Beinett, Fatigue Test Engineer
- R. A. Stuchmer, Creep Test Engineer

This is the final technical report and summarizes the technical activities from June 1, 1971 through November 30, 1973.

This technical report has been reviewed and is approved.

> Kenneth E. Hofer, Jr., Materials Engineering and **Building Technology Section** Project Engineer

For The Commander

albut abouted.

Materials Engineering Branch Systems Support Division Air Force Materials Laboratory

ABSTR CT AND SUPPMARY OF RESULTS

The present program was initiated to generate data on the effect of various environments on the physical, thermal, and mechanical properties of three resin matrix composites: (AVCO 5505/Boron, Modmor II Graphite/Parmoo 5206 and Courtaulds HMS Graphite/Hercules 3002M) and two metal matrix composites (6061 Aluminum/Boron and 6A1-4V ~ Titanium/BorSiC). The resin matrix systems were procured in the form of prepreg tapes and laminates and specimens were fabricated at IITRI. The metal matrix composites were fabricated by vendors in laminate form and supplied to TITRI 6 % and a supplied to TITRI 6 % and a supplied

The environments included steady state humidity conditioning for two exposure periods, cyclic humidity conditioning which included the effects of thermal shocks and the effect of photodegradative exposures, and steady and cyclic thermal exposures.

Part I of this report described the material procurement, materials specifications, laminate fabrication, quality control and material quality assurance tests, and presents the test procedures in detail. Part II of this report presents a complete summary of the data and results of all static, fatigue, creep and thermo-physical properties of the five composites.

On a material by material basis the following conclusions were reached:

AVCO 5505/Boron

There was a general deterioration in the baseline tensile, compressive and in-plane shear strengths of AVCO 5505/Boron with increasing temperature. The elastic moduli of the 0° properties were relatively unaffected up to 350°F but the transverse (90°)

and in-plane shear moduli decreased with temperature. The $[0/45/135/0/\overline{90}]_8$ baseline tensile and compressive moduli were relatively unaffected up to $350^{\circ}F$.

The steady state humidity conditioning caused the strengths of AVCO 5505/Boron to fall below the baseline values particularly at elevated temperatures (by up to 30% for 350°F compression). This occurred for all orientations and in tension, compression and shear (the differences were generally of the order of 10%). The elastic moduli of AVCO 5505/Boron were reduced to a small extent (generally 2-4%) by the steady state humidity conditioning; the strongest effects were noted for the in-plane shear (10%) and transverse moduli (30%). The steady state conditioning increased the moduli of the $[0/45/135/0/\overline{90}]_g$ laminates by 5-10%.

High humidity and thermal shock as indicated by the thermo-humidity cycle results had the same effect as steady state humidity conditions but had greatest impact on the room temperature static strength results (up to 20% for tension of 90°). The largest degradatory effects were obtained for combined humidity and ultraviolet on both strengths and modulus of all three orientations although a mixed effect was noted at elevated temperatures (losses up to 50% of the 90° compressive strength were seen).

The steady state humidity conditioning degraded the fatigue performance of AVCO 5505/Boron composites (losses of approximately 10%). The thermo-humidity cycle degraded the fatigue performance of AVCO 5505/Boron at the higher cyclic levels thus shifting the S-N curves downward and rotating the curve about the low cycle levels (losses up to 25% were encountered at high cycle levels). Accelerated weathering had the least effect on the fatigue behavior.

The steady state humidity conditioning had a detrimental effect on stress-rupture behavior of 0° AVCO 5505/Boron (loss of 25%) but enhanced the stress rupture behavior of $[0/45/135/0/\overline{90}]_8$ composites by approximately 10%.

Steady thermal conditioning enhanced the strength (up to 8%) and modulus (up to 6%) of AVCO 5505/Boron. The cyclic thermal conditioning had a mixed (but moderate rather than severe) effect on the strength and moduli of all three orientations.

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The interlaminar shear strengths were decreased by humidity conditioning (by up to 25%) but were unaffected by both steady state and cyclic thermal conditioning.

The AVCO 5505/Boron composite fatigue behavior was degraded by 10% at all temperatures by steady-state thermal conditioning. The stress-rupture behavior was improved by up to 5% by steady state thermal conditioning. Similarly stress-rupture improvement (20%) and fatigue degradation (10 to 15%) were shown for prior exposure to cyclic thermal preconditioning.

Modmor II Graphite/Na mco 5206

A general reduction in the tensile, compressive and inplane shear strengths with increasing temperature was demonstrated for the Modmor II Graphite/Narmco 5206 composites (except for the $0/45/135/0/\overline{90}$, laminates). The elastic moduli of the 0° orientation remained relatively unaffected although the tensile modulus increased slightly with increasing temperature up to 350° F. The shear modulus and the tensile and compressive moduli of the 90° composites decreased with increasing temperature. The $[0/45/135/0/\overline{90}]_{S}$ tensile modulus also increased with temperature. Residual stresses are suspect in this behavior.

Steady state humidity conditioning affected the moduli of the Modmor II graphite/Narmco 5206 system the least of the three resin matrix composites studied. The strengths of this system decreased below the values of the baseline strengths particularly at elevated temperatures (up to 35%). The exception to this behavior was the 0° tensile strength which showed some improvement (about 20%) with prior steady state humidity conditioning.

The thermo-humidity cycle conditioning influenced the static behavior in a manner similar to that produced by steady state conditioning except that the effects were worse at higher temperatures (up to 25% degradation). The greatest effects on modulus and strength were observed for the combined humidity and ultraviolet conditioning particularly at room temperature.

The thermo-humidity cycle degraded the fatigue behavior of the 90° Modmor II Graphite/Narmco 5206 more substantially at the higher cyclic levels, thus shifting the S-N curves downward and rotating the curve about the low cycle levels. The accelerated weathering had the least effect on the fatigue S-N curves (less than 10%). Steady state humidity conditioning degraded the fatigue behavior as well.

The stress rupture characteristics of Modmor II Graphite/
Narmco 5206 were benefited by steady state and cyclic humidity
conditioning. Residual stresses in the composites become suspect.

The static response of Modmor II Graphite/Narmco 5206 to steady state and cyclic thermal conditioning was quite similar to that shown for AVCO 5505/Boron. The interlaminar shear strengths decreased for humidity conditioning but were unaffected by either steady state or cyclic thermal conditioning. The

fatigue behavior of Modmor II Graphite/Narmco 5206 was improved with prior steady state thermal exposure. Cyclic thermal exposure, on the other hand degraded the fatigue behavior but appeared to be highly dependent on cyclic levels. The stress-rupture behavior was not affected substantially by either steady state or cyclic thermal conditioning.

Courtaulds HMS Graphite/Hercules 3002M

The baseline static tensile strengths of all three orientations of Courtaulds HMS Graphite/Hercules 3002M composites increased with increasing test temperature by approximately 20%. The presence of degradatory residual stresses derived in the cure process becomes a suspect in this behavior. The 0° tensile modulus of this composite increased by 10% up to 350°F. The compressive and shear moduli exhibited mixed behavior over the temperature range. The baseline 90° moduli decreased by 10% with temperature. The $[0/45/135/0/\overline{90}]_{S}$ moduli showed a straight increase of 20% in tension and a compressive decrease of 20%.

Steady state humidity conditioning caused a decrease of 15% in Courtaulds HMS Graphite/Hercules 3002M composite static strengths except in the case of the 0° tensile strengths which improved by 20%. The moduli of this composite was the most affected by steady state humidity conditioning (up to 50% for in-plane shear). The thermo-humidity cycle affected the strengths of this composite greatest at elevated temperatures (the 90° strength loss was 75%). The combined effects of humidity and ultraviolet light on static strengths were greatest at room temperature. Accelerated weathering also affected the elastic moduli of this composite.

The fatigue performance of Courtaulds HMS Graphite/
Hercules 3002M was degraded by steady state humidity conditioning

at high test temperatures and enhanced at room temperature.

Again as in the case of the other resin matrix composites, the thermo-humidity cycle had substantial effects on the S-N behavior of this composite, while the accelerated weathering cycle had the least effect.

Also, the effect of steady state and cyclic humidity conditioning on the stress-rupture properties of Courtaulds HMS Graphite/Hercules 3002M composites was beneficial as in the case of the other two resin matrix composites.

The strengths and moduli of Courtaulds HMS Graphite/
Hercules 3002M composites increased by 10% with steady state
thermal conditioning, while the cyclic thermal conditioning
decreased the static properties by up to 20% in the case of
compression. The interlaminar shear strengths were unaffected
by either steady state or cyclic thermal conditioning.

The fatigue behavior was degraded by 10% steady state and cyclic thermal conditioning. The creep and stress-rupture were affected only slightly by both steady state and cyclic thermal exposure.

6061 Aluminum/Boron

The steady state thermal exposure reduced the 0° tensile strength of 6061 Aluminum/Boron by up to 20% while cyclic exposure to the same temperature reduced the tensile strengths by about 25%. The corresponding 0° compressive strengths were approximately the same as those of the tensile strengths when compared on total exposure time basis. The 90° tensile and compressive strengths were also reduced (by as much as 35%) for similar exposure periods.

Elevated temperatures reduced the tensile, compressive and fully reversed fatigue behavior of 6061 Aluminum/Boron composites, but not severely except in the case of the 90° orientation where up to 50% of the fatigue strengths were lost.

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6A1-4V Titanium/BorSiC

Reductions in the 0° tensile strength of 6Al-4V titanium/BorSiC were approximately 10% for steady state and cyclic thermal exposures. The corresponding compressive strength losses were approximately 10 to 20%, the highest losses occurring at the highest test temperatures. The transverse or 90° tensile strengths degraded up to 50% for steady state and 60% for prior cyclic thermal exposures.

Corresponding compressive strengths losses were of the order of 10 to 15%.

General

Some general commentary on the resin matrix composite behavior is also in order.

The moisture weight gain for all three composites depended only on the total time of high humidity exposure and was not affected by the intervening high or low temperatures, drying periods or U.V. exposures.

The thermal expansion characteristics of the resin matrix composites are dependent on the presence of absorbed moisture in the resin.

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SECTION I

1.0 INTRODUCTION

The objective of this program was to generate basic information on the effect of various environmental variables on the physical, thermal and mechanical properties of advanced composites suitable for application to primary aircraft components. The program encompassed the following tasks:

1.0 Generation of Physical, Thermal and Mechanical Properties of Boron/Epoxy and Graphite/Epoxy composites (Resin Matrix Studies).

The evaluation of the resin matrix materials was further subdivided into the following activities:

- 1.1 material procurement
- 1.2 laminate fabrication
- 1.3 quality assurance testing
- 1.4 baseline data establishment in the following specific areas:
 - 1.4.1 tension
 - 1.4.2 compression
 - 1.4.3 in plane shear
 - 1.4.4 interlaminar shear
 - 1.4.5 flexural tests
 - 1.4.6 fatigue
 - 1.4.7 creep and stress rupture
 - 1.4.8 thermo-physical properties thermal expansion, thermal conductivity
 - 1.4.9 density

- 1.5 exposure of samples to elevated humidity environments
- 1.6 exposure of samples to elevated temperature environments
- 1.7 data generation on the samples exposed to items 1.5 and 1.6 and tested similar to items 1.4.1 through 1.4.8
- 1.8 selective testing of coated samples analysis, correlation of data and reporting activities.
- 2.0 Generation of similar data for Boron/Aluminum and BorSiC/Tital fum Composites (Metal Matrix Studies). The evaluation of the metal matrix composites followed the same general process except that the effect of saidity exposure was not examined.

I thirt is IV. Tables I through III show the overall resinmat. I marrial programs that include baseline tests consisting of unexposed specimens, humidity exposure tests and thermal exposure tests. Table IV shows the metal matrix material program including both baseline and thermal exposure tests.

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* Numbers in paramethesis indicate instrumented Species

TABLE 11

RESIN NATRIX HUMIDITY EXPOSURE DATA PROCRAM

(THIS TABLE APPLIES FOR EACH OF THE FOLLOWING MATERIALS:)

(1) BORDON/A"CO 5505 (2) INTRICOL 11/MAINTON 5206 (3) COUNTAULIDS NOS/HERCULES 3002M)

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" Numbers in parenthesis indicate instrumented specimens

The Steady State Humidity Conditioning Consisted of 96% RM, 120°F for the stated time period see Section 2.1.4.1

The most middity and Accelerated Weathering Cycles are defined in Section 2.1.4.2

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TABLE 111

RESIN MATRIX THERMAL EXPOSURE DATA PROGRAM

(TILS TABLE APPLIES FOR EACH OF THE POLLOGING MATERIALS:)

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^{* !:}mbers in parenthesis indicate instrumented specimens

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^{**} For the details of the steady state thermal conditioning, see Section 2.1.4.3

^{***} Could thermal conditioning involved thermal changes from 100°F to the stated temperature and back to 100°F at a mate of one cph for the stated comber of cycles, see Section 2.1.4.4

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Inter! minar	Steady	Interiguinar Steady 260°F'100 hra		-	,	5	: 		-	,	1	: ! ! .			,			,	٠
	Steady	Stredy 260°F/500 hrs	۳	۳		c		•						,				•	æ
	Steady	Steady 350"F/100 hrs	m	•	,	ı	٠	,		,	,			,	•	•	•	•	ø
	Steady	Steady 350'F 500 hrs	3	•	-		•	• !		,	. :	, ;	-) - !	_	,	,		•	9
	Cyclic	Cyclic 260'F'50C Cyc	6		•	٠.	,				,		,	,		•			9
	Cyclic	Cyclic 260°F/1000 Cv	سم	-	,	ų	1		,	1		•	1					,	•
	Cyclic	300 CCS13-050		ű	•	£			•	,			,					,	9
	Cyclic	Cyclic 350'F/100' Cy	~.	1	۳	9		1				,		•				•	.
Fatigue	Steady	Steady 260°F 500 hrs	5	ş	,	10		ı		,	٠,	s.	,	01	,	٠			50
R = 0.1	Steady	Steady 350°F 500 hrs	٠,	1	iri	0.1		•	1	,		,	- 1	10	,	•	-	•	20
	Cyclic	260°F'500 Cvc			1	10		•		,	ۍ.	v		01				•	20
	Cyclic	Cyclic 260°F/1909 Cv		,		10	1	,	,	,	<u>د</u>	5.		1				,	20
	CVelle	350.5.30C CAL	רט		·r	10	1		1	,	٠.	,	,	01	,				20
	Cyclic	Cyclic 350 F 1000 Cv	·^	•		13	•	•	,	-	٠.		,	16	•	1		•	20
Creep and	Sready	Sready 260°F'500 hrs	,	515	1.215	10/10)	,	•		-	,	\$13.	5(5)	10(10)				•	20(20)
Stress	Steady	Steady 350'F 500 nrs		715)	5(3)	10(10)	,	,			'	5(3)	5(5)	10(10)	,	,	_	•	70(20)
Rupture	کرد	Cyc. 1 260 F 500 Cyc	,	5(5)	3(5)	(01.)01				,		503	_	10(10)				,	20(20)
	Cyclic	Cyclic 2607F 1000 Cv		515)	5:3:	101101	,	1		,	1	5(5)	5(5)	10(10)				,	10(20)
	Cyclic	Cyclic 350°F/500 Cyc	ı	5(5)	5(5)	10(10)				,		5(5)	5(5)	10(10)	•	,		,	20(20)
_	Cyclic	Cyclic 350°F/1000 Cy	,	5(5)	5(5)	(01)01	ı			,		5(5)	5(5)	10(10)				,	20(20)
										1									1

* Lors in parenthesis indicate instrumented specimens

** For the details of the steady state thermal conditioning, see Secti 21.4.3

*** Cyclic thermal conditioning involved thermal changes from 100°F to the stated temperature and back to 100°F at a rate of one c.p.h. for the stated number of cycles, see Section 2,1,4,4

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Table IV

METAL MATRIX DATA PROGRAM

		NO. OF	SPECIMEN	S TO BE TES	NO. OF SPECIMENS TO BE TESTED AT VARIOUS TEMP.*	TEMP. *
	Thermal Conditioning of Specimens	6064 A1/Boron 0° 90°	/Boron	6A1-4V -	T1/BorS1C 90°	TOTAL
Tension:	None	20(12)	20(12)	20(12)	20(12)	80(48)
	Steady State @ T-max** for 100, 500 & 1000 hrs.	. 60(36)	60(36)	(98)09	60(36)	240(144)
	Cyclic @ T-max for 100, 500 & 1000 cycles***	60(36)	60(36)	60(36)	(96)	240(144)
Compression:	None	20(12)	20(12)	20(12)	20(12)	80(48)
	Steady State @ T-max for 100 & 500 hrs.	40(24)	40(24)	40(24)	40(24)	160(96)
	Cyclic @ T-max for 500 cycles	20(12)	20(12)	20(12)	20(12)	80(48)
Flexure:	None	20	20	20	20	ပ စ
Int. Shear	None	10	10	10	10	07
Fatigue (R = 0.1,-1,16)	None	09	09	.09	09	240
Creep and Stress Rupture	None	20	20	20	20	80
Thermal Exp.*** & Conductivity	None	10	10	10	10	07
Test T T-max	RT, 600	A1/Boron RT, 160°F, 400°F, 600°F	600°F	T1/Bors1C RT, 400°F, 600° 800°F	T1/BorS1C 400°F, 600°F, 800°F F	
ATTY TEMP. Kange	1 07C-	4 00/ 04		1 07C-	7.00 F	

**** Temp. Range
**** Thermal cycles consisted of a thermal change from 100°F to the stated temperature and back to 100°F at one CPH for the stated number of cycles. Numbers in parenthesis indicate instrumented specimens.

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SECTION II

2.0 TECHNICAL DISCUSSION

2.1 Resin Matrix Studies

2.1.1 Materials

Advanced composite materials have been under intensive development because of their promise for aircraft structural weight savings, improved capability and potential lower cost compared to conventional structural materials such as aluminum, steel and titanium. The advanced composite materials possess a high strength-to-weight and stiffness-to-weight ratios.

Three resin matrix material systems were selected for study in this portion of the program.

The boron/epoxy system selected was the AVCO 5505/Boron prepreg material. This system was extensively characterized at room temperature and at several elevated temperatures. However, little or no data existed for the effect on the material properties of long-term aging in high humidity and elevated temperature environments. This program included several conditioning environments which will be of interest to designers with boron/epoxy components as flying hardware.

In the time intervening between the purchase of the AVCO 5505/Boron prepreg for use on this program and the completion of this report, the prepreg material was substantially upgraded in average tensile strength. Therefore the values for the AVCO 5505/Boron composites are somewhat lower than can be expected from the newer materials, however, the degradation of the material as a percentage of the ultimate strength will be of value to the designer. In the text, the AVCO 5505/Boron composite summary curves are shown as percentages of the baseline room temperature values.

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Two graphite/epoxy systems were a so selected:

- 1) Modmor II/Graphite Narmco 5206 (a high strength system), and
- 2) Courtaulds HMS Graphite/Hercules 3002M system (a high-modulus system a stiffness of 25 x 10^6 psi).

The raw material was supplied in the three-inch wide tape form for all three systems.

The specifications, to which the systems were ordered and fabricated, were:

- 1) For boron/AVCO 5505 system: General Dynamics specification F.M S.-2001A "Advanced Composite Materials Specifications." The specific type raterial was Type II "Heat Resistant to 420°F. AVCO 5505 has qualified under this specification.
- 2) For Narmoo 5206/Modmor II graphite system: McDonnell Douglas Corporation specification DMS-1936B and all amendments "Tape Unidirectional High Modulus Graphite Filament" was employed. The specific type of interest was Type 1 Continuous filament tape of specified width Class 2 (35 x 10⁶ psi modulus filaments and 195,000 laminate U.T.S.). The Modmor II/Narmoo 5206 system has been qualified by McDonnell Corporation under class 2. 3) For Hercules 3002M/Courtaulds HMS graphite:
 McDonnell Douglas Corporation Specification MMS 546, "Type III Graphite/Epoxy Prepreg Material," and IITRI Specification 0316, "Type HMS Graphite/Epoxy Prepreg Material."

Copies of the three specifications were presented in the Annual Report (AFML-TR-72-205, Part I) and are not repeated here.

2.1.2 Material Procurement, Quality Assurance, and Processing
The following quantities of prepreg tape we e ordered
and received for use on this program: -

AVCO 5505/Boron - 45 lbs.

Modulite 5206 Type II Modmor II/
Narmco 5206 Graphite - 38 lbs.

Hercules 3002M/Courtaulds HMS
Graphite - 55 lbs.

The incoming materials were checked for quality assurance in accordance with the specifications listed in section 2.1.1 of this report. The quality assurance requirements and quality certification reports were presented in AFML TR-72-205, Part I and are not repeated here. A summary of the quality assurance test results are shown in Table V.

Note that the elevated temperature 0° tensile strengths of the Courtaulds HMS Graphite/Hercules 3002M are higher than the room temperature tensile strengths. It is common to this particular system. A more detailed study of this phenomenom is made later in the discussion of the program results.

2.1.3 Fabrication of Laminates

The resin matrix laminates were fabricated using an autoclave to provide the pressure and temperature cycle required.

The autoclave, with internal dimensions of 5'3" in length and 1'8" in diameter provide for the fabrication of either one large plate or several smaller plates simultaneously. The movement of the aluminum heating plate into and out of the autoclave was facilitated by a trolley. The autoclave itself is TABLE V

QUALITY ASSURANCE TEST DATA FOR RESIN

ACCEPTANCE PANEL	CE PAREL	THICKNESS OF FIFTEEN PLIES		O. FLEXIME			90° FLEXURE	URE	1808	HORIZONTAL SHEAR	3
Paterial		Max Non Kax	RT	260°F	350:F	RT	3.09Z	350⁴F	ㅂ	260°F	350.7
-	Penel	in (in) (in.	(ks1)	(ks1)	(ksi)	(ksi)	(ks1)	(ksi)	(481)	(ks1)	(ks1)
Beron/Aves 5305	Gen Dyn PMS-20(1 requirements	180.7970.7970.	275	195	170	13.0	10.0	6.0	13.0	7.0	5.0
	Vendor Q.C. Batch 419	•	306	,	227	8.21	•	15.6	15.0	•	9.0
	IITAL Q.C. Batch 419	510./810./710.	263	240	218	14.0	•	9.5	13.2	12.0	9.1
Mcdmor II Graphite/ Narmco 5206	MAC AIR DMS 1935 Requirements	1	195	,	•	•	•	•	14.0		•
	Vendor Q.C.		6.205	,	1	•		•	16.8		
	IITRI Q.C. Batch 342	•	195.1	•	•		•	•	14.0	•	•
Ccurtaulds NS Graphite/ Hercules 3002H	MAC AIR HAS 548 Type III Requirements	- /505/ -	0° Tensile Modulus Room Temp. 26 x 10° psi	0° Tensile Strength Loca Temp. 100 ksi	O Tensile Strength 350°F 100 ksi	90' Tensile Strength doom Temp. 5,000 pei	•	90° Tensile Strength 350°F 3,500 psi	•	•	3
	Vendor Q.C. Batch 110	- /5/07/ -	32.8 × 10 ⁶	138 ks1	•	5,440 pst	•	3,679 psi			•
	IITRI Q.C. Batch Lilo	- /505/ -	29.0 x 10 ⁶	119 1241	136 kai	5,300 psf	,	4,670 pet	'		

permanently mounted on a steel frame. The heat cycle (maximum capacity 550°F) was automatically controlled. There was a provision for two separate vacuum systems that were used at the same time for fabricating two plates simultaneously. Air pressure up to 100 psi was obtained directly from the air line in the fabrication laboratory.

The preliminary layup procedures followed were developed for the autoclave process. (For the purposes of description in this section, a laminate will mean any composite of several layers of fibers, although the fiber may all be in the same direction.) The tape was first removed from the freezer storage area but was not unwrapped until it had reached ambient conditions. This was done to prevent moisture condensation on the tape surface. The tape was cut to required lengths using a conventional paper cutter and was stacked to the appropriate orientation. (The AVCO 5505/boron laminates required an additional layer of plain woven fiberglass scrim-cloth placed over the entire laminate.) After all plies had been stacked the plate was ready for cure, if convenient, or storage. (Green uncured laminates were sealed in a Mylar bag and stored in a freezer prior to cure if a delay was encountered.)

A stainless steel caul plate, approximately three inches longer and wider than the boron laminate, was used during the curing process. A sheet of TX-1040 separator sheet of the same size as the boron laminate, was placed directly on the stainless steel plate. Next the green laminate and a second separator sheet was added. The aggregate was covered with fiberglass bleeder cloth which was also trimmed to the size of the green laminate. A chloroprene dam consisting of 3/8 inch wide strips of chloroprene was placed around the aggregate. A Mylar perforated sheet was then added. A sheet of 181 fiberglass cloth was then placed on top this stack. The complete package was then placed

on the heater plate in a vacuum bag. Before the cure cycle was initiated, full vacuum was applied to the package, any leaks v re corrected, and a check were made to insure that there were no wrinkles on the laminate.

The cure cycles and postcure used for the three material systems were described in AFML TR-72-205, Part I. Following the cure the individual plates went to specimen cutting, tabling or environmental conditioning processes as appropriate. Individual specimens were inspected visually for flaws and delaminations. Composite describes and fiber and resin volume percentages were determined as escribed in AFML TR-72-205, Part I. The data for individual laminates are presented in Tables VI through VIII.

The densities shown in Tables VI through VIII were determined using the gravimetric process. The values for the densities of the fibers and matrices were obtained from the tape suppliers. No void contents are shown in Tables VI through VIII. This does not imply that the composites were void-free but of low voids. Several inherent inaccuracies are present in resin dissolution methods currently available thus leading to void contents with errors of 100% or greater.

2.1.4 Conditioning Treatments

The various conditioning treatments, to which the composite materials were exposed are described in this section. The equipment and procedures followed in the accomplishment of these conditioning treatments are found in AFML TR-72-205, Part I and are not repeated here.

VOLUMETRIC MEASURES OF FIBER AND MATRIX CONTENTS

IN BORON/AVCO 5505 COMPOSITES

Fiber Orientation	No. of Plies	Specimen Number	Specimen Length (in.)	Specimen Width (in,)	Density of Composite Gm/cc	Fiber Volume (Percent)	Resin Volume (Percent)	Fiberglass Volume * (Percent)
0.	6	N-1005	1.018	0.972	2.000	49.87	43.77	6.36
		N-1007	2,000	0.250	2.001	49.96	43.70	6.34
		N-1008	2.000	0.250	2.013	50.75	42.89	6.36
		N-1009	2.000	0.250	1.996	49.41	44.08	6.51
		N-1011	2.000	0,250	1.992	49.25	44.34	6.41
		N-1012	2.000	0.250	1.995	50.00	44.08	5.92
			AVERACI	E	2.000	49.87	43.81	6.32
90 °	В	N-1002	2,050	0.441	1.980	48.60	45.25	6.15
		N-1013	2.000	0.250	1.960	47.26	46.77	5.97
		N-1014	2. 00 0	0.250	1.962	47.36	46.58	6.06
		N-1015	2.000	0.250	1.961	47.31	46.77	5.92
		N-1017	2.000	0.250	1.959	47.11	46,82	6.07
		N-1018	2.000	0.250	1.985	48,70	45.20	6.10
			AVERAGI	Ξ.	1,968	47.72	46.23	6.05
+ / 5°	b	₹-10;2	2.000	0.222	2,010	50,44	42.98	6.58
		N-1023	2.000	0.250	1.979	48.43	45.45	6.12
		N-1024	2.000	0.250	1.992	49.31	44.50	6.19
		N-1025	2.00C	0.250	1.993	49.41	44.36	6.23
		N-1026	2.000	0.250	1.992	49.39	44.44	6.17
			AVERAGE	2	1,993	49.40	44.35	6.25
10/55/135/0/ 9 0,),_9	N-1027	1.553	0.561	1.980	48,74	45.23	6.03
	•	N-1028	2.000	0.250	1.979	48.71	45.36	5.93
		%-1029	2,000	0.250	1,959	47.14	46.89	5.97
		N-1030	2.900	0.250	1.978	48,60	45.50	5.90
		N-1031	2.000	0.250	1.966	47.75	46.40	5.85
		N-1032	2.000	0.250	1.985	49,00	44.91	6.09
		N-1033	2.000	0.250	1.996	49.71	44.15	6.14
		N-1034	2.000	0,250	1.985	48.70	45.18	6.12
		N-1034	2.000	0.250	1.972	47.55	56.48	5 97
		N-1036	2,000	0.250	1.973	48.19	45.8	5.99
		N-1037	2,000	0.250	1,969	47.77	46 08	6,15
		8-1038	2,000	0.250	1.973	47.93	45.86	6.21
		N-1040	2.000	0.250	1.967	47.36	46.33	6.31
		N-1041	2,000	0 250	1.965	47.44	46.35	6.21
		N-1042	2.000	7,.250	1.964	47.65	45.51	5.84
			AVERAGE	-	1.974	48.15	45.80	6.05

^{*} of carrier glass scrim cloth

Table VII

VOLUMETRIC MEASURES OF FIBER AND MATRIX CONTENTS
IN MODMOR II GRAPHITE/NARMCO 5206 COMPOSITES

Fiber Orientation	No. of Plies	Specimen Number	Density of Composite (gm/cc)	Fiber Volume (percent)	Resin Volume (percent)
0,	6	M1105	1.491	53.15	46.85
		M1106	1.521	60.46	39.54
		M1107	1,515	58.81	41.19
	}	M1108	1,517	58.79	41.21
		M1109	1.493	54.21	45.79
		M1110	1.481	51.70	48.30
		M1111	1.496	54,48	45.52
		M1112	1.485	51,45	48.55
		AVERAGE	1.499	55.38	44.62
	15	M1101	1,513	58.09	41.91
	10	M1147	1,503	55.97	44.03
90 -	8	M1102	1,516	58,72	41.28
]	M1103	1,474	49.80	50.20
		M1104	1.501	55.55	44.45
		M1113	1.473	61.03	38.97
		M1114	1,504	55,13	44.87
		MILIS	1.467	47.27	52.73
		M1116	1.479	50.74	49.26
		M1117	1,516	58,71	41.29
		M1118	1,504	56,57	43.43
		M1120	1.486	51.84	48.16
		AVERAGE	1,492	54.54	45.46
+ 45°	8	M1122	1.490	53.47	46.53
		M1123	1,496	54 .55	45.45
		M1124	1,475	49.84	50.16
		M1125	1.484	52.04	47.96
		M1126	1,488	52,72	47.22
		AVERAGE	1,486	52.53	47.47
0/45/135/0/ 9 0}	9	M1127	1.479	50,76	49.24
5		M1128	1.473	49.64	50.36
		M1129	1.471	49,22	50.78
		M1130	1.481	51.28	48,72
		M1131	1.465	52,11	47.89
		м113°	1,480	51,14	48,86
		M1133	1.465	47.92	52.08
		M1134	1.47:	49.52	50.48
		M1135	1,495	54.25	45.75
		M1136	1.505	56.49	43.51
		M1137	1,489	52,95	47,05
		M1138	1,483	48,68	51.32
		M1139	1.488	52.83	47.17
		H1140	1.479	50 -82	49.18
		M1141	1,498	54.87	45.13
		M1142	1.477	50.41	49.59
		H1146	1,472	69,36	50.64
		AVERAGE.	1,481	51,30	45.70

Table VIII

VOLUMETRIC MEASURES OF FIBER AND MATRIX CONTENTS
IN HERCULES 3002M/COURTEULDS HMS GRAPHITE COMPOSITES

Orf ntation	No. of Plies	Specimen Number	Density of Composite (gm/cc)	Fiber Volume (percent)	Resin Volume (percent)
0.1	6	C1205	1.593	49.72	50.28
		C1206	1.568	45.73	54.27
		C1207	1,585	48,44	51.56
, [C1208	1.573	46.50	53.50
		C1209	1.605	51 .61	48.39
		C1210	1.589	49.07	50.93
		C1211	1.574	46.72	53.28
		C1212	1.567	45.58	54.42
ŧ		AVERAGE	1,581	47.92	52.08
l	10	C1247	1,507	51.91	48.09
ae.	8	C1202	1.578	47.3%	52.66
·		C1203	1 601	51.95	49,05
		C120a	1,569	45,90	54.10
		(121)	1.55+	43.35	56.65
:		C1214	1,580	47.63	52.37
:		C1215	1,572	46.37	53.63
		C1216	1,591	49.38	50.62
1		61217	1,594	49.85	50.15
		C1218	1,584	48,25	51.75
		C1219	1,569	45,88	54.12
		AVFRAGE	1.579	47.49	52.51
± 45°	8	C1222	1,613	52.86	47.14
		C1223	1.589	49,06	50.94
i 1		C1224	1.573	46,43	53.57
ĺ		C1225	1,584	48,26	51.74
,		C1226	1.578	47.31	52,69
<u> </u>		AVERAGE	1.587	48.78	51.22
10/45/135/0/ 90 1	G,	C1221	1 589	49.06	50.94
,		C1227	1,599	48,10	51 .9 0
		1 C1228	1,579	47,46	52,54
		€1230	1,550	42.85	57.15
		İ (1231	1,583	48,10	51.90
		€1232	1,539	41.11	58,89
:		C1233	1,560	44.44	55.56
'		· C1234	1,553	48.86	51.14
ļ		C1236	1,552	43.20	56.80
		C1237	1,531	39,85	60.15
:		i (123)	1.550	42.87	57.13
:		C1239	1.566	45.37	54.63
		C1240	1,566	44,45	55.55
; !		C1261	1.572	46.35	50.00
ļ		C124	1.555	43.66	56,34
		C126.	1,570	46 04	53,96
		AVERAGE	1,563	45.11	54.89

In addition, a comparison base of data was obtained against which the effects of these various conditioning treatments might be measured. The extent of this baseline data program was described in Section I, Table I. The individual baseline data for the three resin matrix systems are found in Appendices I through III.

2.1.4.1 Steady State Humidity Conditioning

The steady state humidity conditioning of specimens includes 500 and 1000 hr. (3 weeks and 6 weeks) exposure to 98% ± 2% relative humidity and 120°F (see Table II). This exposure is the same as that recommended by Mil Handbook 17.

The specimens which were subjected to humidity exposure were prepared as follows:

- 1) All specimens were finish machined and the appropriate room temperature or elevated temperature tabs were bonded prior to initiation of the preconditioning treatment. For elevated temperature tests subject to prior humidity exposure the tab adhesive was Metalbond 329. For room temperature tests subject to prior humidity conditioning the adhesive was FM 1000.
- 2) All specimens for static and creep tests were instrumented (as required) with electrical resistance foil strain gages. The gages were protected with M-coat resin coating taking care to cover a minimum area.
- 3) The edges of the samples were not protected since protection could not be guaranteed to be only to the edges and not to the surfaces of specimen.

- 4) The samples were individually weighed prior to insertion in the chamber.
- 5) Each sample was arranged in the chamber to permit maximum exposure to the moisture-laden air as it flowed from the inlet orifice to the chamber.

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These steps were followed to permit rapid testing of the samples after removal from the chamber. Upon removal from the chamber, the specimens were reweighed, wires were attached to the strain gages and the specimens were tested within 8 hours of removal from the chamber. For certain long term fatigue and creep tests, where the tests were held up for a longer time due to machine unavailability, the samples were sealed in a protective vinyl, moisture proof container. These samples were then reweighed, prior to testing, to determine if moisture loss had occurred.

2.1.4.2 Cyclic Humidity Conditioning

2.1.4.2.1 Thermo-Humidity Cycle

Table II listed two cyclic humidity conditioning exposures for resin matrix composites. The first humidity cycle was the Thermo-Humidity cycle selected from a review of previous aerospace practices. The Webber Environmental Chamber was again used for the humidity exposure.

The details of the Thermo-Humidity cycle employed are: (1) The total time period for the cycle was 500 hours. (2) During this period the specimens were placed in the environmental chamber and exposed to a relative humidity of $95\pm2\%$ at $120^{\circ}\pm5$ °F except for one and one half hour each work day of the week when they were taken out and subjected to thermal shock. (3) This shock treatment consisted of exposing the

specimens for one hour at ~65°F in a cold chamber followed by an exposure of one half hour at 250°F in an oven. (4) During The weekend the specimens remained in the environmental chamber continuously exposed to the humidity conditions mentioned above.

The frost conditions on the samples after exposure to -65°F were noted and some sample delaminations occurred after removal from the 250°F portion of the cycle.

All appropriate specimens were strain gaged in the same manner as the steady-state exposure and were wired after exposure prior to testing. The test specimens were made ready for testing within eight hours after removal from the test chamber as was done for the steady state humidity conditioning exposures.

2.1.4.2.2 Accelerated Weathering Humidity Cycle

The second humidity cycle was an accelerated weathering cycle. An Atlas Twin ARC Weatherometer, Type D as specified in ASTM G23-69 was employed for these tests. All panels and/or specimens were exposed in the weatherometer to the following operation schedule. The recommended practice for this equipment was as described in ASTM D1499-64 and ASTM G23-69. The apparatus was operated 5 days per week, and each 2-hour cycle of operation was divided into periods, during which the panels and specimens were exposed 102 minutes to light without water and 18 minutes of light with water spray. The test specimens remained undisturbed during the remaining 2 days of the week.

The exposure procedures followed were as follows:

The black panel thermometer unit was placed in the test panel rack and with the light on and the water off, the thermometer regulator was set so that the temperature of the thermometer read $145^{\circ} \pm 5^{\circ} F$, when the thermometer was at the point where the maximum heat was produced as the panel rack revolved around the light.

The water supply was adjusted so that the pressure of the water at the spray nozzle was between 12 and 15 pounds per square inch so that the water struck the specimens in a fine spray in sufficient volume to wet the entire surface of the specimens upon impact.

New carbons and clean filters were installed in the light assembly and the weatherometer was started. At the end of the burning period, the old carbons were removed and the decomposition ash was cleaned from the carbon holders and other parts of the light assembly, and the filters were washed with detergent and water. The position of the test panels and specimens were transposed to provide a uniform distribution of light in a vertical plane over the entire surface of the test specimens. New carbons were installed, the filters were replaced and the weatherometer restarted. These operations were repeated after each burning period of the light until the test specimens were exposed for a time period of 500 hours including weekend rest periods. (This resulted in a 360 hour active exposure time plus 140 hours of rest periods.)

2.1.4.3 Steady State Thermal Conditioning

For steady state thermal exposure conditioning conventional circulating air ovens were used to obtain exposures at 260°F for time periods of 100 and 500 hrs. The samples were arranged to get uniform distribution of air circulation over the specimens without localized hot spots.

2.1.44 Cyclic Thermal Conditioning

Thermal cycles from 100°F to 260°F to 100°F and from 100°F to 350°F to 100°F were adopted for cyclic thermal conditioning. Exposure of tes- samples for both 500 cycles and 1000 cycles were undertaken. A cyclic rate of one cycle per hour was established.

2.1.5 <u>Testing Specimens and Test Procedures</u>

This section briefly lists the test specimens and procedures utilized for generating the data during this program.

A detailed description of the test specimens, specimen fabrication procedures and test equipment is found in Appendix II of AFML TR-72-205, Part I.

2.1.5.1 Tensile, Fatigue and Creep Specimens

The same specimen configuration was utilized for tension, fatigue (R=0.1) and tensile creep tests. In addition in plane shear properties were determined using a \pm 45° tensile test. The IITRI straight-sided tab ended coupon was utilized for these properties. After environmental conditioning, each static tensile specimen was fitted with three electrical-resistance foil strain gages.

2.1.5.2 Compression Testing

Two types of specimens were employed for compressive testing. The first was the sandwich beam compression specimen which was utilized only in the generation of baseline data. The second specimen was a coupon specimen commonly known as the Celanese specimen which is an adaptation of the IITRI tensile coupon with longer tabs, reduced gage section and a narrower width. The coupon test fixture was the TITRI compression coupon test fixture.

(All comparative performance results are shown using the coupon test data for the baseline and conditioned curves).

2.1.5.3 Flexural and Interlaminar Shear Tests

The specimens used for all flexural testing was the fifteen ply, coupon universally used for testing advanced composites. Specimens were loaded in a 3 or 4-point bending fixture. Elevated temperature tests were conducted in a Missimer circulating air oven and loads were applied in tension to a flexural test rig.

The maximum interlaminar shear strength of oriented fiber composites was determined on short beam shear specimens.

Elevated temperature tests were performed with the assistance of the fixture described above.

2.1.5.4 In-Plane Shear Properties

The in-plane shear stress-strain curve was determined from a \pm 45° angle ply laminate tested in uniaxial tension supplemented with data from the 0° and 90° tests, and the incrementation of the \pm 45° tensile stress strain curve.

2.1.5.5 Fatigue Tests

The fatigue tests (R = 0.1) were performed at a cyclic rate of 1800 cpm, employing occentric weight mechanical dynamic-load applicators.

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2.1.5.6 Creep and Stress Rupture Tests

The creep equipment consisted of 32 tensile stands located on a vibration-free floor. Each stand was provided with a set of tensile grips enclosed in individually controlled ovens. The ovens are capable of achieving specimen temperatures of up to 800°F. A jig was used to align and grip the specimens prior to installation on the creep stands. For the creep stands employed, the load multiplication factor was 10:1.

2.1.5.7 Thormophysical and Bonsity Proporties

The linear expansion was measured by an automatic recording dilatometer similar to that described in ASTM Designation: C337-57. The dilatometer used had an accuracy of more than 99% and a reproducibility within \pm 2%.

Thermal conductivity measurements were made using the steady state longitudinal heat flow technique. The sample consisted on ten $3/64 \times 1/2 \times 2$ -inch laminates sandwiched together to form a $1/2 \times 1/2 \times 2$ -inch conductivity specimen. Data are obtained from ambient room temperature to $350^{\circ}\mathrm{F}$ in air for three specimens in each of three laminate orientations. Densities of the laminates were determined by the gravimetric method,

2.1.6 Static Properties

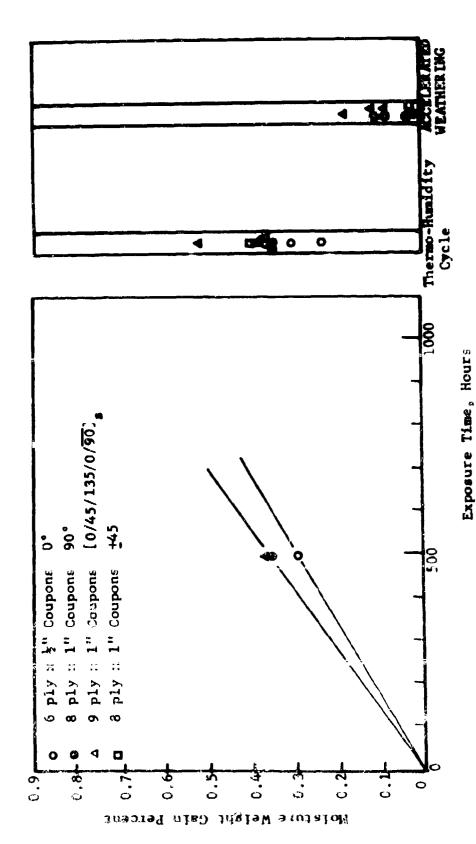
2.1.6.1 Baseline Data

The static baseline data are found summarized in Appendices I through III including average stress strain curves in tension compression, and shear for 0° , 90° and $\left[0/45/135/0/90\right]_{\rm g}$ laminates. The data were obtained from strain gages and were reduced and plotted using computer plotting routines. To average the values of stress and strain obtained from three tests conducted at a given temperature, a program (least squares) to fit a curve to the data was used as a sub-routine to the plotting program.

2.1.6.2 Effects of Humidity Conditioning

The steady state exposure of the three resin matrix composite materials to 98% relative humidity resulted in moisture pickup by the exposed uncoated samples. Fig. 1 shows the moisture pickup versus time for AVCO 5505/Boron. This figure is an aggregate of moisture pickup for three orientations three thicknesses (ply thickness) and two widths of sample so that the ratio of surface area to volume of the samples varies over a substantial range and the ratio of exposed fiber ends to surface area also varies.

Figures 2 and 3 also present the moisture pickup versus time for the Modmor II/Narmco 5206 Composite and the Courtaulds HMS Graphite/Hercules 3002M epoxy composites respectively. The moisture pickups are presented as a percentage of the original weight of the specimens. In plotting these gains for the four different humidity environments account was taken of the various orientations, specimens sizes etc. (see legend on each figure). Thus while the surface area to volume ratio for a nine ply $[0/45/135/0/90]_S$ laminate may remain virtually the same as a six pty $[0/35/135/0/90]_S$ laminate, the exposed Those challed on the $[0/.5/135/0/90]_S$

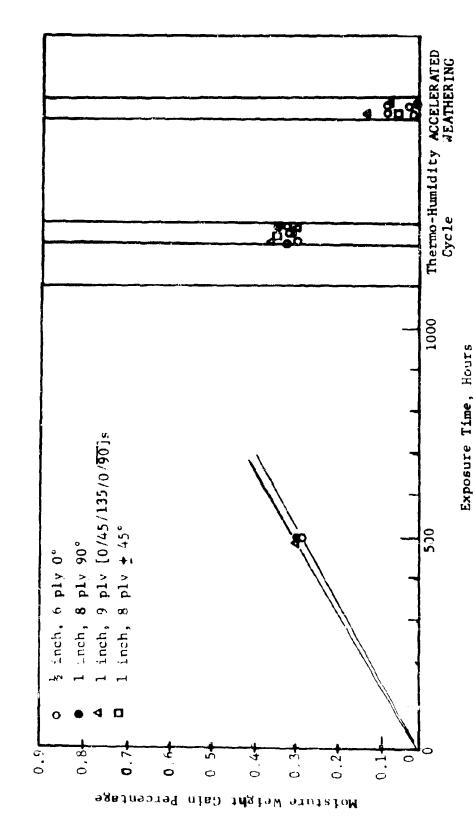


MOISTURE WEIGHT GAIN PERCENTAGES FOR VARIOUS HUMIDITY CONDITIONING FOR AVCO 5505/BORON COMPOSITES rig.

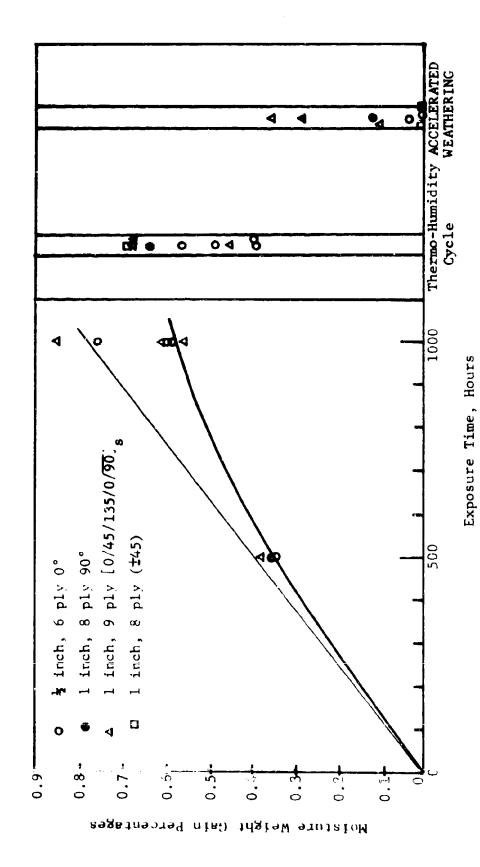
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MOISTURE WEIGHT GAIN PERCENTAGES FOR VARIOUS HUMIDITY CONDITIONING FOF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES Fic.



MOISTURE WEIGHT GAIN PERCENTAGES FOR VARIOUS HUMIDITY CONDITIONING FOR HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES ന Fig.

laminate provide more potential entry paths for moisture to enter the specimen.

Groups of specimens of a given type were inserted at various times into the humidity chamber on their appropriate schedules. Therefore several different points appear at the same total exposure time. Each point represents an average of from 10 to 20 specimens of the type indicated. Thus the variability of moisture pickup from group to group can be obtained from Figs. 1 - 3 as well. Figs. 1 and 2 do not show any 1000 hour steady-state moisture pickups. This data was not obtained.

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The results for the Thermo-Humidity Cycle and the accelerated weathering cycles show marked differences between specimen orientation. By examining the AVCO 5505/Boron spread for the Thermo-Humidity cycle one sees that the 0° specimens percentage weight gain falls to the lower side of the spread of data while the 90° and $[0/45/135/0/\overline{90}]_S$ specimens generally lie at the top of the spread of data (indicating higher moisture pickup percentages). The same qualitative remarks apply to the accelerated weathering mo sture pickup data even though the mean and spread of the accelerated weathering data are smaller.

In general, the Thermo-Humidity cycle data corresponds to approximately 500 hour— f constant humidity exposure and the accelerated weathering data corresponds with approximately 50 to 150 hours of constant humidity exposure (for AVCO 5505/Boron). Slightly smaller weight gains were recorded for the Narmoo 5206/Modmor II Graphite than were recorded for the AVCO 5505/Boron system. However, qualitatively the relationship of fiber orientation to moisture pickup remained the same. Similarly the correspondence of the constant relative humidity to the Thermo-Humidity cycle and the accelerated weathering cycles remained the same.

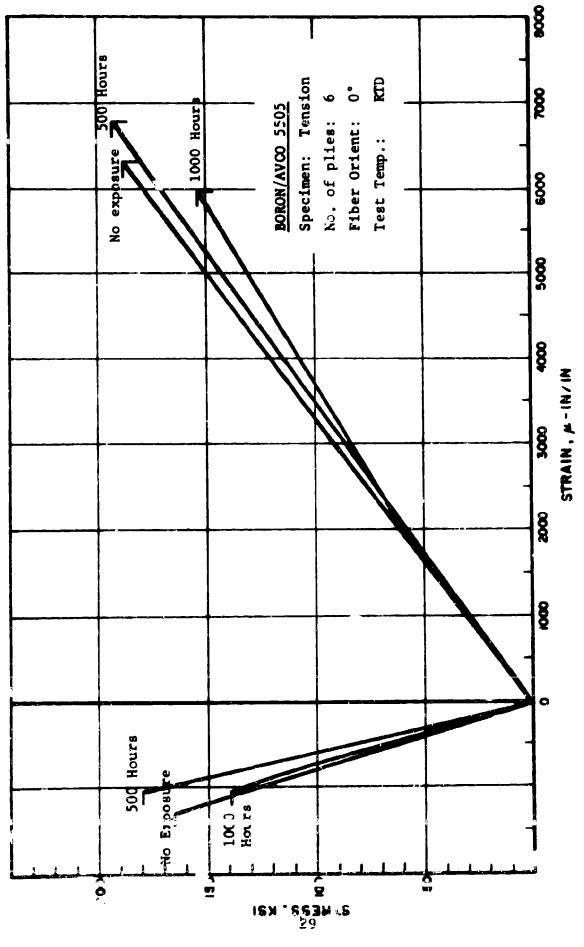
The largest moisture pickups and greatest data spreads were found in the Courtaulds HMS Graphite/Hercules 3002M system. Here the moisture pickup for the Thermo-Humidity cycle exceeded the moisture pickup for constant humidity at 500 hours, corresponding more closely with a constant humidity exposure of approximately 800 hours. Similarly, the accelerated weathering moisture pickup was greated but less than in the Thermo-Humidity cycle, and corresponded to constant humidity exposures ranging from a couple of hours up to 500 hours. The greatest number of data points fell nearer the bottom of this range (at or near 75 hours exposure on the constant humidity moisture pickup curve).

These correspondences with the constant humidity moisture pickup curves were not surprising since, in fact, the total exposure time for the Thermo-Humidity cycle to 98% RH at 120°F (the constant humidity exposure) was 500 hours less than the 1-1/2 hours per day times 15 days or approximately 478 hours exposure. Similarly the net exposure time for the accelerated weathering samples to high moisture was,

15 x $(\frac{18}{60})$ x 12 periods - 54 hours of net exposure time to moisture with 306 hours of light and heat plus 140 hours of inactivity out of a total of 500 hours in the exposure cycle.

In summary the cyclic humidity conditioning treatment: produce moisture gains approximately the same as that for the net moisture exposure time during the constant humidity exposures.

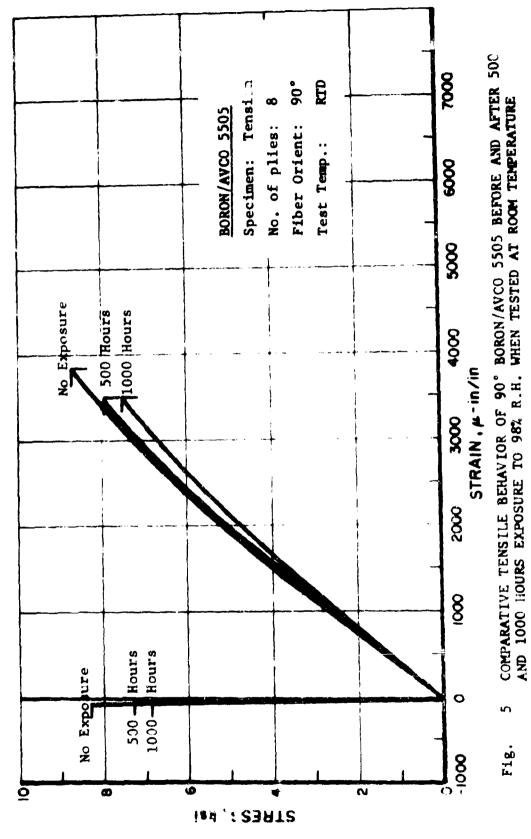
The room temperature longitudinal tensile stress-strain behavior of AVCO 5505/Boron composite material is shown in Fig. 4 after 500 and 1000 hours exposure to 98% RH. The transverse tensile stress-strain behavior of AVCO 5505/Boron is shown in Figs. 5 to 7 for room temperature, 260°F and 350°F respectively. It is apparent from these curves that the effect of moisture is



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COMPARATIVE TENSILE BEHAVIOR OF 0° BORON/AVCO 5505 BEFORE AND AFTER 500 AND 1000 HOURS EXPOSURE TO 98, R.H. WHEN TESTED AT ROOM TEMPERATURE N.

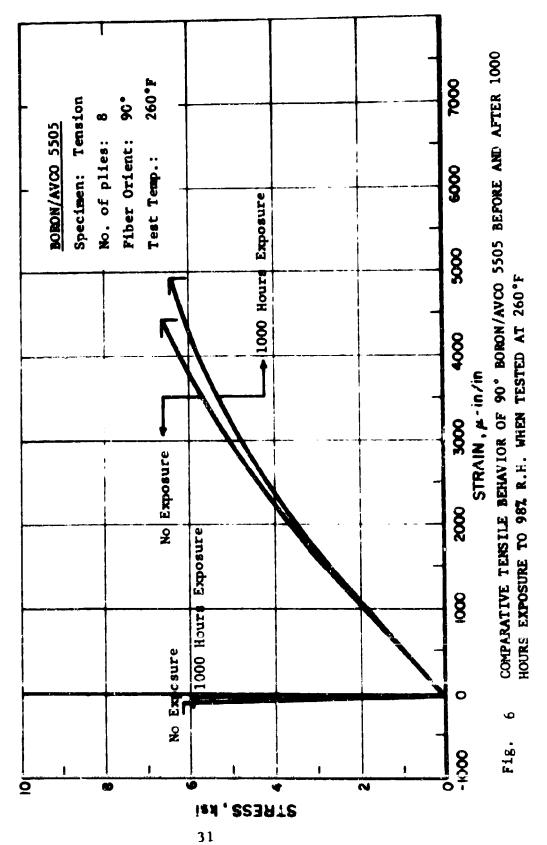


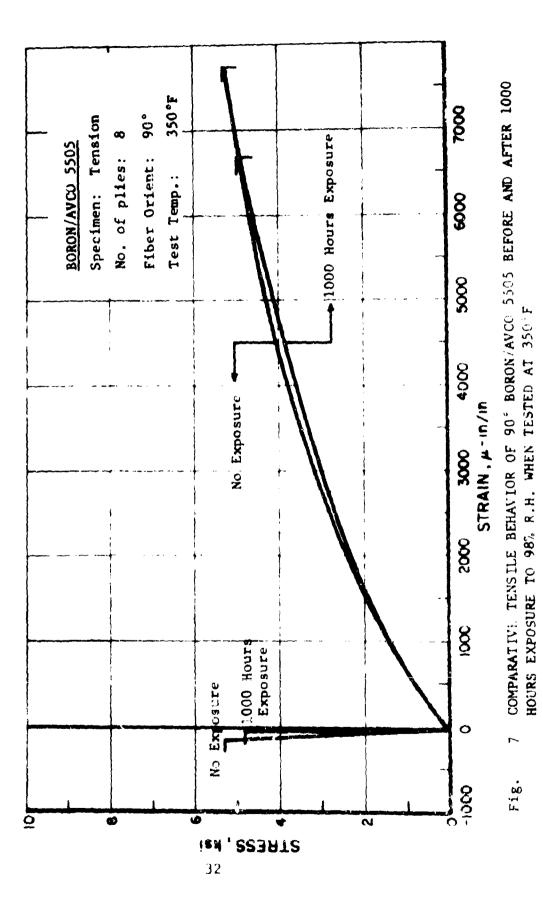
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Fig.





to generally reduce the strength (and ultimate strain) capabilities of the 90° composite. However, the stress-strain characteristics, and particularly, the initial modulus, were not radically altered by the exposure. The effect on cross-ply laminates is shown in Figs. 8 and 9 where no significant changes in strength or modulus are observed.

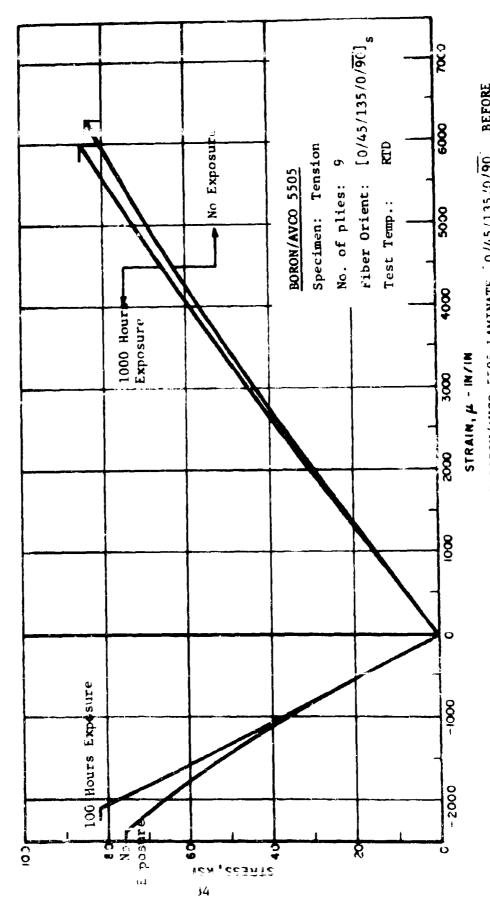
The in-plane shear behavior as affected by moisture is shown in Fig. 10. Here there is a gradual loss in modulus, a slight reduction in strength and a slight gradual increase in ultimate strain capability. This 'oss in modulus is reflective of a matrix and interface change and does not indicate a change in fiber modulus.

The Thermo-Humidity cycle delaminated several of the samples after a hundred hours of exposure. The back-to-back high-low thermal changes were chiefly responsible. This effect was most noticeable in the high-modulus graphite and seemed to be least present for the boron/AVCO 5505 epoxy composites. This effect is caused more by the high differential thermal expansion present in the graphite/epoxy composites compared with that in the boron/epoxy composites.

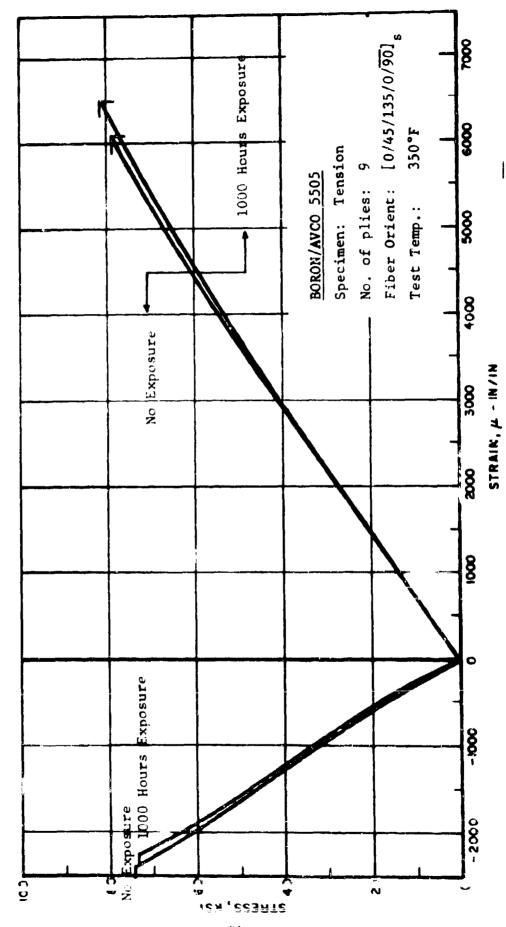
The greatest damage was sustained by the Hercules 3002M/Courtaulds HMS Graphite system. Damage was noted in some of the $[0/45/135/0/90]_8$ systems with delamination clear to the end of the sample. Where such damage was detected, the samples were tested and the delamination noted in the data tables in the appendices to this report.

The Thermo-Humidity cycle is discussed in (1)*.

^{*} Numbers in parenthesis refer to the References at the end of this report.

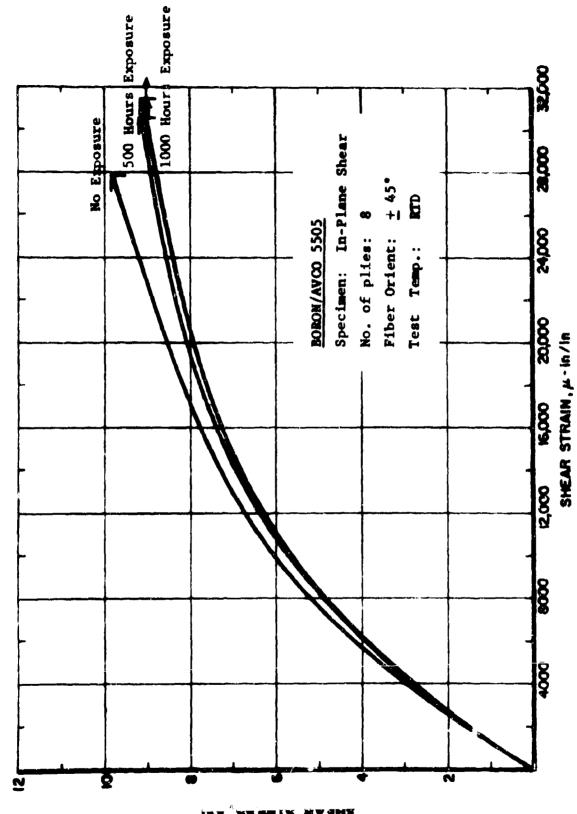


COMPARATIVE TENSILE BEHAVIOR OF BORON/AVCO 5505 LAMINATE [0/45/135/0/90] BEFORE AND LETER 1000 HOURS EXPOSURE TO 98, R.H. WHEN TESTED AT ROOM TEMPERATURE (d) (d)



COMPARATIVE TENSILE BEHAVIOR OF BORON/AVCO 5505 LAMINATE (0/45/135/0/90 BEFORE AND AFTER 1000 HOURS EXPOSURE TO 987 R.M. WHEN TESTED AT 350°F ġ, F. 19.

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COMPOSITE BEFORE AND AFTER 500 AND 1000 HOURS EXPOSURE TO 987 R.H. WHEN TESTED AT ROOM TEMPERATURE COMPARATIVE SHEAR STRESS-SHEAR STRAIN BEHAVIOR OF BORON/AVCO 5505 10 Fig.

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The accelerated weathering cycle contained a U.V. exposure contribution which was not present in the other humidity cycles. A great deal has been published on the correlation between laboratory exposure times and field times, or on the relationship between exposure duration in one area versus exposure duration in another area. The basic photodegradative process is fairly simply stated: the weathering mechanism is a process of chemical change in which the ultraviolet radiation is the source of the energy for these changes and the air and/or water provides the oxygen etc. for the chemical change. The photodegradative efficiency of the solar energy is inversely related to the wavelength, the shorter or U.V. wavelengths causing the greatest damage.

Accelerated weathering cycles are discussed more extensively in references (2) through (13).

Several parametric crossplots illustrating the effect of moisture pickup on the mechanical properties were prepared. Figures 11 to 13 show the effects of moisture on strengths of 0°, 90° and $\left[0/45/135/0/\overline{90}\right]_{\rm S}$ laminates of AVCO 5505/Boron composite material. Figures 14 to 16 show the effects of moisture on the elastic moduli of these three composites.

AVCO 5505/Boron with temperature for the baseline data as shown in Fig. 11. Similar effects are seen for the 0° compressive and 0° shear strengths. With the exception of the room temperature tensile strengths, Figs. 11a -c show that the strengths generally decrease for 500 hours exposure to 98% RH with additional decrease after 1000 hours exposure. The room temperature tensile strengths are probably too low in Fig. 11a. The cyclic humidity conditioning resulted in generally greater

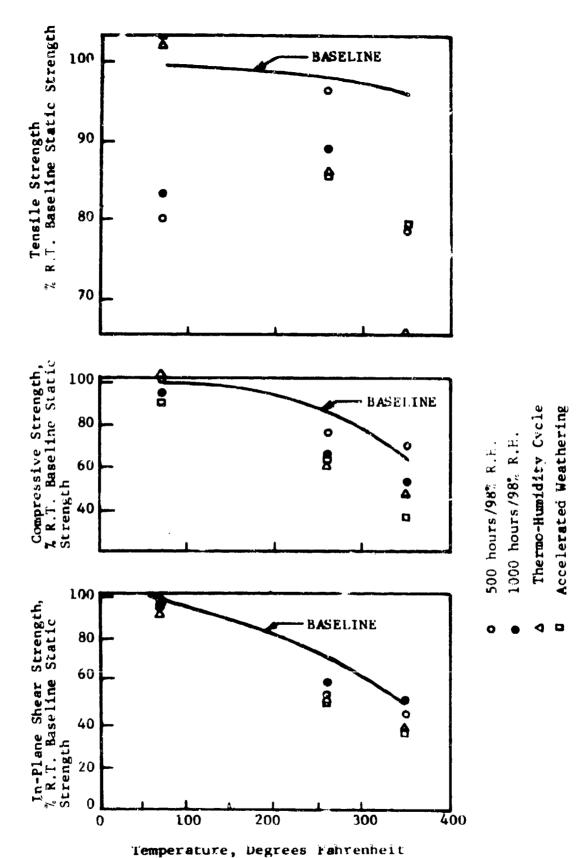
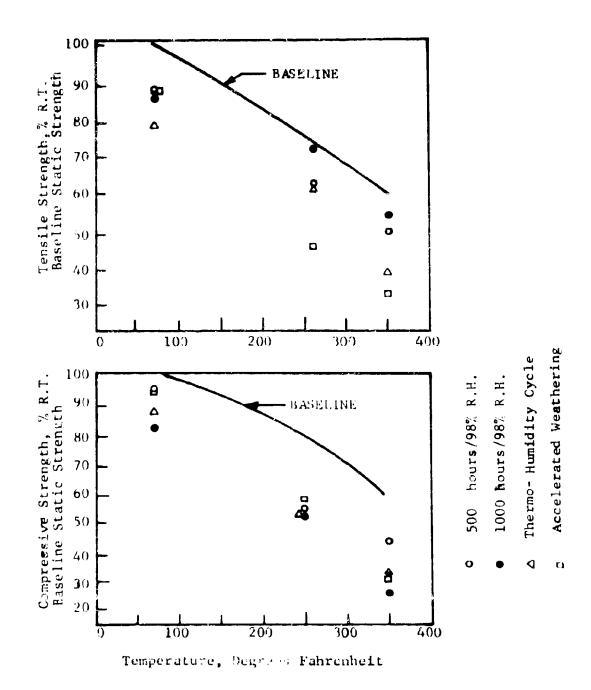
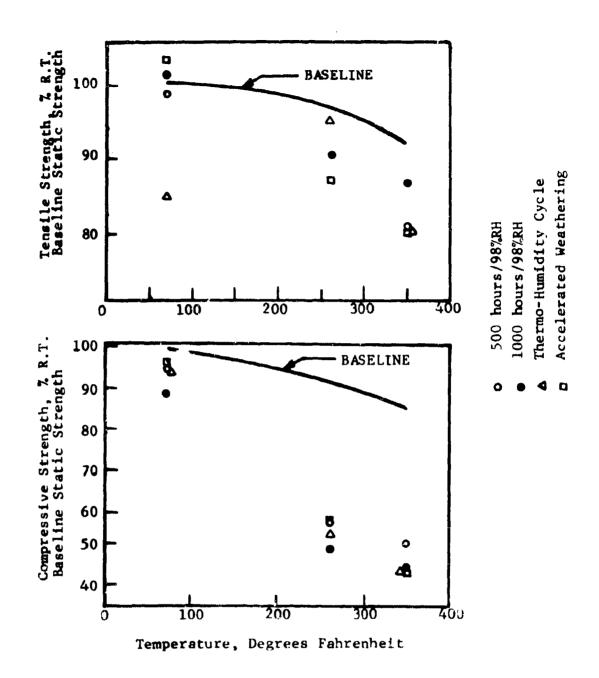


Fig. 11 EFFECT OF HUMIDITY CONDITIONING
ON STRENGTHS OF AVCO 5505/BORON COMPOSITES - 0°



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Fig. 12 EFFECT OF HUMODITY CONDITIONING ON STRENGTHS OF AVCO 5505/BORON COMPOSITES - 90°



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Fig. 13 EFFECT OF HUMIDITY CONDITIONING ON STRENGTHS OF AVCO 5505/BORON LAMINATES [0/45/135/0/90]s

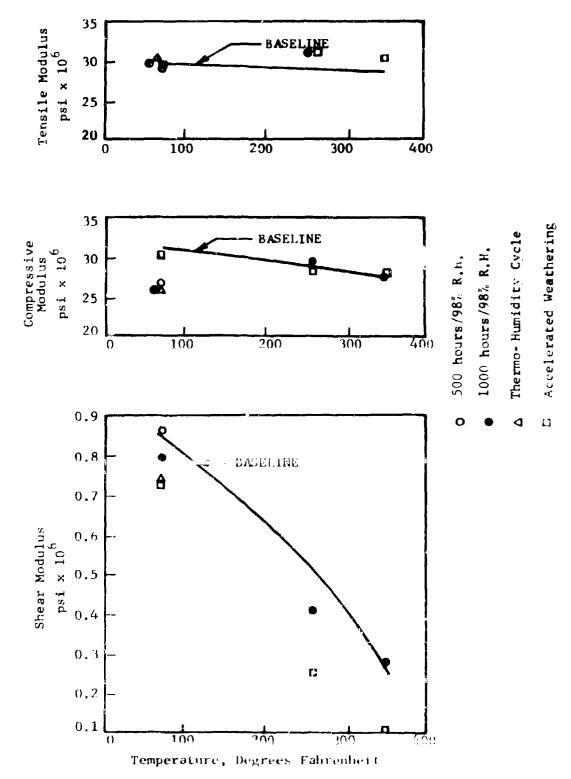


Fig. 16 EFFECT OF HUMIDITY COMDITIONING ON ELASTIC MODULI OF ALCO 5505/BOROR COMPOSITES -0

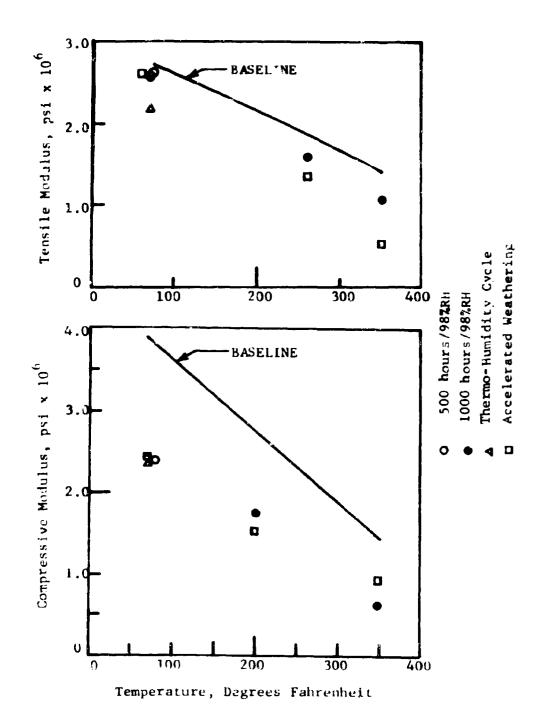


Fig. 15 EFFECT OF HUMIDITY CONDITIONING ON ELASTIC MODULI OF AVCO 5505/BORON CCMLOSITES 90°

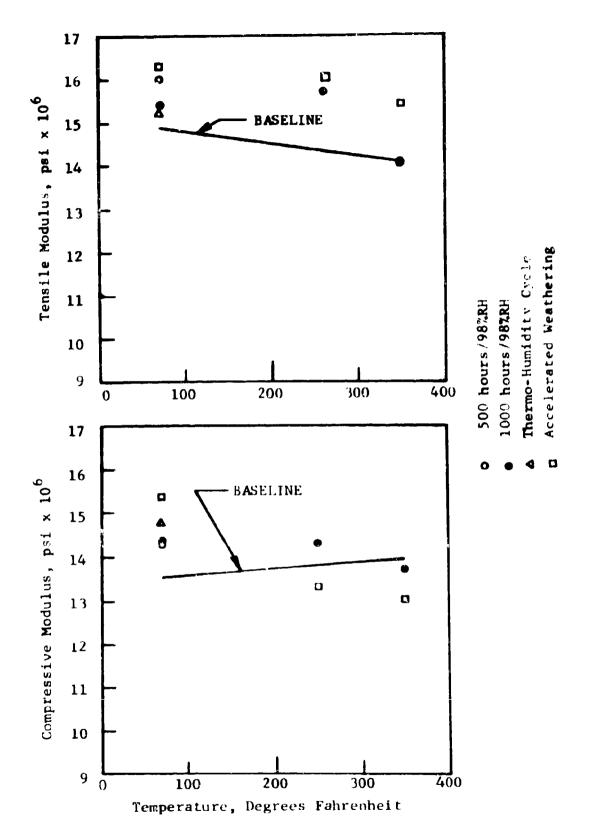


Fig. 16 EFFECT OF HUMIDITY CONDITIONING ON ELASTIC MODULI OF AVCO 5505/30RON COMPOSITES [0/45/135/0/90]s

strength reductions than did the steady state exposures. The transverse strengths are shown in Fig. 12a - c. Again the cyclic humidity exposures affected the tensile and compressive strengths more than did the constant humidity exposures.

The laminate strengths are plotted in Fig. 13. The elevated temperature compressive properties of the $[0/45/135/0/\overline{90}]_8$ laminates were severly affected by both the steady state and cyclic humidity exposures.

The 0° elastic moduli were not as substantially affected by humidity environment (See Fig. 14) as were the 90° and laminate elastic moduli (See Figs. 15 and 16 respectively). The crossply laminate, $\left[0/45/135/0/\overline{90}\right]_8$, stiffnesses increased over baseline values as a result of the humidity conditioning. The rate of stiffness decrease with temperature did not change for the 0° and 90° composites as it altered for the laminates.

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With regard to the compression strength plots it should be noted that the coupon compressive values were used for these comparisons as in all following comparisons.

Similarly cross plots were made for Modmor II Graphite/ Narmco 5206 composites (Figs. 17 - 22) and Courtaulds HMS Graphite/ Hercules 3002M Composites (Fig. 23 - 28).

In several ways, the two graphite composites behaved similarly. The 0° tensile strength for both Modmor II/Narmco 52° and Hercules 3002M/Courtaulds HMS Graphite Composites increased over the baseline 0° strength at room temperature although the latter's strengths at elevated temp ratures full below the baseline strengths. The inplane shear strengths for the two materia's fell close to the baseline values over the entire temperature range. Furthermore, the inplane shear strengths

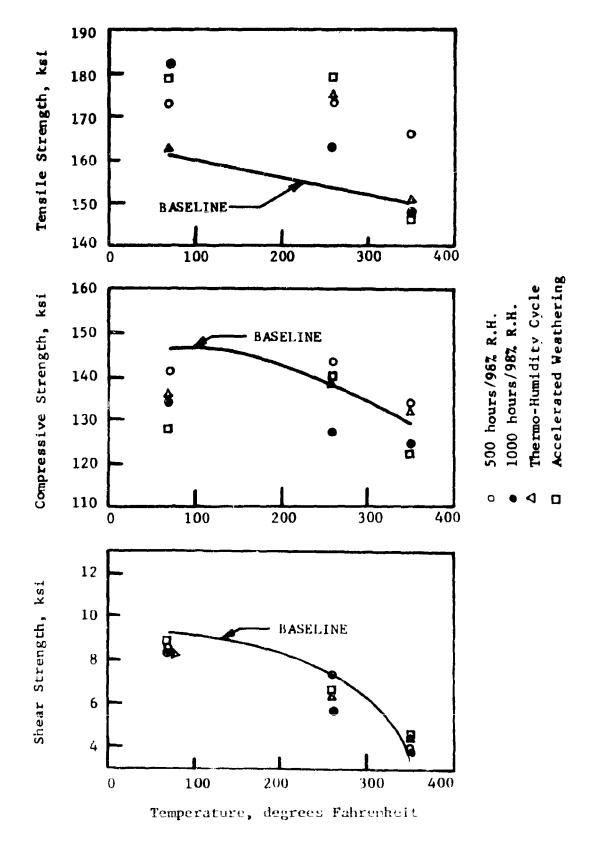


Fig. 17 EFFECT OF HUMIDITY CONDITIONING ON THE STRENG HS OF NARMCO 5206/MODMOR II GRAPHITE COMPOSTIES -

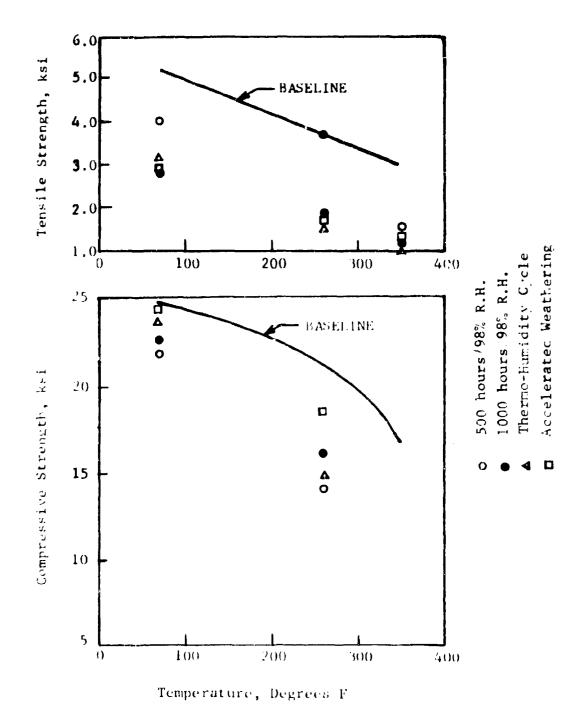


Fig. 18 EFFECT OF HUMIDITY CONDITIONING
ON THE STRENGTHS OF NARMCO 5206/MODMOR II
GRAPHITE COMPOSITES - 90°

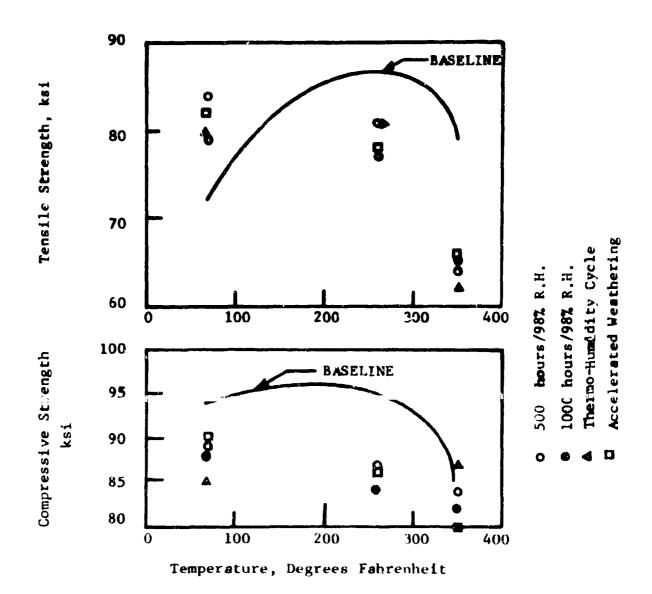
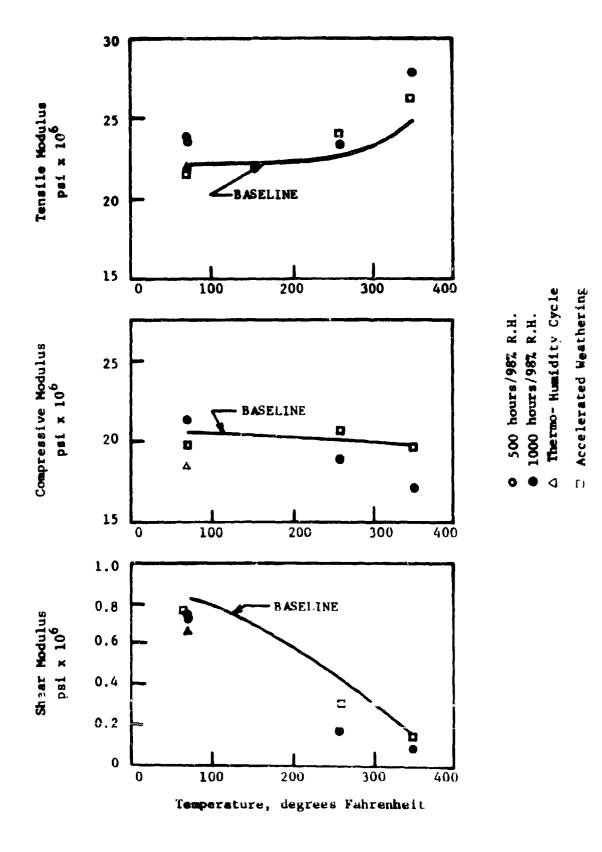


Fig. 19 EFFECT OF HUMIDITY CONDITIONING

ON THE STRENGTHS OF NARMOO 5206/MODMOR 11

GRAPHITE COMPOSITES - [0°/45/135/0°/90]



7ig. 20 EFFECT OF HUMIDITY CONDITIONING ON THE ELASTIC MODULI OF NARMCO 5206/MODMOR II GRAPHITE - 0° -45-

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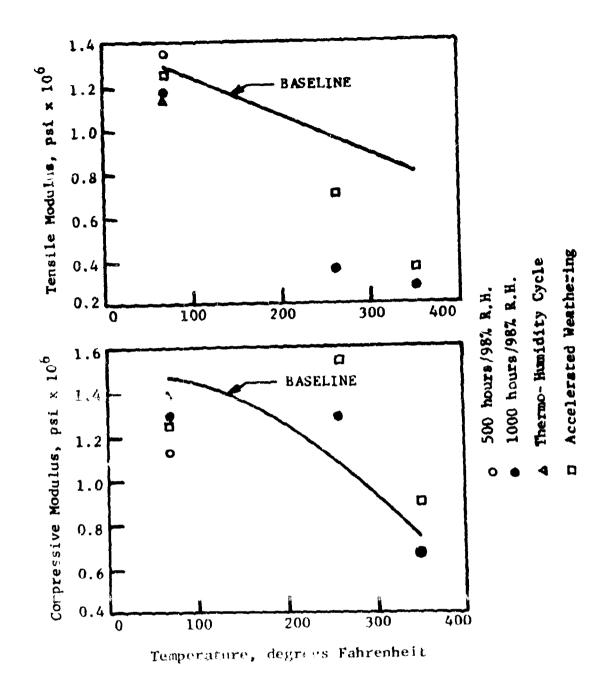


Fig. 21 EFFECT OF HUMIDITY CONDITIONING ON THE ELASTIC MODULI OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITE - 90°

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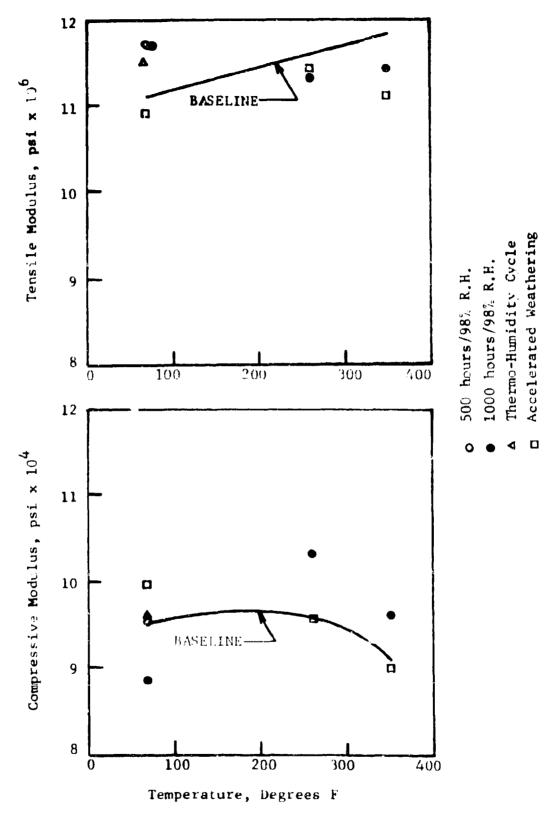


Fig. 22 EFFECT OF HUMIDITY CONDITIONING ON THE ELASTIC 'MODULI OF NARMCO 5206/MODMOR II GRAPHITE [0/45/135/0/90]

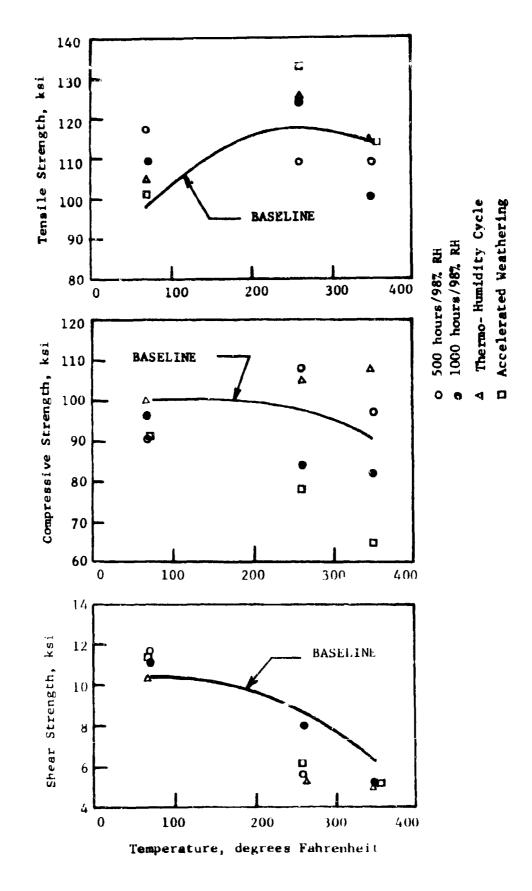
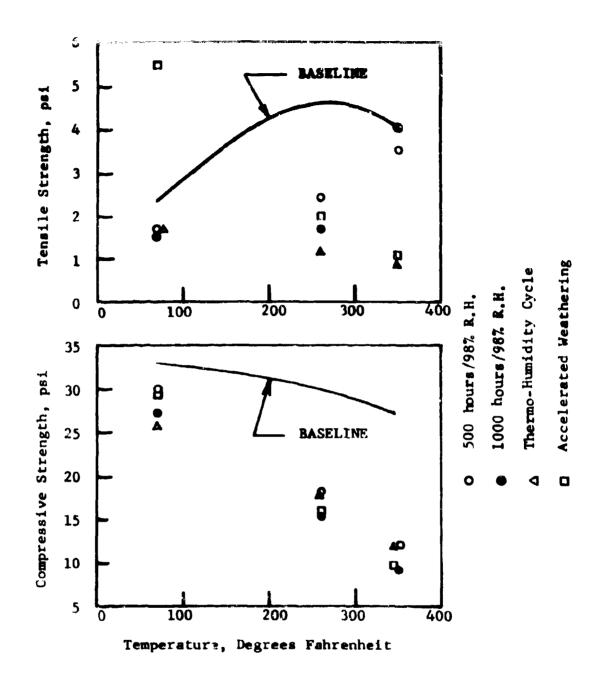
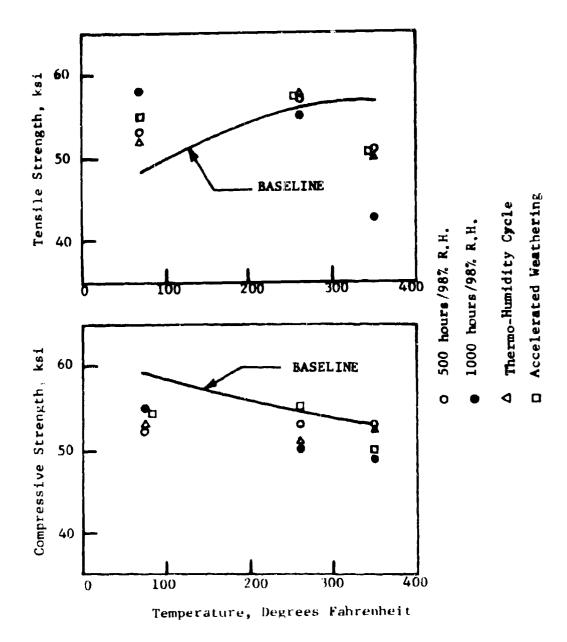


Fig 23 EFFECTS OF HUMIDITY CONDITIONING ON THE STRENGTHS OF HERCU: S 3002M/COURTAULDS HMS GRAPHITE COMPOSITES = 0°



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Fig. 24 EFFECTS OF HUMIDITY CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/ COURTAULDS HMS GRAPHITE COMPOSITES - 90°



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Fig. 25 EFFECTS OF HUMIDITY CONDITIONING ON THE STRENGTHS OF HERCHIES 3002 M/COURT AULDS HM S GRAPHITE COMPOSITES - (0/45/135/0/90)s

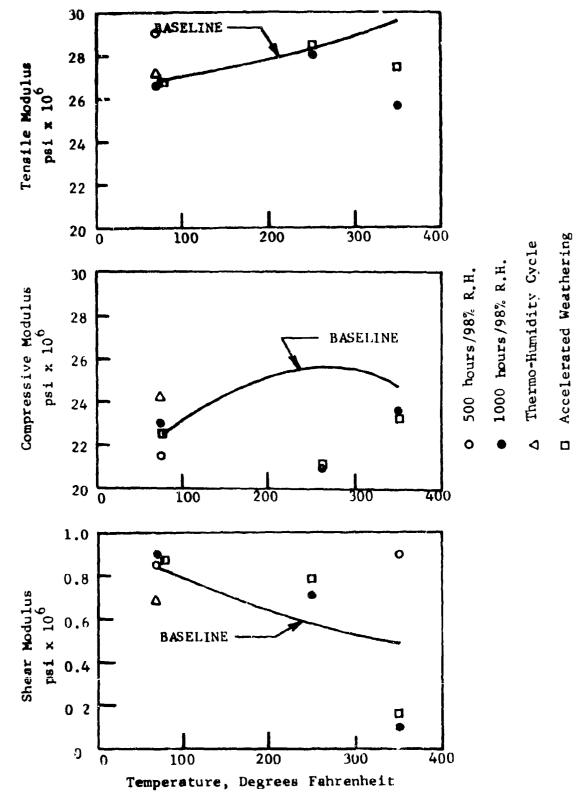


Fig. 26 EFFECTS OF HUMIDITY CONDITIONING ON THE ELASTIC MODULI OF HERCULES 3002 M/COURTAULDS HMS GRAPHITE COMPOSITES - 0°

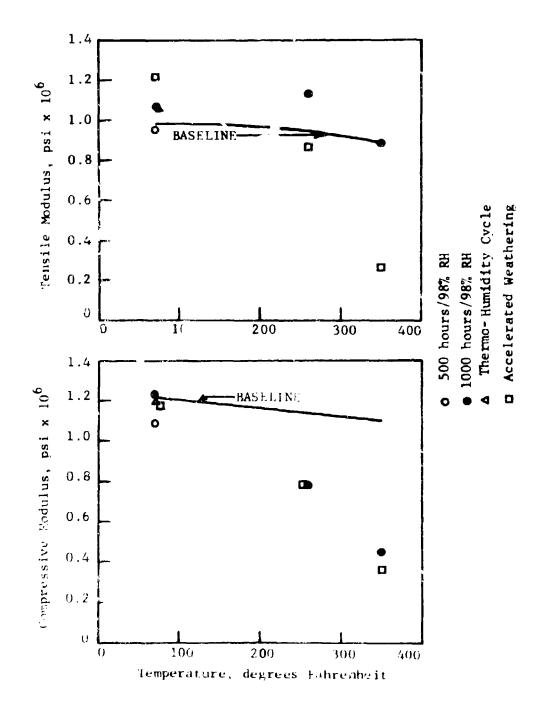


Fig. 27 EFFECT OF HUMIDITY CONDITIONING ON THE ELASTIC MODULI OF HURCHLES 30024/COURTAVEDS 9MS GRAPHITE COMPOSITES - 90:

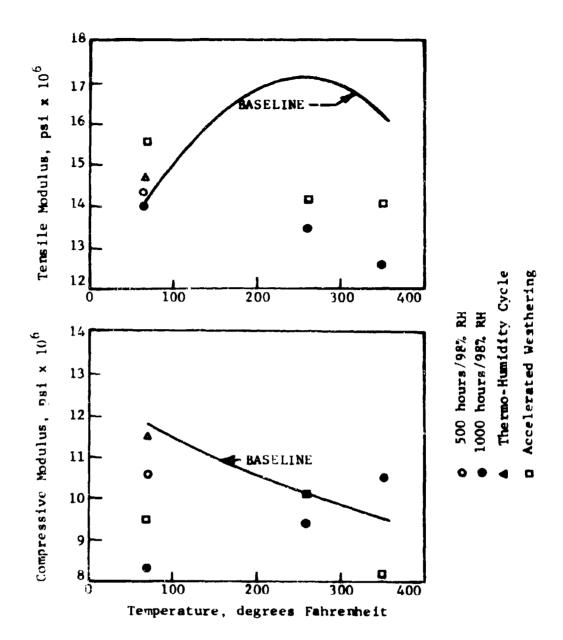


Fig. 28 EFFECT OF HUAIDITY CONDITIONING ON THE ELASTIC

MODULI OF HERCULES 3002M/COURTAULDS HMS GRAPHITE

COMPGATTES - [0/45/135/0/90]

dropped rapidly with temperature increase. The 0° compressive strengths for both materials fell below the baseline values at room temperature but at elevated temperatures some of the conditioning treatments resulted in slight increases in the compressive strengths. Finally the laminate baseline tensile strengths of both materials increased with increasing temperature. influence of he humidity conditioning on the laminate was to increase the tensile strengths at room temperature, show tensiles at 260°F fairly close to the baseline values and show a decrease at 350°F. The tensile strengths of the laminates showed an increase in the rate of change with temperature as a result of the humidity conditioning (See Figs. 19 and 25). This effect was previously noted for AVCO 5505/Boron (See Fig. 13) but much less so than in the two graphite laminates. The 90° baseline compressive strengths of both graphite materials were affected by temperature and the rate of strongth decrease with temperature was higher as a result of prior humidity conditioning. The compressive strengths of the two graphite/epoxy composites were not influenced substantially by either temperature or prior humidity conditioning. The Modmor 11/Narmco 5206 Graphite laminate was affected proportionately less than was the Hercules 3002M/Courtaulds HMS Graphite laminates.

The moduli of the two graphite composite systems are shown in Figs. 20 - 22 and Figs. 26 - 28 for the Modmor II/
Narmoo 5206 and the Hercules 3002M/Courtaulds HMS Graphite respectively. The O' is a language of compressive strength of Modmor II Graphite/Narmoo 5206 remained constant over the temperature range but the Hercules 3002M/Courtaulds HMS Graphite O' compressive strength increased with increasing temperature. The baseline implane shear strength of both materials decreased

transverse moduli and the laminate compressive baseline moduli decreased with increasing temperatures. Similarly the baseline tensile moduli of the laminate, $[0/45/135/0/\overline{90}]_s$, increased with increasing temperatures. Humidity conditioning did not substantially change the 0° moduli of either material. The 90° and $[0/45/135/0/\overline{90}]_s$ moduli were affected substantially by the humidity conditioning. No real differences between steady state and cyclic humidity conditioning were noted for the graphites as they were in AVCO 5505/Boron as far as moduli alterations were concerned.

In summary, the prior humidity conditioning affected both unidirectional and laminate properties. In those cases where low residual stresses were present (as evidenced by a monotonic decreasing strength versus temperature curve), the presence of humidity conditioning generally decreased the strengths. In those cases where substantial residual stresses were present (as evidenced by a peaking or increasing strength versus temperature curve), the humidity conditioning frequently led to an increase in the strength of the composite. In addition it is evident that the amount of moisture absorbed by the composites depends on the total time exposure to high moisture, regardless of the intervening high temperature, low temperature, drying time or U.V. exposure.

On the basis of the static humidity results the Thermo-Humidity Cycle was selected for some additional studies. A limited test program was then initiated to ascertain the effect that moisture protective coatings might have on the static mechanical properties of composites subjected to this high humidity cycle.

Accordingly, aerospace companies were contacted to ascertain the most appropriate coatings for the composites. Air Force Spec. Mil-C-83286 and MACDAC Spec. MMS-420 were utilized to procure the coatings. A polyurethane coating was selected*. This particular system required a four hour drying period at 77°F and was fully cured in 7 days.

The coated samples were then statically tested. Table IX presents a summary of the test results. Unfortunately tape supplies of the system affected the most by the Thermo-Humidity cycle (namely Hercules 3002M/Courtaulds HMS Graphite) were exhausted and new supplies were unavailable in time for the coating tests.

2.1.6.3 Effects of Thermal Conditioning

The exposure of the resim matrix composites to steady state temperature affected the composites differently depending on the material. The Avco 5505/Boron composites appear in general to:

- 1) increase in stiffness slightly (in fact the entire stress-strain curve shifts slightly to the left)
- 2) increase in strength
- 3) have a slightly reduced ultimate strain capacity.

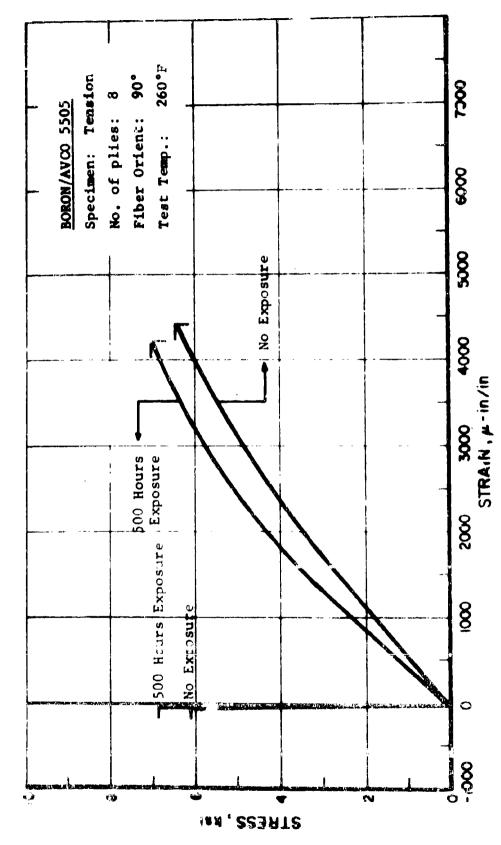
This behavior is illustrated in Figs. 29 and 30 for the laminae (90° tension and in-plane shear) which are most sensitive to prolonged exposure to elevated temperature. The steady-state temperature exposure would appear to be acting as an additional post cure to the AVCO 5505/Beron composite laminae. The laminate behavior is shown in Figs. 31 and 32 and appears to be less severe.

 ^{*} Super Desothane, A product of DeSoto, Inc.

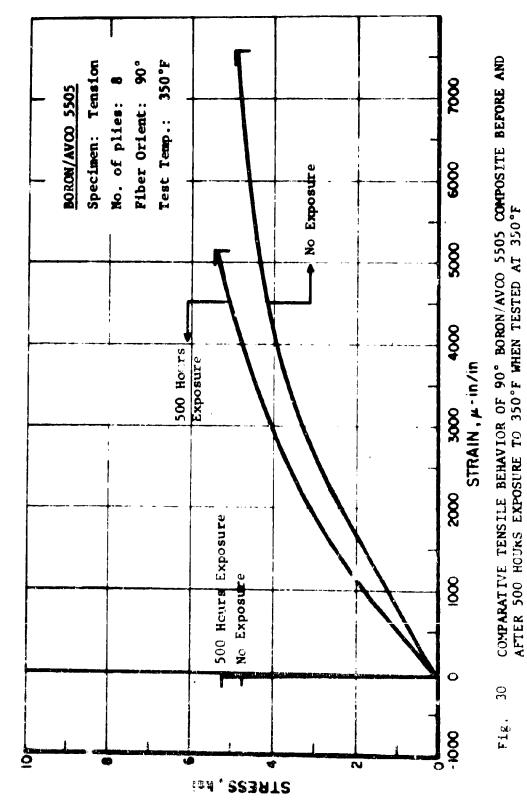
TABLE IX

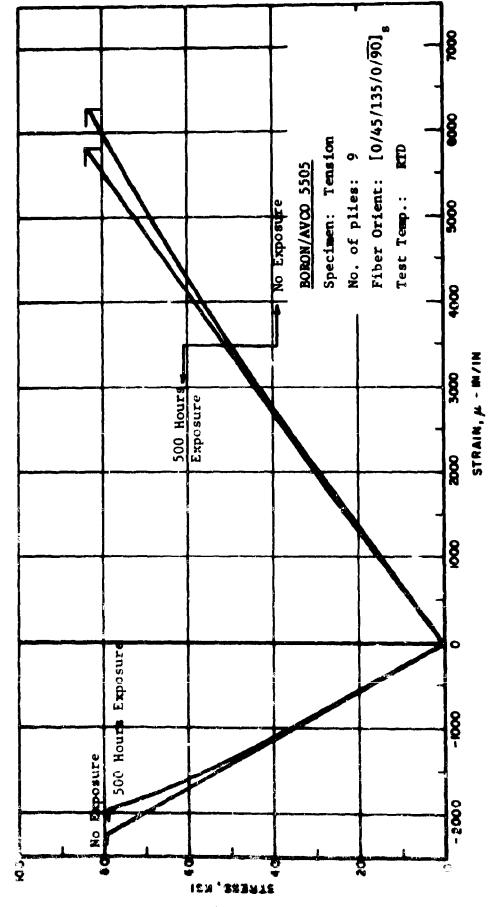
SUMMARY OF TENSILE TESTS AT ROOM TEMPERATURE
ON VARIOUS COMPOSITES COATED WITH SUPER DESOTHANE
POLYURETHANE AND SUBJECTED TO THE THERMO-HUMIDITY CYCLE

SYSTLM	ORIENTATION	CONDITION	dult (ksi)	^ε ult (μ-in/in)	E (ps1 x 10 ⁶)	, (fn/in)
Avco 5505/ Buron	0 0	Bare Coated Bare/Th-Hum Cycle Coated/Th-Hum Cycle	183 185 188 e 186	6420 6200 6360 6180	29.6 29.6 29.6 29.1	0.23 0.25 0.17 0.22
Varmeo 5206/ Modmer II Graphite	°Û	Bare Coated Bare/Th-Hum Cycle Coated/Th-Hum Cycle	161 171 163 e 170	6920 6840 7230 6960	22.5 21.9 22.0 22.0	0.30 0.27 0.24 0.26
Varnco 5206/ Modmor II Graphice	[0/45/135/0/ <u>90</u>]	Bare Coated Bare/Th-Hum Cycle Coated/Th-Hum Cycle	72 70 80 e 73	6610 6580 6820 6950	11.1 10.8 11.5 11.1	0.38 0.32 0.42 0.43

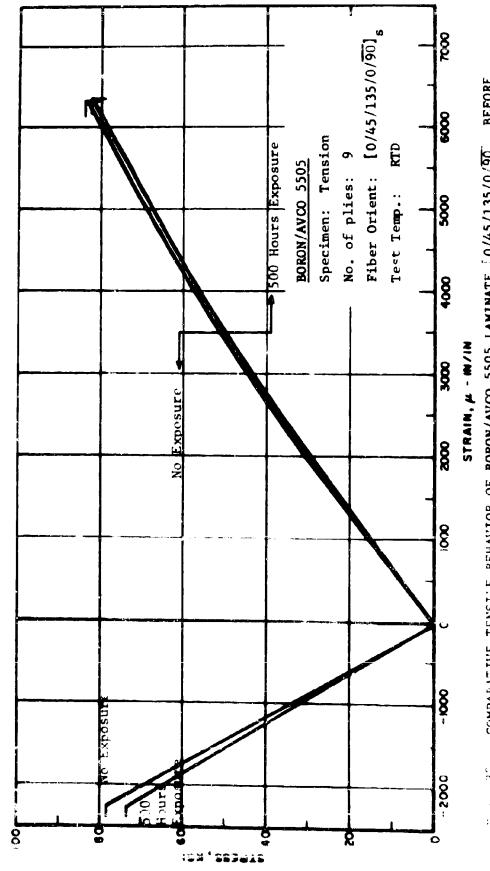


COMPARATIVE TESSILE BEHAVIOR OF 90° BORON/AVCO 5505 COMPOSITE BEFORE AND AFTER 500 HOURS EXPOSURE TO 260°F WHEN TESTED AT 260°F F1g. 29





COMPARATIVE TENSILE BEHAVIOR OF BORON/AVCO 5505 LAMINATE [0/45/135/0/90], BEFORE AND AFTER 500 HOURS EXPOSURE TO 260°F WHEN TESTED AT ROOM TEMPERATURE 33 (1. 14 14,



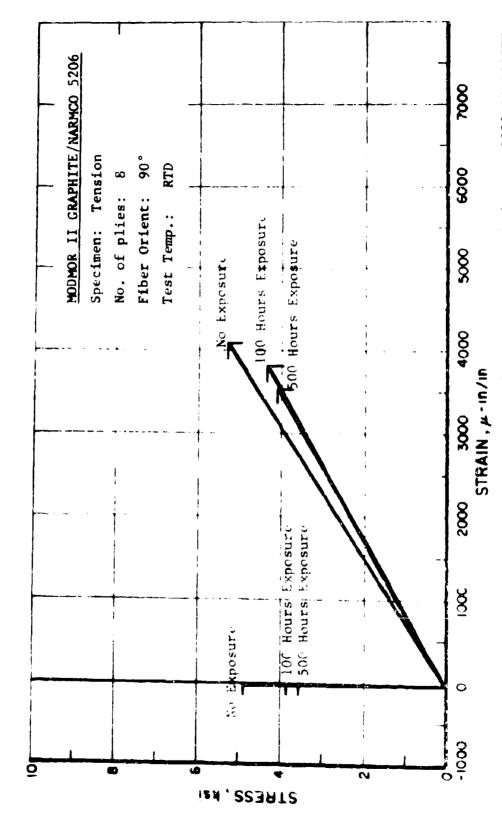
COMPARATIVE TENSILE BEHAVIOR OF BORON/AVCO 5505 LAMINATE [0/45/135/0/90] BEFORE ANT AFTER 500 HOURS EXPOSURE TO 350°F WHEN TESTED AT ROOM TEMPERATURE () ()

Prolonged exposure of Modmor II Graphite/Narmco 5206 epoxy laminae (see Figs. 33 and 34) to elevated temperature acts as a typical detrimental factor by decreasing transverse modulus, ultimate transverse strength and ultimate transverse strain capabilities of the lamina. The modulus reduction would appear to be primarily confined to the early portion of the exposure since after 100 hours and 500 hours the two stress-strain curves are coincident. However additional transverse strength and transverse strain capabilities were lost.

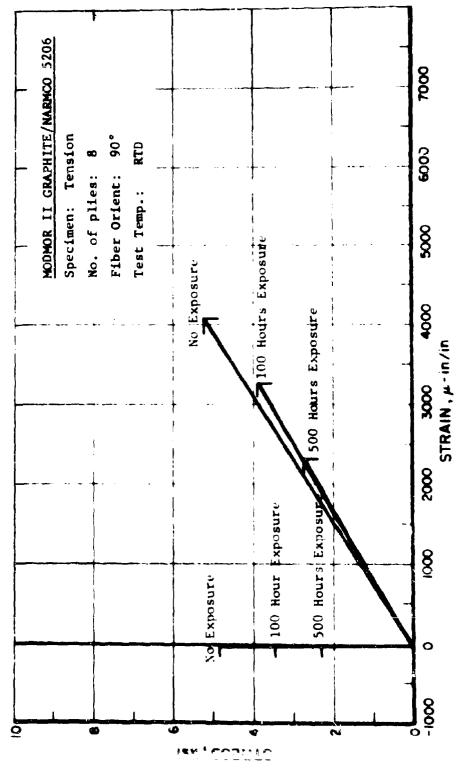
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Several parametric cross plots are available for the purposes of illustrating the effects of steady-state thermal conditioning on the static properties of the three resin matrix composite material: Thus Figs. 35 - 40 present the effects of steady state conditioning on the tensile compressive and shear strength and moduli of 0°, 90° and $[0/45/135/0/\overline{90}]$ AVCO 5505/ Boron composites. In Figs. 35 and 36 the tensile strength of the 0° and compressive strengths of the 90° and $\frac{0.045/135/0.000}{90.000}$ laminates showed a decrease after exposure to steady state thermal conditioning. The 0° compressive strength, the in-plane shear strength of the 0° composites, the tensile strengths of the 90° and $.0/45/135/0/\overline{90}_{1_{\mathbf{S}}}$ showed increases in strength particularly at the elevated temperatures. These latter strengths are more resh sensitive than the former strengths. The steadystate thermal conditioning acts as an additional post-cure on the resin matrix.

The elastic moduli of the 0° AVCO 5505/Boron composites were increased slightly at the elevated temperatures (See Fig. 38) as a result of the steady-state thermal conditioning. In addition, the steady-state thermal conditioning produced medical versus test temperatures more nearly equal between the tension



COMPARATIVE TENSILE BEHAVIOR OF 90° MODMOR II GRAPHITE/NARMCO 5206 COMPOSITE BEFORE ANI AFTER 100 AND 500 HOURS EXPOSURE TO 260°F WHEN TESTED AT ROOM TEMPERATURE Fig. 33



COMPARATIVE TENSILE BEHAVIOR OF 90° MODMOR II GRAPHITE/NARMOO 5206 COMPOSITE BEFORE AND AFTER 100 AND 500 HOURS EXPOSURE TO 350°F WHEN TESTED AT ROOM TEMPERATURE Fig. 34

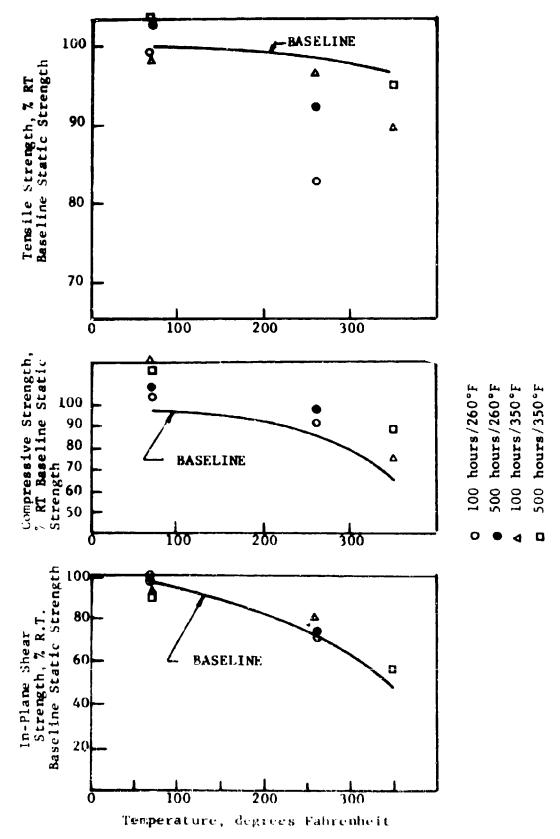
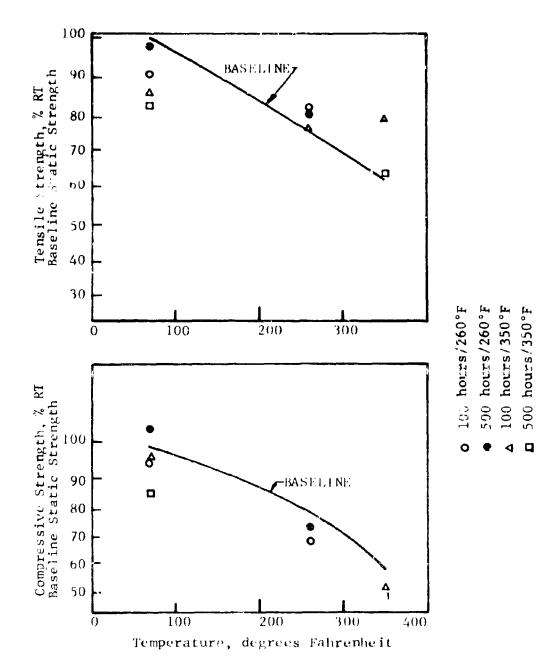


Fig. 35 EFFECTS OF STEAL STATE HERMAL CONDITIONING ON THE STRENGTHS OF AVOID 55 RE/BORON COMPOSITION - 0.3



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Fig. 36 EFFECTS OF STEADY STATE THERMAL CONDITIONING
OF THE STRENGTHS OF AVCO 5505/BORON COMPOSITES - 90°

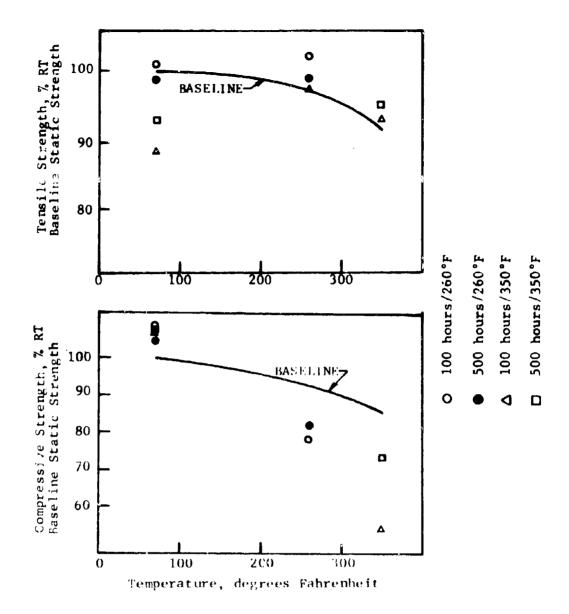
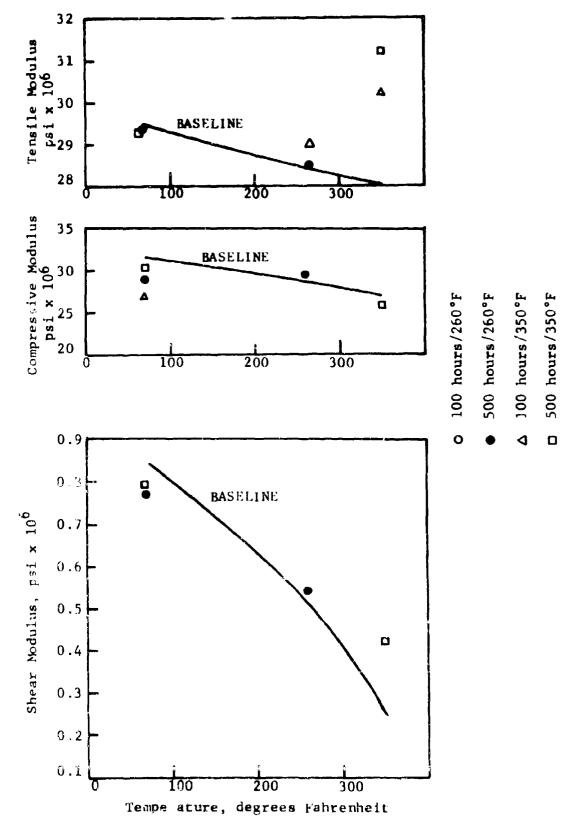


Fig. 37 EFFECTS OF STEP DY-STATE THERMAL CONDITIONING ON THE STRENGTHS OF AVCO 5505/BORON COMPOSITE - $\left[0/45/135/0/\overline{90} \right]_S$



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Fig. 38 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE MODULI OF AVOID 5505/ROBON COMPOSITES - 0°

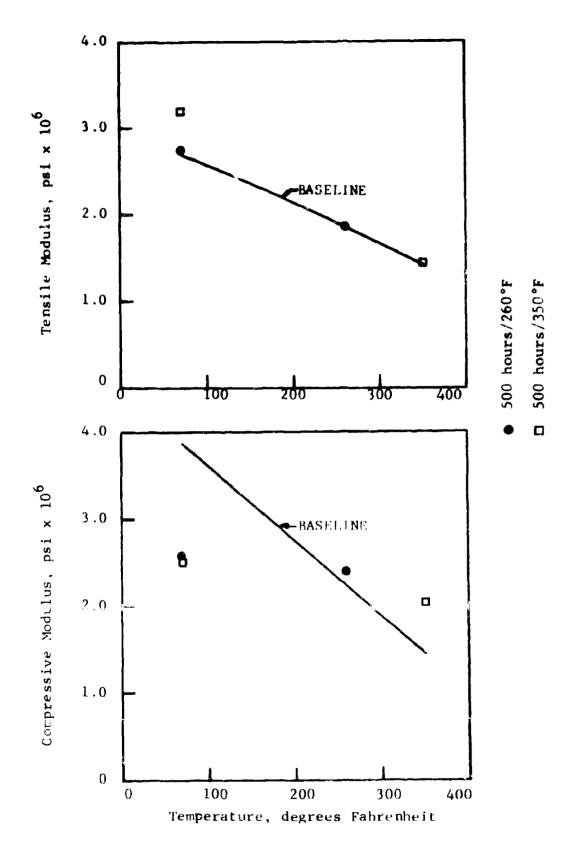
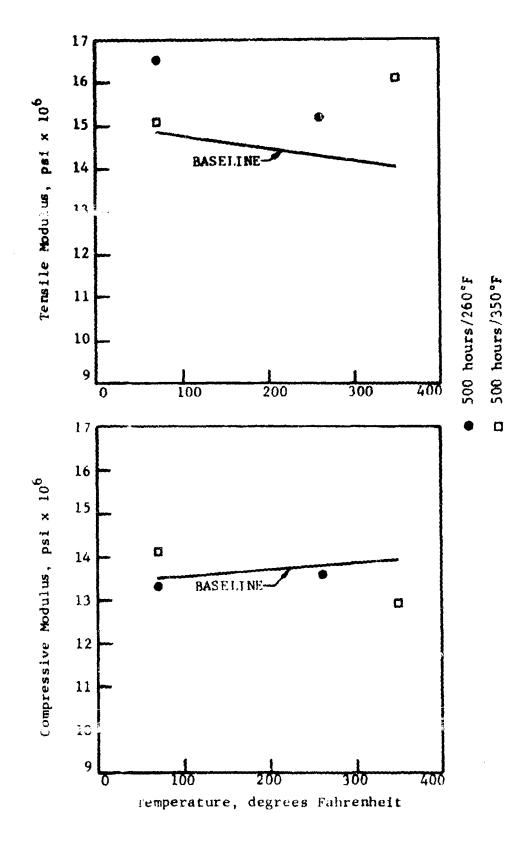


Fig. 39 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF AVCO 5505/BORON COMPOSITES - 90°



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Fig. 40 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF AVGO 5505/BORON COMPOSITES - [0/45/135/0/90] =

and compression modes (See Fig. 39). The tensile moduli of the $10/45/135/0/\overline{90}$ laminates increased over the entire temperature range as a result of prior thermal conditioning while the corresponding compressive moduli decreased at the higher test temperatures.

The effect of steady state thermal conditioning on the strength of Narmoo 5206/Modmor II graphite composites is shown in Figs. 41-43. In general the strengths of this graphite/epoxy composite increased as a result of the steady-state thermal conditioning. The 260°F exposures resulted in small increase: in strength or no change for almost all types of loading and composite orientations. The 350°F exposures showed lower exposed strengths than did the 260°F exposures and often times, as in the case of the 90° tensile strengths, a substantial reduction was detected.

The effect of steady state thermal conditioning on the elastic moduli of Narmco 5206/Modmor II graphite composites is shown in Figs. 44 to 46. The tensile and in-plane shear moduli were not affected substantially by steady state thermal conditioning. The compressive moduli were affected substantially; for all three orientations the 500 hours at 260°F was the worst culprit.

The effect of steady state thermal conditioning on the strengths of Hercules 3002M/Courtaulds HMS Graphite composites is shown in Figs. 47 to 49. In general, for the 0° and $10/45/135/0/\overline{90}$ composites, the tensile and compressive strengths increased above the baseline values at all temperatures. However the 90° tensile strengths were substantially lower, at all temperatures, than the baseline values.

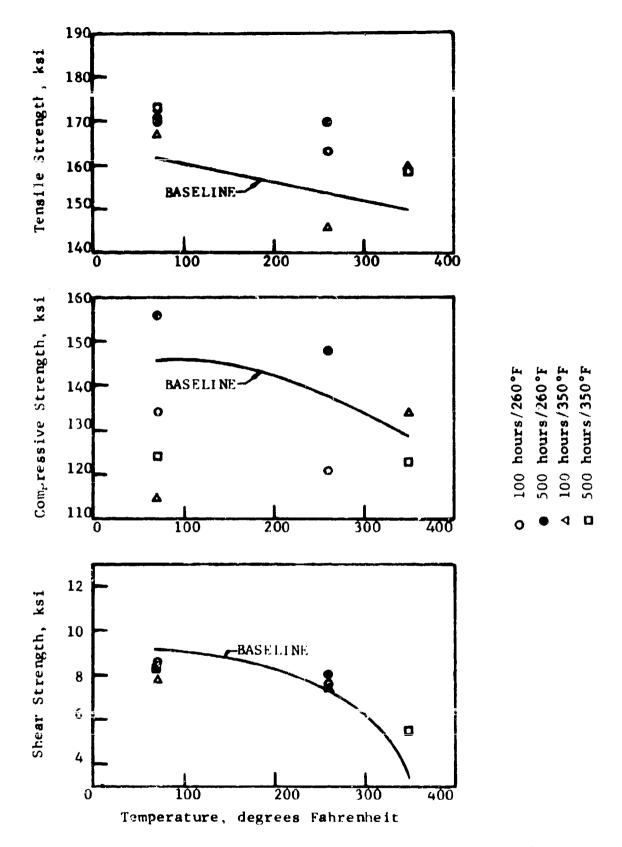


Fig. 41 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE STRENGTHS OF NARREO 5200/HODBOK II GRAFATTE COMPOSITES - 0°

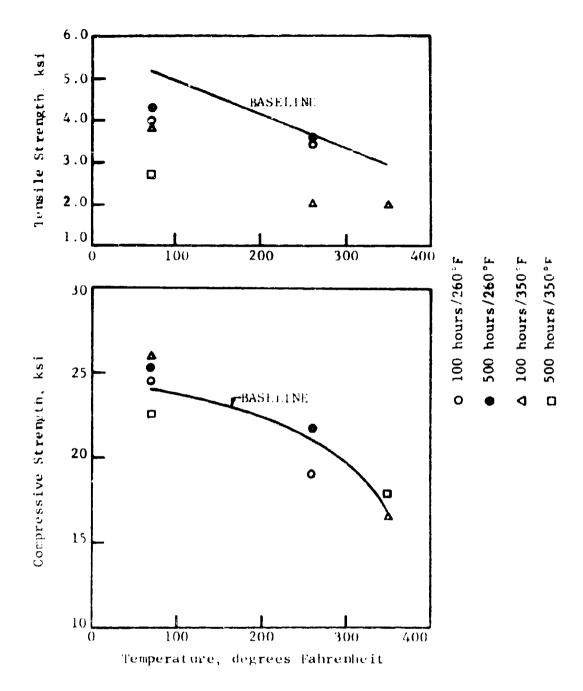


Fig. 42 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE STRENGTHS OF NARMOO 5206/MODMOR II GRAPHITE COMPOSITES - 90°

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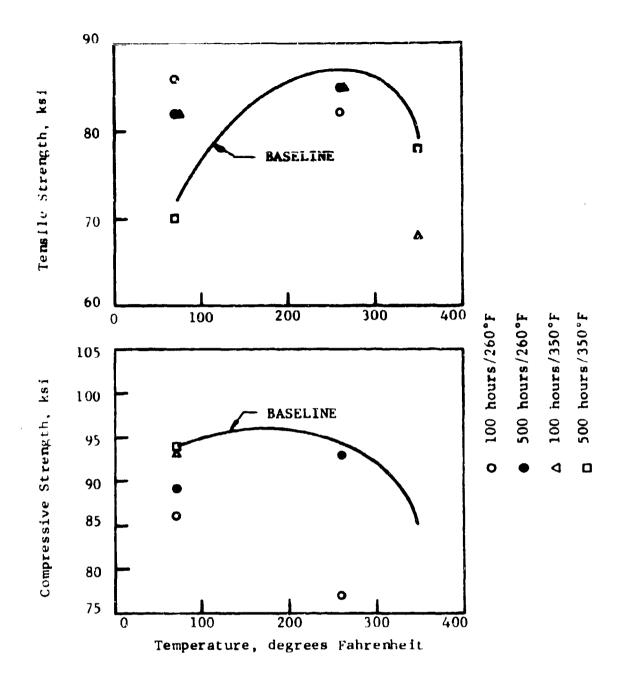
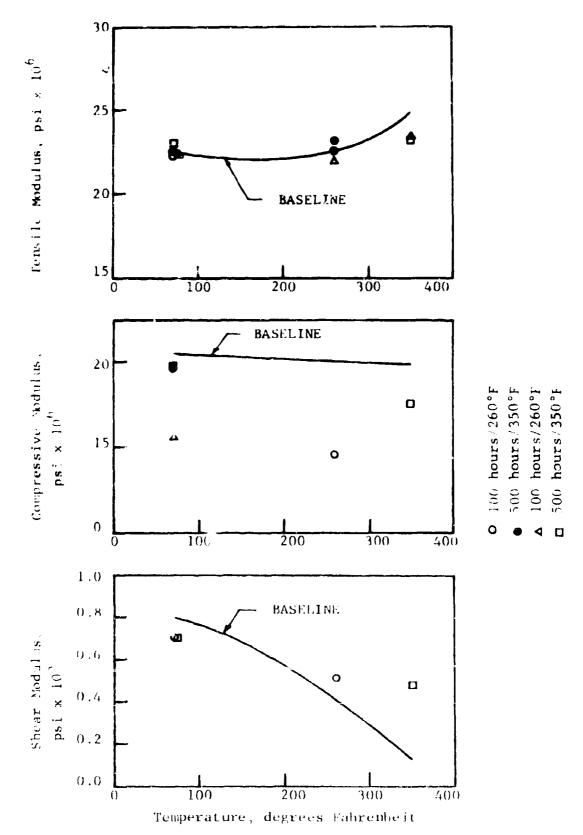


Fig. 43 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE STRENGTHS OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - [0/45/135/0/ $\overline{90}$] s



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Fig. 44 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF NARMOO 5206/MODMOR II GRAPHITE.

COMPOSITES - 0°

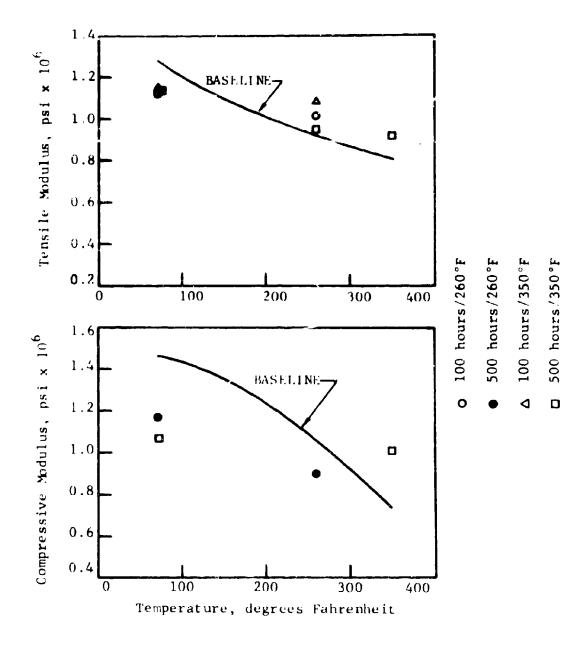


Fig. 45 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF NARMOO 5206/MODMOR II GRAPHITE COMPOSITES - 90°

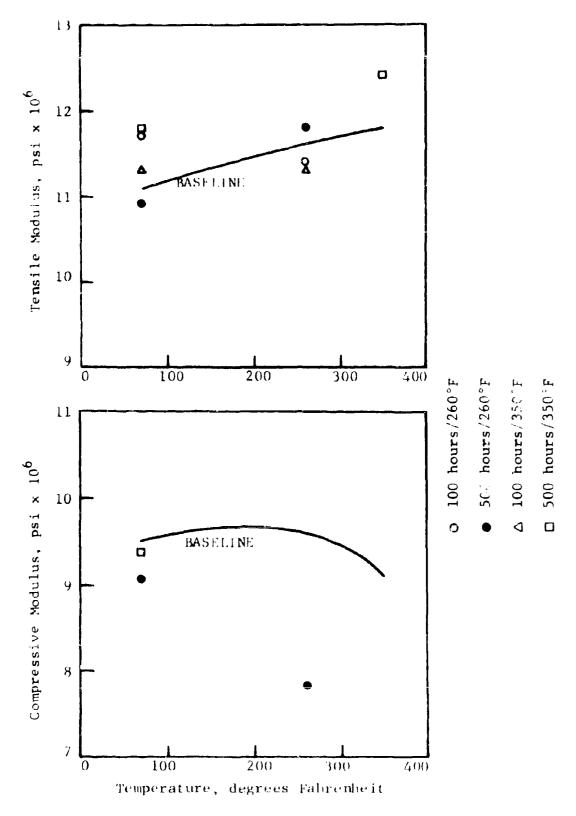


Fig. 46 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULL OF NARMOO 5206/MODMOR II GRAPHITE COMPOSITES - [0/45/135/0/90]_S

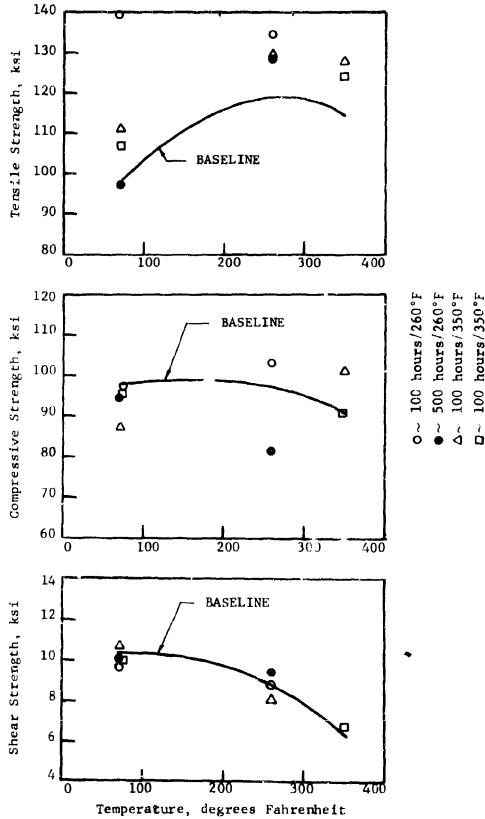


Fig. 47 EFFECTS OF STEADY STATE THERMAN CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPUSITES - 0°

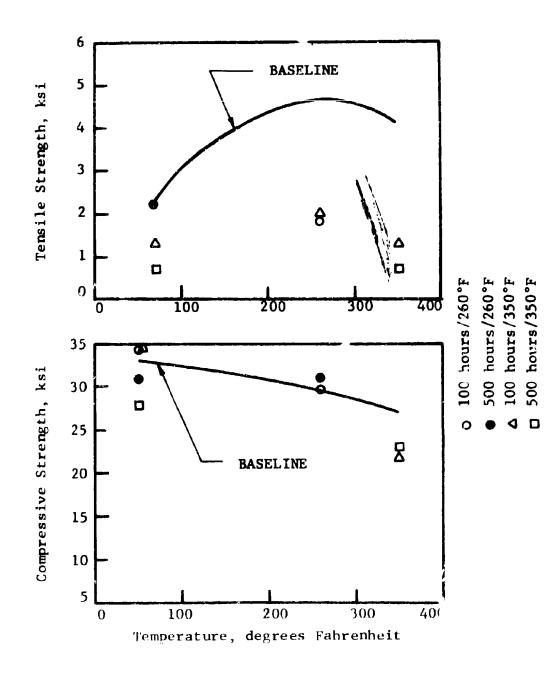
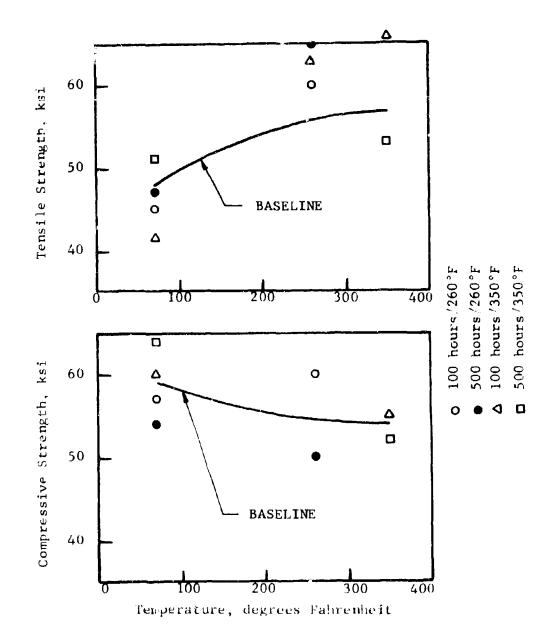


Fig. 48 EFFECTS OF STEADY-STATE THERMAL CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 90°



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Fig. 49 EFFECTS OF STEADY STATE THERMAL CONDITIONING ON THE STRENGTHS
OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - [0/45/135/0/90]

Figures 50 to 52 show the effect of thermal conditioning on the clastic moduli of Hercules 3002M/Courtaulds HMS graphite. Very little change in the elastic moduli of the 0° composites was evident. The $\left[0/45/135/0/\overline{90}\right]_8$ composites showed substantial modulus reduction in the compressive moduli for the 500 hour exposure at 260°F .

Cyclic thermal conditioning effects on the strengths of AVCO 5505/Boron are indicated in Figs. 53 to 55. The most substantial changes were in the 90° compression strengths, particularly, the room temperature strengths. In addition the 0° tensile strengths were reduced by cyclic thermal conditioning over the entire range of temperatures.

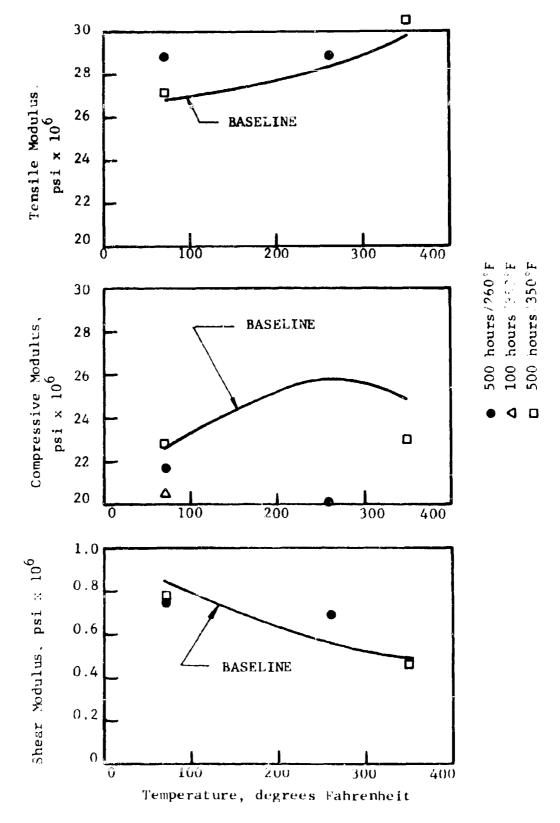
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The effect of cyclic thermal conditioning on the elastic moduli of AVCO 5505/Boron composites is shown in Figs. 56 to 58. Most of the reduction in elastic moduli, from the baseline values took place at room temperature. The exception to this trend was for the compressive moduli of the $[0/45/135/0/\overline{90}]_8$ composites.

The effects of cyclic thermal conditioning on the strengths of Narmoo 5206/Modmor II graphite are shown in Figs. 59 to 61. The in-plane shear strength of Narmoo 5206/Modmor II graphite was altered so as to make the strength nearly constant over the entire range of temperatures. The 0° tensile strength were altered from the baseline strength levels so as to produce an increase in strength with temperature. The most scattered results were again shown for the $10/45/135/0/901_{\rm S}$ composites particularly the compressive strengths. The tensile strengths of the $10/45/135/0/901_{\rm S}$ laminates also became more constant over the entire range than were the baseline strengths.

Modulus changes in Narmoo 5206/Modmor II graphit as a result of cyclic thermal conditioning are shown in Figs. 62 to 64.



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Fig. 50 EFFECTS OF STEADY-STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF HIRCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES- 0"

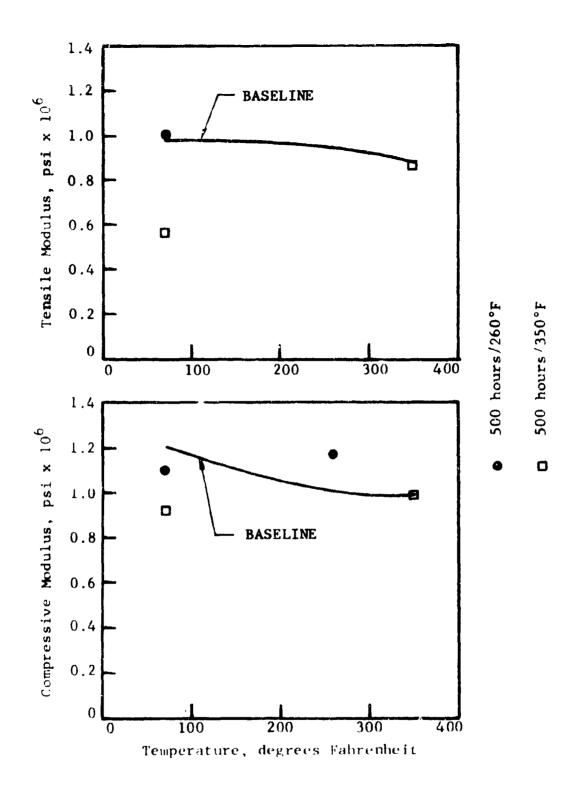


Fig. 51 EFFECTS OF STEADY STATE THERMAL CONDITIONING ON THE ELASTIC MODULE OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 90°

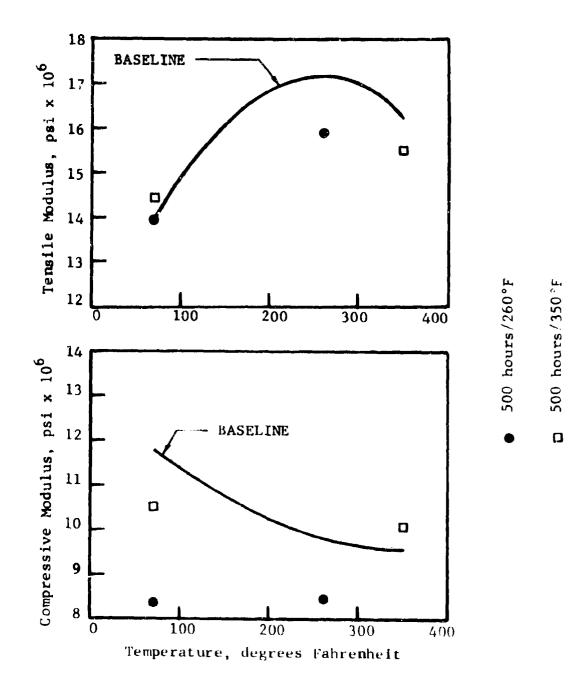


Fig. 52 EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE ELASTIC MODULI OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - [0/45/135/0/90]₈

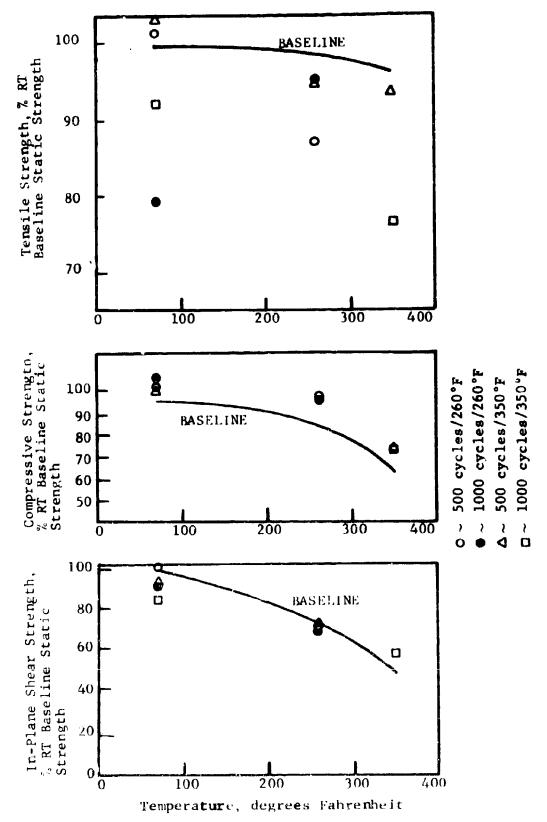


Fig. 53 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF AVGO 5505/BORON COMPOSITES - 0°

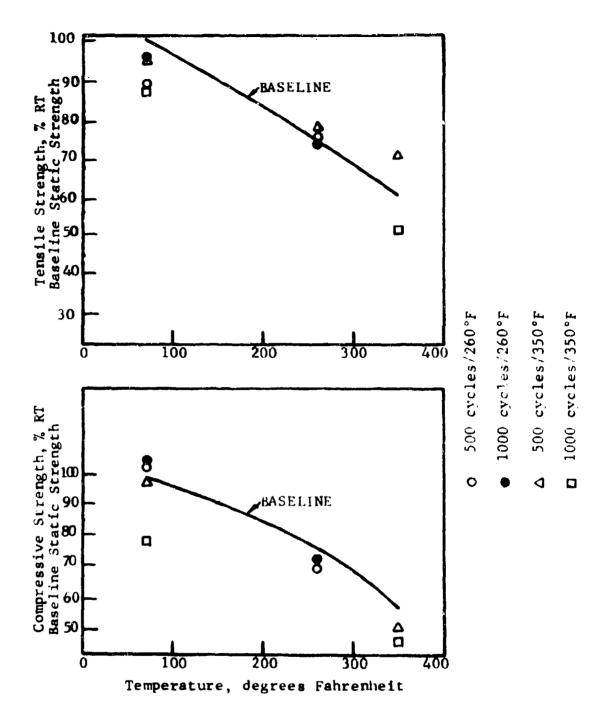
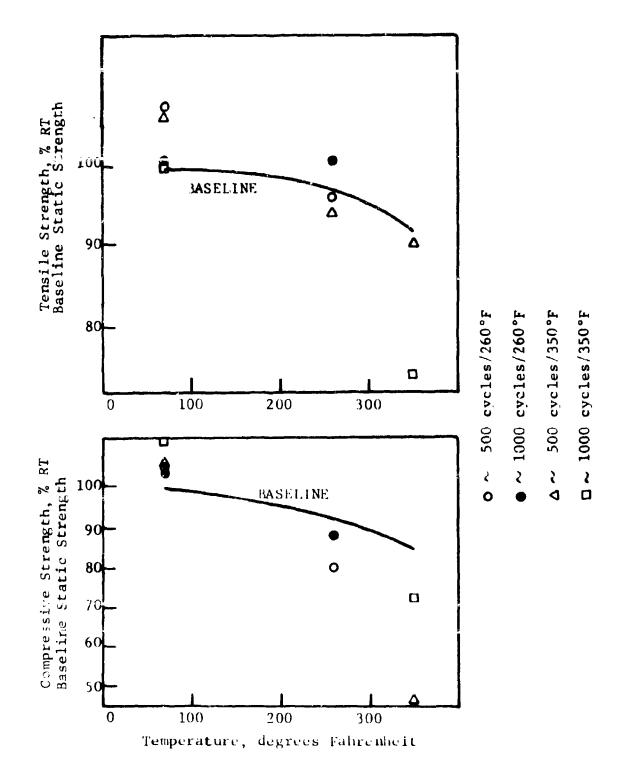


Fig. 54 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHSOF AVOID 5505/BORON COMPOSITES - 90°

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Fig. 55 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF AVCO 5505/BORON COMPOSITES - $[0/45/135/0/90]_s$

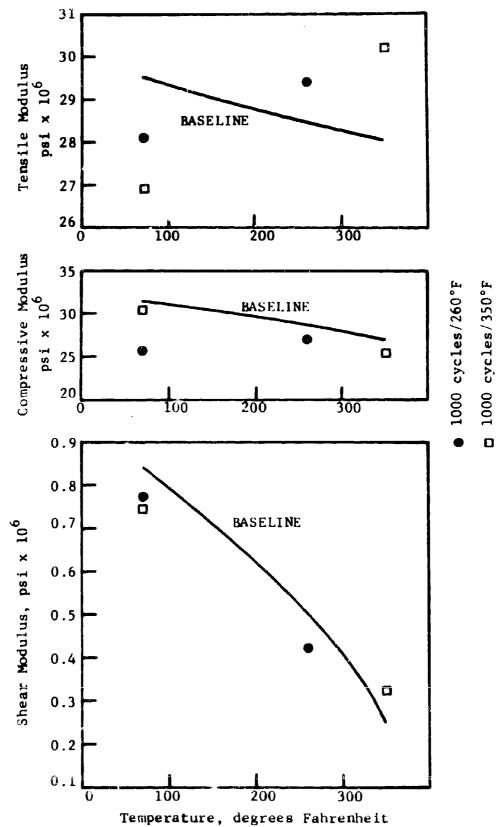


Fig. 56 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF AVCO 5505/BORON COMPOSITES - 0°

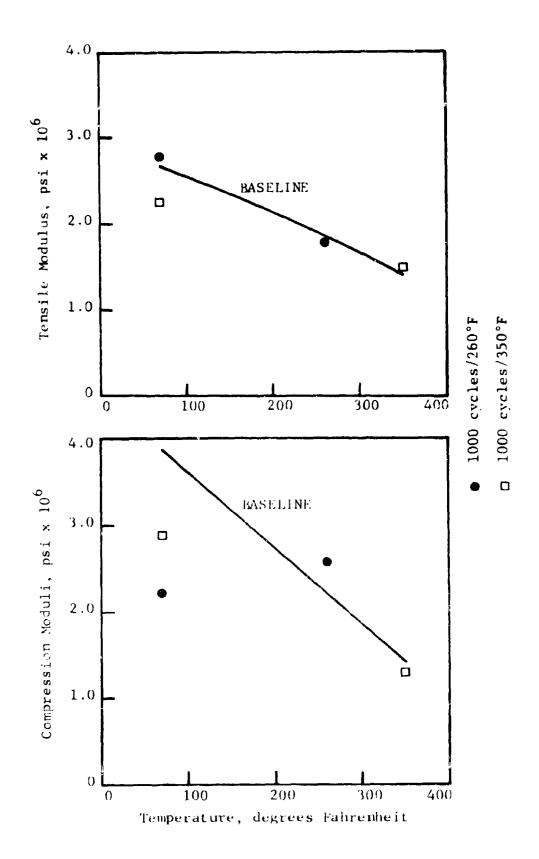


Fig. 57 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF AVEO 5505/BORON COMPOSITES - 90°

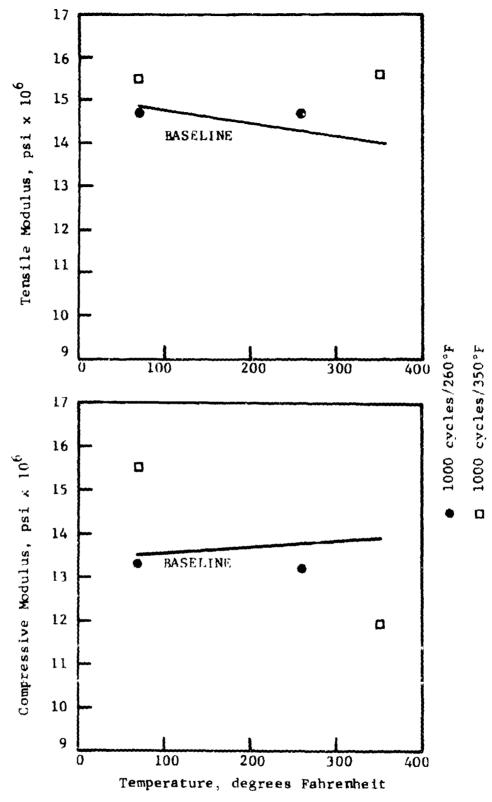
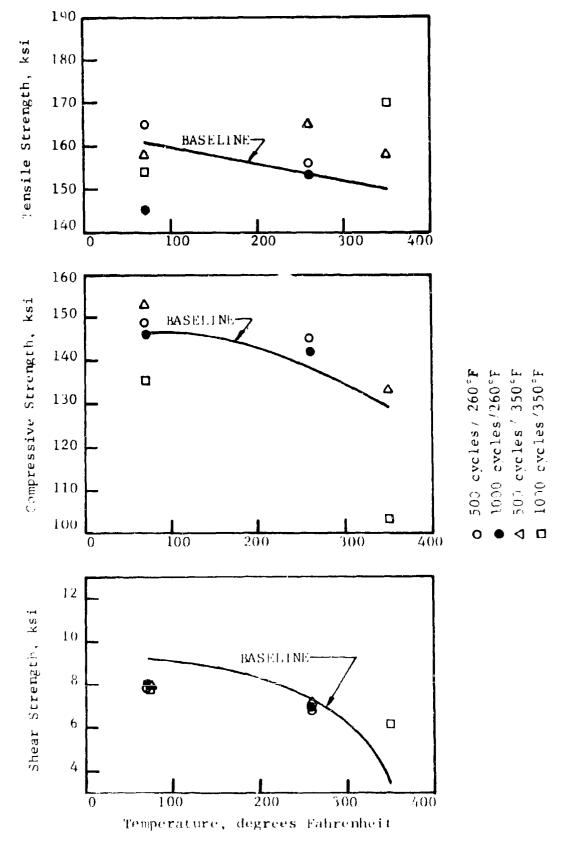


Fig. 58 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF AVCO 5505/BORON COMPOSITES - [0/45/135/0/90]_R



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EFFECT OF CYCLIC THERMAL CONDITIONING ON THE Fig. 59 STRENGTHS OF NARMOO 5206/MODMOR II GRAPHITE COMPOS ITES () o

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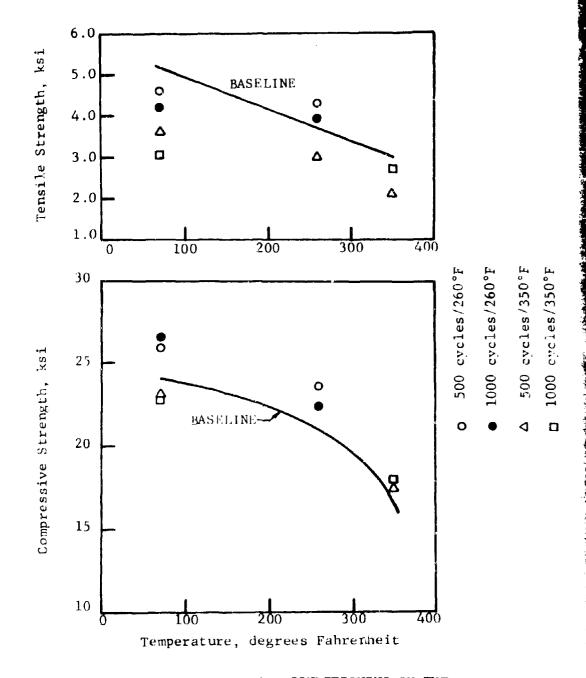


Fig. 60 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF NARMOO 5206/MODMOR 11 GRAPHITE COMPOSITES - 90°

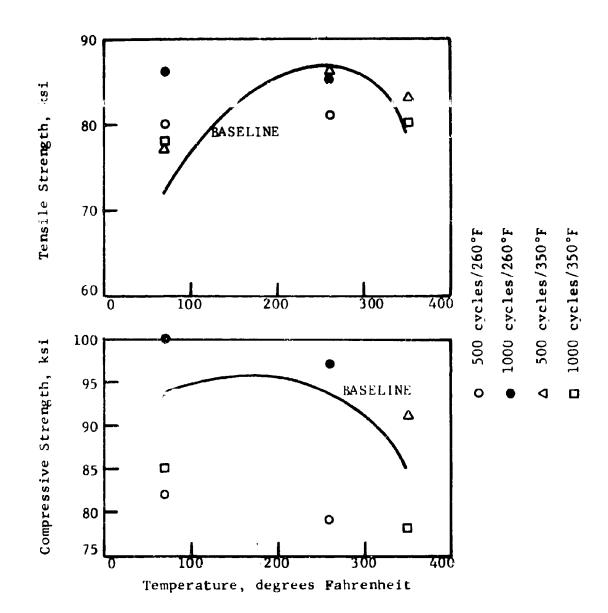


Fig. 61 EFFECT OF CYCLIC THERMAL CONFITTONING ON THE STRENGTHS OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - [0/45/135/0/90]_S

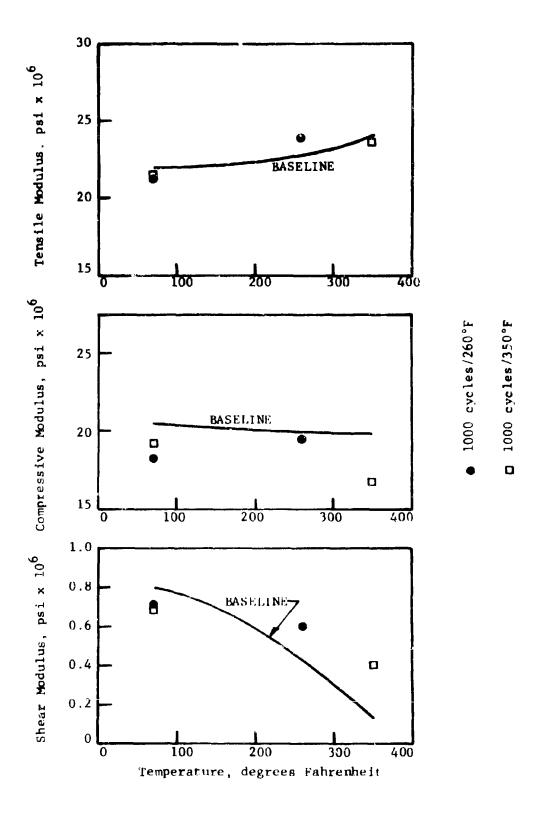
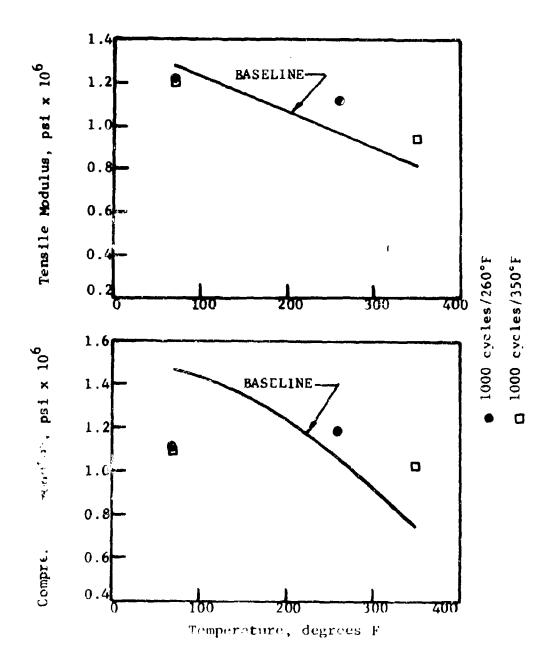


Fig. 62 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF NARMOO 5206/MODMOR II GRAPHITE COMPOSITES - 0°



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Fig. 63 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - 90°

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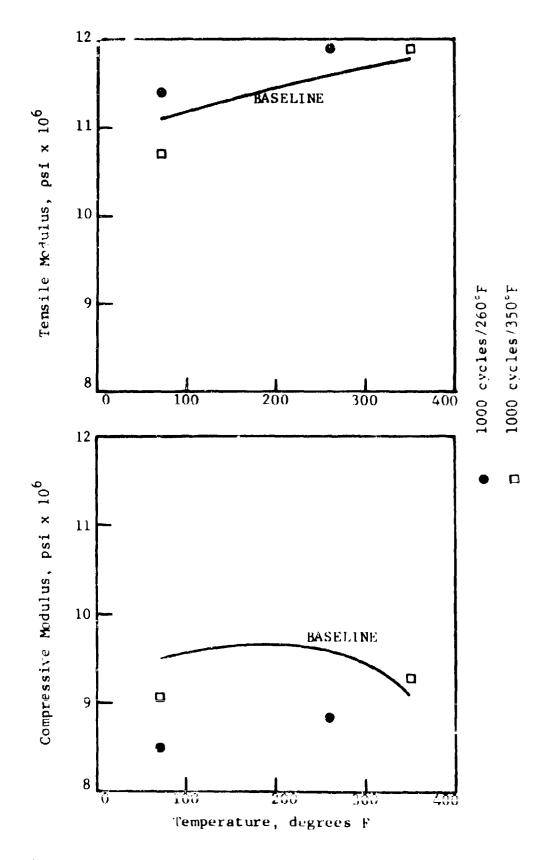


Fig. 64 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF NARMOO 5206/MODMOR II GRAPHITE COMPOSITES - [0/45/135/0/90]

The largest changes from the baseline behavior were evident in the 90° compressive room temperature behavior. In almost every case the elastic moduli became more constant over the temperature range than in the baseline or unconditioned state.

Figures 65 to 67 show the effect of cyclic thermal conditioning on the strengths of Hercules 3002M/Courtaulds HMS graphite composites. The greatest reduction in strength were evident for the 90° tensile strengths at elevated temperatures falling substantially below the baseline strengths. Increases in strengths were determined for the 0° and $\left[0/45/135/0/90\right]_8$ tensile strengths and the laminate compressive strengths.

The elastic moduli of Hercules 3002M/Courtaulds HMS graphite were also affected by the cyclic thermal conditioning as is shown in Figs. 68 to 70. The tensile and compressive moduli of the $\left[0/45/135/0/\overline{90}\right]_{\rm S}$ imminates were clearly affected substantially.

The steady state thermal conditioning generally increased the strength and stiffness of the unidirectional and $\left[0/45/135/0/\overline{90}\right]_g$ composites, decreasing the ultimate strain capabilities at the same time. The transverse strengths were decreased, transverse moduli increased and the strength versus temperature curves altered to a more constant value over the entire temperature range.

The cyclic thermal conditioning also made the variation of strengths and moduli with temperature more constant over the temperature range. The moduli were affected less than steady state exposures and were generally decreased.

2.1.6.4 Effect of Conditioning on Interlaminar Shear

The interlaminar shear strengths of the resin matrix composites are also affected by moisture and thermal conditions in the but to a lesser extent than other mechanical strengths.

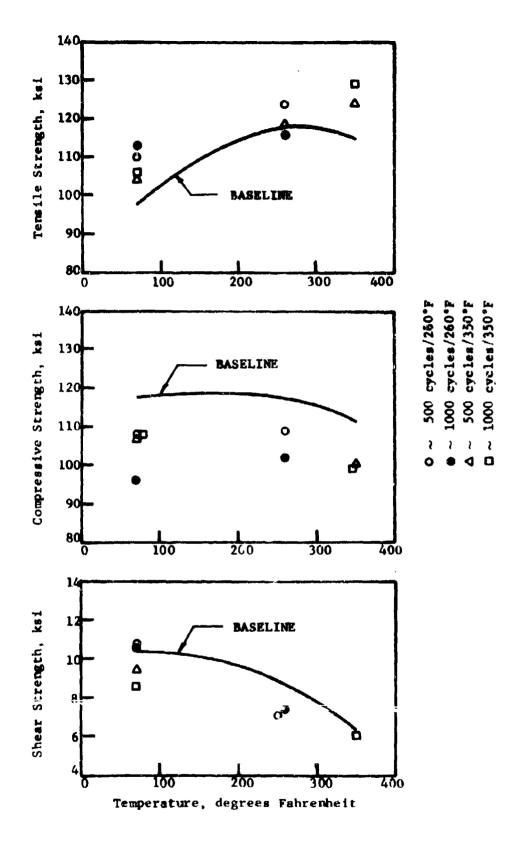


Fig. 65 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 0°

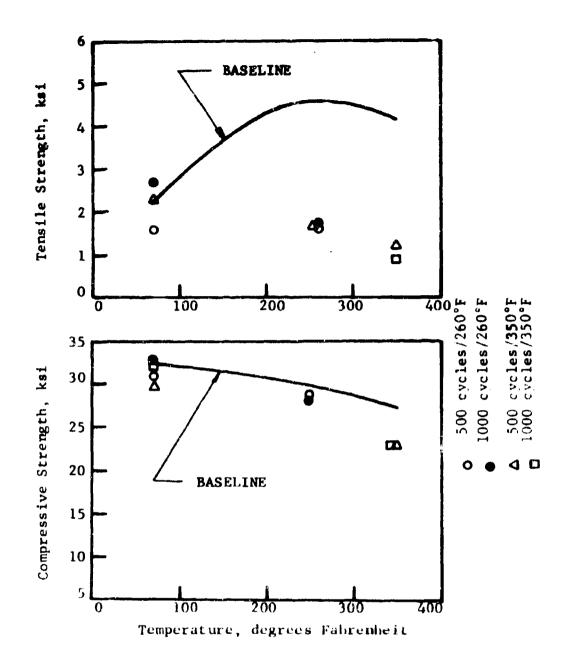


Fig. 66 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 90°

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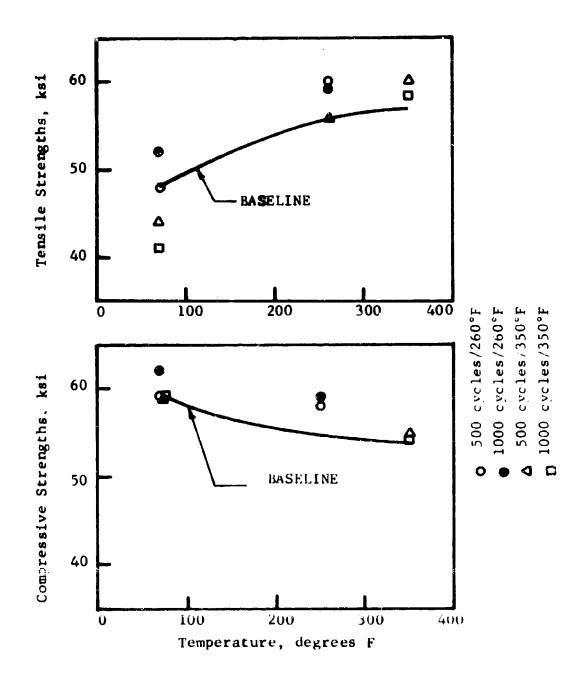
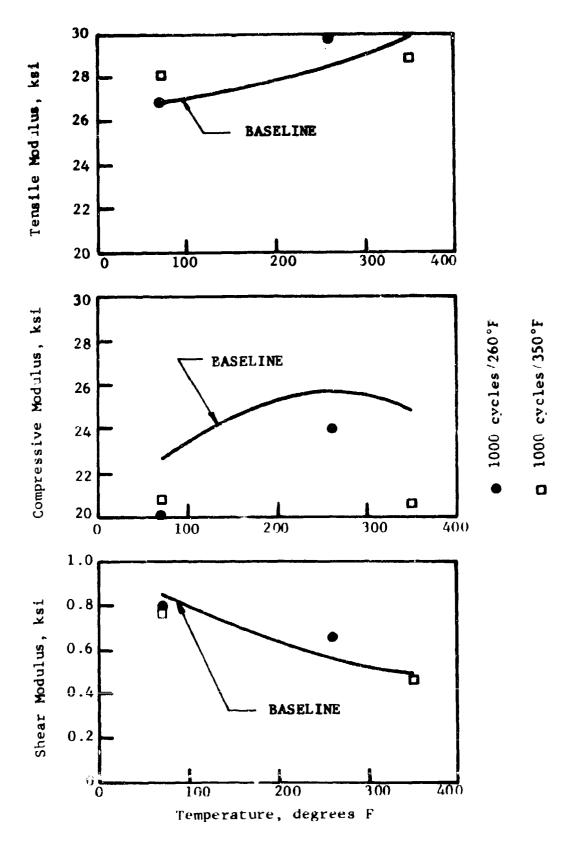


Fig. 67 EFFECTS OF CYCLIC THERMAL CONDITIONING ON THE STRENGTHS OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - [0/45/135/0/90]_S



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Fig. 68 EFFECTS OF CYCLIC THEPELL CONDITIONING ON THE ELASTIC MODULI OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 0°

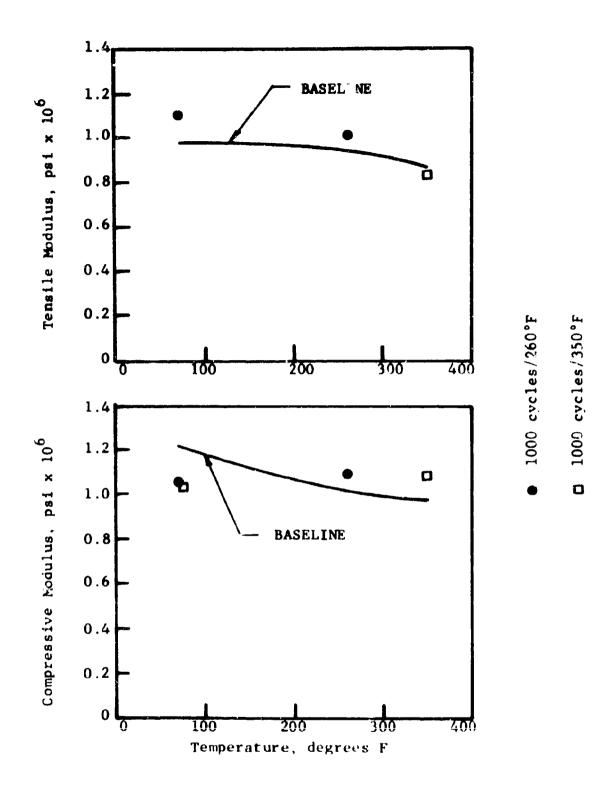


Fig. 69 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF HERCULES 3002 M/COURTA DS HMS GRAPHITE COMPOSITES - 90°

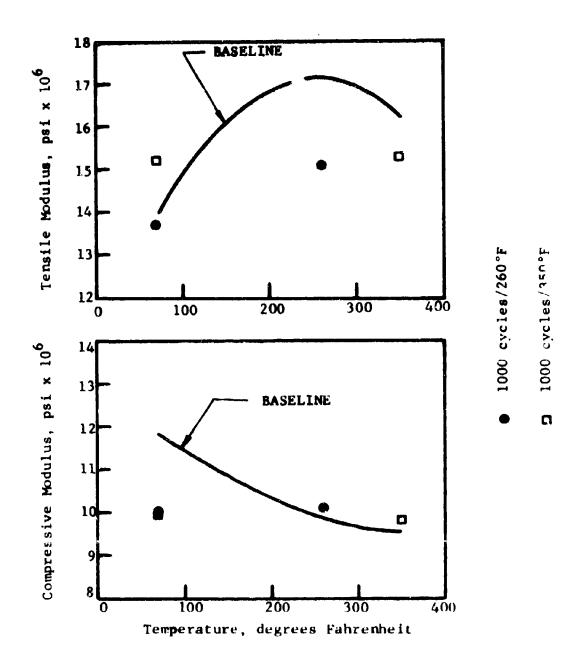


Fig. 70 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE ELASTIC MODULI OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - [0/45/135/0/90]_S

Figures 71, 72 and 73 show how the interlaminar shear strengths are affected by such prior exposures for AVCO 5505/Boron, Narmco 5206/Modmor II Graphite and Hercules 3002M/Courtaulds HMS Graphite composites respectively. All interlaminar shear strengths were obtained on the 0° fifteen-ply short beam specimens.

AVCO 5505/Boron composites show a loss of interlaminar shear strength with exposure to moisture (see Fig. 71a) and all environments result in practically the same loss of interlaminar shear strength.

Constant (noncyclic)temperature exposures (Fig. 71b) increased the interlaminar shear strength. The largest increases were for the least severe exposure (100 hours at 260°F), while the most severe exposure (500 hours at 350°F) increased the interlaminar shear strength the least. In fact the room temperature i.s.s. were least affected while the elevated temperature interlaminar shear strengths were affected substantially more.

Cyclic exposures affected the interlaminar shear strength of AVCO 5505/Boron differently, depending on the peak temperature per cycle. The shear strengths of specimens cycled to 260°F were relatively unaffected by the cyclic exposures whereas some increase or decrease in the interlaminar shear strength was noted for the specimens with 350°F upper temperatures per cycle.

Qualitatively the same effects were noted for the two graphite-epoxy composites (see Figs. 72 and 73) humidity generally decreasing the interlaminar shear strengths over the entire temperature range and mixed effects noted for the steady-state and cyclic thermal conditioning.

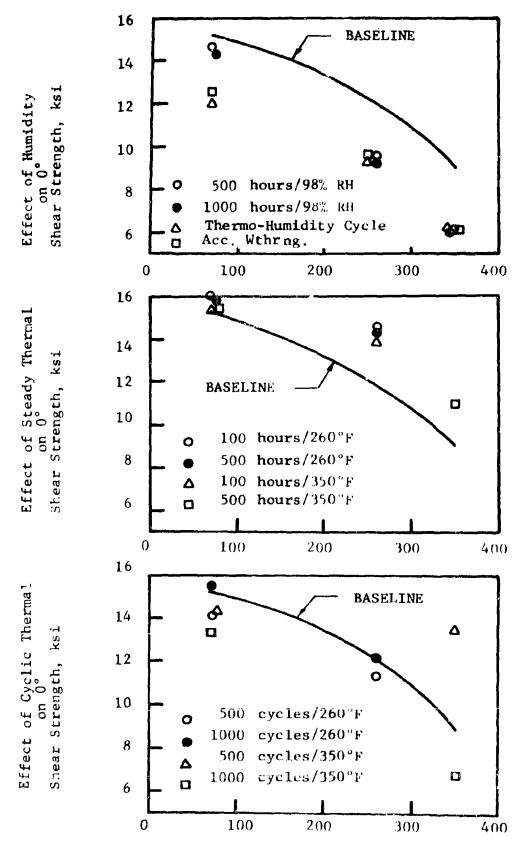


Fig. 71 EFFECT OF VARIOUS ENVIRONMENTAL CONDITIONING ON THE INTERLAMINAR SHEAR STRENGTH OF AVCO 5505/BORON COMPOSITES

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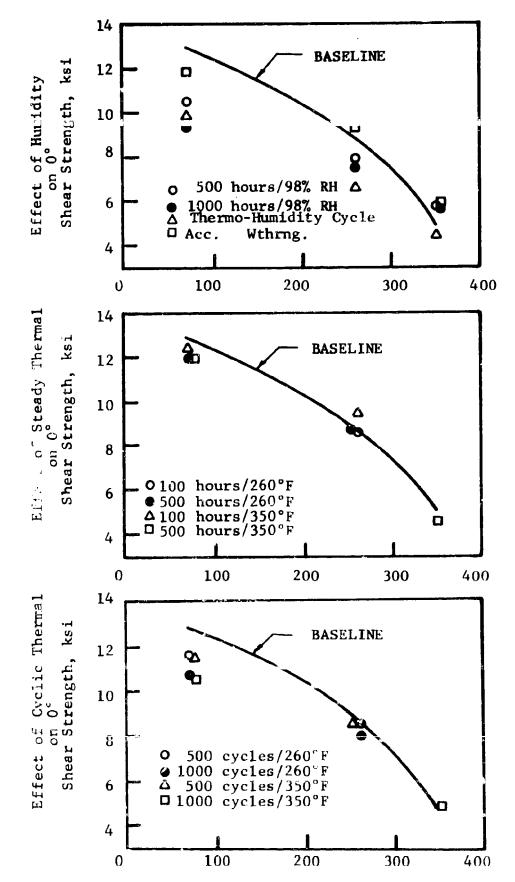


Fig. 72 EFFECT OF VARIOUS ENVIRONMENTAL CONDITIONING ON THE
INTERLAMINAR SHEAR STRENGTH OF NARMOO 5206/MODMOR 11

GRAPHITE COMPOSITES $_{109}$

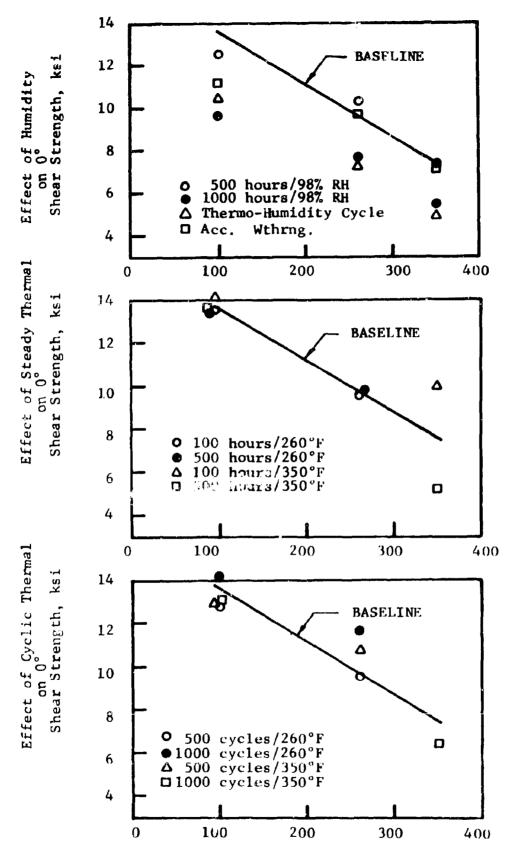


Fig. 73 EFFECT OF VARIOUS ENV CONMENTAL CONDITIONING ON THE INTERLAMINAR SHEAR STRENCTHS OF HERCULES 3002H/COURT AULDS HMS GRAPHITE COMPOSITES

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2.1.7 Fatigue Test Results

2.1.7.1 Baseline Fatigue Data

The baseline fatigue data are presented in S-N plot form in Appendices I through III. It should be noted that there is a general decline in the fatigue resistance of all three composites with increasing temperature.

2.1.7.2 Effects of Humidity Conditioning on Fatigue Behavior

The parametric effects of humidity conditioning on the S-N fatigue behavior is shown in Figs. 74 and 75 for AVCO 5505/Boron unidirectional $[0/45/135/0/90]_{\rm S}$ laminates respectively. The effects are shown at each of the three test temperatures (RT, 260°F and 350°F).

Note that the cyclic humidity conditioning treatments degraded the fatigue resistances of the AVCO 5505/Boron composites considerably more than did the steady-state humidity conditioning treatments.

Similarly the humidity effects on the graphite/epoxy composites are presented in Figs. 76 to 79.

With regard to the fatigue S-N behavior for Narmco 5206/ Modmor II Graphite Composites, it is seen in Fig. 76 note that the ranking is (1) baseline, (2) 500 hours 98% RH, (3) Accelerated weathering, (4) 1000 hours 98% RH and (5) Thermo-Humidity cycle at room temperature. At higher temperatures the 500 and 1000 hours exposures degraded the fatigue behavior more than did the humidity cycles. Similar comments applied to the $[0/45/135/0/\overline{90}]_{s}$ laminates as well (see Fig. 77).

The fatigue degradation due to high humidity conditions for Courtaulds HMS Graphite/Hercules 3002M composites are seen in Figs. 78 and 79. The behavior shown is complex. Some liberal

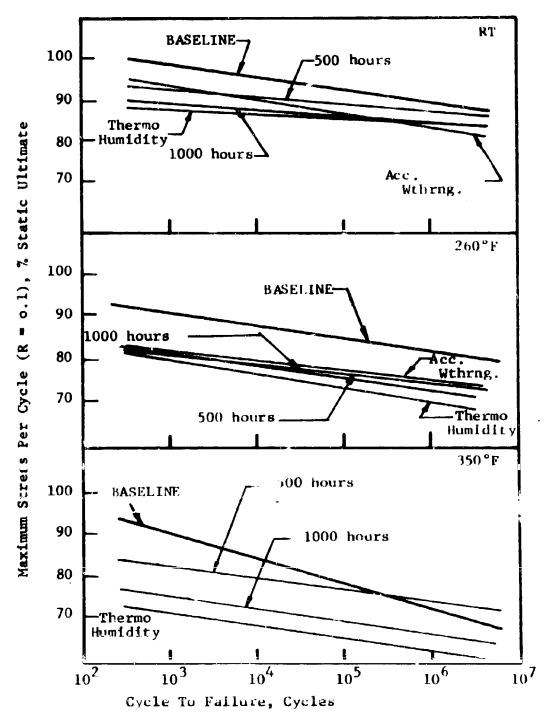
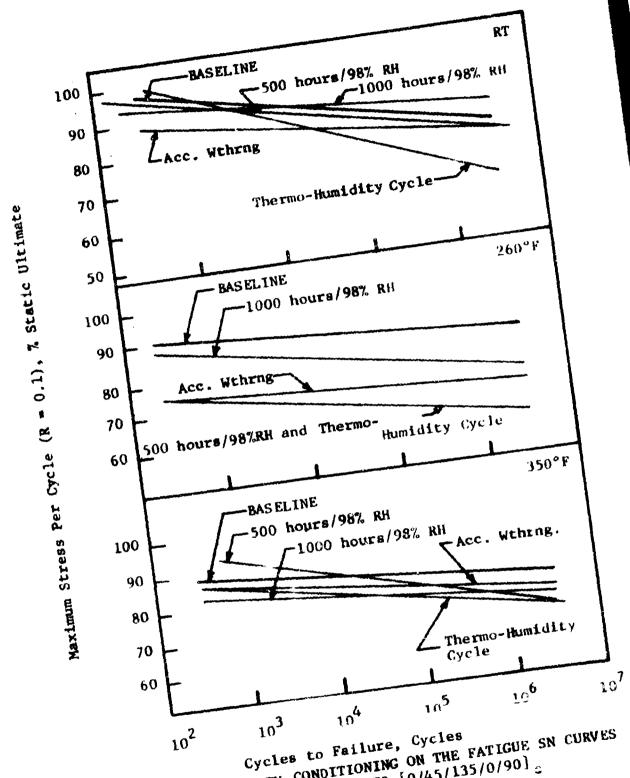


Fig. 74 EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE SN CURVES FOR AVCO 5505/BORON COMPOSITES - 0°



Cycles to reliure, Cycles

EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE SN CURVES

FOR AVGO 5505/BORON COMIOSITES [0/45/135/0/90] Fig. 75

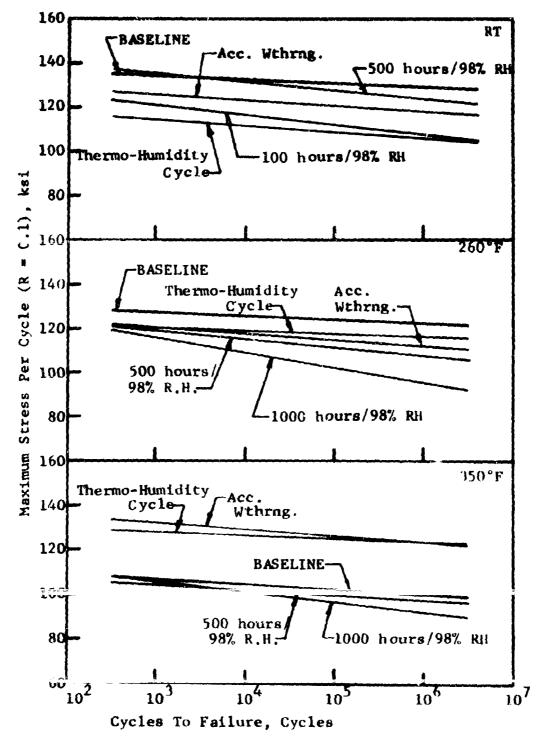


Fig. 76 EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE SN CURVES FOR NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - 0°

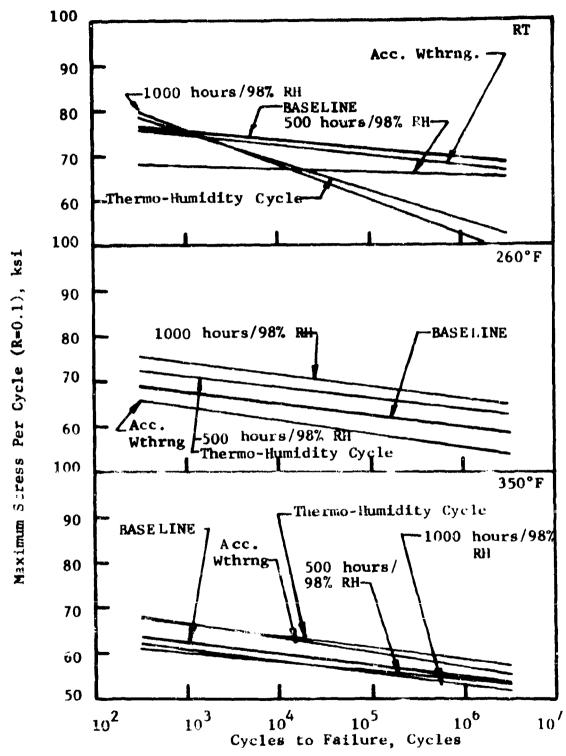


Fig. 77 EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE SN CURVES FOR NARMCO 5206/MODMOR II GRAPHITE COMPOSITES -10/45/135/0/9()s

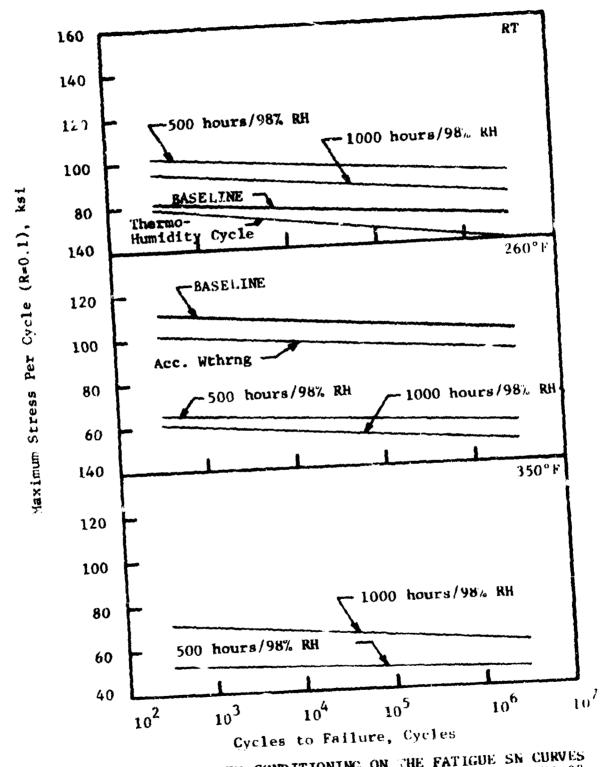


Fig. 78 EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE SN CURVES FOR HERCULES 3002M/COURTAIN DS HMS GRAPHITE COMPOSITES 0°.

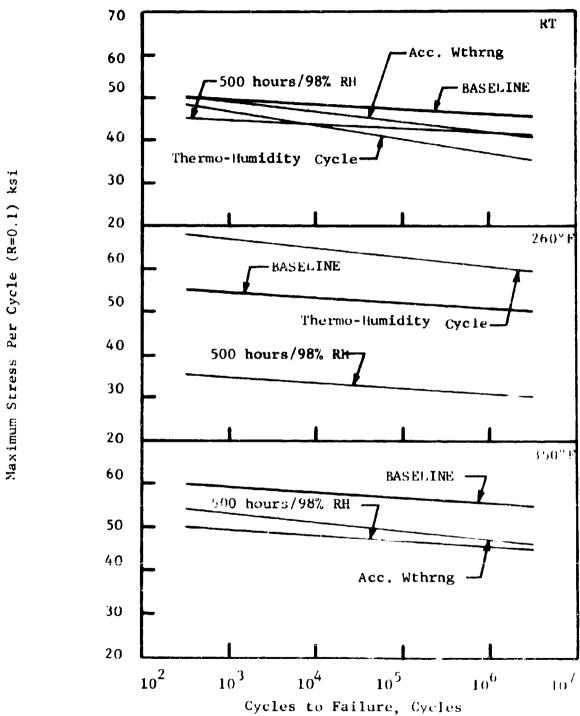


Fig. 79 EFFECT OF HUMIDITY CONDITIONING ON THE FATIGUE S-N CURVES FOR HERCULES 3002 M COURTAULDS HMSGRAPHITE COMPOSITES [0/45/135///90]

interpretation must be applied to the room temperature baseline data which is probably shown too low in Fig. 78. On the other hand, the static strengths rose with temperature and it is possible that the fatigue S-N behavior mirrors this to some extent.

2.1.7.3 Effects of Thermal Conditioning on Patigue Behavior

Several parametric cross plots on the effects of thermal conditioning on fatigue behavior were prepared. The effect of steady-state thermal conditioning are presented as follows:

AVCO 5505/Boron - Figs. 80 and 81

Modmor II Graphite/Narmco 5206 - Figs. 82 and 83

Courtaulds HMS Graphite/Hercules 3002M - Figs. 84 and 85

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The classic degradation of the fatigue S-N behavior with prior steady-state thermal conditioning is seen in Fig. 80 for 0° AVCO 5505/Boron composites. More degradation at the higher cyclic lives is seen in Fig. 81 for the $[0/45/135/0/\overline{90}]_{s}$ composites of AVCO 5505/Boron.

The Narmoo 5206/Modmor II Graphite composites show more complex behavior. Both long term aging effects at the higher cyclic lives and some strengthening effects at the lower cyclic lives are seen in Fig. 82, for 0° composites, particular the room temperature fatigue behaviors. Similar behavior was noted in the case of the $\left[0/45/135/0/\overline{90}\right]_{\rm S}$ laminates although the strengthening was more uniform over the entire range of cyclic lives.

Similar effects are seen for the Hercules 3002M/Courtaulds HMS Graphite composites, (see Figs. 84 and 85).

Cyclic thermal conditioning effects on the fatigue behavior of the three resin matrix composites is shown as follows:

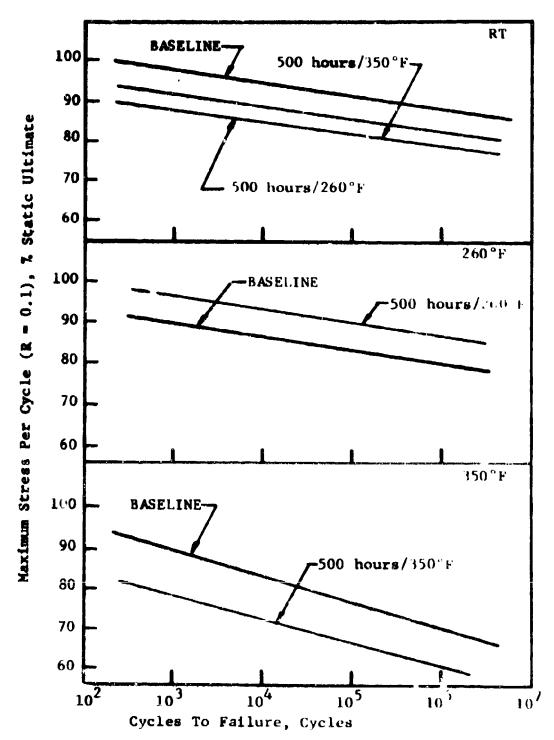


Fig. 80 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR AVCO 5505/BORON COMPOSITES-0°

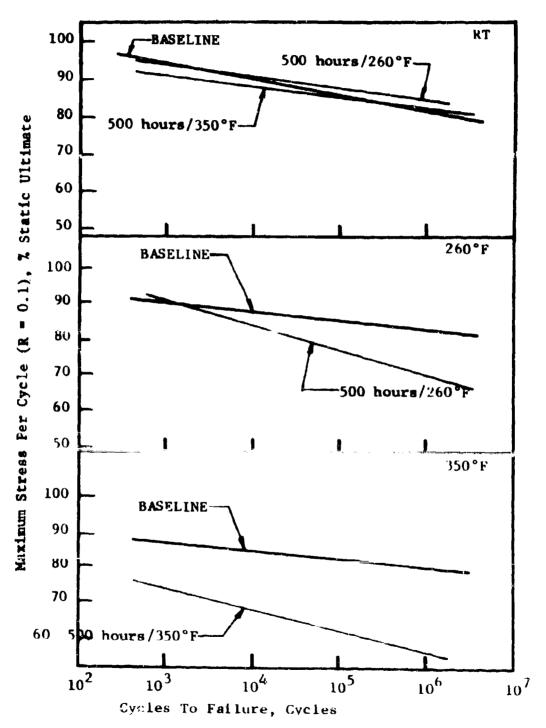


Fig. 81 EFFECT OF SIRADY STATE THERMAL CONDITIONING
ON THE FATIGUE SN CURVES FOR AVCO 5505/BORON COMPOSITES
[0/45/135/0/90]s 120

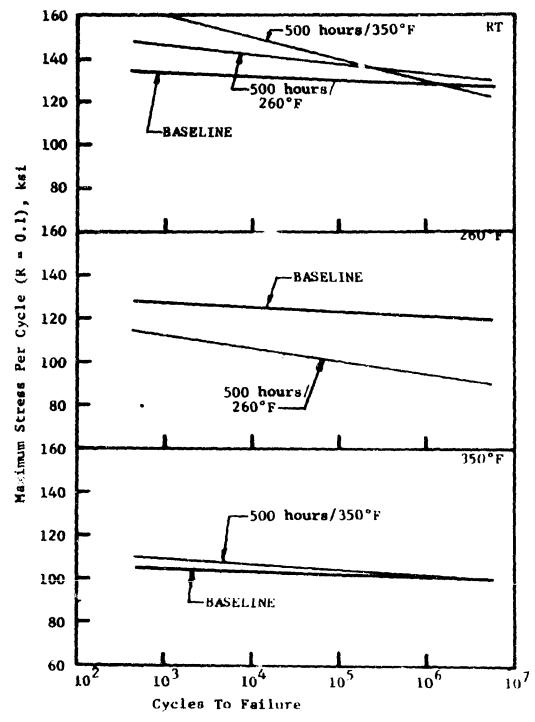
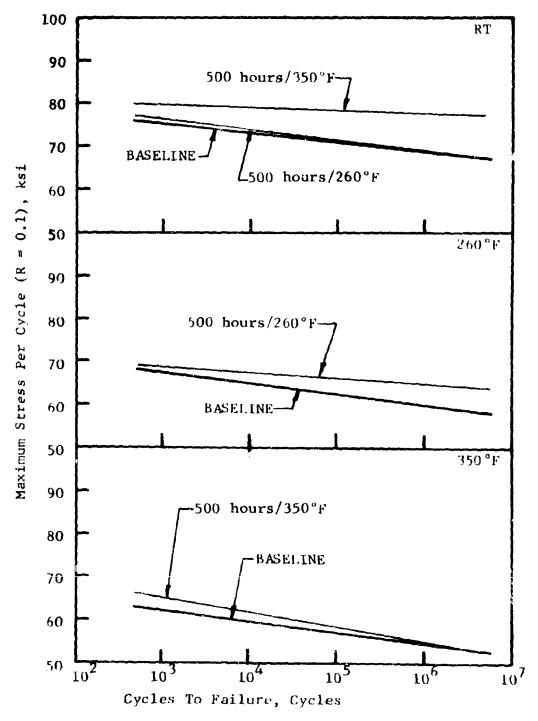


Fig. 82 EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR MODMOR II/NARMCO 5206 COMPOSITES-0°



EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR MODMOR 11 GRAPHITE/NARMCO 5206 COMPOSITES - 10/45/135/0/90s Fig. 83

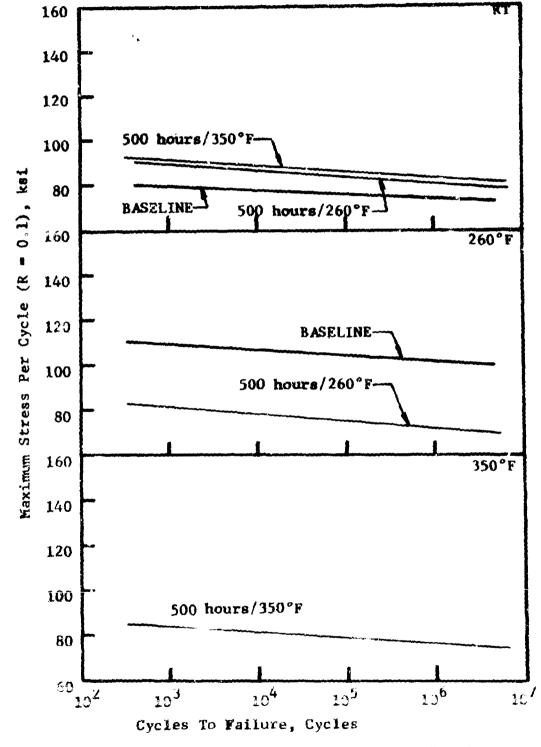


Fig. 84 EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - 0°123

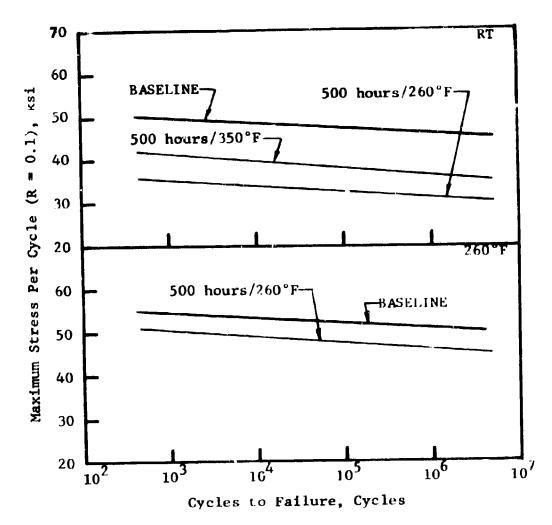


Fig. 85 EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR RERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES [0/45/135/0/90]_S

AVCO 5505/Boron - Figs. 86 and 87

Narmco 5206/Modmor II caphite - Figs. 88 and 89

Hercules 3002M/Courtaulds HMS Graphite - Figs. 90 and 91

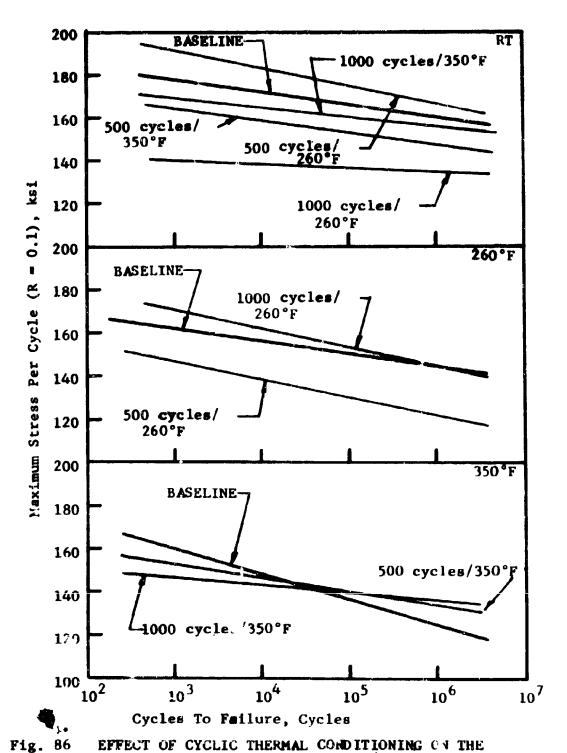
The behavior shows consistant degradation in the following increasing order (1) 260°F for 500 cycles, (2) 260°F for 1000 cycles, (3) 350°F for 500 cycles and the worst (4) 350°F for 1000 cycles. The fatigue data indicates that the materials still retain satisfactory strengths at high temperature after thermal cycling.

2.1.8 Creep and Stress Rupture Test Results

2.1.8.1 Baseline Creep and Stress Rupture Data

loads. Because of the greater creep susceptability of resin matrix composites at elevated temperatures, the creep and stress rupture tests were conducted at elevated temperatures (260°F and 350°F) only. The creep test data were generated in the form of creep strain versus time curves at various percentages of average ultimate tensile stress of the particular material at that temperature. The stress versus time to rupture data was obtained for the composite materials at various percentages of the ultimate tensile stress levels at the two temperatures. Those tests which ran to 1000 hours were terminated at that time and the specimens removed from the test stands. The test results for individual specimens are shown in Appendices I to III. Both stress rupture versus time and creep-time curves are also presented.

Many specimens failed prior to the attainment of the intended load or "during loading." These specimens are so indicated in the tabular presentation of data. Where the majority of specimens for a given conditioning treatment fell into this category, no stress rupture curves were prepared for that particular condition. As this occasionally happened for



EFFECT OF CYCLIC THERMAL CONDITIONING CATHE FATIGUE SN CURVES FC. AVCO 5505/BORON C. MPOSITES - 0°

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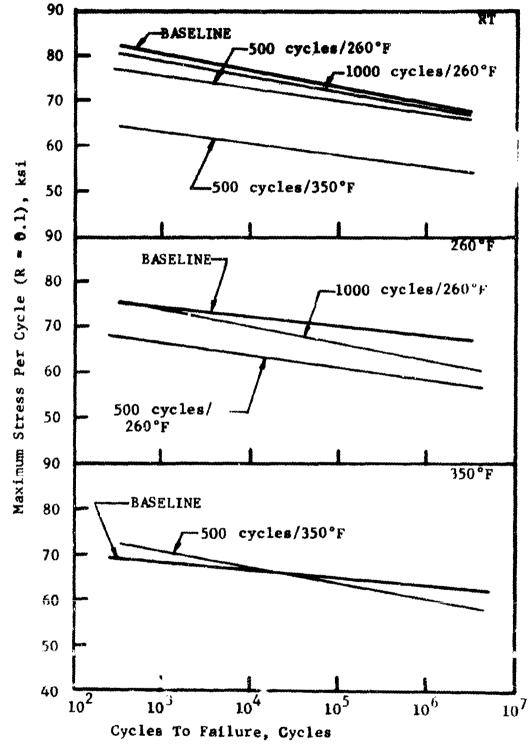


Fig. 87 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR AVCO 5505/BORON COMPOSITES [0/45/135/0/90] 8

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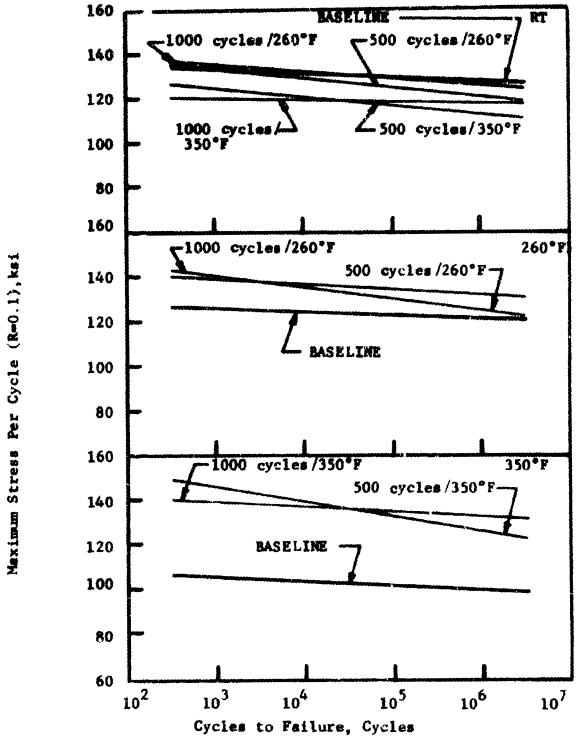


Fig. 88 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR NARMCO 5206/MODMOR II GRAPHITE COMPOSITES -0°.

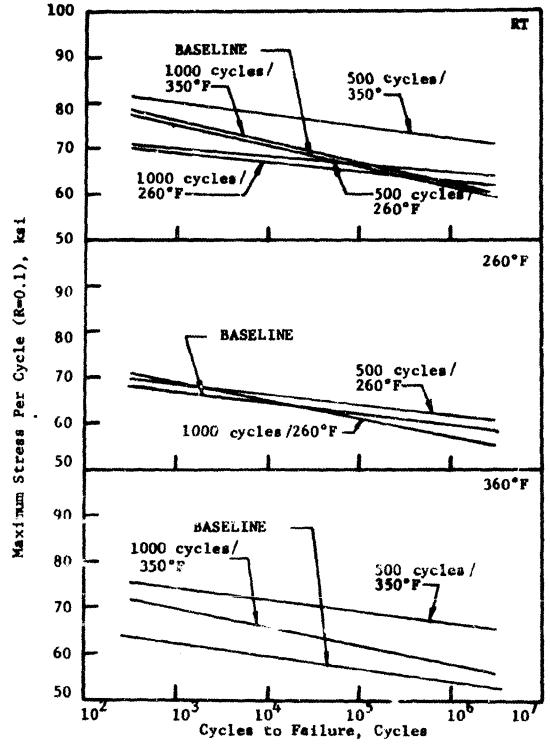
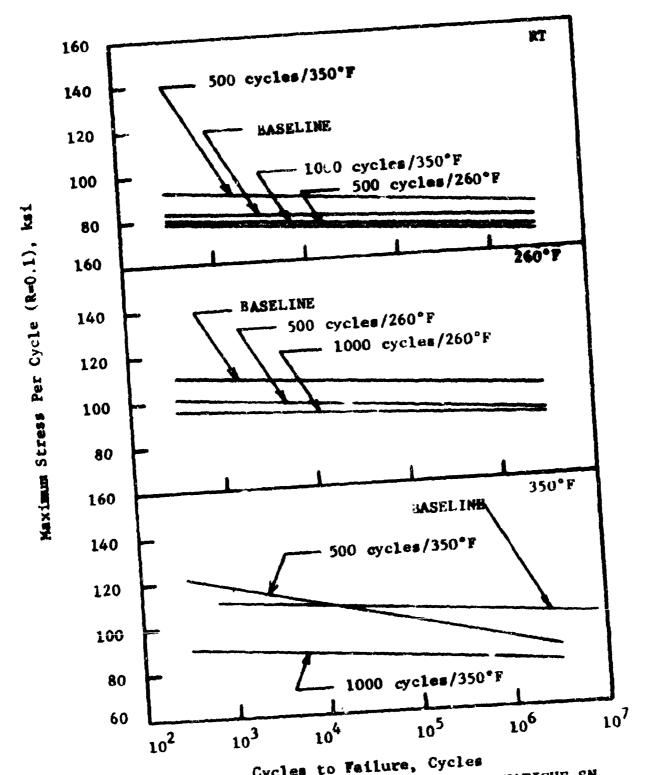


Fig. 89 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE FAZIOUE SN CURVES FOR NARMCO 5206/MODMOR II GRAPHITE COMPOSITES -[0/45/135/0/90]

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Fig. 90 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE FATIGUE SN

CURVES FOR HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES 0°

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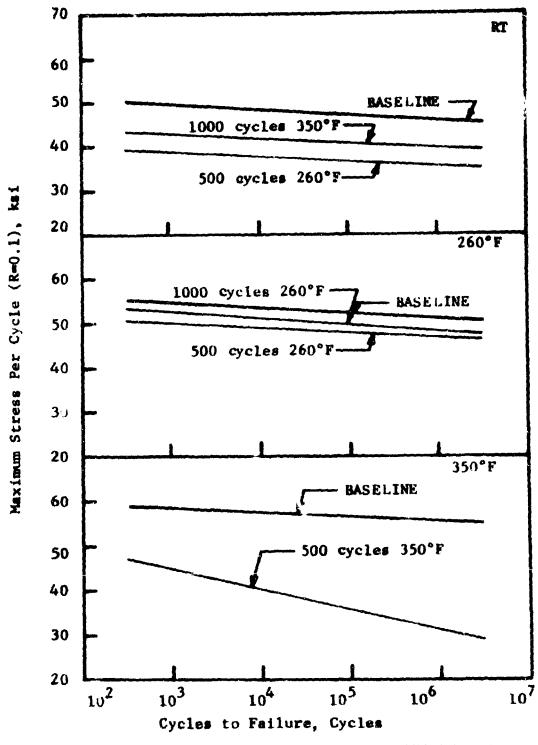


Fig. 91 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE FATIGUE SN CURVES FOR HERCHIES 3002M COURTAULDS HMS GRAPHITE COMPOSITES-10/45/135/0/90).

baseline data as well, some baseline stress-rupture versus time curves are also missing. Where data was obtained over a portion of the time range only, the stress versus time to rupture curves were so indicated.

2.1.8.2 Effects of Humidity Conditioning on the Creep and Stress Rupture Properties

The effect of humidity conditioning, treatments, both steady state exposure and cyclic humidity pretreatments on the creep and stress-rupture properties of AVCO 5505/Boron, Marmoo 5206/Modmor II Graphite, and Hercules 3002M/Courtaulds HMS Graphite composites are presented in Figs. 92 and 93, Figs. 94 and 95, and Figs. 96 and 97 respectively. Both 0° and $[0/45/135/0/\overline{90}]_{a}$ composites were investigated.

Baseline data is not available to compare the effect of himidity conditioning on the stress rupture properties of 0° AVCO 5505/Boron composites at 260°F. Some reduction in the stress-rupture behavior of 0° AVCO 5505/Boron at 350°F is seen in Fig. 92. The stress rupture behavior of the [0/45/135/0/90] laminates at both 260°F and 350°F appears to be only slightly affected (and improved over baseline behavior) after steady state and cyclic humidity conditioning.

A similar set of humidity effects on the stress-rupture behavior of Narmco 5206/Modmor II Graphite Composites is shown in Figs. 94 and 95. All conditioning treatments increased the stress-rupture curves over the original baseline values. The 0° composites were affected the most while the $[0/45/135/0/\overline{90}]_{8}$ laminates were affected the least.

Figures 96 and 97 show the effect of prior humidity conditioning on the stress-rupture behavior of Hercules 3002M/Courtaulds HMS graphite. The only substantial reduction in the

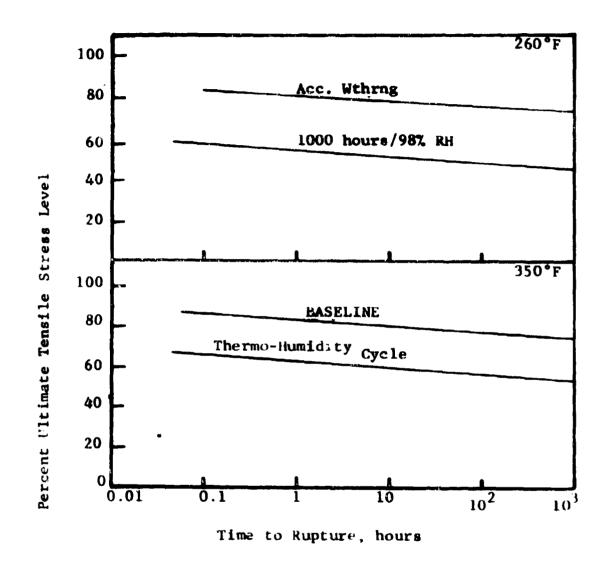


Fig. 92 EFFEC OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVCO 5505/BORON COMPOSITES - 0°

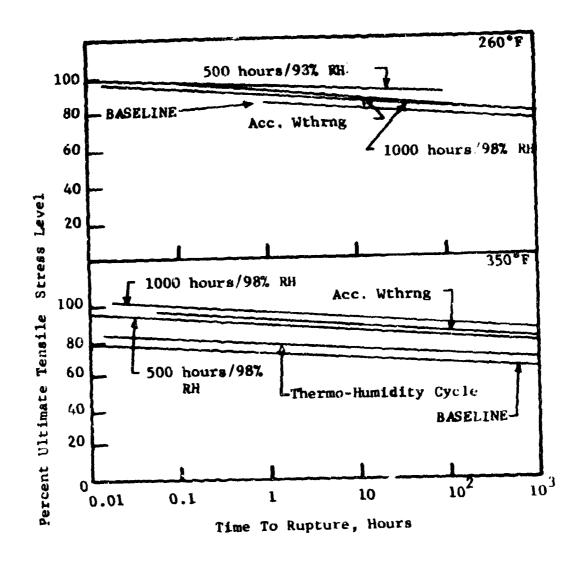
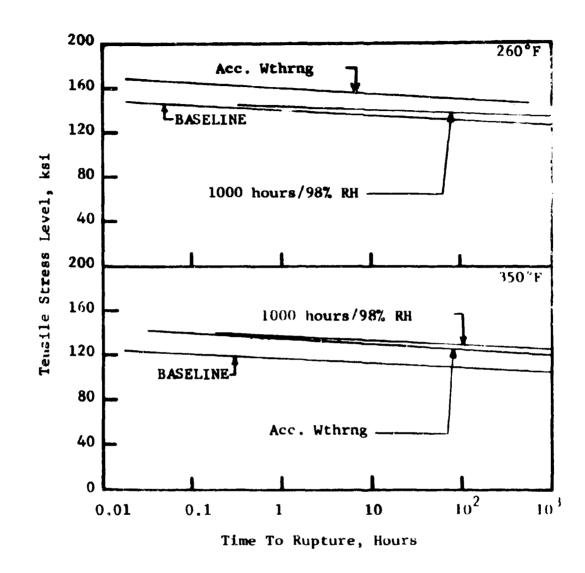


Fig. 93 EFFECT OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVOID DOUD/BURGIN COMPOSITION [0/45/135/0/90]



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Fig. 94 EFFECT OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - 0°

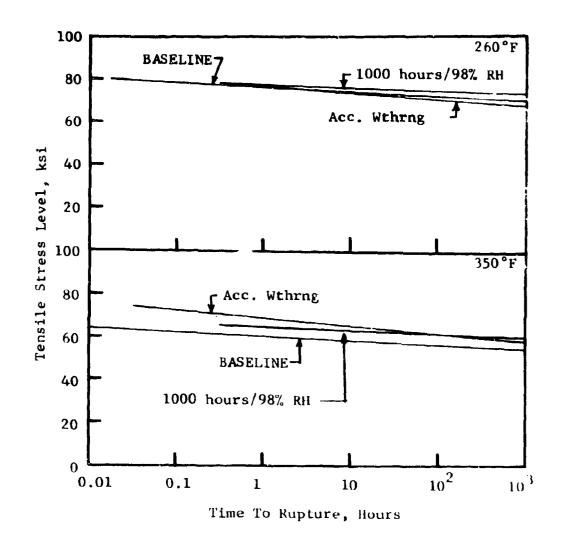


Fig. 95 EFFECT OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - [0/45/135/0/90]s

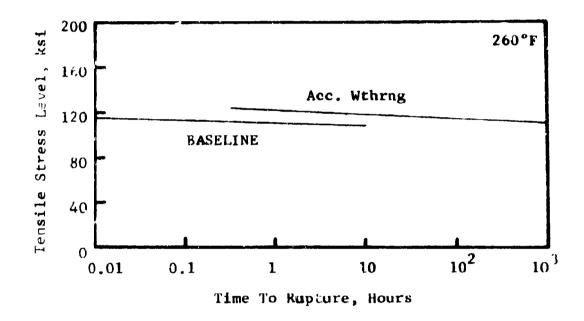


Fig. 96 EFFECT OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF HERCULES 3002 M/COURTAULDS HMS GRAPHITE COMPOSITES - 0°

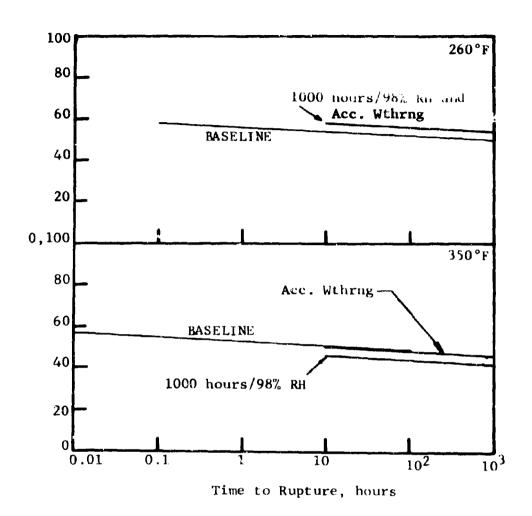


Fig. 97 EFFECTS OF HUMIDITY CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES - [0/45/135/0/90]_s

stress-rupture curves with humidity conditioning for all three materials was detected at $350^{\circ}F$ for the $\left[0/45/135/0/\overline{90}\right]_{g}$ laminates. The accelerated weathering cycle stress-versus time to rupture curves shown in Fig. 97 at $350^{\circ}F$ is coincident with the baseline behavior. However, the 1000 hour at 98% RH data showed a decrease from the baseline values.

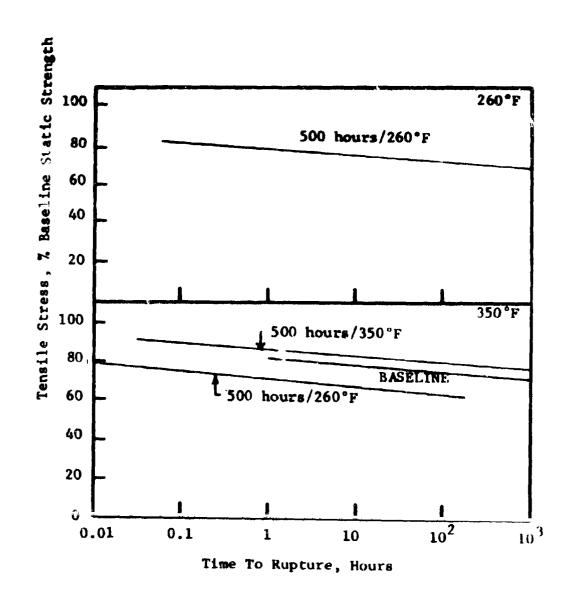
Overall, the stress versus time to rupture behavior of the three resin matrix composites in the unidirectional as well as $[0/45/135/0/90]_8$ laminates was either not affected by prior humidity conditioning or showed a slight improvement in the resistance to rupture under sustained stress.

2.1.8.3 Effects of Thermal Conditioning on the Creep and Stress Rupture Properties

The effects of steady state thermal conditioning on the stress-rupture behavior of the three resin matrix composites are shown in Figs. 98 to 102. Both 0° and $[0/45/135/0/\overline{90}]_{\rm g}$ composites and two temperatures of testing (260°F and 350°F) are presented.

The effect of steady state thermal exposure on the stress versus time to rupture behavior of AVCO 5505/Boron composites is shown in Figs. 98 and 99. In general the exposure at 350°F for 500 hours enhanced the sustained stress properties while the 500 nour exposure to 200°F showed both degradatory and enhancement of the stress rupture properties relative to the unexposed baseline properties. From a logical point of view, it is reasonable to conclude that the thermal exposure may not have enhanced the stress rupture properties but that this conditioning had no adverse affects on the stress-rupture behavior.

The effect of steady-state thermal exposure on the stress versus time-to-rupture behavior of Narmco 5206/Modmor II graphite composites is presented in rigs. 100 and 101. Again the higher temperature exposure (350°F for 500 hours) increased the resistance to failure under sustained load above the 0° baseline



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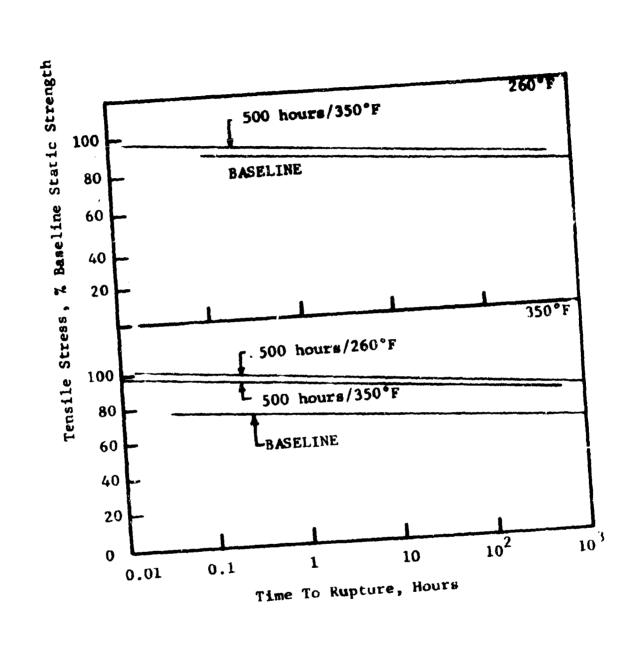
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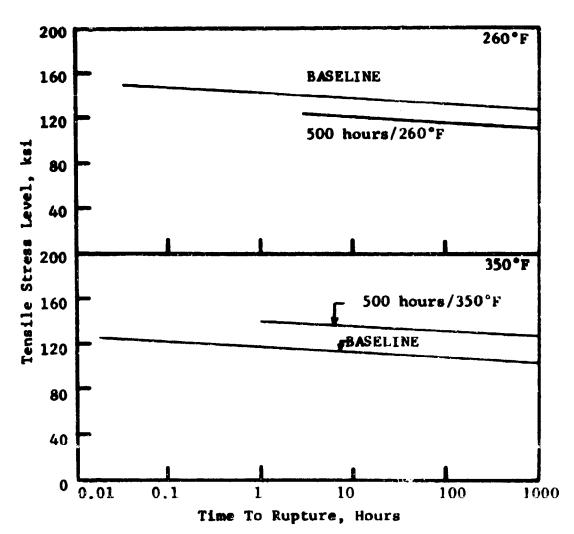
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Fig. 98 EFFECTS OF STEADY-STATE THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVCO 5505/BORON COMPOS TES - 0°



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FIG. 99 EFFECTS OF STEADY-STATE THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVCO 5505/BORON COMPOSITES [0/45/135/0/90]s



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Fig. 100 EFFECT OF STEADY-STATE THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITE - 0°

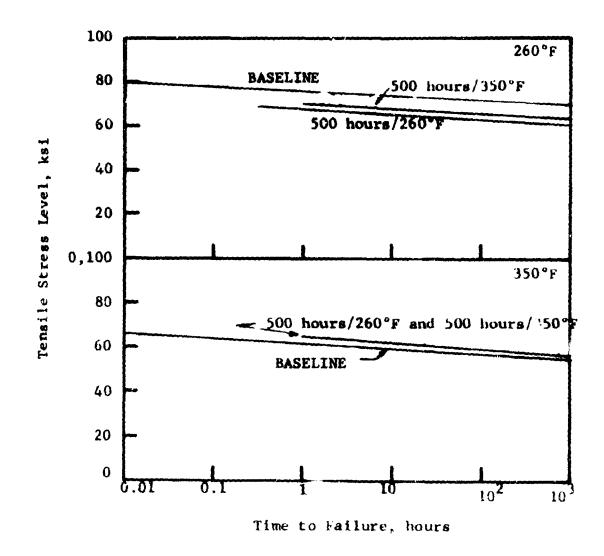


Fig. 101 EFFECT OF STEADY STATE THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR 11 GRAPHITE COMPOSITES - [0/45/135/0/90] s

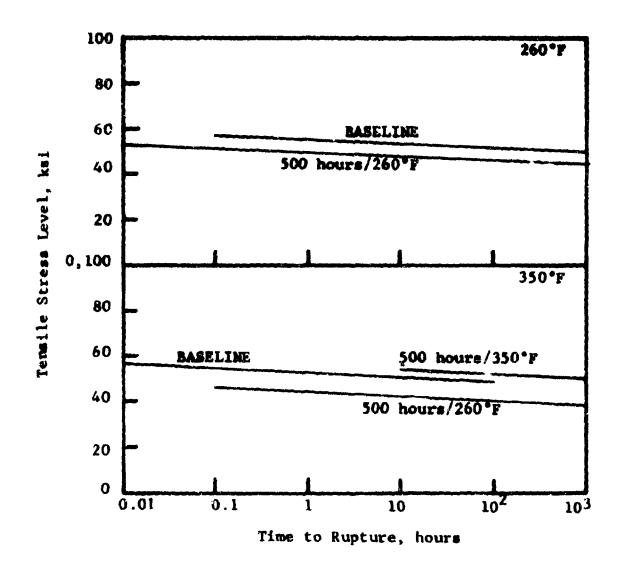


Fig. 102 FFECT OF STEADY STATE THERMAL CONDITIONING ON THE STATES RUPTURE BEHAVIOR OF HERCULES 3002M/COURTAULDS HM 3 GRAPHITE COMPOSITES ~ [0/45/135/0/90]

behavior at 350°F. The 500 hour exposure to 260°F resulted in minor degradation from the baseline behavior.

The 0° stress versus time to failure behavior of Hercules 3002M/Courtaulds HMS graphite is n * shown because the data does not exist or all specimens were 1000 hour runouts. The laminate behavior is shown in Fig. 102. Again the 500 hour exposure to 350°F enhanced the stress-rupture behavior while the 260°F exposure for 500 hours degraded the stress-rupture behavior.

The effect of cyclic thermal conditioning on the stress-rupture behavior of AVCO 5505/Boron composites is shown in Figs. 103 and 104. The rest is showed behavior similar to that of the steady state thermal conditioning. The 500 hours exposure to 350°F showed better performance than both baseline and 500 hours exposure to 260°F. The exception was the $[0/45/135/0/\overline{90}]_8$ laminate creep-tested at 350°F. Baseline data at 260°F for the 0° composites were missing because all baseline coupons were 1000 hours runouts. (See Appendix I - Table XIV.).

Figures 105 and 106 show the effect of cyclic thermal conditioning on the stress versus time to rupture behavior of Narmco 5206/Modmor II graphite composites. The behavior is similar to that for AVCO 5505/Boron composites as discussed above. Finally the stress rupture behavior of the second graphite/epoxy system, Hercules 3002M/Courtaulds HMS graphite is shown in Figs. 107 and 108. No beneficial cyclic conditioning was indicated, all conditioning proving to be degradatory.

2.1.9 Thermo Physical Properties

2.1.9.1 Thermal Expansion

Thermal expansion measurements were made for the three resin matrix composite systems, AVCO 5505/Boron, Modmor II

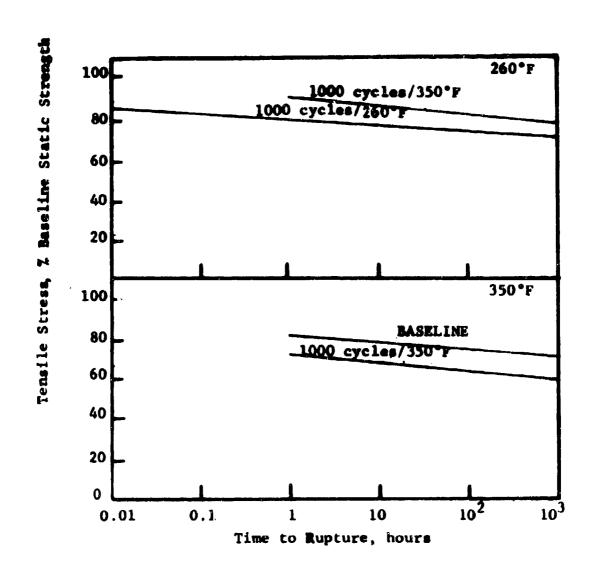


Fig. 103 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVCO 5505/BORON COMPOSITES - 0°

Fig. 104 EFFECT OF CYCLIC THER AL. CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF AVCO 5505/BORON COMPOSITES - [0/45/135/0/90] g

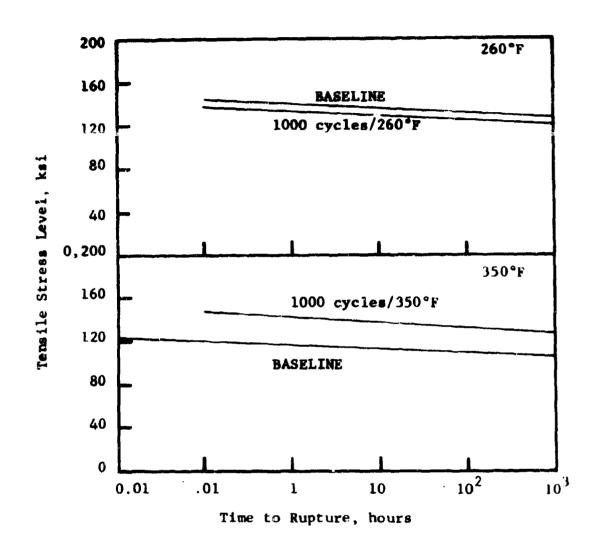
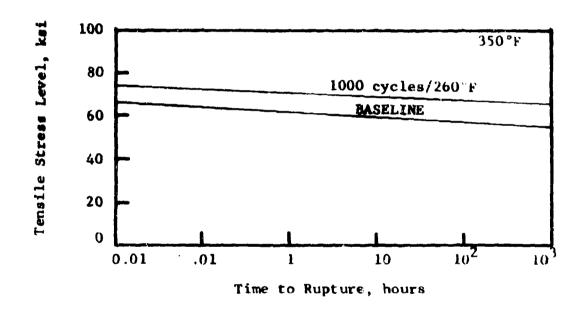


Fig. 105 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR II GRAPHITE COMPOSITES - 0°



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Fig. 106 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF NARMCO 5206/MODMOR 11 GRAPHITE COMPOSITES - [0/45/135/0/90]_s

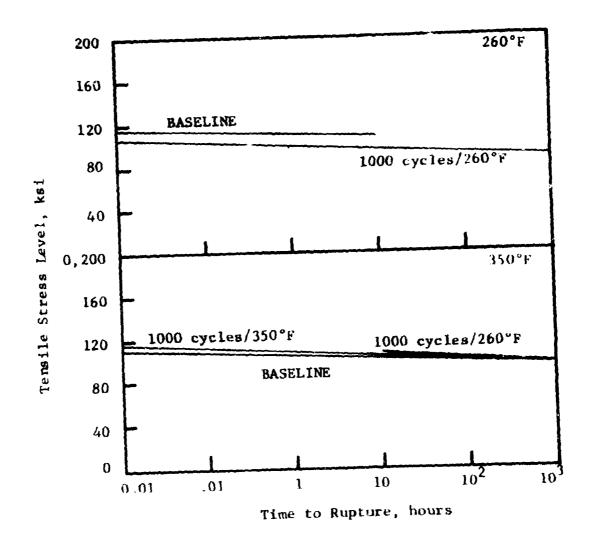


Fig. 107 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE STRESS RUPTURE BEHAVIOR OF HERCULES 3002M/COURTAULDS HMS
CRAPHITE COMPOSITE - 0°

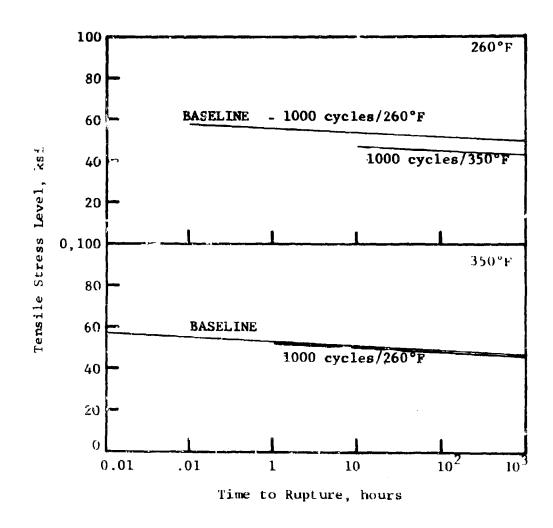


Fig. 103 EFFECT OF CYCLIC THERMAL CONDITIONING ON THE SERESS
RUPTURE DERAVIOR OF RERCULES 3002M/COURTAGLDS HMS GRAPHITE
COMPOSITES [0/45/135/0/90]₈

Craphite/Narmoo 5206, and Courtaulds HMS Graphite/Hercules 300_M. Three samples in each of three fiber orientations (0°, 90°, and $\left[0/45/135/0/\overline{90}\right]_{\rm S}$ with respect to the expansion direction) were tested in air at 4°F/min. from ambient RT to 350°F for each material system employing the NETZSCH Automatic recording pushrod dilatometer described previously (AFML-TR-72-205, Part I).

fiber orientation) a 0.1 to 0.5 percent shrinkage was observed after the first heating/cooling cycle. Stable expansion behavior was observed during the subsequent cycles. This effect was also seen in the 0° and $[0/45/135/0/90]_s$ orientations. The thermal expansion behavior of material of this orientation is much lower. This effect is illustrated in Fig. 109, where percent expansion is plotted against temperature for both heating and cooling cycles for the Courtaulds HMS Graphite/Hercules 3002M material (typical of other materials). Materials experienced slight weight loss during testing, typical weight losses ranging from 0.1 to 0.3 percent.

The instantaneous coefficient of thermal expansion for each resin matrix material and fiber orientation tested was determined for the second cycle stable expansion behavior and is plotted as a function of temperature in Figs. 110 to 112, and tabulated in Table X.

The low expansion of the 0° (longitudinal) fiber orientation composites results from the high modulus fibers restricting the expansion of the low modulus matrix. Since the tensile moduli of the boron and graphite fibers are much higher than tensile moduli of the epoxy resin matrices, it can be predicted from strain compatability considerations that the prope are of the reinforcing fibers control the uniaxial (0°) expansion

Fig. 109 THERMAL EXPANSION BEHAVIOR OF COURTAULDS HMS GRAPHITE/HERCULES 3002M IN THE 90° ORIENTATION

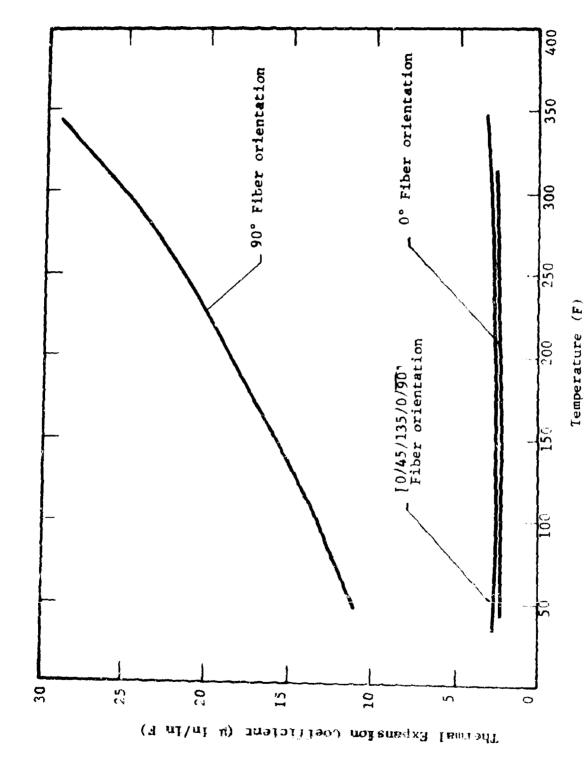


Fig. 110 COEFFICIENT OF THERMAL EXPANSION FOR BORON /AVCC 5505 COMPOSITES

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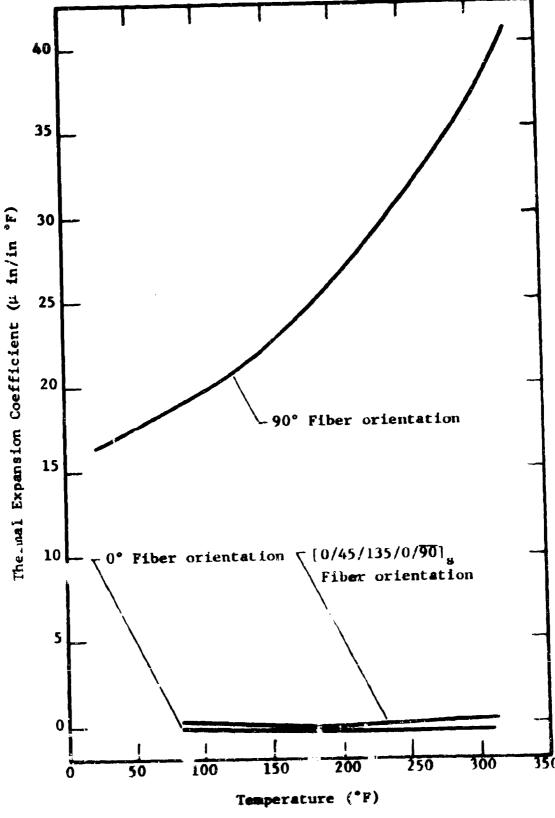


Fig. 111 COEFFICIENT OF THEOMAL EXPANSION FOR COURTABLES HMS GRAPHITE/HERCULES 3002M COMPOSITES

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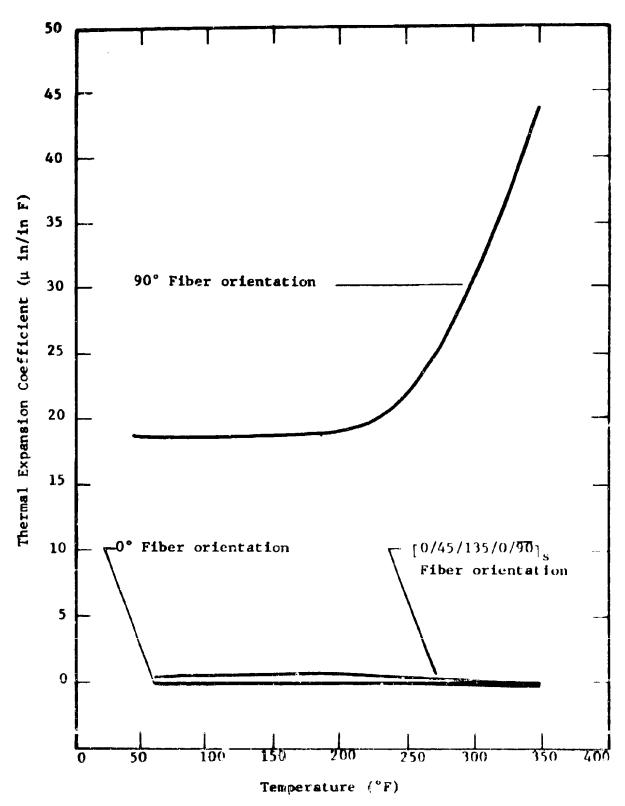


Fig. 112 COEFFICIENT OF THERMAL EXPANSION FOR MODMOR II GRAPHITE/NARMCO 5206 COMPOSITES

Table X

ERAGE INSTANTANEOUS COEFFICIENT OF EXPANSION DATA OF RESIN MATRIX COMPOSITES*

	COEF	COEFFICIENT OF EXPANSION, u in/in °F	F EXPANSI	ON, u in,	/1n °F
Temperature, °F	RT**	100	200	300	350
Boron/AVCO 5505					
0° Ortentation	2.4	2.4	2.5	2.8	3.0
[0/45/135/0/90] _g Orientation	2.6	2.6	2.75	2.95	3.2
90° Orientation	12.3	13.5	18.6	25.1	30
Modmor II Gr./Narmco 5206					
0° Orientation	-0.13	-0.13	-0.20	-0.18	-0.07
[0/45/135/0/90]g Orientation	0.30	0.30	0.30	0.19	0.0
90° Orientation	18.9	18.9	18.9	30.6	43.7
Court. HAS Gr./Hercules 3002M					
0° Orientation	-0.13	-0.20	-0.27	-0.33	-0.45
[0/45/35/0/90] Orientation	0.20	0.23	0.27	0.32	.35
90° Ormentation	18.6	19.5	25.9	36.3	42.2

^{*} Based on 2nd cycle, stable behavior

^{**} Extrapolated

behavior. This was found to occur as evidenced by the data in Table X. Boron fibers exhibit a uniaxial expansion coefficient of 2.7×10^{-6} in/in°F (14), and the AVCO 5505/Boron composite exhibited expansion coefficients ranging from 2.5 - 3.0 in/in°F (from ambient RT to 350°F) in the 0° direction.

Typical graphite fibers used in advanced graphite/epoxy composites exhibit negative expansion coefficients. Both graphite reinforced systems tested exhibited negative composite expansion coefficients in the 0° direction as seen in Table X. It was estimated that the high modulus graphite/epoxy system (Courtaulds HMS Graphite/Hercules 3002M) would have a lower uniaxial (0°) expansion coefficient than the high strength fiber system (Modmor II Graphite/Narmco 5206). This was also observed experimentally, the uniaxial expansion coefficient of the high modulus system being more negative than that of the high strength graphite reinforced material.

These data conform with composite thermal expansion data on boron - and graphite - reinforced epoxy systems found in the literature (14, 15).

The large expansion coefficients for the 90° fiber orientation (transverse) are mainly a result of matrix expansion without restraint effects produced by reinforcement. This increase in importance of the matrix expansion coefficient can be predicted from strain compatability considerations, and is observed in the experimentally generated data summarized in Table X. Data for the transverse (90°) orientation exhibit the general magnitude and temperature dependence of typical epoxy materials as compared to the dependence of the expansion behavior on the reinforcement in the 0° orientation.

Considering both uniaxial and transverse expansion behavior, the reinforcing fibers alone more strongly control the uniaxial expansion behavior than does the matrix alone control the transverse expansion behavior. Thus the fibers have a stranger influence or transverse expansion behavior than the influence of the matrix or uniaxial expansion behavior. These observations are in concurrence with predictions based on stress equilibrium and strain compatability analyses.

In the $[0/45/135/0/\overline{90}]_s$ fiber orientation of each material tested the composite expansion coefficient is also low, indicating the few 0° plies present offer significant restraint to the composite. The angled plies also offer significant reinforcement according to literature data (16).

2.1.9.2 Effect of Absorbed Moisture on Thermal Expansion Behavior of Resin Matrix Composites

It has been demonstrated that each epoxy resin matrix composite investigated absorbs water vapor during shelf storage, as evidenced by weight gain, which results in unstable thermal expansion behavior upon initial heating and cooling, with more stable behavior in subsequent thermal cycles. This behavior has been observed in similar materials systems (15). In particular, the graphite/epoxy composites showed this behavior most clearly.

The unstable first cycle, stable second cycle expansion behavior for the transverse orientation Courtaulds HMS Graphite/
Hercules 3002M material was presented before in Fig. 109. To indicate the role that absorbed water vapor has on this phenomena, a sample, not praviously tested, of the same material was exposed

to a 350°F environment for 63 hours. A thermal expansion test was then conducted on this material, the result of which is shown in Fig. 113. The prolonged temperature exposure has resulted in the elimination of the unstable first cycle behavior. This identical sample was then subjected to 98% R.H. (relative humidity) for 812 hours, until the sample regained its pre-350°F cure weight. Subsequent thermal expansion testing indicated unstable first cycle and a more stable second cycle expansion behavior as presented in Fig. 114. This behavior is similar to that shown in Fig. 109, indicating that the absorbed moisture is responsible for the observed unstable expansion behavior.

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2.1.9.3 Thermal Conductivity Results

Thermal conductivity measurements were made on three reinforced epoxy systems, AVCO 5505/Boron, Modmor II graphite/Narmco 5206, and Courtaulds HMS graphite/Hercules 3002M. Three samples in each of three fiber orientations (0°, 90°, and [0/45/135/0/90], with respect to the heat flow direction) were tested from ambient RT to 350°F employing the guarded steady state longitudinal heat flow method previously described AFML-TR-72-205, Part I.

Thermal conductivity results for these resin matrix materials are presented as a function of temperature in Figs. 115 to 117. For all three materials systems the thermal conductivity in the 0° direction (parallel to fibers) is higher than in the transverse (90°) direction, with the minud ply $[0/45/135/0/90]_g$ orientation data folling in between. The straight-line representation of the data shown for each material orientation was derived from a linear least squares data analysis. Typical thermal conductivity data scatter for these composite

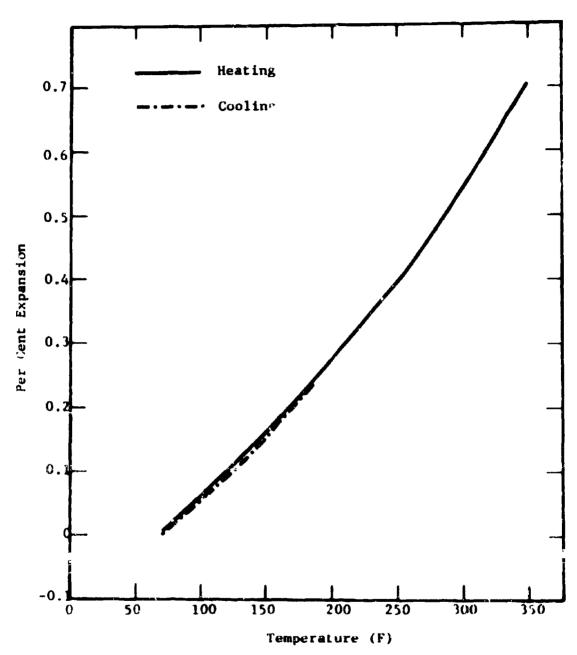
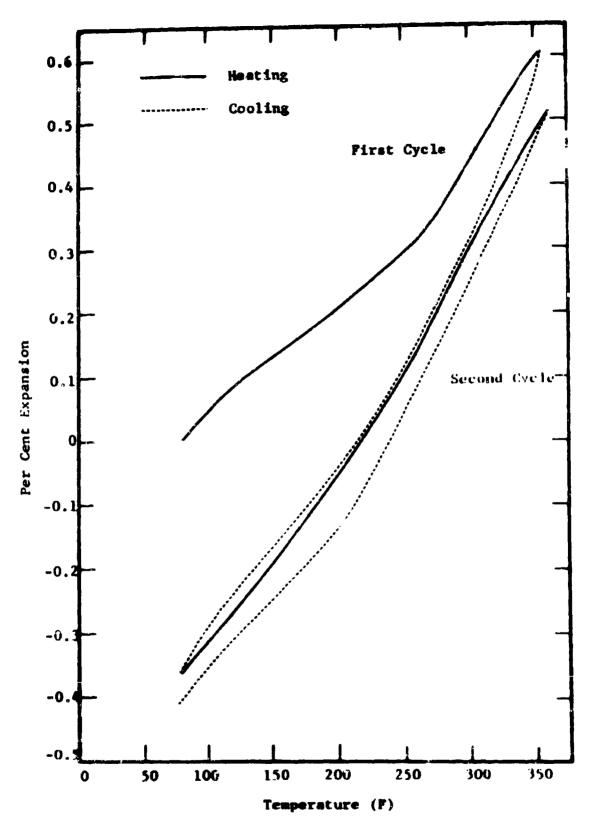


Fig. 113 THERMAL EXPANSION BEHAVIOR OF GRAPHITE REINFORCED RESIN COMPOSITE (COURTAULD'S HMS GRAPHITE/HERCULES 3002M, 90° ORIENTATION) CURED AT 350°F for 63 HOURS



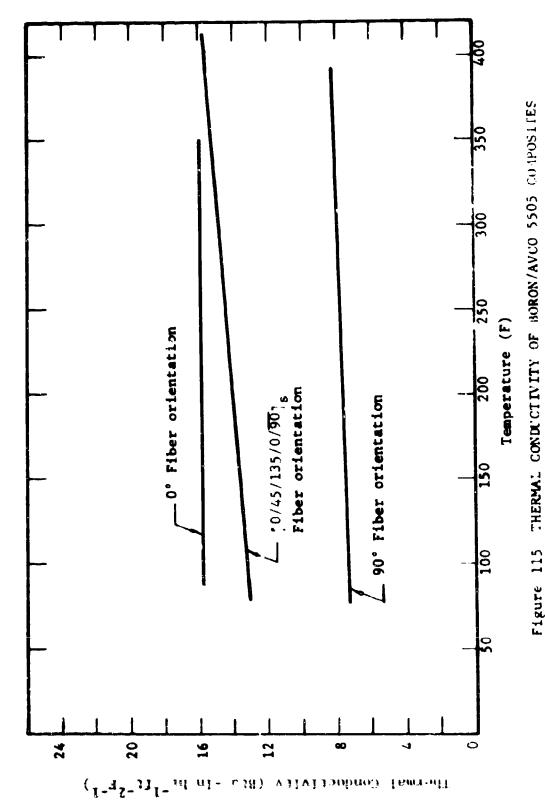
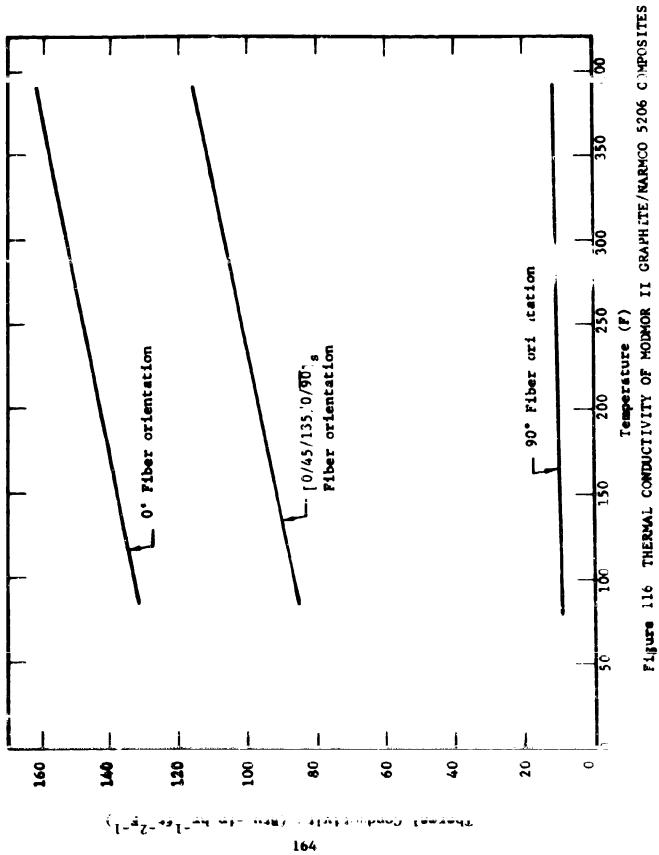


Figure 115

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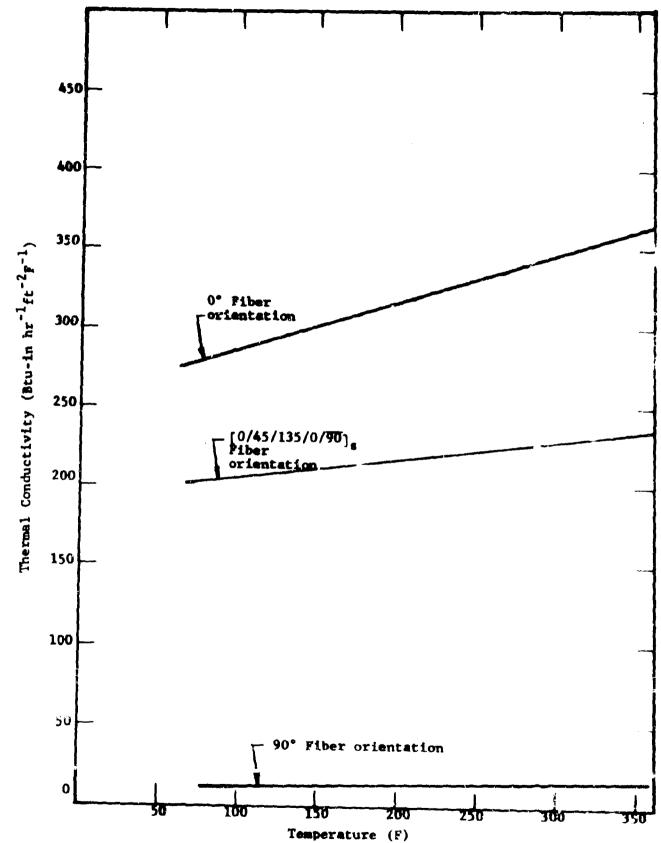


Figure 117 THERMAL CONDUCTIVITY OF COURTAULDS HHS GRAPHITE/HERCULES 3002m COMPOSITES

materials ranged from \pm 4 to \pm 12% maximum deviation from the linear representation shown. Data for the uniaxial Courtaulds HMS Graphite/Hercules 3002M material, however, exhibited \pm 20% variation from the linear representation, owing to sample variability.

The Courtaulds HMS Graphite/Hercules 3002M materials (high modulus fibers) exhibited thermal conductivity data substantially higher than the Modmor II Graphite/Narmco 5206 materials (high strength fibers). Although the properties of the Hercules 3002M and Narmco 5206 Matrices are not readily available, this result is possibly due to the increase in uniaxial fiber thermal conductivity with increasing uniaxial tensile modulus that has been observed in other graphite reinforced epoxy composites (17).

The AVCO 5505, Boron exhibited lower thermal conductivity than either of the two graphite reinforced composites studied. Although the properties of the respective matrix materials are not readily available, this result would not be unexpected owing to the lower thermal conductivity of boron fibers as compared to graphite fibers.

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Both graphite reinforced materials exhibited substantially higher thermal conductivity directional anisotropy than the borom reinforced material. This is presumably due to the greater difference between fiber and matrix conductivity for the graphite reinforced materials as compared to the borom reinforced composites.

In general, good agreement was obtained in comparing the experimentally generated data for the 0° and 90° fiber orientations of the three resin matrix systems studied with data derived from analytical prediction techniques employing familiar parallel and series thermal analogies (16).

2.1.1.0 Fracture Modes

The failure patterns for 0° resin matrix composites depended upon mode and type of loading and on the prior conditioning to a lesser extent. Static tension of the AVCO 5505/ Boron composites produced fractures which generally propagated fully or partially across the specimens in a straight path. The partial fractures also had straight smooth transverse fracture paths which were coupled with longitudinal fractures parallel to the filament directions thus resulting in an overall steplike fracture pattern. Fatigue fracture patterns of the 0° AVCO 5505/Boron composites contained more of such steps with many fibers involved in each transverse path. The creep fracture patterns of the 0° AVCO 5505/Boron were different from both static and fatigue patterns. The creep patterns in the 0° AVCO 5505/Boron composites showed many individual filaments pulled from the matrix in random locations so as to appear as a bundle of broken fibers of various lengths. Environmental conditioning modified these appearances only slightly. Humidity conditioning caused the fatigue and creep fractures to appear less fragmentary. Thermal conditioning caused the fracture surfaces to appear more fragmentary except in the case of fatigue and creep fracture surfaces where the fractures assumed a more straight or transverse crack direction. The two graphite composites behaved similarly to the AVCO 5505/Boron composites except that the fatigue fracture modes were more fragmentary than the corresponding AVCO 5505/Boron fractures.

The failure patterns for 90° composites showed practically no differences between the various fibers. Static tension failures were clear, flat and very nearly lay in plane perpendicular to the direction of loading. The 90° compression

failures (coupon specimens) consistently broke in a fracture plane inclined to the direction of loading. A wedge shaped piece was nearly always broken from the coupons after completion of the test specimens. The fatigue and creep fracture patterns were quite similar to the static tensile patterns for all three resin matrix composites except that the Hercules 3002M/Courtaulds HMS Graphite composites also delaminated.

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Laminate $[0/45/135/0/\overline{90}]_s$, composites generally failed in an irregular path, but straight across the specimens rather than in long steps as in the 0° static tensile patterns. Static compression patterns were so fragmentary that no analysis could be made of the origin and progress of the racking. The fatigue failures generally showed two fractures after testing, but one of these most likely occurred as a result of unrestrained compression followed by bending of the sample following initial specimen failure before the fatigue machine finally stops. Creep failures of the $[0/45/135/0/\overline{90}]_s$ composites also showed double failures on a frequent basis. Considerable delamination of the static tensile, fatigue and creep specimens of Hercules 3002M/Courtaulds HMS graphite composites was evident after testing.

2.2 Metal Matrix Studies

2.2.1 Materials

Although the excellent specific strength and stiffness properties of certain metal-matrix composite materials have been known to the aerospace industry for some time, the application of these materials to structural components has been delayed because of high initial material cost and the lack of fabrication methods that are economical and capable of producing

panels of consistent quality. Two promising methods that are commercially used to fabricate metal-matrix composites were considered for this program.

- 1) Diffusion bonding in which the filaments are encapsulated by hot pressing of the matrix foils in an inert atmosphere.
- 2) Plasma spraying technique where matrix plasma is sprayed on filaments to form composite monolayers (or tapes). This method then requires further processing to produce multilayer laminates.

A substantial number of aerospace applications for metal matrix composites utilize 6061 Aluminum together with the boron fiber. Therefore, this system (6061 aluminum/boron) was selected for characterization in this program.

The second metal matrix system selected was 6A1-4V-Titanium/BorSiC because of its utilization in turbine blade applications. A least two aerospace companies have shown an interest in this material.

2.2.2 Material Procurement

The maximum capability temperature of the aluminum matrix composites is generally defined as 600°F and the titanium matrix composites, 800°F. 5.6 mil boron and 5.7 mil BorSiC were used in the preparation of the laminates. Table IV in Section II presented the test program utilized for evaluation of the metal matrix composites.

Specifications for metal matrix composites did not exist at the time that these two materials were ordered for use on this program. The aluminum/boron composites fabricated contained 50 percent fiber volume. The titanium BorSiC composites contained 45 volume percent fibers. Preparation of the titanium composites was on a best efforts basis.

Inc. The vendor fabricated the material was procured from Amercom Inc. The vendor fabricated the material in laminate form. The composite was diffusion-bonded in vacuum. It was initially assembled by winding boron filaments onto a thin foil of 6061 Aluminum. The filaments were held in place by a "Fugitive binder". The sheets of foil and filaments were then assembled to the requisite lamination, placed in a stainless steel vacuum bag and the bag was evacuated. The binder was eliminated at a low pressure and temperature under a dynamic vacuum. The heat was then raised to the pressing temperature and consolidation was carried out in the solid state under pressure. Following the consolidation the fully consolidated laminate was removed from the bag trimmed and chemically cleaned prior to delivery.

6Al-4V - Titanium/BorSiC material was fabricated by TRW Inc. These laminates were prepared as follows.

BorSiC filaments were wound on a 16 inch diameter drum mounted in a filament-winding machine. The filament spacing was accurately maintained to provide the desired filament volume percent. The filaments were drawn through a glass nozzle in the process which added a polystyrene binder coating to the fiber. The collimated fiber mat is next cut and inserted between two titanium foils. This monolayer was then placed between two stainless steel or molybdenum separator which is coated with graphite and boron nitride antiadhesive coatings. The assembly

is then placed inside a stainless-steel capsule which is then evacuated. Following this the capsule is hot pressed which breaks down the polystyrene into gaseous decomposition products and these are removed by a dynamic vacuum. When the bonding temperature is reached, the pressure is increased and the assembly is bonded for a period of time. Following this the load is reduced the monolayer is removed and the surface is etched to a 50 fiber volume percent thickness. Then the monolayers were stacked between 20 mil thick doubler plates and the new assembly is subjected to pressure and elevated temperature. Thus two distinct diffusion bonding operations are used in the overall process.

2.2.3 Metal Matrix Material Test Specimens

Figure 118 presents the specimen geometries of the various metal matrix test specimens employed in this program.

Referring to the Fig. 118a the tension, and tensile fatigue and creep specimens were similar to the IITRI straight sided tab ended coupons used for the resin matrix studies with 2 in. gage lengths and 4 inches long. The specimen shape was arrived at by machining of the 20 mil doubler plates bonded during plate fabrication on top and bottom surfaces instead of bonding of a tab on the laminate as with resin matrix composite tests specimens. With the removal of 18 mil foil layer on either surface, the specimen thickness was approximately 44 mils.

Figure 118b shows the 15-ply compression and R=-1 and 10 fatigue coupon geometry which was obtained by machining in a manner similar to the tension specimens. The specimen had a gage length of 1/2 in. and with the removal of 18 mil cover on either side, the test section thickness came to about 110 mils.

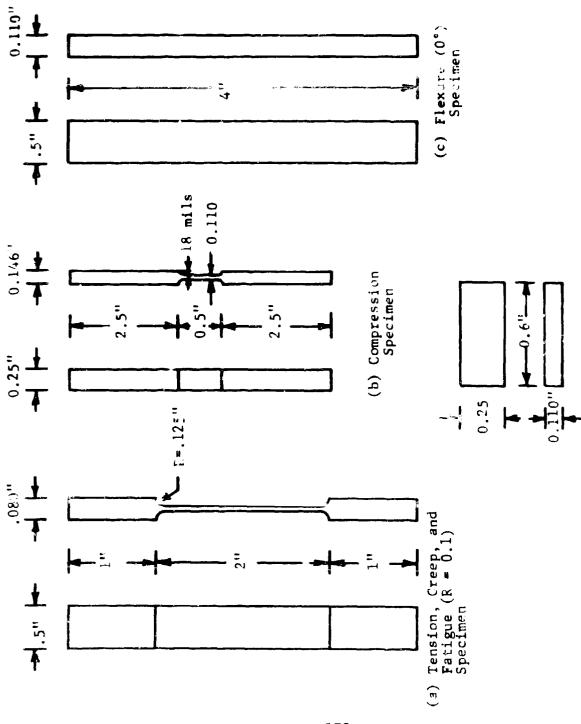


Fig. 118 METAL MATRIX TEST SPECIMENS

(d) Interiamine Shear Specimen

The flexural specimens of 0° and 90° fiber orientations were the same as those for the resin matrix composites. These oppositions had 15 plies with a thicknesses of 110 mils after the removal of the cover. These specimens were tested in a manner similar to the testing of the resin matrix flexure specimens.

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Interiaminar shear specimens were also 15 mils thick and they were 1/2 in, wide and 0.6 in, long. The test procedure was identical to that for the resin matrix specimens with three point loading.

Thermal conductivity and Thermal expansion specimens were 2 in. by 1/2 in. by 6 plies in thickness. For the former tests, several of these specimens were stacked together to a total thickness of 1 inch.

Density determinations were made on specimens similar in size to those used for thermal property tests. The fiber volume density was measured by gravimetric means.

2.2.3.1 General Specimen Machining Procedures

The 6061 Aluminum/boron panels fabricated by Amercom Inc. had a 20 mil thick 6061 aluminum doubler plates diffusion-bonded to both the top and bottom surfaces of the panels.

The tension and compression specimens required tabs at the grips. By machining away the doubler plates in the test section of the specimens (to a depth of 18 mils) we were left with 18 mil thick tabs on each end of the sample on both sides of the specimen. This provided a composite specimen with a uniform matrix covering the filaments throughout the entire test section. For specimens that did not require the tab thicknesses such as in flexure and interlaminar shear tests, the

doubler plates were machined to remove 18 mils uniformly over the full area of the specimen.

The actual machining work was done as described below:

The blank plates were held in a fixture on the surface grinder to permit two cuts in the longitudinal direction, using a 1/16" cutoff wheel. These two cuts removed the rough edges. The plates were then turned 90° and held in a similar clamping fixture to cut the plates to the specimen length. The reduced section was then ground in the plates using a grinding wheel with the corner radius dressed on both edges. At this point the plates were held in a fixture and the specimens were cut to the proper width with a cutoff wheel. After this operation the specimens were deburred.

A similar procedure was adapted for specimen fabrication from the 6Al-4V - Titanium/BorSiC material which also had a matrix foil cover of 20 mil thickness on either face.

2.2.3.2 <u>Machining Procedure for Titanium/BorSiC Composite</u> Specimens from Diffusion Bonded Plates

The edge condition of each blank plate was examined. If there was any doubtful edge sections, a predetermined amount was removed to assure a uniform density.

This whole operation was done in a steel hold down fixture in a surface grinder, (the clamp bar being 0.015" to 0.030" away from the cutoff wheel) to prevent fiber separation. The wheel utilized was an.

ALLISON, VA-602-M-RA, (1/16" x 3" x 12") operated at 2800 RPM spindle speed. A normal traverse, 0.001" downfeed--water soluble coolant was employed.

Depending on the fiber orientation relative to the specimen geometry, the specimen blanks were removed in the following manner:

- (A) Specimens that required the fibers in the longitudinal direction were cut in strips to the correct width and the full length of the plate. (Using the fixture described above).
- (B) Specimens with transverse fibers were cut to the correct width from sections that were cut from the blank plate-each section cut to the specimen length and parallel to the fiber direction. A smaller plate fixture of similar design was used to cut transverse specimens to the correct width. The wheel utilized was the same as above.

The reduced area was ground in a hold down fixture that accommodated twelve specimens and which clamped the specimens within 1/3?" of the reduced section. Equal amounts were removed from each side. When the specimens were turned over for grinding, a suitable shim was placed in the original reduced area to prevent deflection and to act as a heat sink. The wheel used was a,

NORTON, 37C-60-JVK, (1" x 3" x 12") operated at 2200 RPM spindle speed. A normal traverse, 0.0005" downfeed-soluble coolant was employed.

The radius was dressed on both edges of the wheel. The final grind, on both sides, was at 0.0001" to 0.00015" downfeed.

The amount of stock removed per pass was critical; any increase in the grinding cut caused excessive heat and tended to make the specimen deform upwards into the wheel, exposing the fibers.

Plain specimen blanks (no reduced section) were produced as described above for the specimen blanks.

The material removal of the entire surface of the plain blanks was done individually. A vise with "step jaws" was used to hold the specimens during the grinding operation. An equal amount of material was ground from each surface. The wheel used was as described above for removing material for tensile coupons.

The grinding of large sections resulted in deflection, due to the heat generated at the wheel contact point (even with coolant), and resulted in damaged areas with some fibers exposed. Inspection was required.

2.2.4 Static Test Results for Metal Matrix Composites

2.2.4.1 Baseline Data

Baseline data were generated for both 6061 Aluminum/boron and 6Al-4V Titanium/BorSiC Composites for both 0° and 90° properties, in tension and compression at various temperatures: 70°F, 160°F, 400°F, 600°F and at 800°F for the Titanium/BorSiC. These results are presented in Appendices IV and V. Both tabularized data on strengths and moduli and stress-strain curves are shown there.

2.2.4.2 <u>Effects of Thermal Conditioning on the Static</u> Properties of Metal Matrix Composites

Both steady-state and cyclic thermal conditioning treatments were applied to the two metal matrix composites. The effects of these conditioning treatments are summarized in Figures 119 - 126. In general, the steady state treatments appeared to have a mixed effect on the tensile strengths of the two composites, while the cyclic thermal effects appeared to cause a general degradation of the tensile strengths. Both

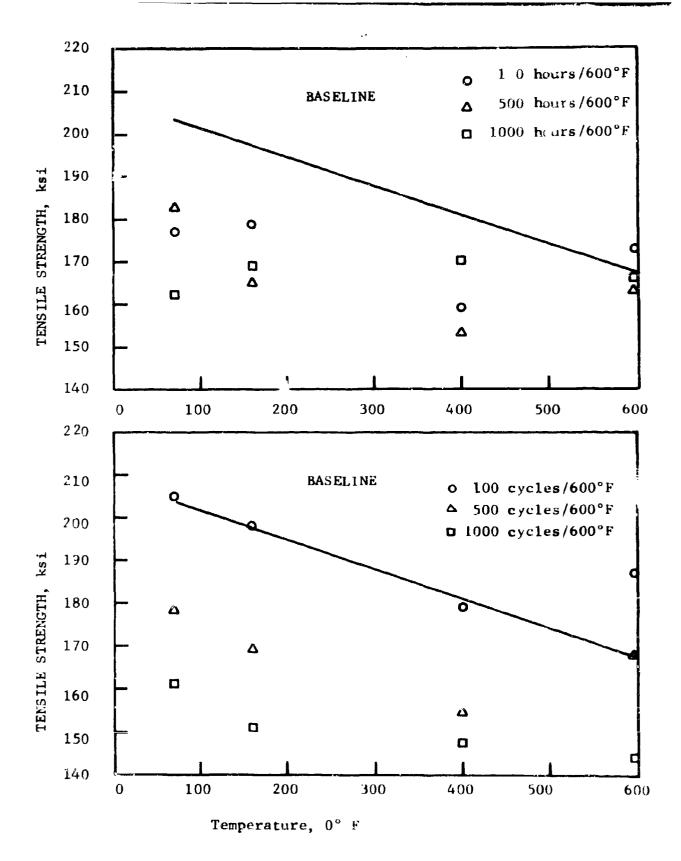


Fig. 119 L. FECT OF PRIOR THERMAL CONDITIONING ON THE TENSILE PROPERTIES OF BORON/6061 ALUMINUM COMPOSITES - 0°.
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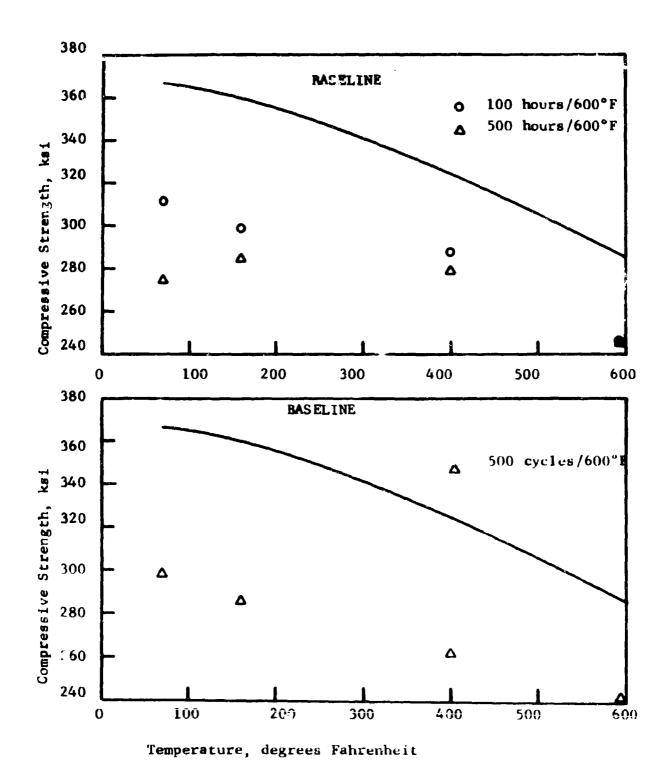
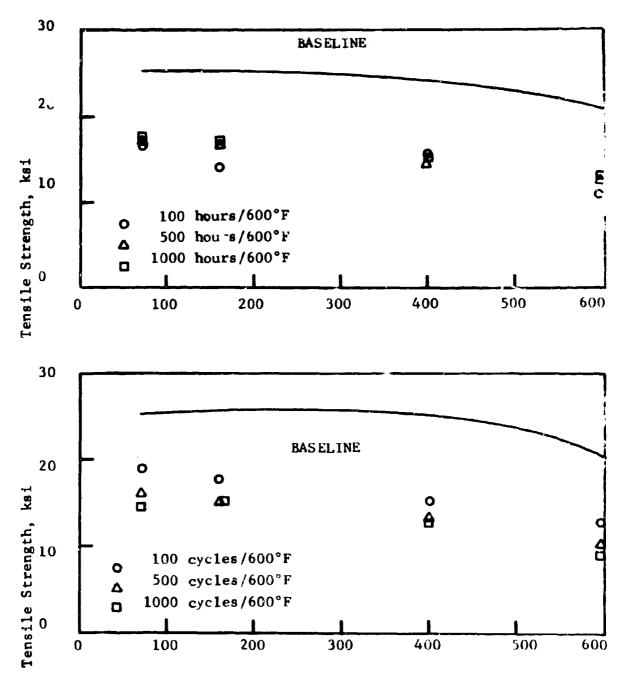


Fig. 120 EFFECT OF PRIOR THERMAL CONDITIONING ON THE COMPRESSIVE PROPERTIES OF BORON/6061
ALUMINUM COMPOSITES 0.



Temperature, degrees Fahrenbeit

Fig. 121 EFFECT OF PRIOR THERMAL CONDITIONING ON THE TENSILE STRENGTHS OF 6061 ALUMINUM/BORON COMPOSITES 90°.

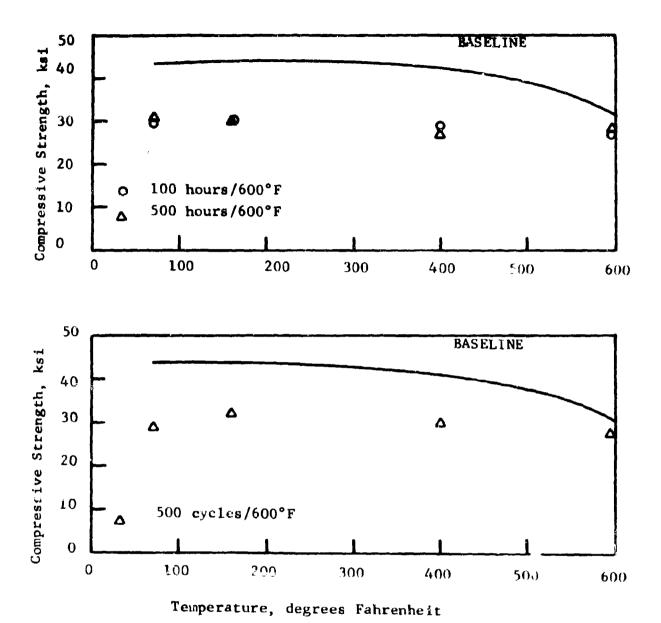
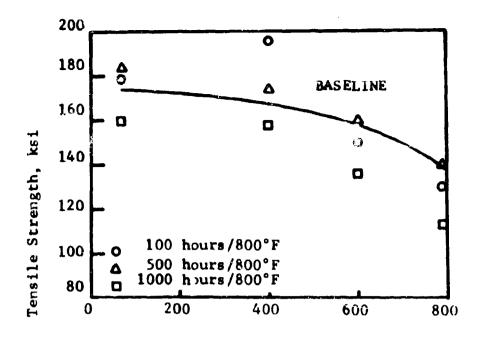


Fig. 1.22 EFFECTS OF PRIOR THERMAL CONDITIONING ON THE COMPRESSIVE STRENGTHS OF 6061 / LUMINUM/BORON COMPOSITES 90°.



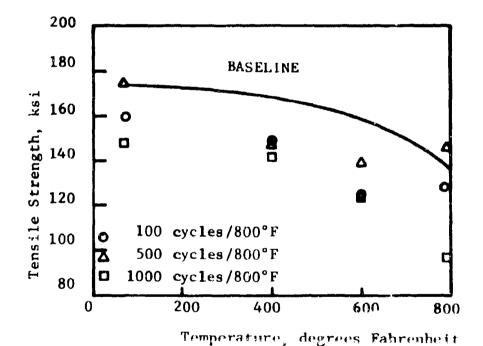


Fig. 123 EFFECT OF PRIOR THERMAL CONDITIONING ON THE TENSILE PROPERTIES OF 6A1-4V TITANIUM/BORSIC COMPOSITES 0°.

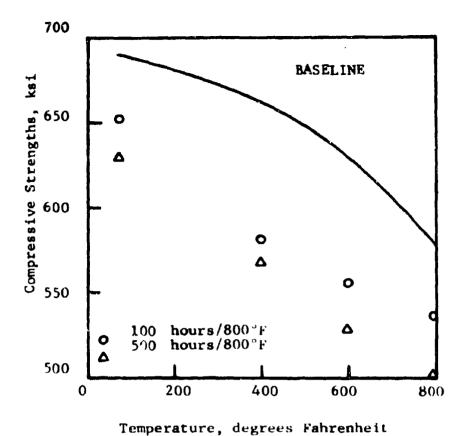
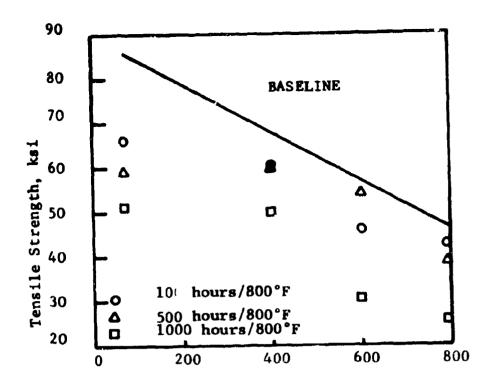


Fig. 124 EFFECT OF PRIOR THERMAL CONDITIONING ON THE COMPRESSIVE STRENGTHS OF 6A1-4V-TITANIUM/BORSIC COMPOSITES O°.



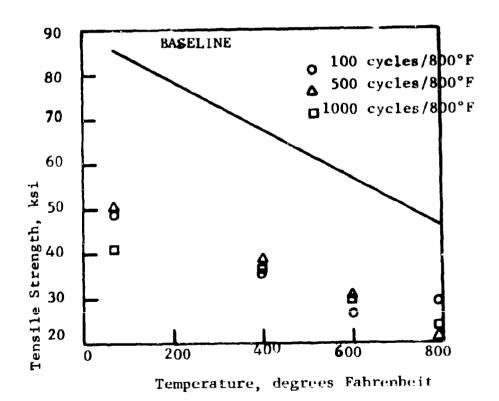
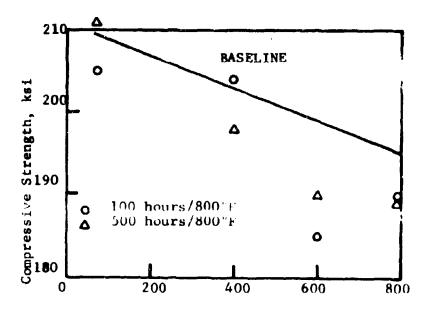


Fig. 125 EFFECT OF PRIOR THERMAL CONDITIONING ON THE TENSIVE STRENGTHS OF 6A1-4V TITANIUM/BOR-SIC COMPOSITE 90°.



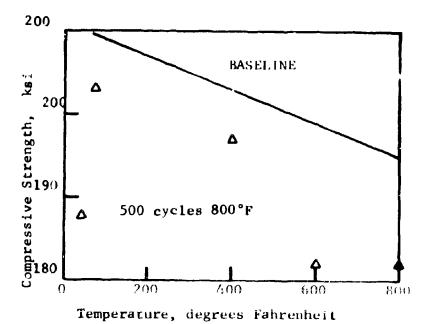


Fig. 126 EFFECT OF PRIOR THERMAL CONDITIONING ON THE COMPRESSIVE STRENGTHS OF 6A1-4V-TITANIUM/BORSIC COMPOSITES: 90°

conditioning treatments appeared to cause a degradation of the compressive strengths.

2.2.5 Fatigue Test Results

Both metal matrix composites were subjected to fatigue at various R ratios (R - 0.1, -1, and 10) at a cyclic frequency $\Phi = 1800$ rpm. No conditioning was applied to the metal matrix composites but the materials were tested at temperatures of 70°F, 160°F, 400°F, 600°F and at 800°F for the 6A1-4V-Titanium/BorSiC Composites. Fatigue in both the longitudinal (0°) and transverse (90°) directions was studied.

The results are presented in Appendices IV and V and in Figs. 551 to 556 and 588 to 593.

The 6061 Aluminum/Boron tensile fatigue (R-0.1) results fell within a very narrow band for all temperatures, with only a slight degradation in strength with increasing temperature of the 0° composite. A much wider spread in the S-N curves was shown for the fully-reversed loading (R-1) and considerable scatter in the compression fatigue (R-1.0) results of the 0° composites. Similar results were evidenced for the 90° 6061 aluminum/boron composites.

The 6A1-4V - Titanium/BorSiC fatigue results show a greater reduction in strength with temperature and a greater scatter in the fully reversed and compression fatigue data for all temperatures and both longitudinal and transverse load-carrying capacities.

2.2.6 Creep And Stress Rupture Results

Both metal matrix composites were also subjected to long term tensile stress-rupture and creep testing. Both 0° and 90° composites were tested in creep at 70°F, 160°F, 400°F, 600°F

and 800°F. The results are presented in Appendices IV and V in tabular form, creep curves and stress versus time to rupture curves.

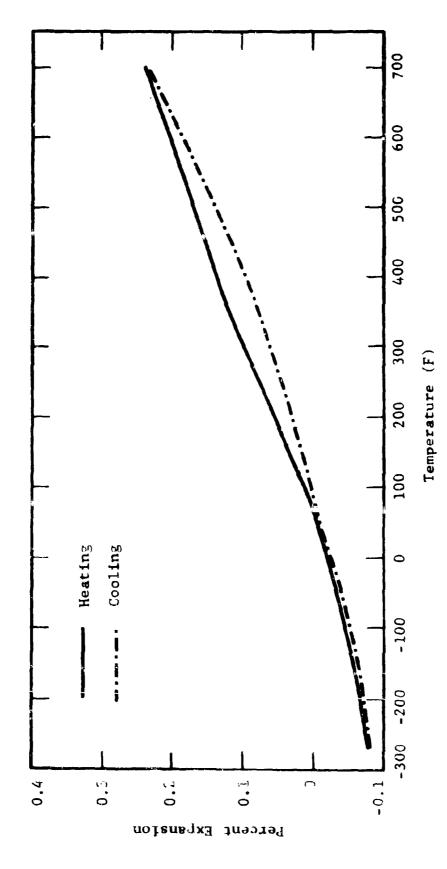
The test results indicate considerable scatter for the 6061 aluminum/boron composites and vary little useful data for the 6Al-4V titanium matrix composites. The 6061 aluminum/boron stress versus time to rupture curves are extremely flat similar to the fatigue S-N curves. In addition it should be noted that the transverse creep strain versus time curves for the 6061 aluminum/boron composites showed a tendency to increase in growth rate at elevated temperatures similar to the familiar aluminum base metal creep curve performance. (The room temperature strains did not increase as quickly as those at the elevated temperatures). The 0° creep strain versus time curves were quite flat out to 1000 hours.

2.2.7 Physical and Thermophysical Properties of Metal Matrix Composites

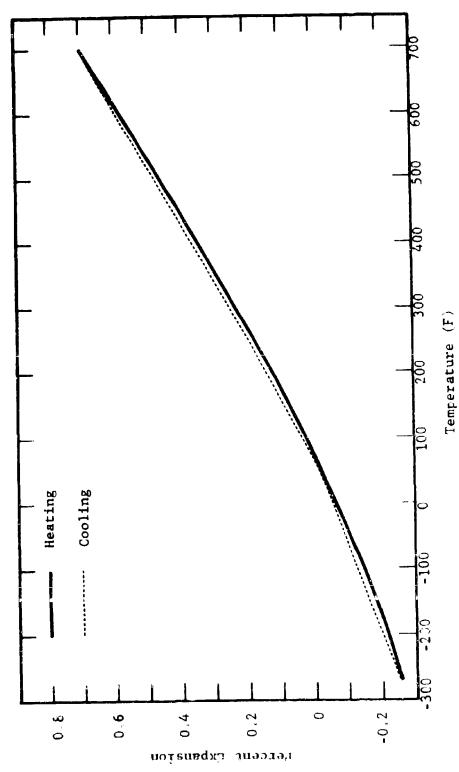
2.2.7.1 Thermal Expansion Test Results

Thermal expansion measurements were made for the two fiber reinforced metal matrix systems, 6061 Aluminum/Boron and 6A1-4V-Titanium/BorSiC. Five samples in each of two fiber orientations (0° and 90° with respect to the expansion direction) were tested in air with the NETZSCH automatic recording pushrod dilatometer described previously. Testing was conducted from -320°F to 700°F for the 6061 Aluminum/Boron material, and from -320°F to 900°F for the 6A1-4V Titanium/BorSiC material.

Typical results for the 6061 Aluminum/Boron and 6A1-4V-Titanium/BorSiC materials are presented in Figs. 127 to 132, where percent expansion is plotted against temperature for both heating and cooling cycles above and below ambient RT. For the



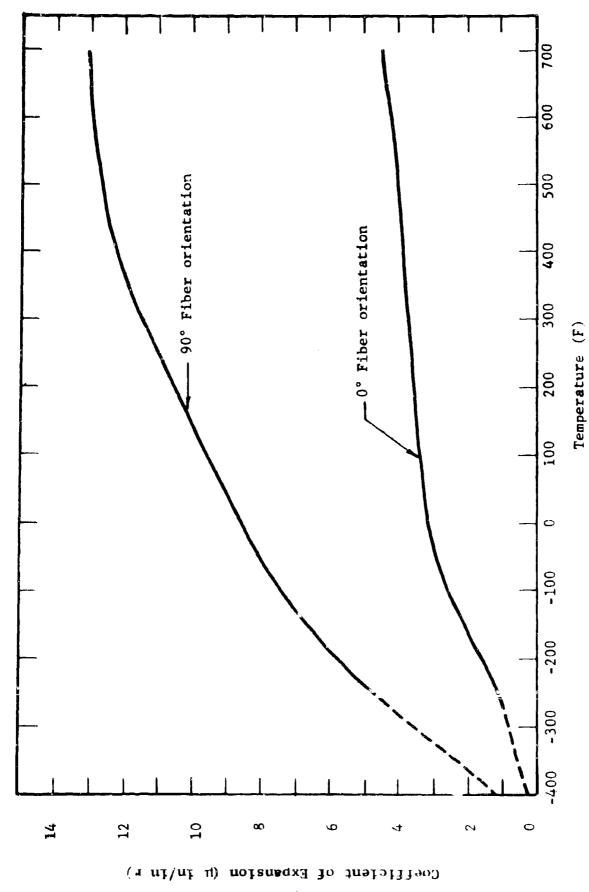
THERMAL EXPANSION BEHAVIOR OF 6061 ALUMINUM/BORON MATERIAL IN THE 0° FIBER ORIENTATION Fig. 127



THERMAL EXPANSION BEHAVIOR OF 6061 ALUMINUM BORON MATERIAL IN THE 90° FIBER ORIENTATION

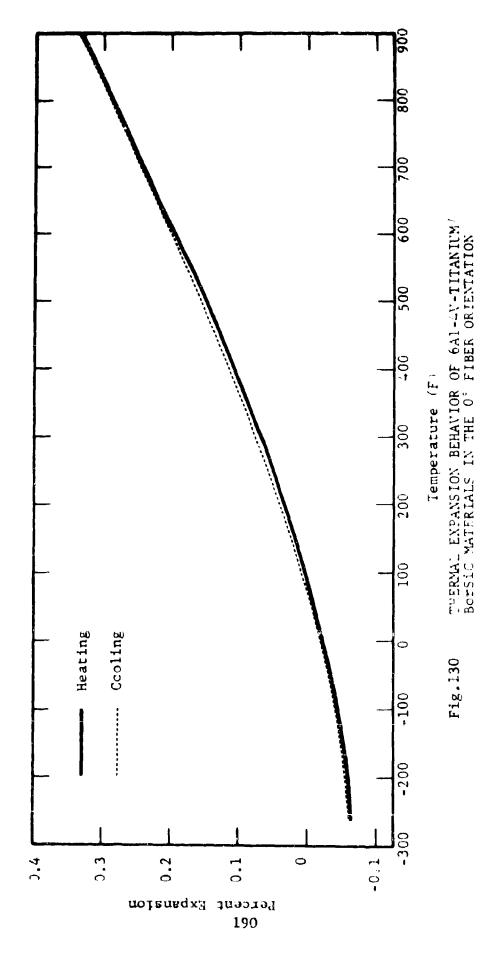
Jig. 128

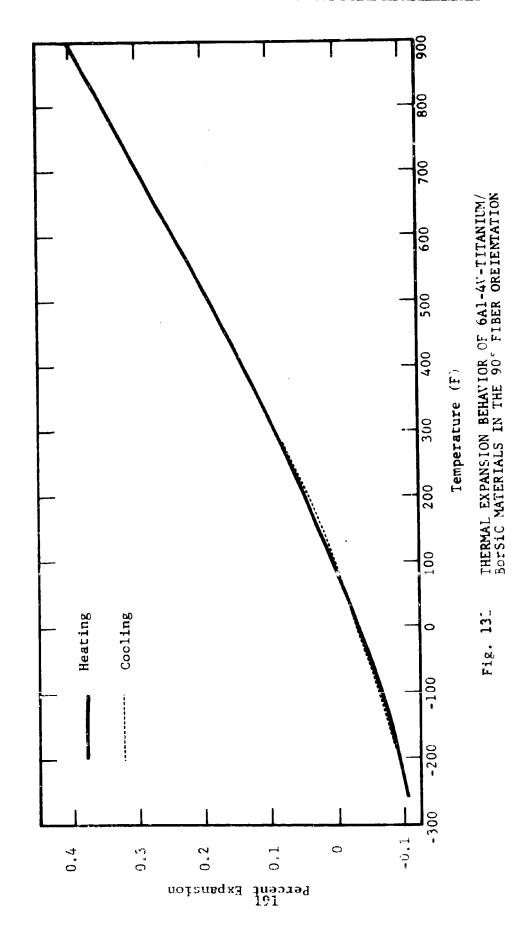
188



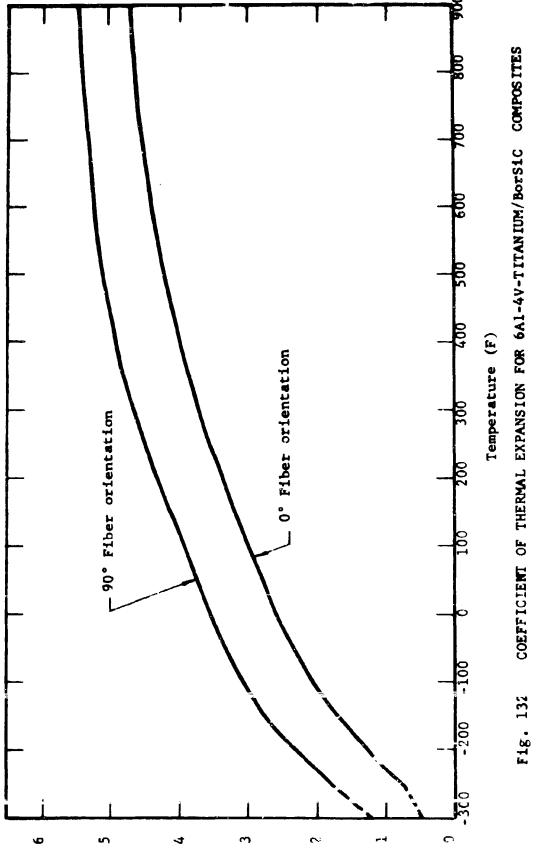
9)

Fig. 129 COEFFICIENT OF THERMAL EXPANSION FOR 6061 ALUMINUM/BORON COMPOSITES

一般の通過の表現を使うなというとは、 これのでは、 これのできない はない はんしゅう これの これのこと




大学の大学をあるというである。



(**4 ur/ur т) цохвиво**мя у**яшха**ны, до энатэт<u>гд</u>вор **192**

6061 Aluminum/Boron system the overall expansion from -320°F to 700°F was 0.3 percent or the 0° direction and 0.95 percent or the 90° direction. Over the temperature range -320°F to 900°F the 6Al-4V Titanium/BorSiC materials exhibited 0.39 percent and 0.5 percent expansion in the longitudinal (0°) and transverse (90°) directions, respectively. For both metal matrix systems tested good sample-to-sample reproducibility was observed, with no unstable expansion behavior such as was obtained for the resin matrix materials.

The instantaneous coefficient of thermal expansion was determined for both metal matrix systems or the longitudinal and transverse directions and is presented as a nunction of temperature in Figs. 131 and 132 and is summarized in Table XI.

In the 0° direction (parallel to fibers) expansion coefficients are low due to fiber reinforcement effects. Expansion data for the uniaxial (0°) 6061 Aluminum/Boron metal matrix material are similar to the resin-matrix AVCO 5505/Boron system previously discussed. However, the boron fibers more strongly control the uniaxial expansion behavior in the resin matrix system. This can be predicted by considering the relative fiber and matrix moduli in both metal - and resin - matrix systems.

The uniaxial (0°) expansion coefficients of the 6A1-4V Titanium/BorSiC material were similar to those observed for the 6001 Aluminum/Boron system. Both materials were boron fiber reinforced which controlled the uniaxial expansion behavior.

The larger expansion coefficients for these metal matrix materials in the transverse (90°) direction represent relatively unrestrained matrix expansion, the boron fibers offering only minimal reinforcement. The transverse 6061 Aluminum/Boron material expansion coefficients are similar to those of the

Table Af

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AVERAGE IN TANTANEOUS (GENERALE) THERMAL EXPANSION DATA FOR METAL MATRIX COMPOSITIES

		COEFFIC	IENT OF	COEFFICIENT OF EXPANSION, uin/in°F	ION, IL	In/ta°F	
TEMPERATUIE, °F	-300*	-100	100	300	200	700	906
Ę		i 1	9			,	
-	68.0	2.55	30.0	3.75	90.	07.7	:
90° orientation	3.55	7.33		9.70 11.50	12.70	13.10	;
		•					
6A1-4V-Titani m/Borsic							
0° orientation	05.0	2.05	2.05 3.00	3.70	4.18	4.55	4.73
90° orienta:ion	1.20	3.05 3.92	3.92	49.4	2.08	5.30	5.47

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*extrapolated

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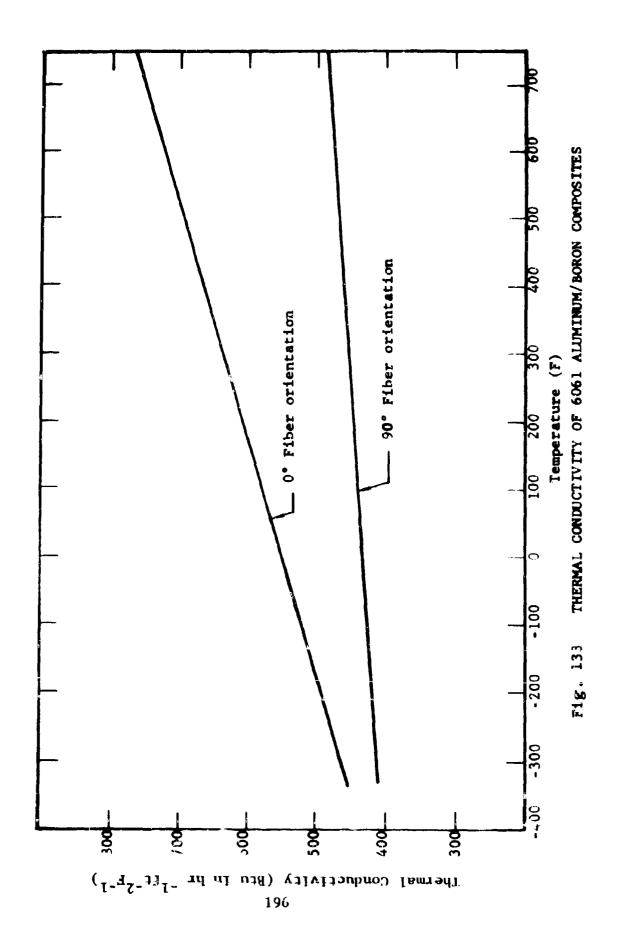
6061 Aluminum matrix ($\sim 13~\mu\text{-in/in}^\circ\text{F}$ at RT), and the transverse expansion coefficients of the 6A1-4V-Titanium/BorSiC material are similar to those of titanium ($\sim 4.7~\mu\text{-in/in}^\circ\text{F}$ at RT). In both materials transverse expansion coefficients were slightly lower than for their respective matrix materials only, due to the contribution of the reinforcing fibers in the transverse direction.

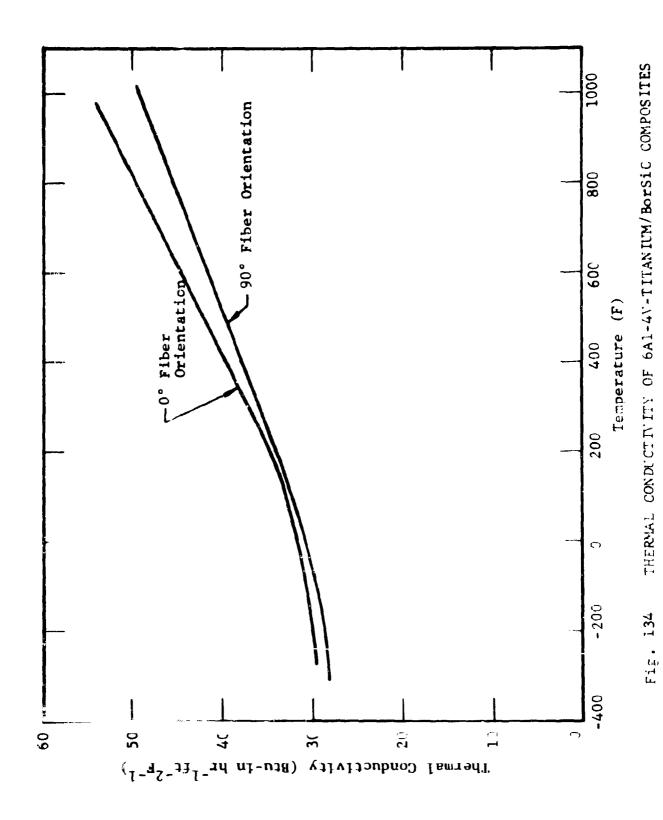
The 6Al-4V titanium/BorSiC material exhibited less thermal expansion anisotropy than the 6061 Aluminum/Boron materials. This occurred presumably because the individual fiber and matrix component expansion coefficients were closer for the 6Al-4V-titanium/BorSiC system.

2.2.7.2 Thermal Conductivity of - Metal Matrix Composites

Thermal conductivity measurements were made on the two metal matrix composites: 6061 Aluminum/Boron and 6Al-4V-Titanium/BorSiC. Five samples in each of two fiber orientations, longitudinal (0°) and transverse (90°), were tested. Testing was conducted to 700°F and 900°F for the aluminum matrix and titanium matrix materials, respectively.

Thermal conductivity results are presented in Figs. 133 and 134, where the rual conductivity is plotted as a function of temperature for both the longitudinal and transverse orientations. The thermal conductivity parallel to the fiber reinforcement was higher than in the transverse (normal to fibers) direction. The straight-line representations shown for each material/orientation are the result of linear least equares data fits. Data scatter for these metal matrix materials was roughly \pm 4 to \pm 12% (maximum variation), with some evidence of sample-to-sample variability.





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The thermal conductivity of the boron reinforced aluminum material was higher than for the boron (coated with Silicon carbide) reinforced titanium matrial due to the higher matrix thermal conductivity. A lower degree of directional anisotropy was observed for the 6A1-4V Titanium/BorSiC materials, which agrees with the thermal expansion results where lower directional anisotropy was also observed for the 6A1-4V Titanium/BorSiC material compared to the 6061 Aluminum/Boron material.

2.2.7.3 Densities

The densities of the metal matrix composites were also determined using gravimetric and leaching processes. These determinations were made to verify the fabricator's stated fiber densities. The average fiber density, from three plates of the 6061 aluminum/boron was 49 %. Similarly the average fiber density, from three determinations of the 6A1-4V-Titanium/BorSiC was 46 %.

2.2.8 Fracture Modes of Metal Matrix Composites

The 0° 6Al-4V-Titanium/BorSiC composites exhibited tensile fracture surfaces which were rough but lay in planes relatively transverse to the load direction. No steplike fractures were evident, as was the case for resin matrix composites, for the baseline data. Similar results were obtained for all levels of steady-state thermal exposure. However those samples which had been cyclically exposed to thermal conditioning showed both delamination and multiple step fractures particularly in the surface plies. The 0° compression failures of all 6Al-4V-Titanium composites were impossible to analyze because of severe crushing, multiple fractures, metal smearing and occasional delamination particularly those tested at elevated temperatures. Interlaminar shear and flexural fail are modes at room temperature

appeared to be as is normally encountered for resin matrix composites.

At room temperature the 90° tension fracture in 6A1-4V-Titanium/BorSiC composites appeared similar for the baseline, steady-state thermal and cyclic thermal conditioning. Failures are transverse to the load direction, generally lie in a plane perpendicular to the load direction. There is often a shear lip on one side of the specimen if there was sufficient metal from the cover plates remaining after grinding of the surfaces. Compression failures of the 90° 6A1-4V-titanium/BorSiC composites were flat and lay in a plane always at 45° to the load direction. At room temperature little or no delamination was evident although metal smearing was present and may have hidden this delamination. At elevated temperatures (above 400°F) the failures were quite different. The fractures surfaces were generally out of plane but more perpendicular to the load direction and included some overall out-of-plane curvature to the gage section This latter phenomenom may indicate buckling of the 90° 6Al-4V-titanium BorSiC composites at elevated temperature due to the combined effects of modulus reduction and some delamination which is partially evident looking at the side of the fractured samples. The prior cyclic thermal conditioning and steady state thermal conditioning did not alter this failure mode.

The 0° 6061 aluminum/boron tension failures at room temperature were closer to those encountered in the corresponding resin matrix tests. The fracture surfaces were very rough, did not lie in a plane and multiple stepping in the fracture patterns was clearly evident. Prior steady state and cyclic thermal conditioning resulted in a higher percentage of failures

close to the ends of the gage section near the tabs. 0° compressive failures of the baseline 6061 aluminum/boron composites at room temperature showed substantial flow present in the gage section, but whether this was encountered prior to failure or just after fracture is not known. Both steady-state and cyclic thermal conditioning caused less flow to be present as judged from post fracture examinations of the broken specimens.

Both baseline and steady state thermal conditioning of the 90° 6061 aluminum/boron composites resulted in fractures exhibiting considerable plastic flow as judged by post-fracture examinations and some curvature to the entire gage section of the sample. However the cyclic thermal conditioning resulted in room temperature compression failures of the 90° 6061 aluminum/boron composites which were generally flat, and at a 45° angle to the load direction. Occasionally some samples showed a double angle meeting at the center plane of the coupon. Tensile failures of the 90° 6061 aluminum/boron composites were similar to the failures of the 6Al-4V-Titanium/BorSiC composites except that many of the specimens exhibited a short (about 10 fiber diameters) steps in the failure surface.

The 6061 aluminum/boron composite fatigue failures were varied. Tensile fatigue (R = 0.1) of the 0° coupons appeared to follow the static fracture patterns. The compression fatigue (R = 1.0) failures, however, generally were different from the static tests with a short (approximately, 1/16 inch) segment of the 0° coupon broken away from the sample, thus resulting in three pieces after failure. Compression fatigue failure surfaces were normal to the load direction. The fully reversed 6061 aluminum/boron \cup ° composites frequently fractured into

Four or more pieces with the failure surfaces more like the tensile fatigue, i.e. irregular with some steplike appearance. The 90° 6061 aluminum/boron tensile fatigue (R = 0.1) failures contained some fiber failures resulting in a small step of a few fiber diameters on each failure surface. The compressive fatigue failures of the 90° composites frequently resulted in several post fracture pieces. Fully reversed failures were a mixture of these two modes.

The 0° 6Al-4V-titanium/BorSiC composites tested in tensile fatigue showed irregular fracture surfaces but a continuous non-stepped fracture path. The 0° compressive specimens invariably failed at the ends of gage section and considerable damage of the fracture surface took place after the initial fatigue failure. The 0° fully-reversed fractures more closely resembled the tensile fatigue failures. The 90° tensile fatigue failures were flat, planar and perpendicular to the load direction. The compression fatigue fracture surfaces of the 90° 6Al-4V-titanium composites were flat, planar and at 45° to the load direction.

SECTION III

3.0 CONCLUSIONS

In conclusion this program has demonstrated the capabilities of composites in retaining mechanical properties after exposure to various humidity and thermal environments. This data appears at a particularly appropriate time in the evolution of composite technology, when emphasis is being placed on composite reliability and durability. The effectiveness of composites in resisting environmental degradation has been demonstrated.

Several items of particular concern to aerospace desimers and test engineers planning to utilize these materials in preliminary or advanced design were established during this program. These items can be summarized as follows:

The boron/epoxy system was particularly sensitive to moisture conditioning and moisture coupled with high/low temperature shocks. The boron/epoxy strengths were affected to a greater degree than were the moduli. Overall the results showed that the properties of the composite which depend largely on the resin constituent properties were affected the greatest. These included interlaminar shear strength, transverse strength and modulus and compressive strengths (the latter would appear to be a result of resin softening by plasticization which increases the tendency toward microbuckling). A somewhat similar properties loss in the transverse compressive strength was also observed for humidity coupled with ultraviolet radiation (accelerated weathering). Long term (high cyclic level) fatigue performance also was affected deleteriously by humidity coupled with thermal sbocks.

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The high-strength graphite/epoxy system were affected substantially by moisture and moisture/thermal shock conditioning. Again interlaminar shear strength, transverse strength and modulus and compressive strengths all properties sensitive to changes in the resin constituent properties, were affected the greatest.

The high-modulus graphite/epoxy exhibited less deleterious response to moisture than did the boron/epoxy or high strength graphite/epoxy systems. In fact, the presence of residual fabrication stresses in the high-modulus graphite/epoxy system led to some enhancement of the strengths from prior humidity conditioning. However the combined humidity/thermal conditioning radically affected the high-modulus graphite/epoxy transverse strength (losses up to 75%) and fatigue behaviors.

The metal matrix composites exhibited very improved transverse and compressive strength properties over the resin matrix composites. Losses in fatigue strength due to prior thermal conditioning were primarily confined to the transverse direction and were worse for cyclic (thermal) rather than steady-state conditioning.

Upon the introduction of these composite materials into an aerospace component design, the test engineer could obtain a rapid reading on the feasibility of their utilization by an examination of the above properties.

Certain portions of this program have led to other new questions such as the importance and characterization of residual fabrication stresses particularly in the graphite/epoxy systems.

The role of moisture in degrading (or enhancing) the mechanical properties of resin matrix composites is not entirely understood. The complimentary roles of simultaneous heat-cold cycles and ultraviolet are also vague. Although the data appear consistent, the frequently confusing nature of the qualitative and quantitative response implies that, to fully exploit these composites, further study of the fundamental nature of these causative factors would be in order.

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APPENDIX I

DATA SUMMARY FOR AVCO 5505/BORON COMPOSITES

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APPENDIX I

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7	Figs. 242 to 269 Creep Strain Versus Time Curves	271-278

TABLE XII STATIC PROPERTIES SURBARY - A LCO 5505/BURON

Orient at lor	Typo Load	Prior Conditioning	Test Temp. (*p)	E (psi x 10 ⁶)	, (1a/1a)	°alt (ksi)	ευlτ (μ-10. 'tm.)	
0.	Tersion	None	£	29.6	0.23	183	6423	1
•	Teru ion	Mone	260°F	28.6	0.21	181	6383	
•	Tersion	Bone	350°F	28.5	0.22	111	6230	
• 06	Tersion	None	£	2.69	3 .0	8.85	3900	
	Ten sion	None	260°F	1.84	0.03	6.56	2€87	
• 0\$	Tension	Mone	350°F	07.1	0.03	5.32	2.87	
(0/45/135/0/30) ₆	Tension	None	Ē	14.9	0.45	85.1	6280	
[0/45/135/0/30]	Terse i on	Note	260°F	13.9	0.42	82.6	96190	
[0/45/13: /0 96	Tenston	None	350°F	14.1	77.0	78.9	6190	
•	Compress ton	None	· CE	31.5	67.0	362	11,270	
•	Compression	None	Ē	26.6	0.20	18	96.30	
•	Compression	Morre	260°F	27.1	0.30	172	0009	
•0	Compression	Hors	350°F*	29.4	0.52	265	9530	
•0	Compression	None	350 F	7.62	0.17	126	4170	
• 06	Compression	None	Ē	3.55	0.03	29.3	9620	
*	Compress i on	None	Ē	3.89	0.0	35.5	13,620	
•	Compression	Mone	260°F	1.77	0.0	26.8	21,530	
•	Compression	None	350°F*	4.05	00.0	23.5	9959	
•06	Compression	Mone	350°F*	2.15	0.0	19.9	16,300	

Sanderich Blas Beta

TABLE XII STATIC PROPERTY SUMMARY - ALCO

\$505 BURON (COUL' d)

Orient(:ion	Type Load	Prior Confitioning	Test Temp. (*F)	E (psf x 10 ⁶)	, (1n/1n)	dult (kai)	ult (u-1n./in.)
0/45/11:5/6/90	Compression	None	RTD*	18.2	3.°0	236	13,350
06/0/5/17970	Coerression	None	•	13.5	94.0	H.7.6s	14,280
0/43/103/0/90	Compression	Kone	260°F	13.9	0.45	164	13,430
06/0/5/17/57/0	Compression	.V.	350°F*	16.0	0.52	183	11,990
. <u>06</u> /0/5:1/57/0]	Compression	9607	350°F*	13.9	77.0	151	10,790
. 0	In-Plane Shear	None	£	0.84	1	9.7	26,000
0	In-Plane Shear	None	260⁴F	0.50	•	7.1	32,000
0	In-Plane Shear	Mone	350⁴F	0.25	•	6.4	17,000
. 0	Int. Shear	None	RTD	·	•	15.2	1
: 0	Int. Shear	None:	2 60°F	•	•	12.0	•
. 0	int . Shear	Youe	350°F	•	•	9.1	•
0/45/135/0/990	Int. Shear	None	e t	į	•	10.9	•
[0/45/135/0/ 90]	Int. Shear	None	260°F	•	•	7.9	•
.06/0/5ET/57/0]	Int. Shear	None	350°F	•	•	6.4	•
•	Flex	None	E	•	•	263	•
•0	Flex	None	260°F	•	·	240	•
•	Fla	None	350°F	•	•	218	•
,06	Flex	None	Ē	•	•	14.0	•
.06	Flex	- Kone	260°F	•	•	12.5	•
• 06	Flex	None	350°F	•	,	9.5	•
[0/45/135/0/90]	Flex	Hone	E	•	•	107	1
(0/45/135/0/90)	Flex	None	260°₽	•	,	101	•
06/08/138/0/80	Flex	Kone	350%	•	•	93	•

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P. E. V.L. STALE MOSERTES

			-					ı
hrientation	T, pe Load	Prior Conditioning	Test Temp. (*F)	E (ps1 x 10 ⁶ ,	, (16/10)	gult (kaf)	^ε υ1ε (μ-in./in.)	i i
•0	Tension	98% KH /500 Krs.	£13	28.7	0.20	147	5670	1
•0	Tension	98% RH/500 Hrs.	260°F	•	•	177	5670	
• 0	Tension	98% RM/500 Hrs.	350°F	•	•	131	•	
• 0	Tension	987 RH /1000 Hrs.	RTD	29.4	0.21	153	2960	
• •	Tension	98% RH /1000 Hrs.	2 60 'F	30.3	0.21	163	5540	
• 0	Tension	952 RH /1000 Hrs.	350°F	•	•	ı	•	
•0	Tension	Thermo-Humidity Cycle	RTD	29.6	0.17	981	6360	
.0	Tension	Thermo-Humidity Cycle	266 °F	•	ı	158	•	
•0	Tension	Thermo-Humidity Cycle	350°F	,	•	120	•	
•0	Tension	Acc. Wthrmg.	OT.N	9.5.	0.17	190	6710	
• 0	Tension	Acc. Wthrng.	260°F	36.7	0.27	157	9400	
•0	Tension	Acc. Wthrng.	350°F	30.1	0.22	146	C#87	
, 06	Tension	987 RH-500 Hrs.	CTA	2.58	0.01	7.8	3330	
. 3	Tenston	98% RH/500 Hrs.	260°F	•	•	5.6	•	
• 06	Tension	98% PH/500 Hrs.	350°F	4	•	4.4	•	
.06	Tension	987 RH/1000 Hrs.	GT X	2.55	0.02	1.6	3430	
• 06	Tension	98% RH /1000 Hrs.	260°F	1.56	0.01	4.9	4830	
•06	Tenston	987 RH/1000 Nrs.	350°F	1.06	00.00	4.4	6750	
.06	Tension	Thermo-Humidity Cycle	£	2.18	0.0	6.9	3720	
.06	Tension	Thermo-Humidity Cycle	4.09₹	•	1	5.4	•	
• 06	Tension	Thermo-Humidity Cycle	3 50°F	•	•	3.6	•	

ARUJ. ATT. STATTC PROPURETURS STAMBY * AVEO \$505/RORON COMPOSTTES (GARL'd)

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Orientat fon	Two Load	Sujacyty, we act in	(ab)	8 (pst × 10 ⁶)	411.7	'ult (231)	fult (n-ta./fu.)
8 90	Tension	Acc. kthrng	RTD	66.7	0.00	7.8	3480
06	Tension	Acc. Wthrng	760°F	1.32	0.02	4.0	2980
06	Tension	Acc. Wthrng	350°F	0.51	0.01	2.9	6340
. <u>06</u> /0/ <u>32</u> /0/ <u>57</u> 0.	Tension	98% KH/500 Hrs.	ктр	16.0	0.43	8.4	0909
0/45/135/0/90	Tension	98% RH/500 Hrs.	7.097	•	1	•	•
\$	Tension	98% RH/500 Hrs.	350°F	•	1	69	•
S = 1/2/25/1/57/0	Tension	98% RH/1000 Hrs.	RTD	15.4	0.44	98	0209
0/45/135/0/90's	Tension	98% RH/1000 Hrs.	260°F	15.7	97.0	7.1	5820
0/43/133/0/30 s	Tension	98% RH/1000 Hrs.	350°F	14.1	0.38	7.7	5920
3/45/135/0/60	Tension	Thermo-Humidity Cycle	RTD	15.2	09.0	7.2	5070
06/0/51/57/0	Tension	Thermo-Humidity Cycle	260°F	•	•	81	•
s	Tension	Thermo-Humidity Cycle	350°F	•	•	68	•
0/42/135/0/ <u>90</u>	Tension	Acc. Wthrng	RTD	16.3	0.42	88	5830
\$ 06/0/51/57/0	Tension	Acc. Wthrng	260°F	16.1	0.43	74	2020
10/45/135/0/90	Tension	Acc. Wthrng	350°F	15.4	0.56	89	0867

€;

Orientalion	property (US)	Majuaj # Pruoj.			(u, u)	² :1 c (×84)	*ult (1-fn./fm.)
0	Graphession	LY KH 100 HTS.		e e e	0.23	60 i	0768
0,	unissend of	y and aboutes.	1 09-	•	•	150	•
ç Q	C. Tression	43' KH/500 Hrs.	350°F	•	•	159	•
ناد	30 mpression	937 RH/1000 Hrs.	RID	25.7	0.17	186	8290
۽ (Compression	437 RH /1000 Hrs.	260° F	5.95	0.25	129	0977
U °	G. pression	937 RH/1000 Hrs.	150°F	2.5	0.14	106	7 260
0.	hapression	Thermo-Humidity Cycle	r di	25.8	0.15	207	8110
.0	Compression	Thermo-Mumidity Cycle	1.00°F	•	•	119	•
9.6	Scapression	Thermo-Humidity Cycle	35075	•	•	93	
J.	forpression	Acc. Wthmg	нТр	30.2	0.24	179	06.29
3-	Ct. spression	Acc. Wthrng	260 ℃	Å.Ö.	0.28	123	7200
, O	Compression	Acc. Wthmg	.50°₽	28.2	0.15	7.7	2560
90\$	Cc pression	93% RH /500 Hrs.	M. D	2.39	0.0	33.2	17,580
, C c	Compression	937 RH /500 Hrs.	250°F	•	•	18.9	•
433	Compression	937 RN /500 Hrs.	350°F	•	•	15.0	•
• 06	Compression	93% RH /1000 Hrs.	RTD	•	•	28.5	•
937	Compression	930 PH /1000 Hrs.	3€0•F	1.75	0.0	17.9	21,980
93*	Compression	9 2" RH/1000 Hrs.	350°F	79 .0	0.0	8.9	16,840
90.	Compression	Thermo-Humidity Cycle	RTD	2.33	0.01	30.4	17,630
90°	Ccmpression	Thermo-Humidity Cycle	260°₽	•	•	18.2	•
.06			3.US?	•	,	71 4	•

Silatation	Te . L	Prior Conditioning	[est iv n. (*F)	E E C C S S S S S S S S S S S S S S S S	· (in in	"ult (ket)	"ult (x-in.'(n.)
, 06	(Chress())	Acc. withrng	919	2.45	0.0	53.3	15,340
, Or	CC_bress: 53	Acc. Wthrnk	J. 097	1.53	0	20.3	21,620
, 0¢	G. Tression	Acc. Wthrng	3×0×8	6,43	0.0	10.5	19,650
<u> 36,0/581/54 0.</u>	ucijssnad. "J	98 RH:500 Hrs.	G.	14.3	0.10	69:	13,160
	thissaid_n0	98" KF /500 Hrs.	J 04.	ı	•	9.8	•
0 45 135/0/4C	Gerpression	98 IIB 30C Hz :	350°F	,	•	9,	•
	C: _pre-ston	98' an 1000 Prs.	RID	;	0.41	157	9470
	C. pression	98C 2H/1000 Ers.	26078	1	0.54	98	6250
	Compression	98' RE 1000 Brs.	3,05 f	,	6.57	62	0009
20/0/581/57.0.	ucissaud_22	Thermo-Humfdity Ocle	5 T	14.8	0.50	169	10,360
	Compression	Thermo-Humidity Cycle	260°F	r	1	92	•
\$ 35/0/5ET sh 0.	Compression	Therrac-Hunfdity cycle	350°F	ı	•	78	•
.0/22/132/0/ <u>9C</u>	(c-pression	Acc. Wthrng	RTD	7.51	0.50	171	13,160
0 +5 135/0/9C	Compression	Acc. Sthrng	260'F	13.3	0.57	100	7360
<u>36</u> /0/581,55 0,	Commession	Acc. Wthrng	350°F	13.0	0.51	92	6320
,	Ir-Plane Shear	98% RH7500 Hrs.	RTD	98.0	•	6.9	30,000
ő	Ir-Plane Shear	98% RH/500 Hrs.	260°F	•	•	† 5	
ຳດີ	Ir-Plane Shear	98" RH /500 Hrs.	350°F		•	4.5	•
0	Ir-Plane Shear	98° RH/1000 Hrs.	RTD	97.0	•	9.5	30,000
ပံ	Ir-Plane Shear	987, RH/1000 Hrs.	260°F	17.0	ı	6.0	30,000
°	Ir-Plane Shear	987 RH /1000 HTS.	350 °F	0.28	•	5.2	30,000

ANTE ALL STATIC PROPERTIES SIMPLEMENT - AVOD FOLLOWING

Orientation	Type Load	Prior Condictoning	Test Temp. (*F)	E (pst x 10 ⁶)	(in/in)	dult (ksf)	*ult 'u-in./in.)
,0	In-Plane Shear	In-Pisne Shear Thermo-Munidity Cycle	E	0.74	•	9.1	30,M0
•	In-Plane Shear	In-Plane Shear Thermo-Bunidity Cycle	260°F	•	•	5.1	•
• 0	In-P. une Shear	In-F. une Shear Thermo-Burnidity Cycle	350°F	•	•	3.8	•
• 0	In-Plane Shear	Acc. Wihmig	6.7.3	6.72	•	7.6	30,000
•0	In-Plane Shear	Acc. Wthrng	260°F	0.25	•	5.0	30,000
• 0	In-Plane Shear	Acc. Withrag	350°F	0.11	•	3.6	30,000
0	Inc. Shear	967 EH/500 Brs.	£	•	•	14.6	•
•0	Inc. Shear	967 EH 500 Nrs.	4.092	•	•	9.5	•
•	Int. Shear	967 EH/500 Hrs.	350°F		•	9 .0	•
•	Int. smear	96% MH 1000 Nrs.	E	•	•	14.3	1
• 0	Int. Shear	987 BH '1000 Hrs.	260°F	•	ı	9.1	•
•	Int. Shear	96% EH/1000 NES.	330°F	•	•	5.9	•
•0	Int. Shear	Thermo-Bunidity Cycle	£	,	•	12.0	•
• 0	Int. Shear	Thermo-Bunddity Cycle	260°F	•	•	9.2	•
• 0	Int. Shear	Thermo-Bunddity Cycle	350°F	•	•	6.1	Ī
•	Int. Shear	Acc. Wthrag	Ę	•	•	12.5	•
•0	Int. Shear	Acc. Wehring	260 * F		•	9.5	•
•	Int. Shear	Acc. Wthrag	350*7		•	5.9	•

TABLE XII STATIC PROPERTIES

Oriestatica	Type Loud	Prior Conditioning	Tent Tem. (*p)	E (pei x 10 ⁶)	, (1a/1a)	gult (ks1)	*ult (4-fn./in.)
.0	Tension	260°F.160 Nrs.	C.L.			182	,
0.	Tension	260°F/100 Hrs.	260°F	•	•	152	ı
• 0	Tension	260 F. 500 Mrs.	£	29.4	0.18	188	6530
, 0	Tension	260°F 500 Hrs.	260°F	28.5	0.19	169	6210
٥٥	Tension	350°F/100 Hrs.	Ē	•	•	180	•
٥	Tension	350°F/100 Hrs.	260°F	29.0	0.16	177	6180
•0	Tension	350°F/100 Hrs.	350.8	30.2	0.13	791	2000
•0	Tension	350°F '500 Hrs.	£	29.3	0.19	190	0499
•0	Tension	350°F /500 Hrs.	350°F	31.2	0.13	174	0409
•0€	Tension	250°F /100 Hrs.	Ē	,	•	0.8	•
•0€	Tension	260°F/100 Hrs.	260°F	•	•	7.2	•
.0€	Tenston	260°F/500 Hrs.	E	2.75	0.0	6.7	3440
• Of	Tension	260°F/500 Hrs.	₹ 09 ₹	3.1	0.0	7.1	4330
• O.E	Tension	350°F/100 Hrs.	E	•	•	7.6	•
•06	Tension	350°F (100 Hrs.	260⁴₹	ı	•	6.7	•
.0€	Tension	350 F/100 Hrs.	3.060	•	,	6.9	•

TABLE XII STATIC LANGURITES SUMMARY - AVCO 5305/PARKN

Orientation	Type Load	Prior Conditioning	Test 70p.	E (psi x 10 ⁶)	, (10/18)	oult (ks1)	υlt (μ-in./in.)
, 26	Tension	350°F/500 Hrs.	RTD	3.21	0.0	1.3	2560
. 06	Tension	350*F/500 Hrs.	350°F	1.44	0.01	5.6	2110
05/0/51/35/0/	Tension	260°F/100 Hrs.	RTD	•	•	*	•
(٥/45/135/٥ بقت) _	Tension	260 F/100 Hrs.	260 € ₹	•	,	19	•
(0/45/135/0/90)	Tension	260"F/500 Hrs.	RTD	16.5	0.45	ž	5770
[048/138/0/80]	Tension	260 "F/500 Hrs.	260°F	15.2	0.48	1	0009
06/0/51/32/0/30	Tension	350°F/100 Hrs.	CIX	i	•	7.5	•
[0/45/135/6/90]	Tenston	350 F/100 Hrs.	260°F		,	23	
[0/6/2/135/0/90]	Tension	350*F/109 Hrs.	350*F	•	•	62	•
(0/45/135/0/90)	Tenstor.	350*F/500 Hrs.	RŦD	15.1	0.43	62	5820
[0/45/135/0/90]	Tensior.	350°F.500 Hzs.	350°F	16.1	0.43	18	2660
	Compression	260 F, 100 Hrs.	QT.	,	1	207	•
•0	Compression	260 F, 100 Hrs.	260°F	•	•	163	ì
•0	Compression	26C*F/500 Hrs.	E	27.8	0.20	213	0798
	Compression	260°F/500 Hrs.	260°F	29.3	•	195	6710
•	Compression	350 F. 100 Hrm.	e	27.0	0.18	238	9190
•0	Compression	350°F'100 Hrs.	350*F	•	•	149	•
•0	Compression	350 F / 500 Hrs.	e E	30.6	0.20	232	8390
•0	Compression	350 F/500 Hrs.	350°F	26.9	0.24	172	7110

TABLE XII STATIC PROPERTIES SUPPORT - AVCO

Orientation	Daor acki	Pricr Conditioning	Test Temp. (*F)	E (ps1 x 10 ⁶)	, (11/11)	dule (kai)	*ulc (p-40./10.)
.06	Corpression	260°F/109 Hrs.	7. T.T.	•	•	33.2	•
,06	Compression	260°F/100 Hrs.	260°F	•	•	23.7	1
60 م	Compression	260°F/500 Hrs.	E E	2.57	0.0	37.2	16,830
8	Compression	260°F/500 Hrs.	350°F	2.40	0.01	25.2	19,350
. 26	Compression	350°F/100 Hrs.	£	ł	1	33.6	ı
, 26	Compression	350°F/100 Hrs.	350°F	ı	•	18.1	1
ື ງ6	Compression	350°F/500 Mrs.	RTD	2.50	0.0	29.3	13,550
ູ ງ6	Compression	350°F/500 Hrs.	350°F	2.05	0.0	17.9	17,370
.c/45/135,0/90.	Compression	260°F/100 Hrs.	£	ı	•	194	•
106/0/132/0/ <u>30</u>	Compression	260°F/100 Hrs.	260°F	•	1	137	•
E/45/135, 3/90]	Compression	260°F/500 Hrs.	E	13.3	0.51	186	12,760
3.06/0, SE1/57/0	Compression	260°F/500 Hrs.	260°F	13.6	0.51	147	11,390
[0/45/135/0/ 30]	Compression	350°F/100 Hrs.	£		•	190	•
[0 <u>6</u> /0, set/sh/o]	Compression	350"F/100 Hrs.	350⁴F	1	1	93	ı
*(<u>04</u> 6/0,'\$E1/\$7/0)	Compression	350 °F /500 Hrs.	E	14.1	97.0	161	13,850
(06) 0/ \$2 // \$7/ U	To be a second	350°E 600 Hrs	4.05E	12.9	0.38	128	0696

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TABLE XII STATIC PROPERTIES SUPPARY - AVCO 5505/BORUN COMPCITES (Cont'd)

Orientation	Type Load	Prior Conditioning	Test Temp. (*F)	E (ps1 x 10 ⁶)	, (1n/1n)	^d ult (ks1)	^e ult (μ-fn./fn.)
0.	In-Plane Shear	260°F/100 Hrs.	RTD	,		6.6	•
.0	In-Plane Shear	260 F/100 Hts.	260°F	t	ı	7.1	•
٥٥	In-Plane Shear	260°F/550 Hrs.	RTD	77.0	•	8.6	24,000
٥.	In-Plane Shear	260°F/500 Hrs.	260'1	0.54	ı	7.4	30,030
•0	In-Plane Shear	-/100 Hrs.	RID	ı	ı	9.2	¢
•0	In-Plane Shear	JOU'F/100 Hrs.	260 °F	•	4	0.0	
• 0	In-Plane Shear	350°F/500 Hrs.	RID	0.79	•	9.1	21,670
•0	In-Plane Shear	350"F/500 Hrs.	350°F	0.42	•	5.6	30,000
• 0	Int. Shear	260"F/100 Nrs.	ATD	•	•	16.7	,
• 0	Int. Shear	260 F/100 Hrs.	260°F	•	•	14.6	•
•0	Int. Shear	260°F/500 Hrs.	RTD	•	•	15.1	•
•0	Int. Shear	260°F/500 Hrs.	260°F	ľ	•	14.3	٠
•0	Int. Shear	350°F/100 Hrs.	RID	•	•	15.4	•
•6	Inc. Shear	350"F/100 HEB.	260°F	•	,	13.8	•
•0	Int. Shear	350 F/500 Hrs.	ET.	•	•	15.3	•
•	Int. Shear	350"F/500 Hre.	350*F	•	•	11.0	•

TABLE XII STATIC PROPERTIES

Orientation	Type Load	Prior Conditioning	Test Tap.	E (pet x 10 ⁶)	(12/12)	"alt (bat)	**************************************
.0	Tension	260°F/500 Cy.	£	•	•	**	•
• 0	Tension	260 F/500 Cy.	7.097	•	•	160	•
,0	Tension	260 °F/1000 Cy.	E	28.1	0.18	146	9460
•0	Tension	260°F/1300 Cy.	7.0%	29.4	0.10	175	6010
•0	Tension	350*F/300 CF.	£	•	•	189	•
0.	Tension	350*F/500 CF.	260°F	•	•	174	•
•0	Tension	350°F/500 Cy.	350*F	J	•	172	•
• 0	Tems i on	350°F/1000 Cy.	£	28.9	0.19	169	62.90
,0	Tension	350*F/1000 Cy.	350°F	36.2	0.15	141	5%0
• 06	Tens ion	260 *F/500 CF.	Æ	•	•	7.9	•
• 06	Tension	260 °F/500 Cy.	7-042	,	•	6.7	•
• 06	Tension	260 F/1000 Cy.	£	2.79	0.0	6.5	3410
, 06	Tension	260°F/1000 Cy.	J.092	1.78	0.03	•••	4300
.06	Tension	350*F/500 C7.	£	•	•	4.4	•
.06	Tension	350*F/500 Cy.	1,092	•	•	6.9	•
• 0	Tens ton	330*F/500 Cy.	350*7	•	•	6.3	•
.06	Tension	350°F/1000 Cy.	£	2.28	0.0	7.7	3150
. 06	Tension	350*F/1000 Cy.	350°F	1.46	0.0	9. 4	9000

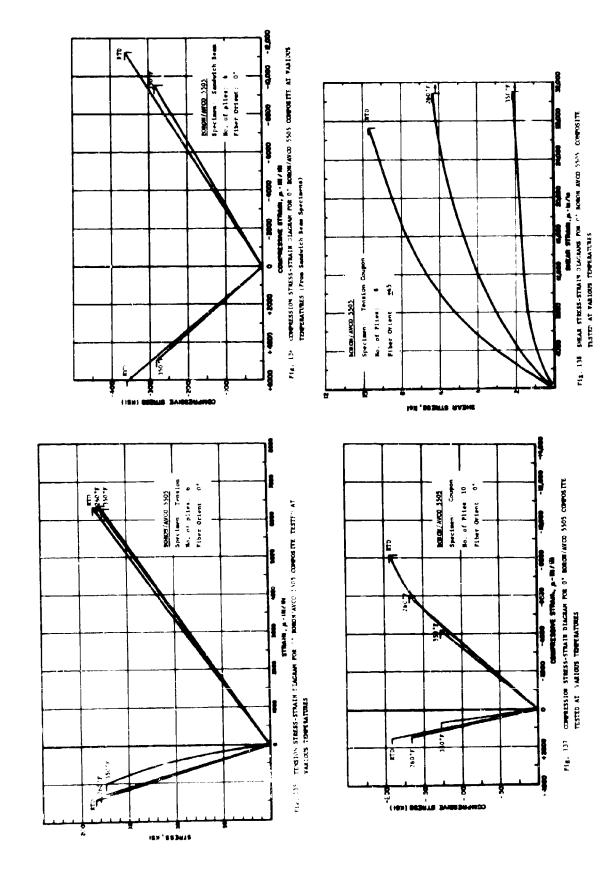
TABLE NOT - STATIO OF COME TON (MARK) - A100
Orien ation	Type Load	Prior Comditioning	Test lemp.	E (pst x 10 ⁶)	(in/in)	dult (ket)	^e ult (μ-in./in.)	
[0/43/135/0/80]	Tension	260°F:50°1 Cy.	RTD			92		
[C/45/135 '0/ ^E]	_ension	260°F/500 Cv.	250°F	•	•	82	•	
<u>" []/0, 5[]/57.0]</u>	Cension	250°F/1000 CV.	RTD	14.7	0.40	*	6070	
12 -1/133/0/ET.	Tension	260°F/1000 3y.	260°F	14.7	0.40	*	6070	
10/45/135/0/EE	_ension	350 F/500 Cy.	RTD	•	•	91	•	
[C1/65/135/0/J]	ension	350°F/500 Cy.	260°F	•	•	90		
[0,45/135/0/[2]	Tension	350°F/500 Cy.	350°F	•	•	11	•	
[[/43/135/0/[]]	Tension	350°F/1000 Cy.	ET.	15.5	0.41	88	6110	
[C/45/135/0/[].	Tension	350°F/1000 Cy.	350°F	15.6	0.43	52	4010	
[,] ن	Compression	260°F/500 CV.	Ę	•	•	208	•	
٥	Compression	260°F/500 Cy.	260 ° F	•	•	199		
ر.	Compression	260"F/1090 CY.	Q.E.	25.5	0.19	215	8920	
ື	Compression	260°F/1000 Cy.	260°F	27.0	0.23	18 18	7340	
ຶບ	(compre-ston	350°F/500 Cy.	£	•	•	202	•	
. ט	Compression	350°F/500 Cy.	350 °F	•	,	149	•	
ຶນ	(oupression	350°F/1000 Cy.	67	30.3	0.20	123	8750	
_e 0	Compression	350°F/1000 Cy.	350°F	25.4	0.28	149	5670	

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DABLE XII STATIC PROPERTIES SUMMENT - AVCO

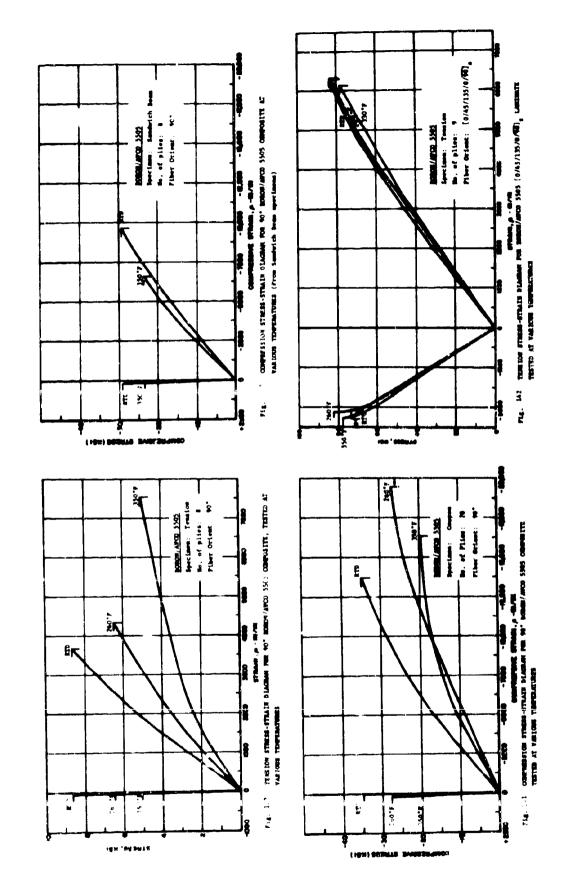
SUMMENT - AVGO \$505/WOROM COMPOSITES (CORE'4)

Orientesia	Type Load	P-for Cond.t.foning	1 t 1 (1)		(41/41)	Tale (Jan)	⁰ ult (#-18./18.)
• 06	Compression	260*F/500 Cy.	Ē	•	•	11 %	•
• 06	Compression	260°F/500 Cy.	7.0%	•	•	4 .7	,
•06	presition	260*F/1000 Cy.	£	27.72	0.01	76.8	20,000
• 06	Compression	260 F/ 1000 CF.	7.0%	2.57	0.01	3.8	10,940
• 06	Compression	350*F/500 Cy.	E	,	•	¥.3	
	Compression	350*F/500 Cy.	350*7	•	•	17.9	•
• &	Compression	330*F/1000 CF.	£	2.87	0.01	17.1	12,830
• 0\$	co resuton	350 F/1000 CF.	350*7	1.30	0.0	16.4	12,360
[0/43/135/0, 35]	Compression	260°F/500 Cy.	E	•	1	183	
[0/45/135/0/30]	Compression	260*F/500 CF.	7.092	•	•	142	•
[0/45/135/0/30]	Compression	260 *F/1000 Cy.	£	13.3	3.0	*	14,330
[0/45/135/0/30]	Compression	260 °F/1000 CF.	4.0%	13.2	0.37	156	12,600
[0/45/135/0/ ²]]	Compression	350°F/500 CF.	£	•	1	18	•
[0,45/135/0/,0]	Compression	350*T/500 Cy.	350'T	•	•	3	•
[0/45/135/0/35]	Compression	350*F/1000 CF.	£	15.5	0.49	133	15,140
[0/47/135/0/37]	Compression	330*F/1000 Cy.	350.7	11.9	0.53	128	11,630



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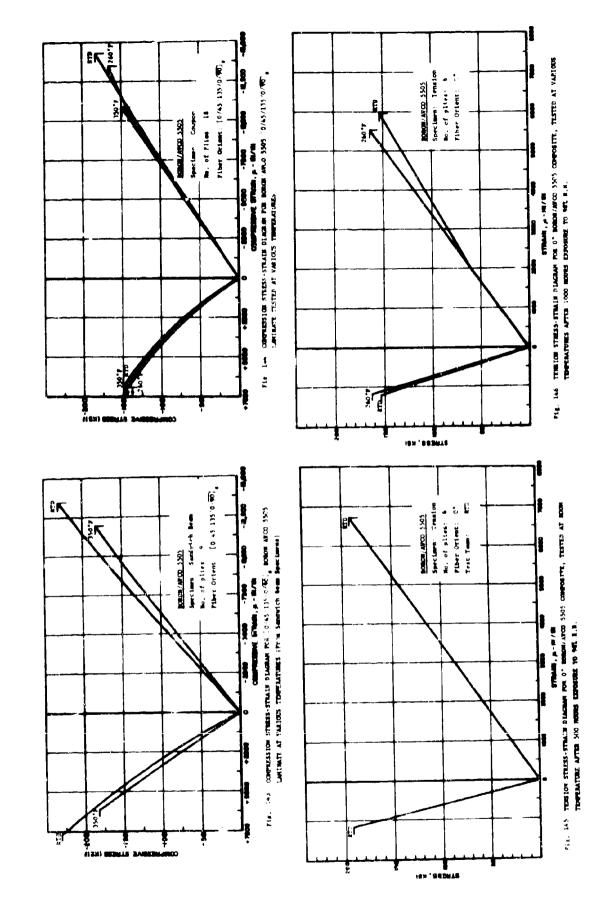


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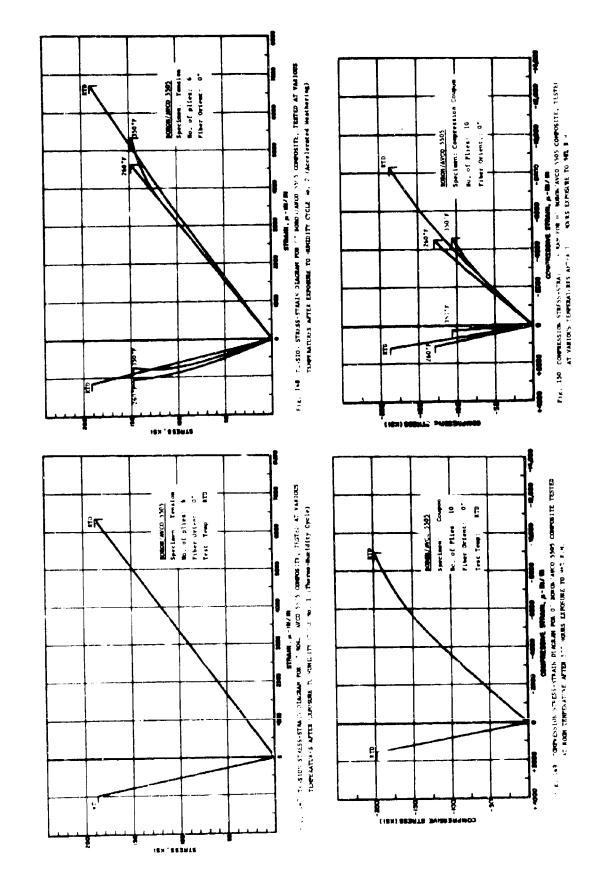
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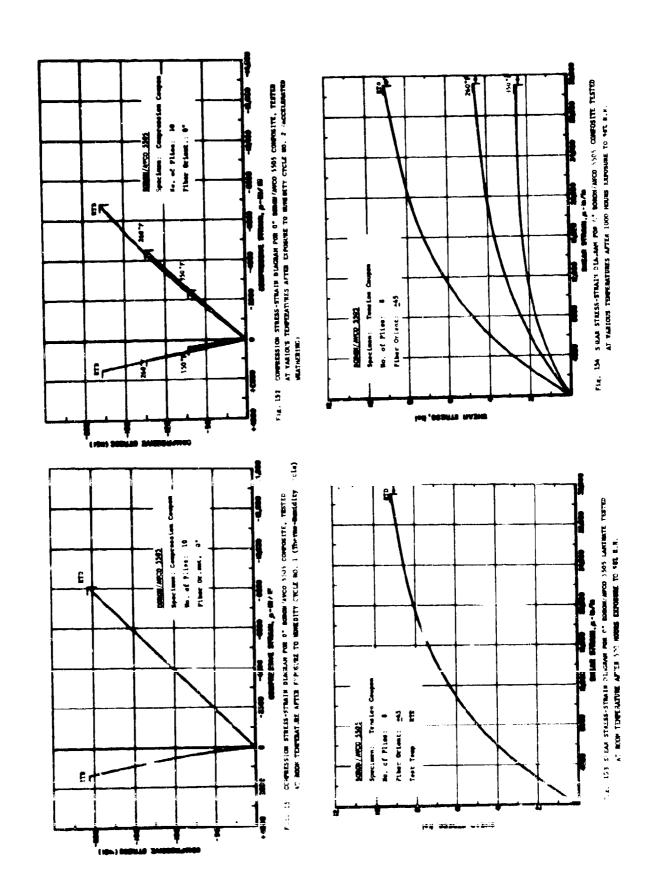


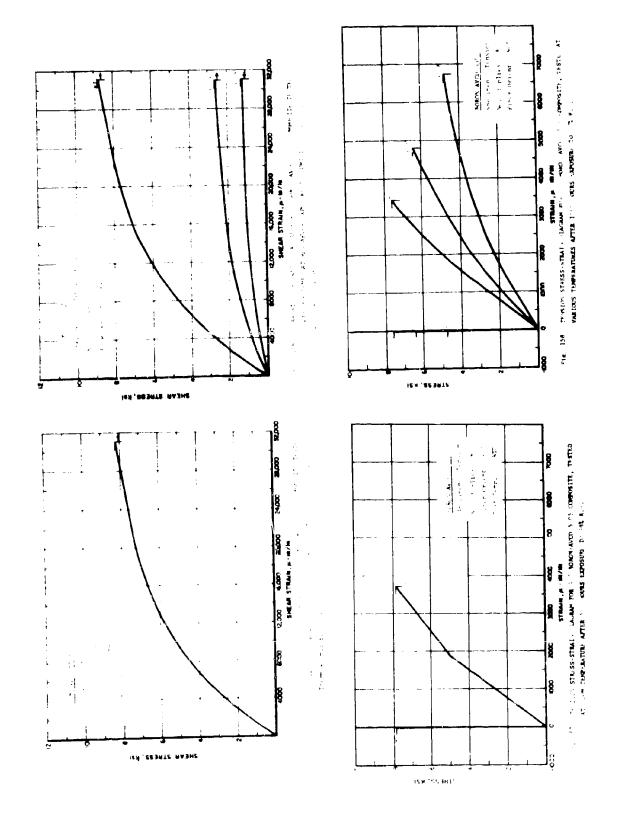
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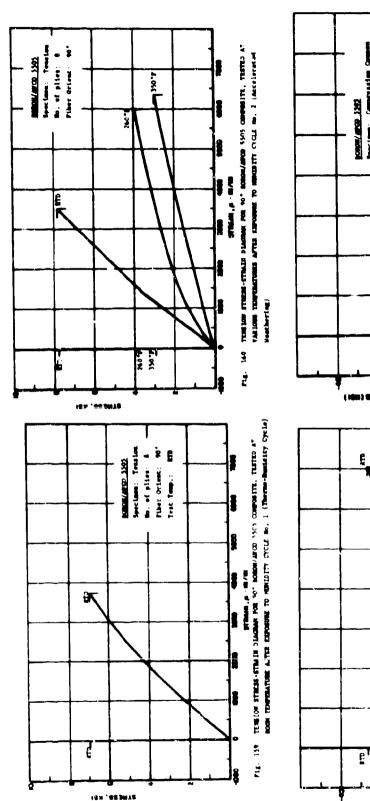
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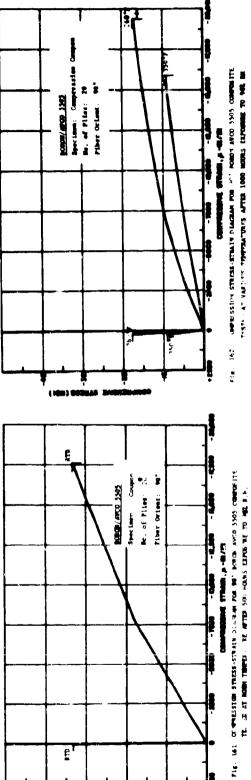






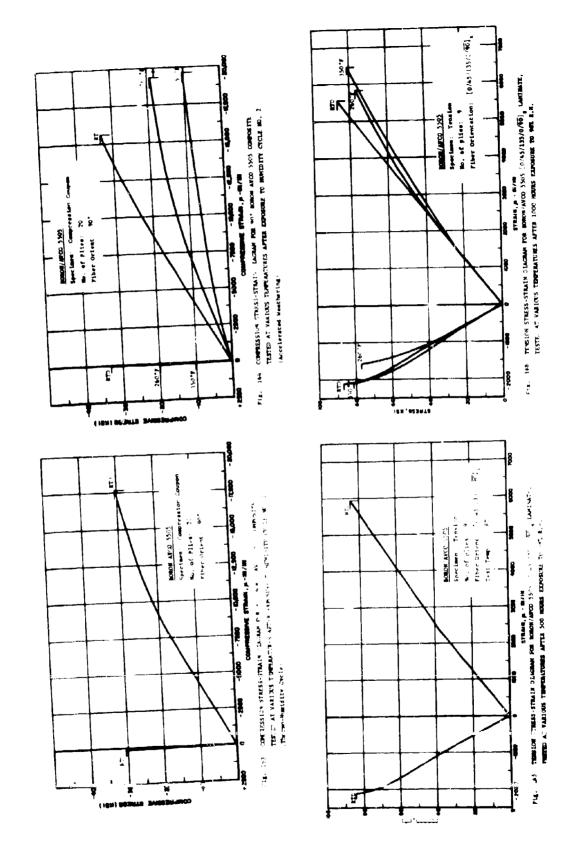
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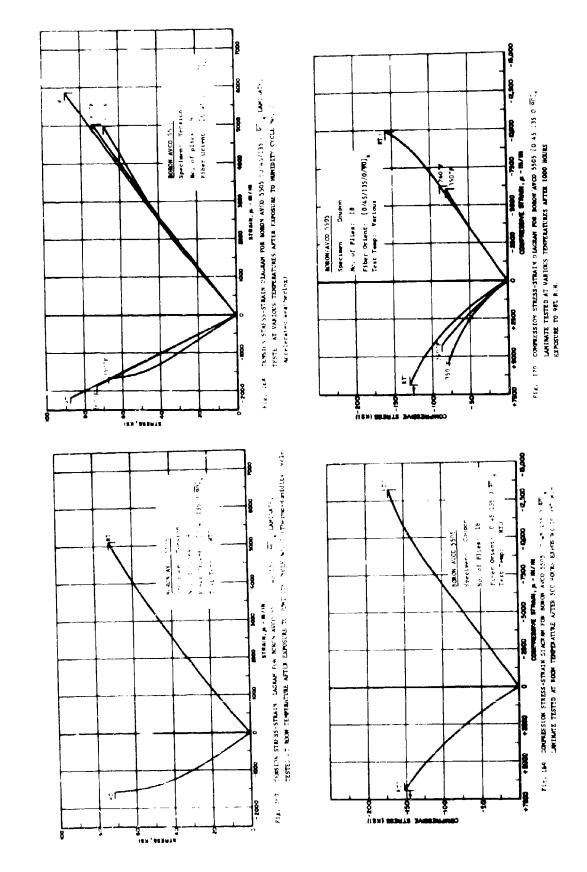


CHERCASION STREAM-STRAIN DIAGRAM FOR NOT HORD AND \$500 CHERCATTE. Ftg. 162

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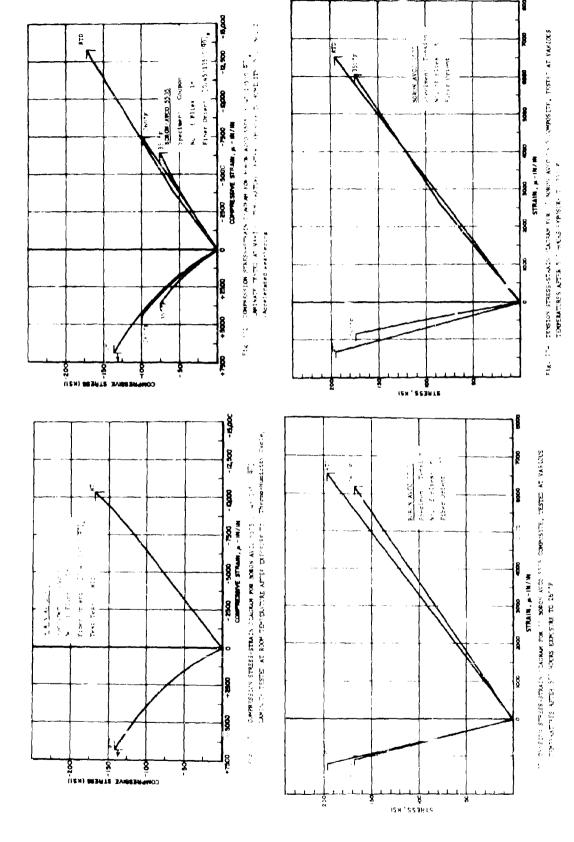
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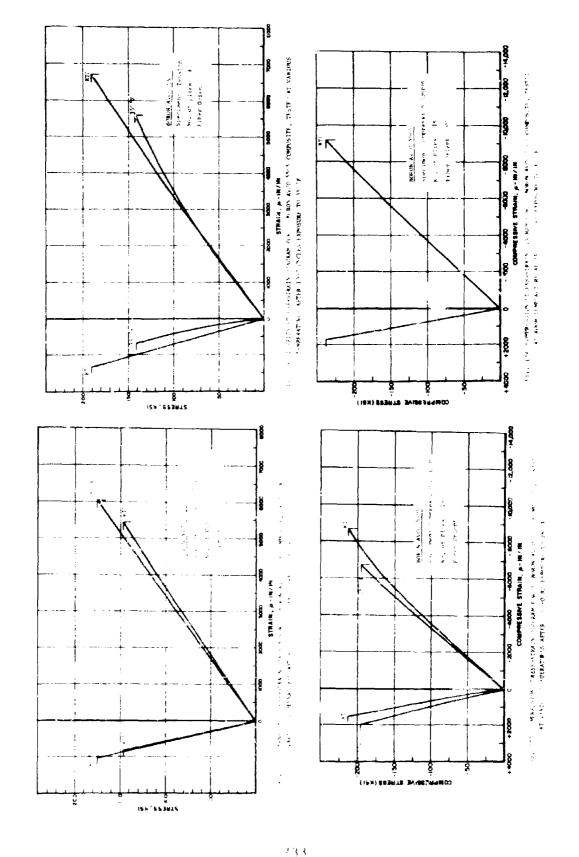


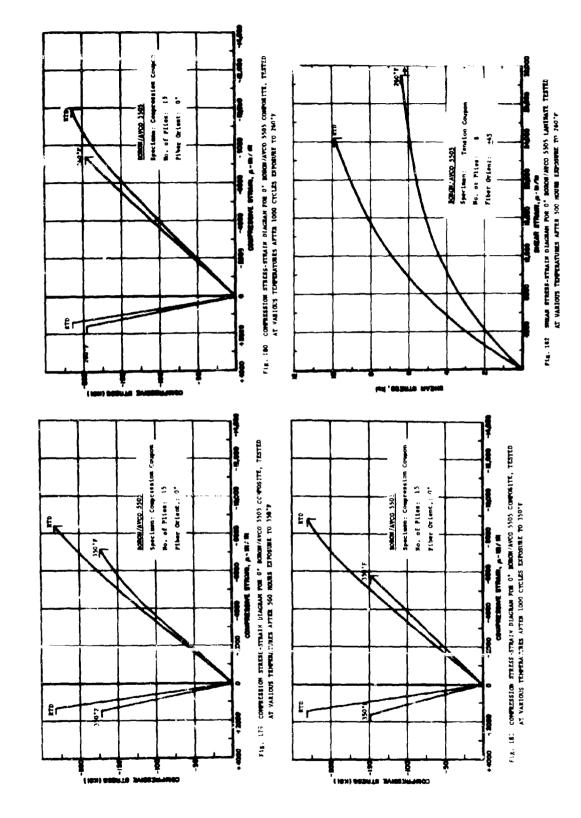
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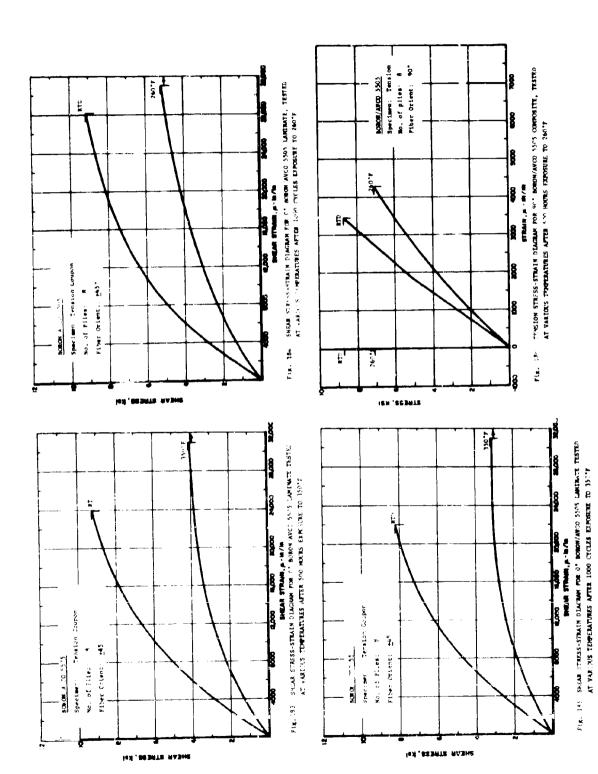






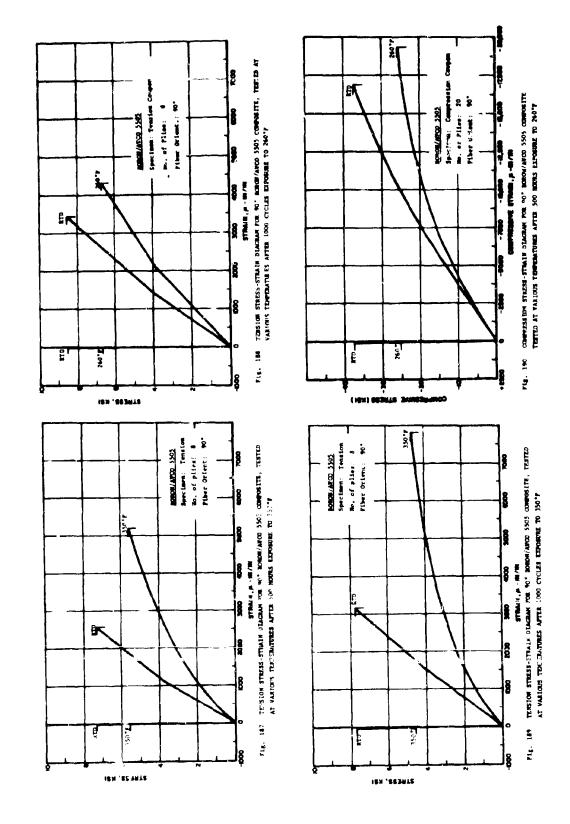
Ĉ,

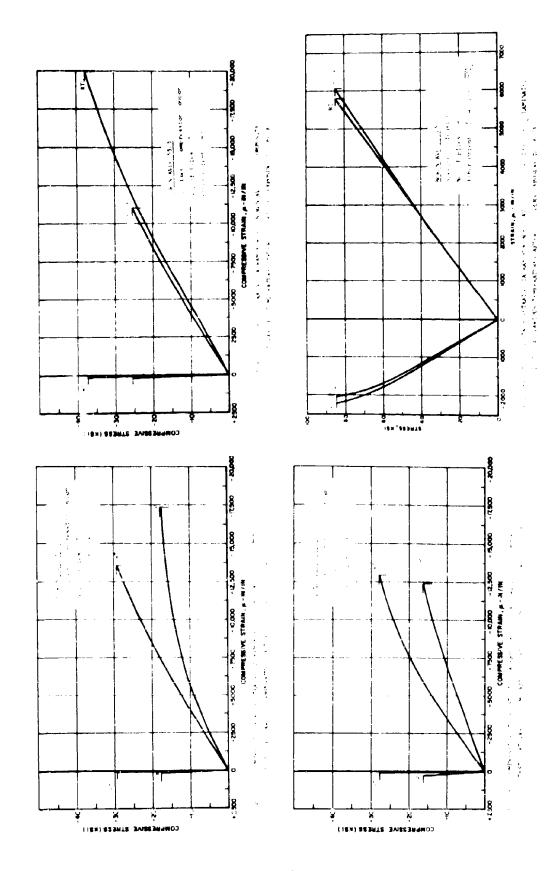
~:



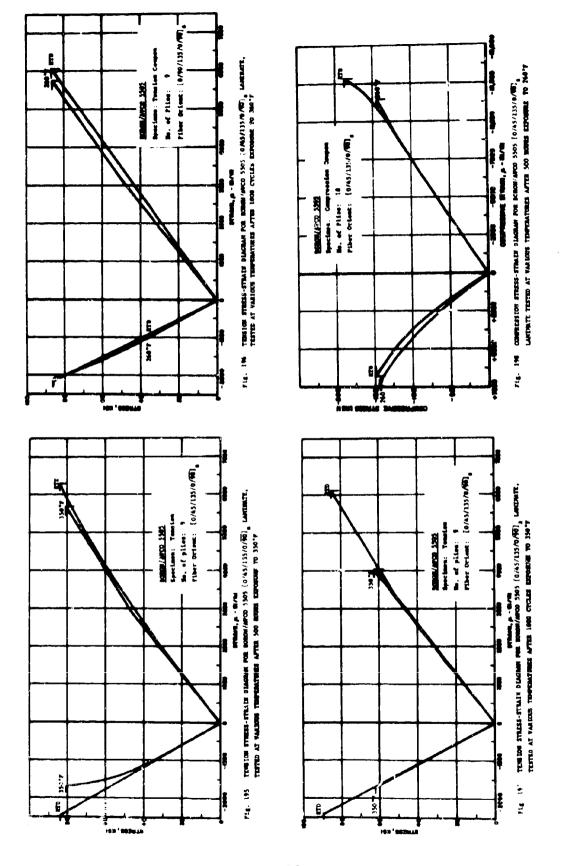
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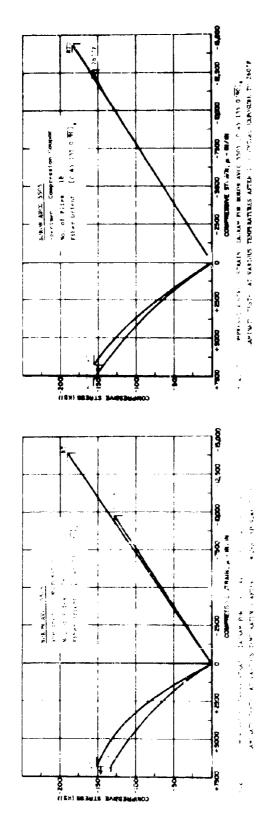
Š

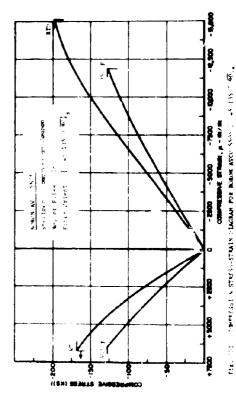




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LAMINARY TESTON AT VARIOUS TEMPERATURES ANTER 1 OF LYCLEY EXPOSURE TO 156°F

TABLE XIII FATIGUE PROPERTIES SUMMARY - AVEO 55/5/60/BON CORPOSITES

					-					
								Cycles		
							Cycles	App 1 Ced		
200	Thickness		2017	PRIOR COMDITIONING	Test	Stress Level	3 ;	without	Resident	
Parther	(Pites) (In.)	Ortentation	1ype	Duration	ĖĐ	(Tale) (hat)	(cycles)	(cycles)	(test)	Comment
3-45001W	A - 0 632		900%	:	1	1 2	710 000			
WINDSA.F.	1000	•		1				•	•	•
3.VC001K	0 - 0.032	> (E S	•				•	•	•
MICON -ED	٠	. D	Mone		E	3	8. 8		•	•
1100JA-E	6 - 0.032	•0	Kone	•	Ē	175	98.	•	•	
N1005A-E:0	6 - 0.032	•0	None	•	Ê	91	37, 980 1, 37, 980	•	•	
M1005A-E:1	6 - 0.032	٥	None		e	152	100	•	•	•
N1005A-E12	•	•0	Kone	•	6	175	8	•	•	•
N:005A-E:3	•	•	, Contract	•	6	170	24,000	•	•	•
4100SA-F14	٠		Hone	•	Ē	01-	20,000	•	. (. 4
1100Ca		•				176)	1	•
13-4701X	260.03	•		•	2 (707		•	•	•
7-W007W	6 - 0.032		Mone	•	Ē,	X	3,473	•		•
N1005A-E3	6 - 0.032	, O	None		Ē	3				•
#1002-6	240 0 - 8	ş	a de la constante de la consta	•	8	9	9	•	•	•
00018	440	:			1				,	
1-700TH		2	ž,	•	} {	7.	3	•	•	•
9-700TN	10.0	3	900	•	2	0.0		•	•	•
M1002-9	3 - 0.044	. 8	Ž	•		0.4			•	
M1002-10	440.0 - 60	<u>,</u>	None		E	4.5	423,000		•	•
M1002-11	440.0 - B	• &	agog.	•	E	0.4	•	10,425,000	F. 7	•
K1002-12	8 - 0.044	•06	None		£	5.5	10,000	•	•	•
M1002-13	6 - 0 Own	.06	None	•	E	4.4	1,130,500	•	•	
H1002-14	9 - 0.044	• 8	None		e	5.0	98.19		•	•
N1902-15	8 - 0.044	• 8	Mone	•	Ē	o. .	9,000	ı	•	•
A-A7001W	90	[0/65/135/0/96]	9	ı	Ē	3	161 000	,	•	Tek Ares Pellure
		Total Control (Control)			i	; ;				
M1027A-7	0.030	[0/45/135/0/90]	None	•	e	*	8	•	•	•
M1027A-8	9 - 0.050	[04/0/501/59/0]	None	•	B	8	378,000	į	•	•
M1027A-9	9 - 0.050	[0/45/135/0/90]	Mone	•	£	~	13,000	•		•
N1027A-10	9 - 0.050	[0//2/135/0/30]	Kone	•	E	ĸ	88°,	,	•	
M1027A-11	9 - 0.050	[0//45/135/0/90]	Mone	•	E	8	1	•	ı	L fate Pailure
			,		1	**				The state of the s
N1027A-1	9 - 0.050	0/45/135/0/90]	aco.	,	8	8		•	•	Tab Area Fallure
M1027A-1	9 - 0.050	[0/42/135/0/90]	ğ	•	ê	2	•	16,100,000	63.0	•
K1027A~1	9 - 0.050	[0/45/135/0/90]	None None	•	6	2	000	ı	•	•
W1027A-4	0.000	[0/45/135/0/90]	None		6	3	•	12,900,000	o. X	•

TABLE XIII FATIORE PROPERTIFS CHORES - AVIO \$505 ROPERTIFS

								Cycles	Applied		
Specimen	Thickness		No.	FIOR CONDITIONING	<u>.</u>	Stress Level	Sere!	to Pat 1	without	Les (des)	
n contract	(Pites) (la.)	Orlentation	K G	Duret lon	(£)	'ult)	ult) (ka1)	(cycles)	(c)cles)	(101)	Commonst
#1007A-11	6 - 0.033	•0	E CO	ŧ	260.7	32	150	00			
M1007A-12	6 - 0.033	•0	MCDe	•	260°F	83	120	120.000			
#1007A-13	١	• 0	Mone		260°F) 2 6	155	350,000	•		• (
#1007A-14	e - 0.033	•0	ACD e	•	Z60 °F	69	150	265.000			
M1007A-15	1	•0	Mone	•	260°F	000	160	12.000			. (
M1007A-16	1	•0	Pone	•	260°F	&	155	000		•	٠,
W100"A-17	6 - 0.033	•0	Mcne	•	260 °F	98	145	145,000			•
#1007A-E1	ı	•0	K Cne	•	260°F	87	157	149,000	•		Tab Area Failure
M1007A-19	1	•0	#CDC	•	260°F	60	148	725,000	•	•	Tab Area Zatlute
N1007A-20	6 - 0.032	•	K re	•	260°F	11	140	244,000	•	•	
#101% -11	2 9 0 0 3	\$	1		4.070	7.		Ş			
	•	· •			1.046	9, 4	۰ ۵	3 8			
		2		,	200	2;	٠.	3.5			
	200	2 8		•	1.067	1 5	.	200°			
	1	2		,	1 707	10	4	36.1			
	•		900		Z66 F	52	3.4	53,000			
	4 - 0 - 0 + 4 - 1	• O	MC Je	•	260°F	53	3,5				
	ŧ	.	e::a	•	260°F	53	3.5				
M1013 -18	E - 0.044	• \$	Mene	•	260°F	Š	3.3	229,000	•		
M1016 - 1	8 - 0.045		Mene		260°F	94	٣		2.2 x 10"	9.9	Tab Area Failure
M 016 2	E - 0.045	•	2 00	r	260°F	6.4	3.2	1.451 x 10°			
K10338- 2	640.0 - 5	[0/45/135/0/90]	10.34	•	260 ℃	592	70	11,000	•	•	•
M10338- 3	640.0 - 9	[0/45/135/0/30]	Mene	1	260°F	79	65	11,000	•	•	•
M10338- 4	640.0 - 5	[0/45/135/0/30]	ar.o	•	260°F	85	5	2,000	•	•	•
H1773(38- 5	640.0 - 5	[0/45/135/0/90]	Mone	•	260 °₽	6,	65	175,000	•	ı	•
1103[B- 6	690"0 - 6	[0/45/135/0/90]	aco g	ŧ	260 ⁴F	9.1	75	2,000	•	•	•
M1.358- 7	9 - 0.049	[0/45/135/0/90]	1 0.76	•	2€0. 1	81	7.2	62,000	•	•	•
M10378- 8	6 - 0.049	[0/45/135/0/90]	3	•	260°F	87	72	186,000	•	•	Þ
W20338- 9	9 - 0.050	[0/45/135/0/90]	Mone	•	260 °₽	91	75	10,000	•	•	•
W10338-10	6,000 - 6	[0/42/135/0/90]	Bo ne	ı	7.09Z	82	89	738,000	•	•	•
W10338-11	9 - 0.049	[0/45/135/0/30]	Mone	•	260°F	82	89	1,265,000	•	•	•
)									

	Thickness		PRIOR C	PRIOR CONTITIONING	Hear	Stress Level	ī	Cycles	Cycles Applied without	Kesiduel	
Marber	(Pifes) (In.)	Orientation	Type	Curation	ĖE	(Xoult) (kst)	(kst)	(cycles)	(cycles)	(keil)	Comme
N1007B-1	6 - 0.033	.0	None		350°F	ş	160	3,000		,	
N10078-2	,	ပံ	None	•	350°F	06	160	2,000	•	•	•
N1007E-3	٠	.0	Mone		350°F	80	155	000.	•		•
N 0078-4	•	•0	None	,	3.00°F	80	155	••	•	•	Impediate tab
											failure
N10073-5	6 - 2.033	0.	None		350°F	85	150	2,000		•	•
N10073-6	6 - 0.033	ċ	None	•	350°F	8 2	14.5	13,000	•	•	•
N10078-7	6 - 0.033	• 0	None	•	350°F	76	167	,	•	•	,
8-87001N	6 - 0.033	• 0	None	•	350°F	85	150	•	•	•	Immediate tab
						į					fel lure
M10078-9	6 - 0.033	•0	Mone	•	350°F	66	140	9,000		•	•
N1307B-10	6 - 0.033	0.0	Mone	1	350°F	73	130	511,000	•		1
31014-3	9790	•06	Mone	•	3.05E	7.5	7	3,000			
41014-4	**************************************	.06	None	•	350 F	7.5	4	3,000			
X1014-5	440.0	,06	None		350°F	ጵ	C	268,000			
41014-6	340.0 - 8	•06	Mone	,	350°F	53	2.8	1,789,000			
1-91011	490.7 - 8	.06	None		350°F	\$	3.5	1,000		-	Excess epoxy not taken
11014-8	8 - (.043	•06	None	•	350°F	69	3.6	≈			off. Spec. broke while
11014-9	6 - (.043	• 06	Mone	,	350°F	9	3.2	135,000			Ful junca
41014-10	8 - 0.043	•06	None	•	350°F	9	С.	10,000			
11014-11	8 - 0.043	• 06	Mone		320.6	62	۳. ن	11,000			
111014-12	740') - B	• 2	None		350-1	9	7.7	7,000			
1:1034A-1	670.3 - 6	[0/45/135/0/90]	Mone	•	350 °F	0	63	668,000	•	1	•
171034A-2	670'0 - 6	[0/0/51/37/0]	None		350°F	6 2	65	554,000	•	•	•
F1034A-3	6 - 0.049	[0/45/135/0/90]	None	•	350°F	82	65	10,000	•	,	•
4-A4601 4	640.0 - 6	[0/45/135/0/90]	None		350°F	2	70	7,000	٠	•	•
X 1034A-5	9 - 0.049	[0/45/135/3/90]	Mene		350°F	8	70	7,000	•	•	•
K1034A-6	050'0 - 6	[0/45/135/0/90]	Kone	•	350°F	88	67	7,000	1		•
K1036A-7	690'0 - 6	[0/42/132/0/90]	Mone	•	350°F	*	99	9,000	•	•	•
8-YAC	9 - 0.050	[0/48/135/0/90]	Mone	1	350 ₽	85	67	10,000	•	•	•
H1034A-9	6 - 0.049	[0/45/135/0/90]	Mone	•	350.7	28	99	13,000	,	•	•
N 0344-10	690'0 - 6	[0/45/135/0/90]	Mone	•	350°F	3	*	16,000	1	,	•
F 0344-11	640.0 - 6	[0/45/135/0/90]	None		350.1	\$	2	24,000	•	•	
-											

·	Thebase		2	PRIOR CHEMICAL	1081			Cycles to	Cycles Applied without	Les (der)	
Pocc Leases Marie or	(Plies) (In.)	Orientation	Ē	Deretion	į.	(T'ult) (ksi)	(F.8.E)	Feilure (cycles)	Failure (cycles)	Strength (kal.)	Comment
410658-7	6 - 0.033	•	/ Ha 1200	500 Hrs.	£	2	165		000		Tab Area Pailines
11005B-1	6 033	•0		500 Hrs.	6	B 2	158		210,000		
K10058-	, ,	• 0		500 Hrs.	9						-(Specimens broke
#10058-10 #10058-11	, , , ,			500 Hrz.	6						Aduring .
11-10-01		, ,		1 200		į					Tablication
	6 - 0.033	5 6			1.090	6	2 2	. 8			Immediate Failure
	6 - 0.033	o c	/ HE 196		260°F	2 62	7 P	900			
M100 88-18	6 0,033	•0	/ HOT 196		260°F	2	135	2,000	•		
61-E5JUTH	6 - 0.033	.0	_	500 Hrs.	260°F	7.2	127	. 1	3.176 x 100	135.5	Tab Failure
#1009A-15	6 - 0.032	.0	/ KM 136	500 Hrs.	350°F	104	150		4		Ismediate Failure
H10C94-16	6 - 0.033	•			350°F	8	55	,	2 x 10°	150.3	
MICK JA-17	6 - 0.032	• •	_	500 Brs.	350 . F	101.5	145	3,000			
M10K9A-18	6 - 0.032	•0			350°F	97.5	140	900			
K10C9A-19	6 - 0.032	• G	/ E		350°F	ま	135	9			
M10C 58-12	6 - 0.033	• 0	/ H21 1296	1000 Hrs.	e E	104.5		1,000			
K10C58-13	6 - 0.033	,0		1500 Hrs.	£	101.0		13,000			
H10C 5B-14	6 - 0.933	• 0			e E	2		903,000			Tab Feilure
M10C 58-15	6 - 0.033	•	/ HE 196	1000 Hrs.	Ê	8	153	62,000			
MOC 58-16	6 - 0.033	•0		1000 Ars.		2		3,000			
#10C3B-20	6 - 0.033	÷	/ 821 126	1000 Brs.	260°F	92		000,4			
F100%-1	6 - 0.033	• 0	_	1000 Ers.	2.09Z	2		90.			
F10C 9A-2	6 - 0.033	•	7 226	1000 Ers.	260°F	2	9	9			
F10C 94-3	6 - 0.033	•	_		260 F	2		000 81			
110CSE-1	6 - 0,033	•	_		260	44		2,492,000			
FLOK 94-20	6 - 0.032	.	_		350 F	*		8			
F10C98-1	6 - 0,033	•	/ E 186		350 F	•	135	86.			
110/31-2	€ . 0.032	•	7 116	1000 Hrs.	320.6	*	221	8			
110C-38-3	6 - 0.033	• •	7 126		350°F	*	120	000			
F10C 98-5	6 - 0.033	•	7 126		350 F	*	113	2,000			
						*	data	o date not evailable	•		
							חונ				

* o.1. data not available

nt nt			17.e		ıre	Tab Failure Immediate Failure
Comment			Tab Failure Tab Failure Tab Failure		Tab Failure	Tab Feilure Immediate F
Residual Strength (ksl)		154.3			168.8 163.1	
Cycls Applied Without Failure (cycles)		2.1 × 10 ⁶			2.25 × 10 ⁶ 2.791 × 10 ⁶	
Cycles to Failure (cycles)	2,000 90,000 27,000 2,000 3,000	2,000 5,000 6,000	12,000 8,000 12,000 13,000 314,000	30,000 45,000 9,000 176,000	51,000	1,000 236,000 9,000
Stress Level	85 160 82.5 155 84 158 81.5 153 79 148	98 155 92 145 87.5 138 82.5 130 76 120	108 130 104 125 100 120 96 115 92.5 110		89 140 83 130 79.5 125 81.5 128	82 120 79 115 72 105 75.5 116
1 (a)		260°F 260°F 260°F 260°F	350 F 350 F 350 F 350 F		260°F 260°F 260°F 260°F	350°F 350°F 350°F
NIOR COMBITIONING TPE Duration	Thermo-Humidity Cycle Thermo-Humidity Cycle Thermo-Humidity Cycle Thermo-Humidity Cycle Thermo-Humidity Cycle	midity Cycle midity Cycle midity Cycle midity Cycle	midity Cycle midity Cycle midity Cycle midity Cycle midity Cycle	Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng.	Wihrng. Wihrng. Wihrng.	Wthrng. Wthrng. Wthrng.
Type	Thermo-Humidity Thermo-Humidity Thermo-Humidity Thermo-Humidity Thermo-Humidity	The ruo-Bandity The ruo-Bandity The ruo-Bandity The ruo-Bandity The ruo-Randity	Thermo-Humidity (Thermo-Humidity (Thermo-Humidity (Thermo-Humidity (Thermo-Humidity (Acc. W		Acc. W
Orientation	60000	, , , , , , , , , , , , , , , , , , ,	00000			0000
Thickness (311es) (In.)	6 - 0.033 6 - 0.033 6 - 0.033 6 - 0.033 6 - 0.033	6 - 0.033 6 - 0.033 6 - 0.033 6 - 0.033 6 - 0.033	6 - 6.033 6 - 6.033 6 - 6.033 6 - 6.033 6 - 0.033	6 - 0.033 6 - 0.033 6 - 0.033 6 - 0.033 7 - 0.033	1 1 1 1	6 - C.033 6 - C.033 6 - C.032 6 - C.033
Spectmen	N1G05B-17 N1G05B-18 N1G05B-20 N1G05B-20 N1G05B-21	N1009A-5 N1009A-6 N1009A-7 N1009A-8 N1009A-9	20068-5 N10098-6 N10098-8 8-850CIN 8-850CIN	M1005C-1 M1005C-2 M1005C-3 M1005C-4 M1005C-5	N1009A-11 N1009A-12 N1009A-13 N1009A-14	N10098-10 N10098-11 N10098-12 M10098-13

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1 1 2 2	Thickness		PRIOR	PRIOR CONDITIONING FEST	rest	Stress Level	wel	Cycles	Cycles Applied Without	Residuel		
Number	(.nl) (saild)	Orientation	adái	Duration	Ē	(T ³ ult) (ksi)	(ks1)	(cycles)	(cycles)	(kai)	Comment	
•			ī		į							
11.(283-5	100		•		<u>.</u>	<u>3</u>	75	6,000			Tan Area Failure	
NEC253-5	19010 1 1				<u>C</u>	-6	20	20 ,000				
X10283-7	()	-	Thermo-	Thermo- Humidity Cycle	RTD	93	<i>:</i> 9	36,000				
N1C283-8		-	The Tayl	Humidity Cycle	ET.	\$	4.5	000			drift ware a waitline	
Y10188-9	GC:0-5	=	1200	Dermo- Humidity Gyale	č.	63	09	764,000			Teb Area Failure	
X1036A-1	(F)	=		Humidity Cycle	ь: ;	i c	ý	,			1	
N. 136A-D	on in	-	1000	Humicity	2b. F	62	5 5	•	7 47 - 10t	6 97	יייייייייייייייייייייייייייייייייייייי	
N.C.363-3	or or	-	The 1120-	Humidity Cycle	260°F	1 de	:	000 07	21.7	7.2,		
XI - 364-	250.0 - 6	-	-OBCTOR	Humidity Cycle	360 F	74.5	9	000				
71136A-5	(1) (1) (1) (1)	2	Therman	Therew- Humidity Cycle	2e0.F	3	5	15,000				
V. F 3 " A- !		-	-65735	Termo- Humidity Ovels	350 °F	52	30		2.062×10^{6}	58.0		
		Ξ		Franchista Carlo	3.50 F	or or	Ç	86.000				
1 - 5 - 5 - 5 - 5 - 5 - 5 - 5 - 5 - 5 -	. (.) . (.)	=		Heridity	350 F			000				
	1 m	÷			350°F	92.5	٠ (000°				
X10378-5	1	-	Thermo-	Chermo- Humidiry Cycle	350°F	26.5	200	78,000				
C1-68001X	50° - 6	-	Acc.	Acc. Withrog.	KTD	980	70	7, 000				
1000	190°C - 6	:	7.0▼	thrus	RTD	1	, y	672,000			Tah Estlute	
	5 - 5.053	=	Acc	Vihing.	RTD	77.5	99	377,000			3	
0-76201N	140°C = c	=	Acc.	Wthrng.	CT.	83.5	73	2,000				
X1129A-3	. (C)	£	Acc.	Athrng.	gta	9 8	70	691,000			Tab Failure	
7.1.36A-5	3 € 0,352	<u>.</u>	Acc.	Wthrng.	260°F	81.5	09	3,000				
X1036A-7	10.05		Acc.	Wthrng.	260°F	74.5	35	1			Immediate Failure	
X1036A-5	5 - 0.051	:	Acc.	Vchrng.	260 F	72	53	1,412,000				
X10363-3	בורים. ביים ביים	-	Acc.	uthrng.	260°F	77.5	56	7,000				
N.C36A-19	4 - 0,051	r	Acc.	Wthrag.	260°F	76	26	38,000				
		:			35005	9	4	000 170				
Nic3/A-0	. `	:	Acc.	Henring.	3500	8 8	20	000,17				
MECS/A1	0000	:	, 200 100		7,056	7 7 8	3 0	75.100			F	
0-8, 6378		a	, oc.	יייין דיווקי	7007	2.00	7 4	30,00			ido ratture	
V1037A-9	;;				350 5	2.5	7 (200.				
07-4-6072	5 - 5	S 04: 1, CF*, C*, ()	Y 00.	weneng.	2000	č Č	6	14,000				

The second of th

,		;							Cycles		
0 m	Thickness		PRIOR CO.	PRIOR CONTITIONING	Test	Stress Lovel	ove1	F84 1114	Without F. Turk	Residual Strength	
- Car	(Pifes) (In.)	Orient at 1.	Type	Duration	(4.)	(Toute [kst]	(kst)	(cacles)	(cycles)	(kai)	Comment
1* 1 40 0 1 X	5E - 2	ن	a speak	F . 500 Hrs.	Ê	c T	7.	1.000	•	•	
0 - 400 N		. c			. IX	Ľ.	1.41	20,000	•	•	Tab Linea Failume
-190/IF		٠. ت		500	¥	ű	3	£30,00r			Tab Irea Tailure
7 000 X	E - 0 - 4	Ť		F 500 Hrs.	7	ď,	165	000		•	•
4 DG 4	£ {\(\frac{1}{2}\)}	ن.	Steady 260°F	200	ж. :	č	551	000°4a.	1	•	
7 - 14 - 3	Ck. 9 - 9	<u>.</u>	Steady 350 F	F 500 Mrs.		S. SX	-1-	1,000	,	٠	Tak grea Failure
1900 E		ċ		. 500	C.F.	úć	161	.)00		•	
4 - V90C IN	200 0 - 4	č		005	CTX	oʻ,				•	Isl Srea Failure
7 - 1 400 R	6 - 11, 132	, O		604	∴ £	7.5	ن ا (5.671 × 10°	1	•	
8 90C ×	N 132	င်			KTD	92	-1 -1	15,000	,	•	Tab area Failure
Ø : 1 4 ± 2 : 1	4 . 0 . 4	,0	Ccc11c 260°	F . 500 Cvc.	. T.	90	191	1,400,000			
31-1900 N	5 - 0.0 34	.0	Cvc11c 260°F	•	Ē	44	ું દ ્	110,000			
- 906	6 - 0,133	,0		٠,	RT.	10:	190	3,000			
90	6 - 0,333	°0	. velle 260°F	٠,	<u></u>	٥٥. ۶	185	1,000			
F (D06.1-3	h - 0.033	Ċ	Cvc11c 260°1	~ .	E C	41.5	170	206,000			
N (00)	5 - 0.033	11.1	C.clic 260"F	٠.,	KT	41.5	170	į			Immediate failure
2. (OC) X	0 - 0.034	ć	Cyc.11c 260"F	~	CT3	98	091	•	•		Immediate Sailure
4.7.00 T. F	6 - 0.033	. 0	Cyc.11c 260°F		EL	20	130	,	2.021x10°	185.4	Tab failure
N.9001 N	表0.0 - 4	0	Cv:11c 260°1	٠.	و لا	75.5	140	2,000			Tab failure
ਲ : 000 ਪ	A - 0.333	ů	Cyclic 260°F	F / 1000 Cyc.	¢ L K	80.5	150				Immediate Tab Failure
2 · 1 (90%) #	£ - C.033	.0	Cyclic 350°F	/	RT	06	170	1,000			
M (0063-10	6 - 0.034	.0	Cyclic 350°F	٠.	L	85	160	4,000			
N1006 - : 1	4 . 0.033	ئ	Cyclic 350°F	•	ول	82	155	292,000			
N : 006 3- 1 2	6 - 0.033	· o	Cyclic 350°F	-	RT D	83.5	158	3,000			:
K1006:13	7€0. 0 - €	٥.	Cyclic 550*1	`	11	80.5	152	22,000			Tab Failure
# CO6.7 - 1 a	6 - 0,033	•		-	12	42	155	890,000			
M 1006 t- 15	6 - 0.033	•0	Cyc11r 350"	/ 1000	E	94.5	160	137,000			
N 1 0063 16	6 - 0,433	• c	Cyc11c 350"F	7	61	\$7.5	165	8,000	•		
#1000E-17	, 10	0		7	Z	Spec	imen Bro	Specimen Broke During Fabrication	prication		
M 1006:1-18	- 9	.0	Cyc15 .50"F	F / 1000 Cyc.] X	Spec	ושפו מינ	ואפ החדיוות נשי	ILLCALLUM		

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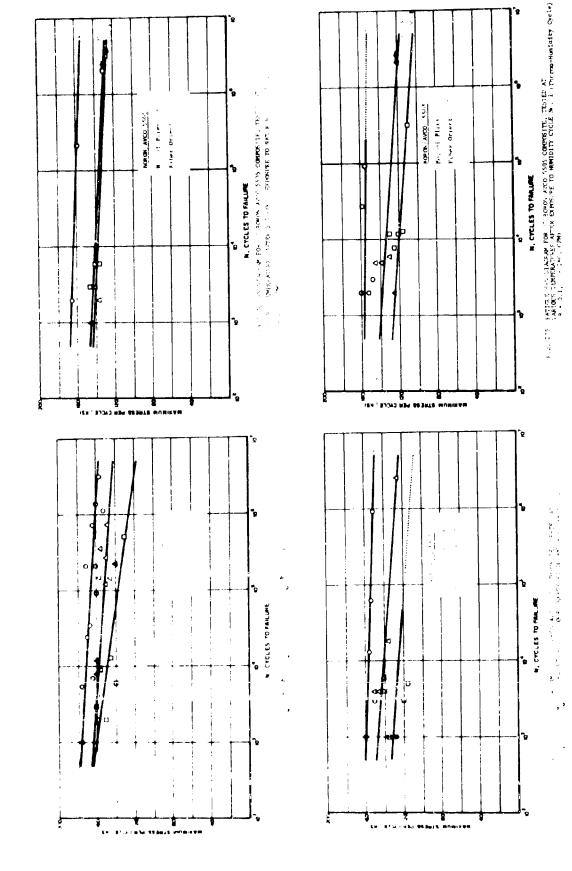
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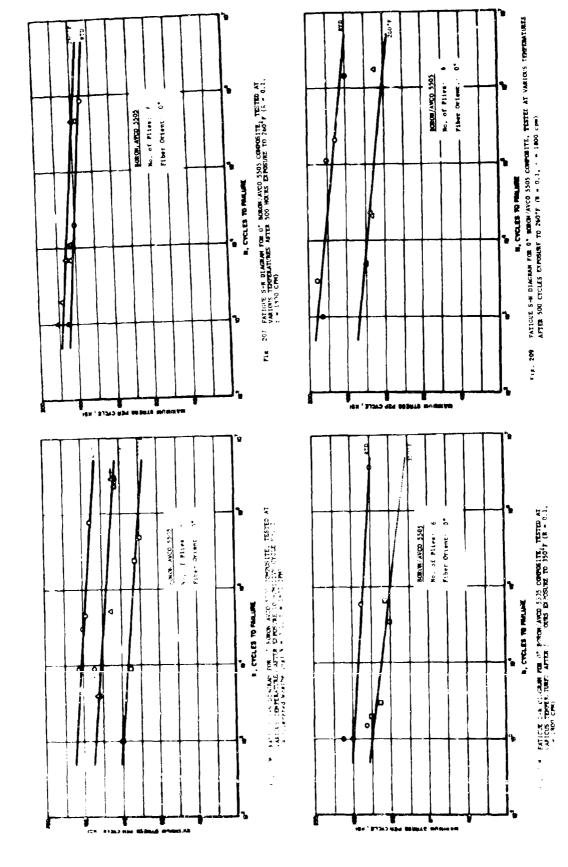
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					1				1			
	\$8.30 ₄₀₇	į		PRIOR	PRIOR COMUNICATION	100		: 	Cycles	Cycles Appli without	Residual	
	(P : cs) (ln.		Orten' #110n	1	Durat ion	ĖĒ	(L'ult) (kst)	(ks1)	failure (cycles)	Failure (cycles)	Streeth (bat)	Comment
- 4000	3 . 0 . 0	0.050	[3/45 3 15 46]	1, 097	/ 500 Hrs.	a L		. 04	000		,	
1.0 GH	2 - 0.051	150		+ ()4.		e	3	2.	261,000	•		.)
- 200	1.0.0	1,0	•	£ 09%	/ 590 H78.	911	97.	76	2.000	•	•	ı
- 500		651	-	1.092	/ 500 H-s.	Ê	ş	ž	2000	•	•	,
- E00	2 . 0.0	051	-	3.047	/ SOO Hrs.	Ë	95.	7.3	14,000			
10XC	. 0.0	050	-	15.13	/ 500 Hrs.	Ė	<i>3</i>	92	000 899	•	ı	1
- 100 0	0.050	0 . 0		÷	W.H. HTS.	Ē		S	000	. •	1 (• •
620	8 - 0.056	050		7507F	, SOO Hrs.	Ê	•		39,000	•	•	, ,
- 1500	050.0 - 6.050	050	•	151	/ 500 Hrs.	Ē	5		1.000	•		
- 100	3.0 . 0	0.0.0		<u>ا</u> الم	/ 500 Hrs.	Ē	ř	ş		2.0 × 10 ⁵	o.08	•
. 116.	•	950	-	: ·¥:	100 CVC	Ē	76	7.7	17 000			Tak Walthurs
- 1160.	٠,	ن کرن	-	. 6	200 CVC	Ê		ž		2 418x10 ⁶	7 62	
- 531A	۲ زان	۵۶ر		4,090	1 SIN CAR	Ė		Ç	,	0 57 12 10		
. 0513	3. J	0.050		3 - 097	/ 5 Cyc	É	 60	,0	2.000		9.0	ISD ATER PRITUE
23.5	J.)) , ()	-	26C°F	/ 500 Cyc	Ê	7,	68	1,504,000			Tab Area Failure
0314	د د	.50	•	3.092	/ 1000 CVC	e P		٧٠	4.000			
. 03LA -		151		26035	/ 1000 CVc	É	,	. 3	3,719,000			Tab Area Failure
0.514	9 ·	050	-	2667F	/ 1000 Cyr	Ê	0 /	NO CO	2,243,000			Tab Area Failure
0314	J\$0.0 · 0.05C	0,50	ž.	260°F	/ 1000 Cyc	Ê	r . 80	7;	3,000			
= ح د	ນ - ເ 0 ວ າ	0 50	-	280°F	/ 1000 Cyc	Ē	80	7.3	178,000			Tab Failure
0.00	\$0.0	150	-	350°F	/ 500 Cyc	Ê	11	ů,	2,000			
- P. CO.	. (.05)	.;		350'F	/ 500 Cvc	Q.	\$	<u>,</u>	10,000			
1 - VACO!	150.0 - 0.051	951	•	350°F	200 CA	Ē	9	., .,	2,000	•		Tab Failure
. M. M.	0.0	150	÷	350°F	ري √ 200 کيد	Ē	<u>ج</u>	.)	•	2.0 x 10°,	75.3	
.0318	2 - 0.C	153	Ξ	350°F	, 300 Cyc	ظَ	6.5	33	•	2.468 x 10°	6.11	
6160	۱ - 0.0	350	-	330.8	/ 1000 Cyc	É	82.5	5	2,000			
10318 - 1	0.0	350	•	35()}	/ 1000 Cyc	Ē	7t.5	65	413,000			
10314	S . 0.050	35	•	350°F	260 00C /	Ē	ä	3	و د :			
. 11605	050.0 - 0.050	8	=	350 F	/ 1000 Cyc	Ê	2	3	421,000			
0314	7 - 0.0	150	[.6/0/51/57/0]	350.4	/ 1000 Cyc	Ê	2	63	7,000			

FABLE VIII FATTOTE PROPERTIES SUMMARY - AVYO 5505 SORIN COMPOSITES

Specials	Thickness (Pites) (In.)	Orientation	PLIOR	PRIOR COMDITIONING Type Duration	ž į E	Stress Level (T ^o ult) (ksf.)	evel (kat)	Cycles to Failure (cycles)	Cycles Applied without Pailure (cycles)	Residual Strength (kai)	Comment
K10404 - 1	80°C - 6	(0. 5/135/0/ <u>90</u>)	Stende	260°F/ 506 Hrs	260 ₽	2	×	3.000			
٠	ı	[0/13/138/0/90]		Š		80	2	000.6			
#1060A - 3	9 - 0,051			8		7.7	9	25,000			
•	٠	-	Steedy			1,	3	195,000	•		
310404 - S	9 - 0,051	[04/25/135/0/90]	Steady	260"F/ 500 Hrs	. 260"F	\$	57		2.014 × 10°	70.7	
				741076							
40 40 E	10.01	0.0000000000000000000000000000000000000	Cyclic	260 /4-045	7.047	73	3:	77,000			
/ - SOLE	•			000 / 1.007		? i	6	,			Immediate Tab Failure
	•	[06/0/cs1/cs/0]		8		٠ <u>٠</u>	2	9,000			
- YOU'L	10.101	[0/1.5/135/0/90]		8		5.69	53	000,064			
01- 1000	•	[06/0/SS1/Cw01	Cyclic	280 F/ 300 Cyc.	. 280°F	75.5	79	66,000			
1 - 8040 K	9 - 0.350	[0//03/135/0/90]	Ceclic	260*F/1000 CVC	260°F	75.5	59	210,000			
N 0408 - 2	## CO	_				87.5	:	47,000			
X10408	•	_					. 20	000			
٠	9 - 0, 350	[0/15/135/0/90]				3	25	•	4. 344 x 106	63.7	
N1040B - 5	•			250*F/1000 Cyc.	. 260°F	81.5	2	11,000			
						ř	,	8			
0 - 0000 H	•				_	7.	2 :	38			
7 - 000	640 C - 6		C. C. C. C.	3 5		9 4	2 2	38			
	٠	100/0/SEL/ST/01		Ş		5 67	3	1 158 000			
	670 - 6	[0/12/133/0/40]		350 Mrs	350°F	Se	45	818,000			
		(00) 0) 36 () 3 () 0)			1000	7	,	6			
	•		Cyclic			16	2 4	98			
				5		; ;	2 4	26,36			
4 4 4		[0/(5/135/6/90]		350 F/ 500 Crc.	350"	200	3 %	200			
	•	0.00/0/5/11/5 //0/		8		. ec	3 3	200			
						3	3	2	•		
8 - A140EM	٠	[0//2/135/0/90]		350*F/1000 Cyc.		*	09	•	2.0 × 10 ⁶	0.08	Tab Failure
1.04th - 7	6 - 0.04	[0/(5/133/0/30]	Cyclife			103	65	382,000			
#.0614 - 8	•	[0/(5/135/0/90]	Cyc Ite	350°F/1000 CY		112	2	22,000			
	9 - 0,049	[0/4.5/135/0/90]		350" F/1000	350'F	109	89	295,000			
01- Y H	٠	[0/(2/132/0/30]		350°F/1000		117	73	14,000			





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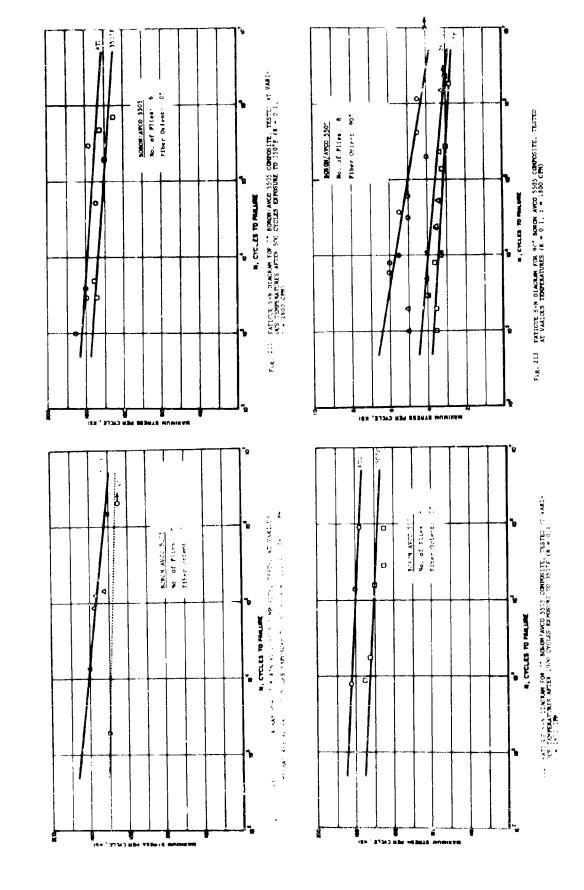
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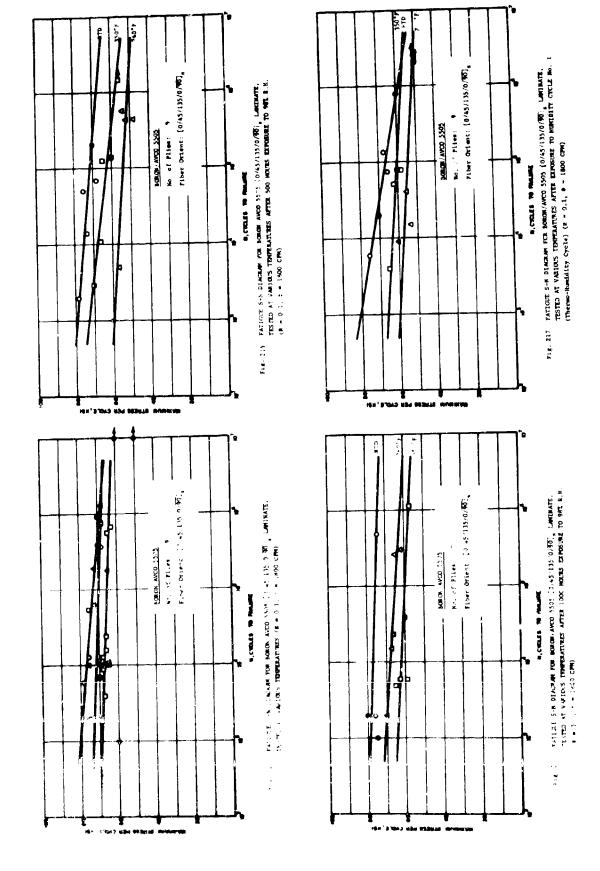
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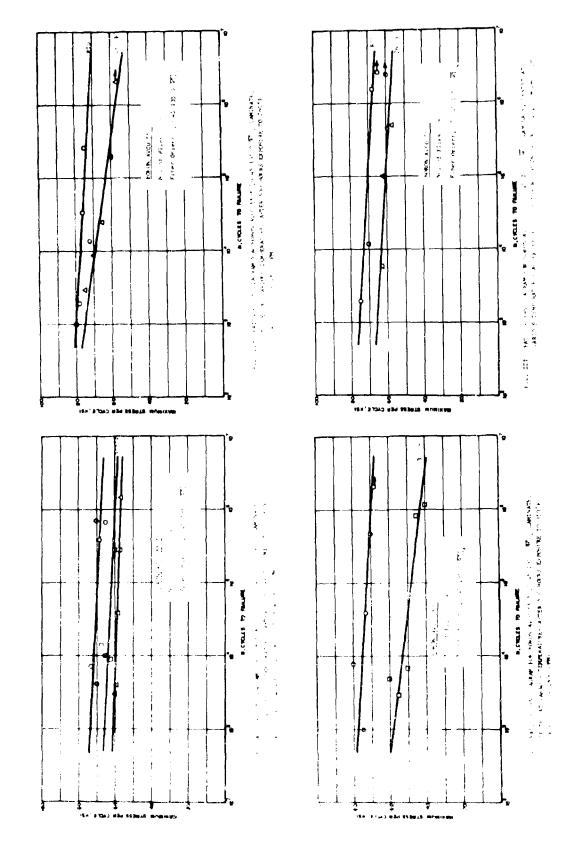
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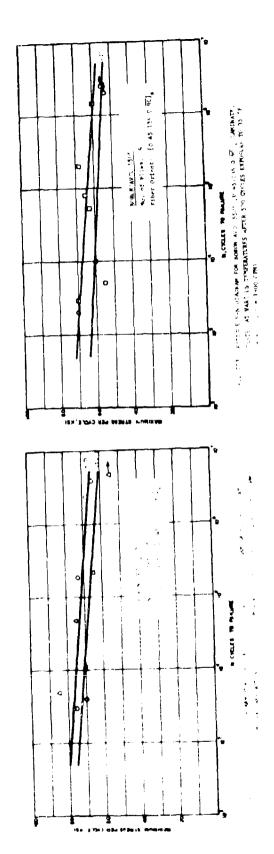
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TANTA XIV. GREEP AME STRESS RUPTEAR PROPERTIES STREET, AND AVOID 5505 FROMON, COMPOS, TTES

								!		
Spe Later Ragber	Thickness (Files) (In.)	: 1 eut at ion	PRIOR O	PLOR COMPITIONING Type Duratics	ž į į	Stres level		Timero Fallure (Hours)	Applied without Failure (Hours)	Comment
H.: 0078-11	6.037		ò		350°E	æ	571	ı	Ş	
W1007F-12	0.031	* ()	3		260	,	7.	1	2	
W10078-13	6 6.031				2.092	,	7 7	,		
11-21 65%	6 0.031	ڻ ن			760.	11		•	9	
FC 900 78 - 15	t 6,030		None		260°F	<i>;</i>	74.16	ı	1000	
#1007E-1	e 0.031	č	NOTE:		260°F	ύL	. 6	,	,	Instantaneous Facilities
#1 000 7A	. 60 3		Acase		760°F	-t	78		1000	
31-8700 DE	5 (.03)	č	Mone		260°F		J		1000	
5 9Z 00 TH	. (c. 031	• €	Mone		7.092	43.5	7.4	1	1000	
#10078-2C	0.031	£	Kone		760°F	1	À	ı	•	Failed during loading
#1006A-1	r 6,03;	·	Kon		350.1	80	6	ı	1000	
H1006A	0.031	i.			38C . F	**	66.3	•	1000	
K1006A-3	110°0	·	Kop		380.1	ř	 []	231	,	
H1006A	1 C 2 C	.,	Motor		320	ă	3.45	•	1001.4	
¥10008A - 5	٠ (٠ ل	ز	H CD.		350°F	36	, •	,	1000	
#1008A	6	*.,	NC P		350°F	7.	12-	6.1	ı	
K1008A-	£ €.02:	د ا	Bran e		350°F	ာ	14.	5,4		
#1006K-4	150.0	٠, ٠	None		320.1	Š	15.	∑9Z	•	
5-7800 LI	0.433	L	Mone		350°F	90	141		1011	
M1008A-10		٠,			350'F	9	<i>y</i>	1	•	Failed during loading
11.048-27	0.03	:,	NO.		350°F	Ĭ.	. 36:	ı	•	Broke during localing
1000 L	F 0.031	. c.	H ope		350 •	æ	14.		,	Broke during loading
1044-3	t (C3)		Hope		350 F	9	Š		•	Broke during loading

電視電視量の機能は関係のでは、これのできない。 との情報の主要を作りますのできる。はないのできませることでは、これでいることできませる。

AREA MIV CREEK AND THE COLD OF THE CASE OF THE SAME

							1'	: !	!	, <u>;</u>	
	Ē	Thicks see		PRIOR	PRIOR COMDITIONING	i i	Stress Level	lers 1	Tica to Fatlure	Applied without Failure	
Part Land	(7116	(P1fes) (In.)	Orientation	Type	Duration	3	(1°ult) (1:sf)	(l:st)	(Hours)	(Hours)	Coment
1 174e	0	30	[06/0/581/59/0]	None		260°F	70	58	ı	1038	
2-14/61	• •	8	W)	5		3 9 0, £	85	70.7	233	•	
1341-1	· o	8	:	None		3.096	80	56.1	•	1080	
7 17 12	•	8	:	None		260°F	7.5	Ç.	•	1001	
5- 1 100			=	None		3€ 397	•	ı	ı	•	Specimen broke during fabrication
* ********	۰	1 20 2	Ξ	į		4,097	ÜΣ	50	,	1006	
2-04-01	• •		:	- None		260°F	Ę.	56.1	979	•	
2 TALOU	•		:	NO.		3.092	,		•		Specimen broke during fabrication
0-440	0	050	:	000		260°F	980	56.1	ı	1006	
01-14CH	•	8	[0/45/135/0/ 9 0]	None		260°F	85	39.7	17.2	r	
		2	(00/0738173770)	ě		260°F	£		•	,	Specimen broke during loading
10.00 AC	·a	1	×	ď		260 F	8		r	•	Specimen broke during loading
20,000	·or	3	:	Noo.		260 F	95		•	,	Specimen broke during loading
7	•	8	:	Nore		260°F	63		•		Specimen broke during loading
S. 6401	· or	649	ŗ	Mone		2 60°F	92		•		Specimen broke during loading
11049-32	σ	840	Ξ	Kone		260°F	45		•		Specimen broke during loading
1 - 436 - 1	٠	2	(06/0/\$8() \$4/J)	Kone		350°F	70	35	20	•	Temp Control malfunctioned (J00°F
- 500		d	#	None		350°F	20	35	1.4		
F-3600	•	2		ou.		330.5	080	Ę.	618		
- XC31	•	0.0	ı	Kone		320°F	75	đ:	.87		
ne354-5	•	0 049	=	None		350°F	73.5	£7.88	r .	•	
4 1961	a	8	ε	N.		350°F	02	55.1			
110.33A-8	•	0	t	None		350 F	25	39.1			
1.01%-B	•	8	•	None		350°F	70	 	6.6		
- M.CO.	•	0	•	Mone		350°F	80	e n (1)	61.4	•	•
1:0354-10	٠	6 do 0	[0/45/135/0/901 ₈	None		350°F	8 0	~		ı	Instantaneous rallure
11049-37	•	0	[06/0/135/0/60]	None		350°F	86		•	,	Broken during loading
11041.21	•	0.0		Kone		350 F	9 6			1	
410a9-26	•	8		M One		350°F	95		1	•	Broken during loading
110-19-28	•	0	=	Mone		350°F	76			•	
1069-29	•	0.0		au de		350°F	85		•	•	Broken during loading
1,00,9-21	•	0,049	[06/0/5(1/59/0]	Mone		350°F		159.3	•	,	Overload

C. E. M.W., CALED AND SIGNESS REPORTED BY HER LESS SUSPANDED AND ANGLOSOF COMPANIETES.

								į	Time	
Spictmen Himber	Thickness (Files) (In.)	Ortentation	PRIOR C	PRIOR COMDITIONING Type Duration	Teat Temp. (*F)	Stress Level (T ^C ult) (ksf)	(kaf)	rick co Failure (Hours)	Applied without Failure (Hours)	Compens
#1 17-11	2 0	\$, V		260°F	20	05	ı	,	rathed during loading
M1 17-14	770	Ç	None		260°F) O	06.5	.033	,	Tab ayes falled
W1.17-15	E 30.00	0,7	Kone		260°F	06	2.90	.016	•	Tab area fatled
41 17-16	\$ 70.0	, 03	None		260°F	ě	3.43	413	•	Strain gage failed
M1.17-17	8 0,045	: 06	None		14. (4.)	980	5.25	•	•	Pailed during loading
81-21. 1k	8 0.045	, ()	None		260°F	80 80	5.72	•	1	load applied too quickly
1 - 601	8 0.045	, c6	Kone		260°F	85	5.58	•	,	Failed during loading
M1 18-2	8 0.043	.06	None		250°F	93	5.45	0.10	٠	
19-7	8	. 06	Kone		260°F	78	5,11	0.008	,	
M1 18-	8 0,045	•06	None		260°F	75	4.92	15.6	1	
W1 - 179-14	8 0.045	• 06	None		260°F	55	3.6		1005.3	Lost Strain Gage on Load
M 178-15	F 0.045	°06	Жле		260°F	65	F		1005.1	Lost Strain Gage on Load
M1:178-5	6 0.045	, O 6	None		350°F	202	2.13	1	•	Failed during loading
W1 178-6		: 06	Kone		350°F	20	2.13	110	•)
11.178-7		-06	Kone		350°F	75	2.28	541	1	
41 178-8	20.00	3	Kone		350 F	75	2.28	867		Strain gage failed after 193 hrs.
11 178-9		• 06	None		350°F	080	2.44	387	•	
W1 173-10		•06	None		350 F		2.44	209		
M1 178-11	8 0,045	° 8	None		350°F	85	2.58	3		
W1 18-72	0.045	.0	Mone		350°F	85	2.58	154.3	•	
		\$	aco.		350 F	8	2.74	126.8	•	
\$ - S	440.0	06	No.		350°F	8	2.74	14.8	•	

			:	!!				:		į į	
pecimen Munber	Th P11	Thickness Plies) (In.)	Ortentablon	add.	Type Duration	Trab.	"tras low!	(ks!)	11 to 545 June 19. cers)	Appl of without Failure (Heurs)	Comment
1-450	4	1807.9		ţ	· L.		,	ŗ	•	,	Tab failure - hrive during ' acting
12 T	٤		•	÷	, : · · · · · · · · · · ·		r	•		ı	TITE
1.048-3	4	11.631	-	ž	1(M) 11.7 .	•	;	÷		•	Tab failure - hrray turing loading
7-4703	4		=	•	i,	:	,	•	,	•	Tab failure - brake during loading
10.8-5	•	E - 1	Ξ	;		-	<i>t.</i>	<i>.</i>	1	ı	Tab failure - 'rake during leading
6.1-8-00.19	•	: -	3	ia 1	1 42	- - - -		-	17.4	1	
- (- K - 30)	٤.	C 34	c	# *		Į. Į		<u>.</u>	¥ \$ -3 *.	1	
_1-6506[1	(b)	c	## C	Total Street	<u>ب</u> ج	:	·.	,	1	Strain gage failed
#1-850C	£	(ε	¥.		-	ŗ	•	,	•	Oven overheated
71-85U(17	4	5000	E	.Ti ;≇ ™ ?	135 44	•		•	ì	1000	
ر. د. او د د د	4	16.	Ξ	7	, , ,	-,	,		Î	,	(wen overheated
11068-7	ı		c	д т	S.M. Here.	٠ + (•	,	1	•	Oven overheated
\$1.04 F - 8			С	I.	SON) HTV.	 	عر	=		•	Oven werheated
\$-43C	•	150.0	c	H.	. O HTA	3311 F	<i>;</i> -	₹	ī	•	Oven overheated
7₺€-13</td <td>1.</td> <td>(.631</td> <td></td> <td>, X</td> <td>SOFT HEN.</td> <td>3 JES</td> <td>.:</td> <td><u>\$</u></td> <td>ı</td> <td>•</td> <td>Oven overheated</td>	1.	(.631		, X	SOFT HEN.	3 JES	.:	<u>\$</u>	ı	•	Oven overheated
81010A-0	1	1	9	E SE T	, and hre,	199	~	<u>,</u>	1	,	Failed during loading
410104-6	£	0.033	ę	7 7	1 you liez.	5.5	;	15.4			
(- V) (0)	4		ç		1: 00 Prs.	100 F	J X	, ,	,	,	Failed during loading
10101-6	4	-1	عر	T.	1000 Hrs	404	í. X	<u>,</u>		•	Failed during loading
5-7.1-7.	4	¥.	-	¥,,,	1000 Hrs.	150 F	J.	1-3	ı	•	Failed during loading
11-841	٢	160.0	~	E A	The printer of the	4 (147	•	145.3	,	•	Split lengthwise
21-84U.	4	, n31	ث	上。由此	The pro-Harddity Cv. L.	14 (14)		· ·	,	•	Split lengthwise
E1-89-13	1	0.431	·ċ	L	THE AMERICA CO. In	3:092	٠	151.6	1		Tab failure - broke during loading
1.048-14	£	0,311	·C	FFT	Haridit.	3,042	yκ	154.5	•	1	Tab failure - broke during loading
.1-450	1		÷	10 L	The pro-Hamildity (Note)	240 8	†**. **	153.2	1		Tab failure - broke during loading

Number	Thick (Plies	Thickness (Plies) (in.)	Orientation	PATOR C	PRIOR CONDITIONING Type Duration	Test Temp. (°F)	Stress Level (2 ^c ult) (ks1)	Level (ks1)	Time to Failure (Hours)	Appled without Failure (Hours)	Comment
N1C48-16	9	0.031		Therra-Ilu	midity Cycle	350°F	65	114.0	1.50	•	Tab failure
N1048-17	•	0.031	.0	There-Hu	midity Cycle	350°F	79	112.8	0.82		Broke during loading - near middle
N10-8-18	ú	0.031	0 °	The Trans-Hu	midity Ovely	3.05k	63	5.011	0.47	•	Broke during loading - near middle
N1648-19	œ	0.031		at-CF-F	midity Cycly	350°F	<u> </u>	108.0	320		
N1048-20	φ.	0.031		Therro-Hu	Therro-Humidity Cycle	350°F	67	117.6	•	ı	Tab failure - broke during loading
M1048-31	φ	0.031	٥٥	The Hu	midity Cvely	350 F	38.5	68.3	•	1000	
95-870 LX	· «c	1100	°C	The Tried	midity Cycle	350 °F	90			1000	
N1048-34	.	0.931	. 0	Thermo-Hu	Thermo-Humidity Cycle	350°F	36	63	117.6	•	
N1 009B - 20	vo	0.033	0,	ACC. W	Wthrng.	₫,697	91	143	419	1	
N10:04-1	ve	0.033	,0	Acc. W	Wthrog.	3,60 ⋅ ₽	68	139	921	,	
N1010A-2	· •	0.033	0,0		Wthrng.	3,09€	87	136	,	1000	
N1010A-3	9	0.033	ô	Ac: W	Wthrng	3.00°E	85	130		1000	
41010A-4	•	0.033	•0	Acc. W	Withing.	3.09 ₹	83	130	11.4	•	
N10104-10	vo	0.034	.0	Acc.	Wthrng.	350 · F	06	131	,	ı	during
N1010A-11	40	0.034	0.		Wthrng.	350°F	92	134	ı		during
N10:04-12	5 0	0.033	•0		Wthrng.	350%	76	137	•	•	during
N1010A-13	vo	0.034	٥٥		Wthrng.	350°F	96	140	•		
N1010A-14	νo	0.034	.0	Acc. W	Wthrng.	320 ₺	80	143	i	•	Failed during loading
N1049-1	œ	0,049	[0/45/135/0/90]	/ EN %86	500 Hrs.	3,092	86	62	.016	1	Broke at tab
2-67018	o o	0.040				260° 5	97	78	2		Broke at Tab
V1049-3	ď	0.048	z	987 R	500 Hrs.	260°F	86	79	•	•	Broke during loading
31049-4								i	,		•
3-670TN	σ,	0.048	÷	987. RE /	500 Hrs.	260°F	80 6	79	-	ı	Broke at tab
410.012	σ	6.40	=		560 Hrs.	350°F	88	61.1	252.2	•	
7-0707	· o	870				350°F	90	62.5	183.2	•	Tab failure - slipped 1/8"
8-67017	۰ ۵	0.048	=	987 R		350°F	93	9.49	5,1	•	
6-6701k	, 6	0.00	=	987 R.	500 Hrs.	356°F	Şó	0.99	239.8	•	
11049-10	6	0.047	[0/45/135/0/90]	987. R.: /	500 Hrs.	350°F	86	68.1	168.9	,	

TABLE XET CREEP AND STRESS RUPITRE PROPERCIES FORMARY AVIO 5505 FORMON CACIPOSITIES

	rícetíon		loading Loading Loading		
Couperit	Oven overheated Spec. broke during fahricetion		Tab failure Specimen broke during loading Specimen Broke During Loading Specimen Broke During Loading Specimen Broke During Loading	Broke during loading Broke during loading Broke during loading	Broke during loading Broke during loading Broke during loading
Tim. Applied without Failure (Hours)	1000	1000	11111		
Time to Failure (Hours)	.043 \$1.5 - -	96.3 0.134 336			1.0 27 min - - 8 min
Leve 1 (ks1)	70.6 69.2 66.1 51.5	71.4 72.9 65.2 67.7 65.4	76.5 78 71.5 67.6 71.5	79.8 79.0 78.2	60.3 66.3 66.3 60.9
Stress Level	92 90 86 80	96 98 93 93 88	95 98 90 90 90	99 98 97 84	99899999999999999999999999999999999999
Test Temp. (*F)	260°F 260°F 760°F 20°F 200°F	350°F 350°F 350°F 350°F	260°F 260°F 260°F 260°F	260°F 260°F 260°F	350 50 FF
PRIOR CONDITIONING Type Duration	/ 1060 Hrs. / 1060 Hrs. / 1060 Hrs. / 1060 Hrs. / 1000 Hrs.	/ 1000 Hrs. / 1000 Hrs. / 1000 Hrs. / 1000 Hrs.	mmidity Cycle mmidity Cycle mmidity Cycle midity Cycle midity Cycle	midity Cycle midity Cycle midity Cycle	
PRIOR O	987. RH 987. RH 987. RH 987. RH	987, RH 987, RH 987, RH 987, RH	Thermo-Humidity Thermo-Humidity Thermo-Humidity Thermo-Humidity Thermo-Humidity	Thermo-Humidity Thermo-Humidity Thermo-Humidity	Thermo-Hunfalty Thermo-Hunfalty Thermo-Hunfalty Thermo-Hunfalty Thermo-Hunfalty Thermo-Hunfalty Thermo-Hunfalty
Oxtentation	[C/45/135/0/90]	*(<u>06</u> /0/\$£1/\$ 7 /0] ""	[0/45/135/0/ <u>90]</u>	" [5/65/135/0/90]	[05/0/511/54/0]
Thickness (Plies) (In.)	0.051 0.052 0.052 0.051	0.051 0.051 0.050 0.051 0.050	0.048 0.048 0.048 0.048	0.048	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2
Thic (Plies	0,000	ው ው ው ው ው	ማ ማ ም ማ ማ	თთი თ	ათთით თათ
Specimen Muncher	N10578-1 N10578-2 N10578-3 N10:78-4 N10578-5	M1058A-1 N1058A-2 M1058A-3 N1058A-4 A1058A-5	N102 9-11 N102 9-12 N102 9-13 N102 9-14 N102 9-15	N1049-31 N1049-34 N1049-35	N1069-12 N1069-13 N1069-19 N1069-20 N1069-22 N1069-38

TABLE XIV CREEP AND STRESS WIPTURE PROPUELLES FINHARY AVOX 5505/BORON COMPOSITES

. c

Congrent	Strain gage failed Strain gage failed		Failed in loading	Pailed in Loading	Failed during loading Pailed during loading Failed during loading Immediate failure Failed during loading	Immediate failure Strain gage failed after .25 hr. Immediate failure Immediate failure
Time Applied without Failure (Hours)	1001 1001	1901 1001 - 1001 1000	1001			1000
Time to Failure (Hours)	167.9 31.6 14.4	116	198	.014 .05 .067		9.
Stress Level (I ^G ult) (ks1)	94 69.1 96 70.6 98 72.1 90 66.2	93 53.6 95 65 98 67 90 61.5 88 60.1	70 118 75 127 80 135 85 144 90 152	95 144 92 140 90 137 88 134 85 129	93 167 94 169 90 161 91 163 89 160	93 162 80 139 92 161 86 153 90 157
1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	260°F 260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F	260°F 260°F 260°F 260°F 260°F	350°F 350°F 350°F	260°F 260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F
PRIOR COMDITIONING	Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng.	Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng.	Steady 260°F/500 Hrs. Steady 260°F/500 Hrs. Steady 260°F/500 Hrs. Steady 260°F/500 Hrs. Steady 260°F/500 Hrs.	Steady 260*F/500 Hrs. Steady 260*F/500 Hrs. Steady 260*F/500 Hrs. Steady 260*F/500 Hrs. Steady 260*F/500 Hrs.	Steady 350°F/500 Hrs. Steady 350°F/500 Hrs. Strady 350°F/500 Hrs. Steady 350°F/500 Hrs. Steady 350°F/500 Hrs.	Steady 350°F/500 Hrs. Steady 350°F/500 Hrs. Steady 350°F/500 Hrs. Steady 350°F/500 Hrs. Steady 350°F/500 Hrs.
Ortentation 7)	[0/45/135/0/90]s		0 0 0 0	0° S S C C O O O O O O O O O O O O O O O O	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	
Thickness (Plies) (In.)	9 0.051 9 0.051 9 0.051 9 0.051 9 0.051	9 0.036 9 0.036 9 0.036 9 0.036	6 0.034 6 0.034 6 0.033 6 0.033 6 0.033	0.034 6 0.033 6 0.033 6 0.033 6 0.034 6 0.034	6 0.033 6 0.033 6 0.033 6 0.033 6 0.033	6 0.035 6 0.035 6 0.033 6 0.033 6 0.033
Coclass	10378-6 10378-7 10378-8 10378-9	3.48.60 1.0384-8 1.0384-8 1.0384-9	1.0116-16 1.0118-17 1.0118-18 1.0118-19 1.0118-20	12012A-1 12012A-2 12012A-3 12012A-4 12012A-5	1,312A-16 1,312A-17 1,312A-18 1,312A-19 1,312A-20	F10128-1 F10128-2 F10128-3 F10128-4

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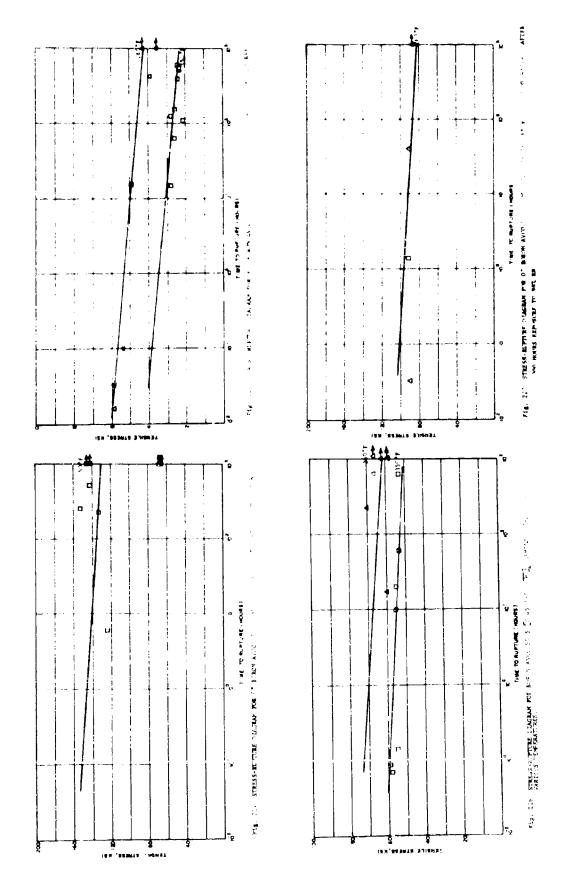
TARLE XIV. GREEP AND STRESS RUFITURE PROPERTIES STOWARY AND \$505 Horder Compositives.

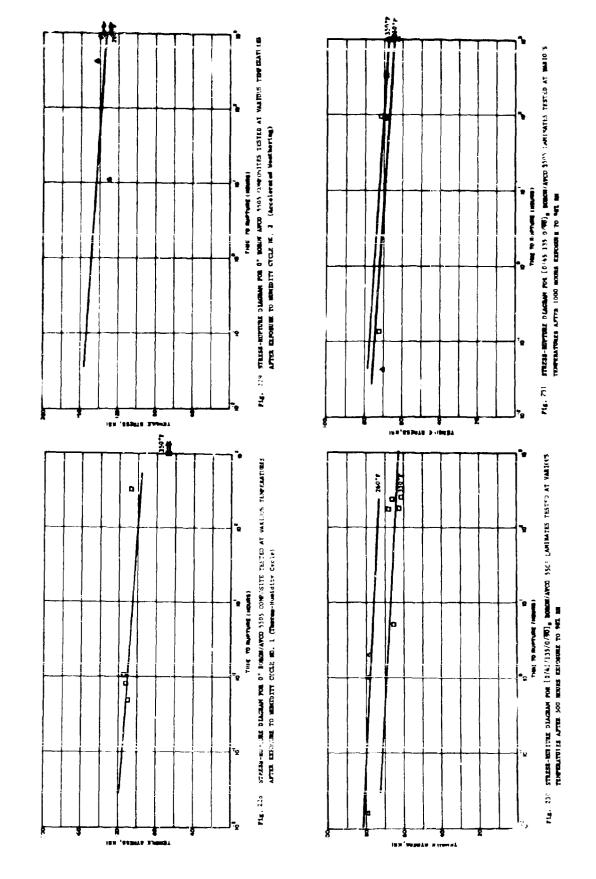
္အခရီ၁	Thie	Thickness		PRIOR COMDITIONING		Teat	Stress Level		Time to Failure	Time Applied without Feilure	
Marcher	(Plies	(Plies) (ln.)	Orientation	Type 2-re	P-ration	(£)	(Toule) (ket)	l	(Hours)	(Horrs)	Coment
M10418-	6	0.050	[0/45/135/0/90]	Steady 260°F/500 Hrs.	Mrs.	260°F	70	58.9			Ovens overheated, spec. delaminates
M10418-	6	6.0	10	Steady 260°F/50	S Hrs.	260°F	7.5	63.1	,	1007	
M0418-	6	0.049	=		N Hrs.	260°F	90	67.3	3.7	1	
H10413-	ታ	0.050	:	Steady 260"F/500	NO HES.	260°F	85	71.6	•	1032	
M1041F-	6	0.049	=		M Hrs.	260°F	06	75.8		•	Immediate failure
M10418-	•	0.051	ī	St : ady 260 F/50	X Hrs.	350°F	;}	6,	910.	,	
*1041B-	6	0.050	:	Steady 260°F/50	Mrs.	350°F	06	75.6	-:	•	
<u>୍କମ</u> ୪ ୦୮	6	0.051	:	Steady 260 F/560 P	So Hrs.	350°F	88	7.7	.33	•	Strain Gage Falled
C-81 401N	6	0.050	:	Stcady 260°F/500	M Hrs.	350°F	85	71.4	.51		Strain Gage Failed
C 81 401N	6	0 051	[0/45/135/0/90]	Steady 260°F/500	Mrs.	350°F	8 0	67.2	7.67	•	Strain Gage Failed
W10428-	0	0.052	[0/45/135/0/90]	Steady 350°F/500	X Hrs.	260°F	95	75.6	.01	•	
N1042B-	6	0.052	•		Mrs.	260°F	6 5	73.2	102		
31042B-	6	0.051		Steady 350 F/500 P	30 Hrs.	260°F	*	74.8	.016	•	
N10428-	6	0,051			M Hrs.	260°F	88	70.0	24.6	•	
31042B-	9	0.051	=		Mrs.	260°F	8	71.6	. 25	1	Strain gage failed
H10428-(ø	0.052	Ξ	Steady 350°F /500 P	Mrs.	350°F	Ł	75.7	910.	•	
K10428-	6	0.051		Steady 350*F/50	W Hrs.	350.7	96	72.5	.25	•	Strain gage failed
H10428-1	ø	0.051	=	Steady 350°F/500 1	N Hrs.	35C*F	88	6.07	16.5	•	,
K10428-5	6	0,050	=	Steady 350 7 /50	N Hrs.	350°F	76	67.7	42.6	•	Strain gage failed after .05 hr.
#1042B-	50	0.050	[0/45/135/0/90]	Steady 350 F /500	NO Hrs.	350°F	8 0	7,49	7.8	•	Strain gat: falled after 1 hour

TABLE XIV CREEF AND CONSISTENCY OF CIPPELLY SOMEON AND CONTRACTORS

Comment Failed in Loading ING ING ING ING ING Strain gage failed Strain gage failed	Failed in LA PRECONDITIONING P
Strain gage failed	
train	
train gage f	Strain gage
train gage faile train gage faile	Strain gage Strain gage
illed in Locaing	Failed in L
9	NING
ailed in Loading	Failed in LA
ven overheated	Oven overhea
10AGHU	

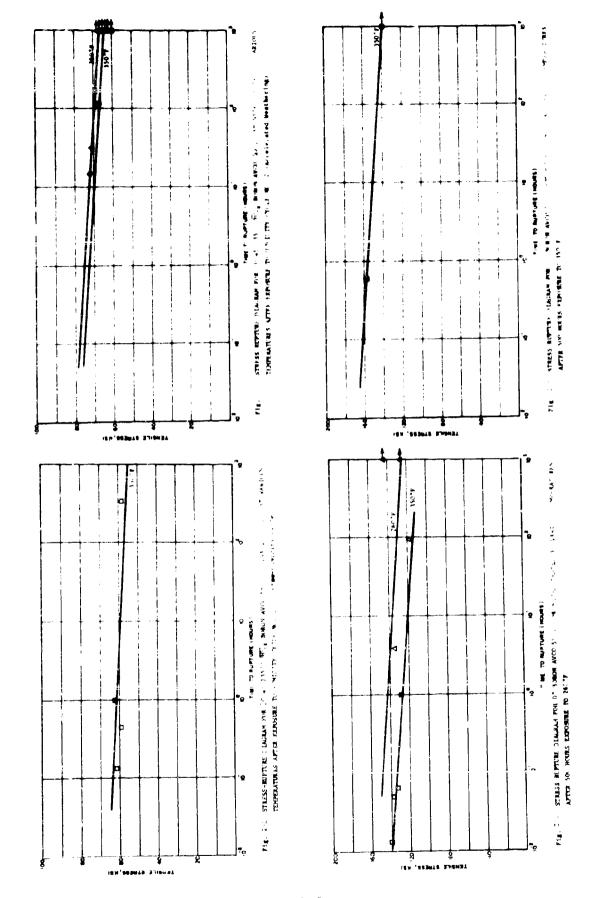
í					Test	ė	•	Time to	Time Applied		
Spectmen in Number (Pii	(Pites) (in.)	Orlentation	Type Duration	Duration	(•)	(Z ^d ult) (ksi)	Level (kai)	Failure (Hours)	Feilure (Hours)	Comment	
6 1-07501	0.050	(0/45/135/0/90		_	2607F	06	96.3		1000		
. 6	0.050			/ 1000 cvc.	260°F	95	70	887	,		
3	0.050	•		_	260°F	86	72.2	•	•	during	
5	0.050	:		_	260°F	\$	70.7			during	
41047C-5 9	0.050	=	Cyclic 350°F	/ 1000 cyc.	260'F	93	68.5	,		Broke during leading	
9	0.051	÷			350°F	85	53.1	25.7	•		
M1-420-7	0.051	=			350°F	83	51.8		1000		
6	0.050	:	Cyclic 350°F	/ 1000 cyc.	350°F	90 90	55		1000		
6	0.050	=			350°F	06	56.3	•	,	Oven overheated	
10426-10 9	0.050	[0/45/135/0/90]	Cyclic 350°F	/ 1000 cyc.	350°F	06	56.2	3.4	,		

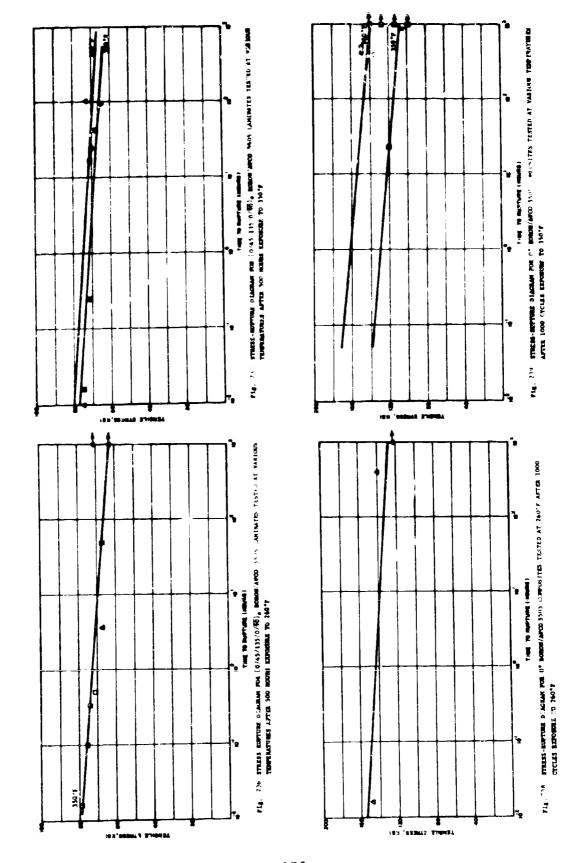


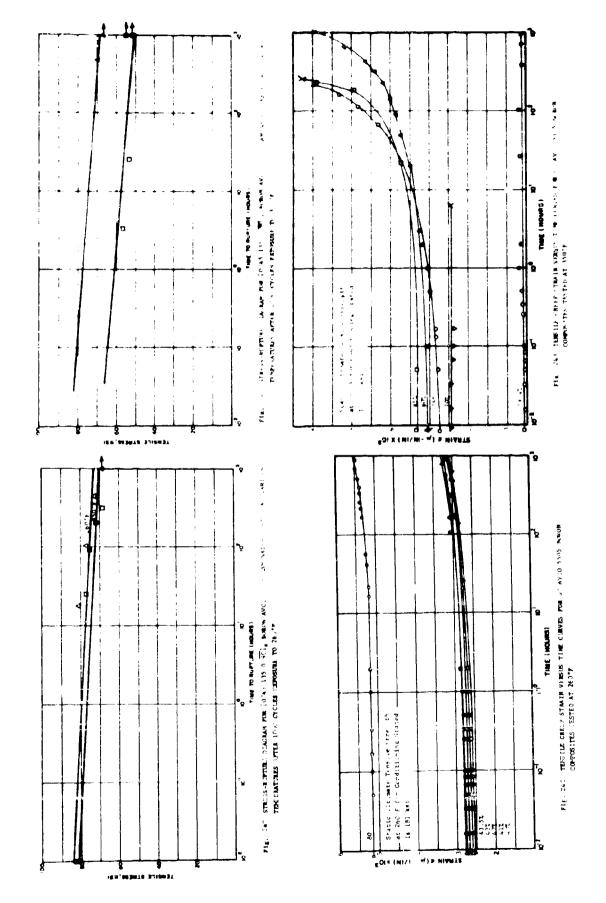


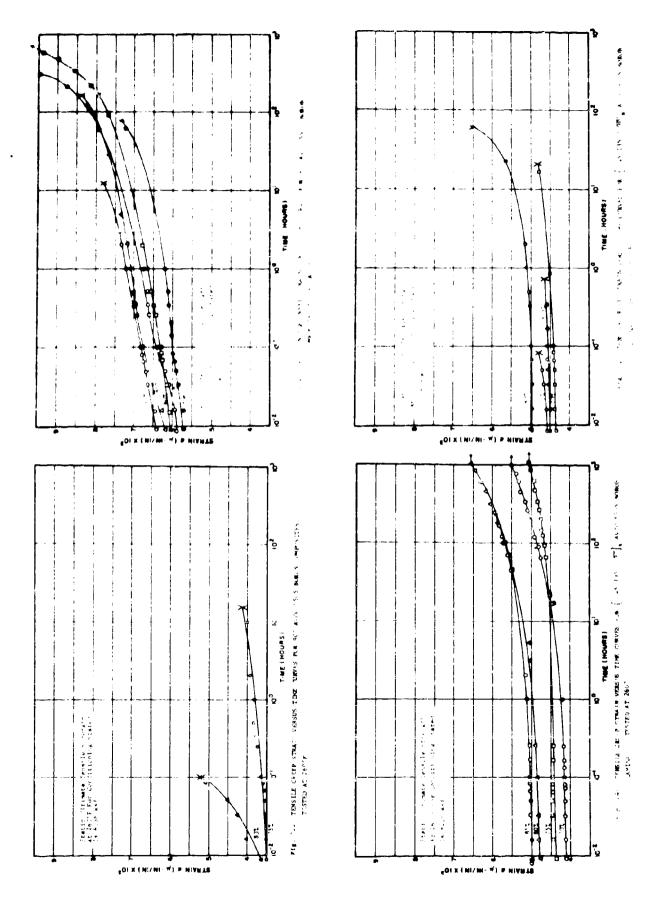
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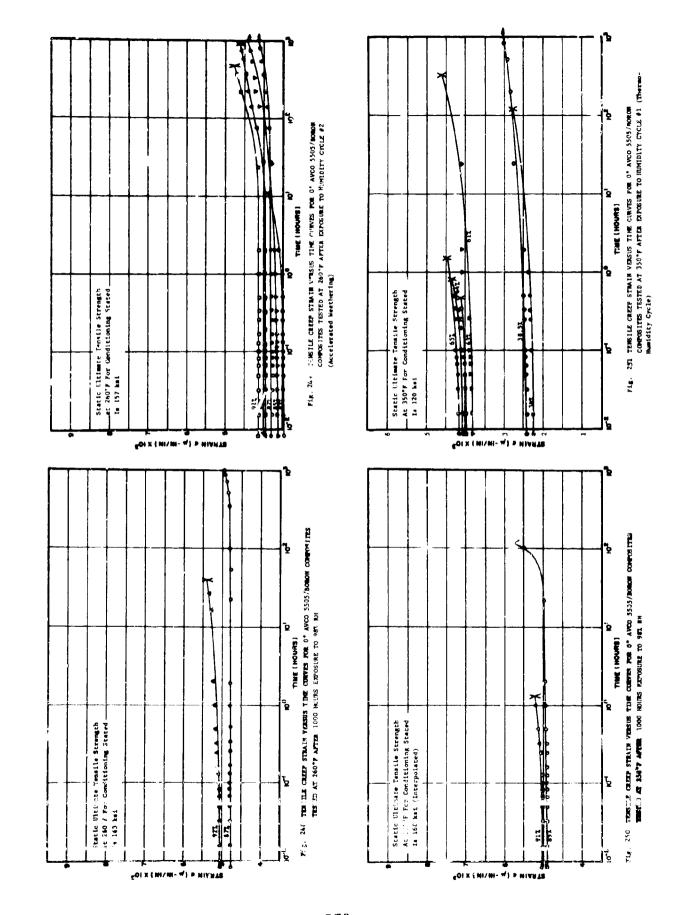
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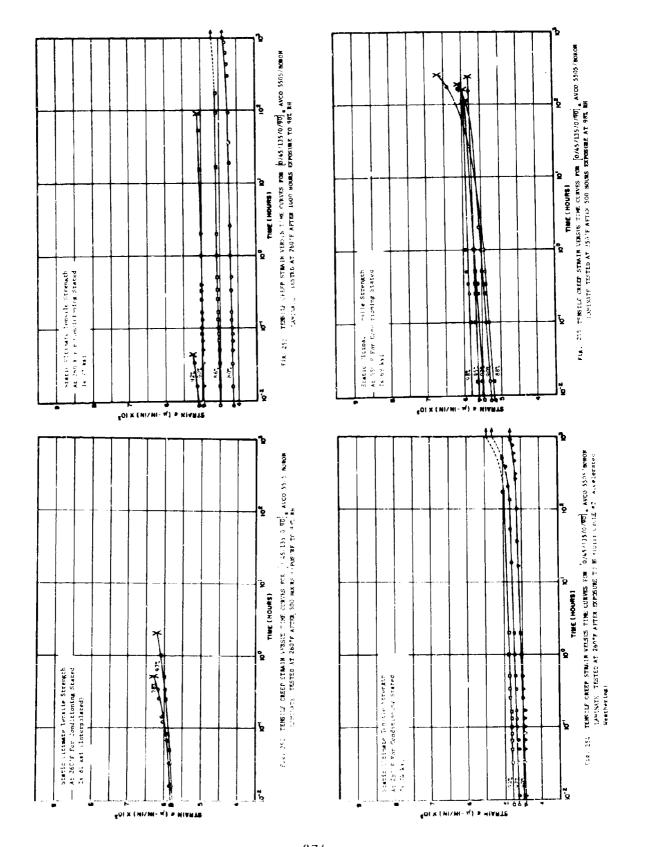
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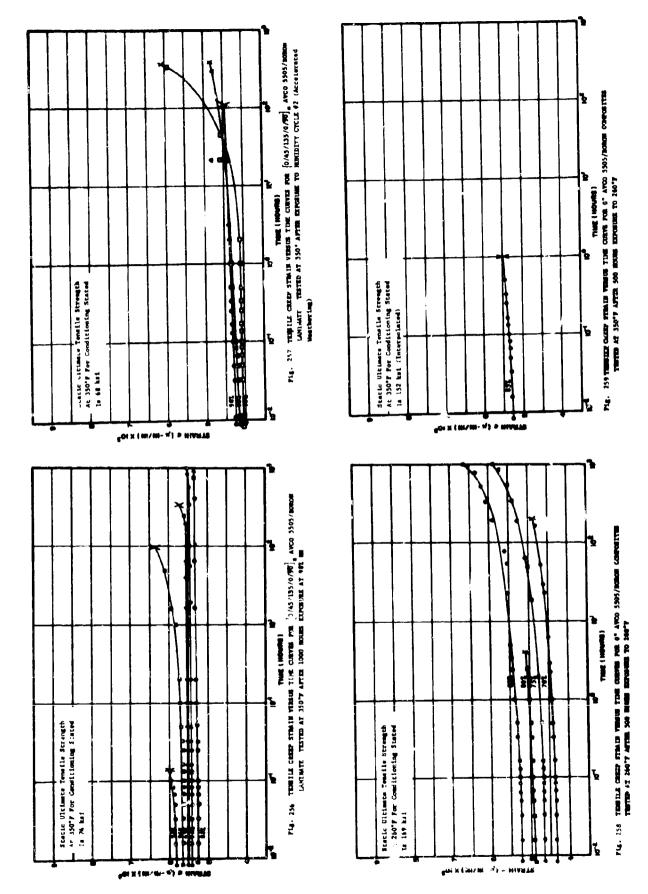


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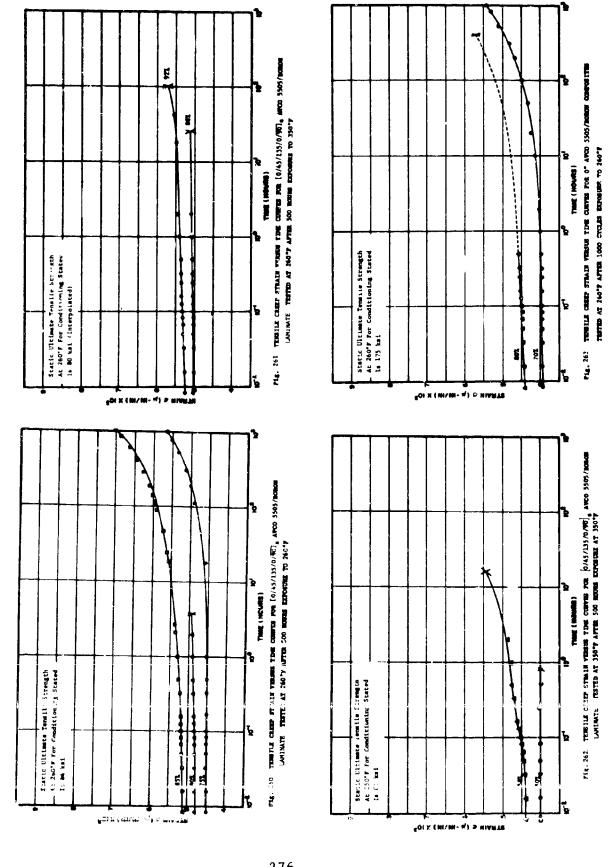
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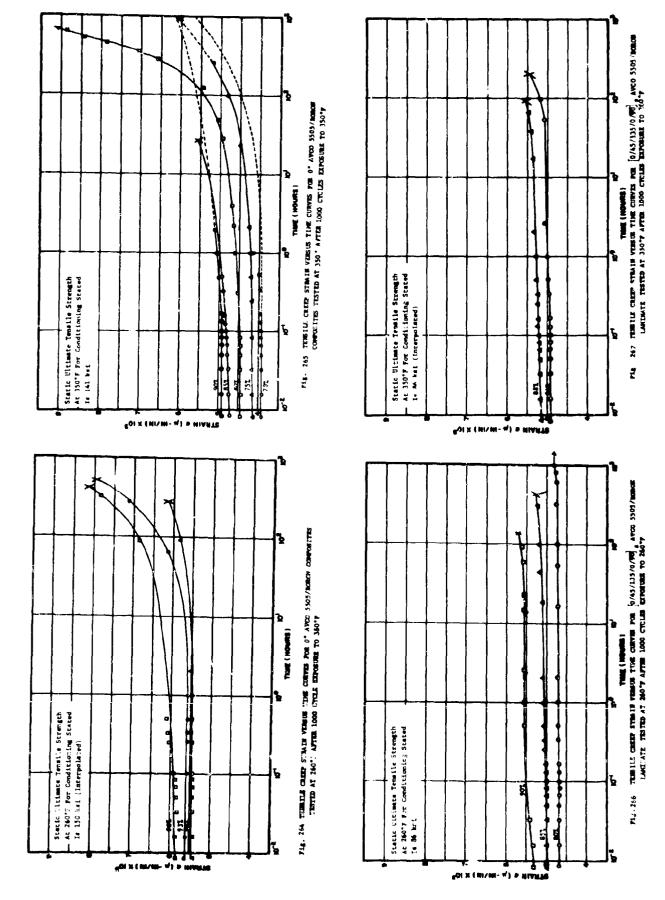
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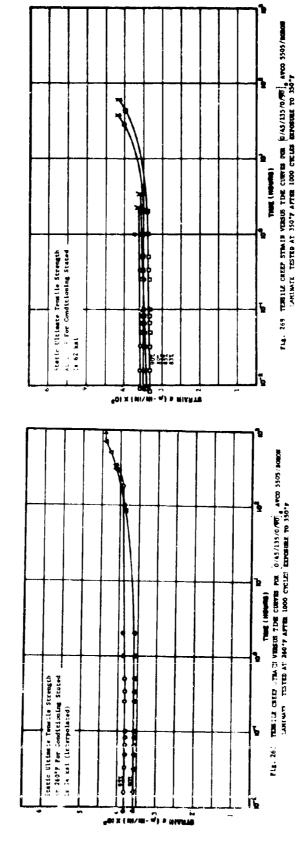
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APPENDIX II

DATA SUMMARY FOR MOI MOR II GRAPHITE/NARMCO 5206 COMPOSITES

HT RESEARCH INSTITUTE

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APPENDIX II

<u>Item</u>	Description	Pages
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STATIC PROPERTY STORMS CARNOOL STORMS CANNOT STORM THE CONTRACT THE CO

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Crientation	Type Coat	60. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1. 1	(F)	. 10°.	, (in/in)	Cult (kg1)	ευις (μ-ία./im.)
.0	Tens ivn	N DC N	KTD	22.5	0.30	191	6,920
ם ב	lension	None	3 ,09 ℃	22.5	0.26	150	6,780
0,	Tension	Nobe	350°F	25.0	0.23	150	6,040
÷06	Tension	57. 1	KIL	1.28	0.05	5.2	7,050
, O 6	Tension	7.00	3°0°£	Ĝ	6.05	3.1	3,130
²06	Tension	Mone	i°05£	0.81	0.05	3.0	4,510
/45/135/0/ <u>90</u>]	Tension	! 95	ETD	11.1	0.38	7.2	6,610
[0.45/135, 0.790]	Tension	3. J.	i,00?	21.6	0.45	87	7,200
45-133-6 <u>97.</u>]	_ension	None	356 8	or 	6.45	79	7,090
Č	ucissarinoj	Total	KŢD≠	£.8.	0.75	141	7,610
90	Corression	one.	RT D=	20.5	0.24	146	7,830
0٠	Compression	None	260°F	20.0	0.37	138	7,570
0.	Compression	Notice	350°F*	18.9	0.66	123	6,790
Û÷	Compression	None Section	320, ċ≠	19.8	0.25	129	6,150
。0 6	Compression	None	KT D*	1.71	0.03	24.7	20,340
306	(ompression)	X556	RTD*	1.47	0.01	24.9	16,970
,06	Conpression	None	260°F	1.06	0.01	21.1	530° 000
, 06	Company as ion	None Second	350 F*	1.20	0.00	17.3	000 °0€<
;					•	,	000

Orientation	Typ. load	Sittle 12 to 1	it Temp. (°F)		, (10/10)	Jult (k41)	ε _{ux} t (μ-in./in.)
[0/45/135/0/90]	Complession	None	# Q **,	· · · · · · · · · · · · · · · · · · ·	7.53	66	10,080
[0/45/135/0/96]	Compression	None	RTD*	 O	15.0	3	11,730
[0/45/135/0/90]	Compression	None	250°F	9.6	65.0	95	9,320
[0/41/135/0/90]	Compression	None	350°F*	7.01	95.0	. .	8,780
[0/45/135/0/90]	Compression	Nune	350°	9.1	٦,4	6.5	9,320
0	In-Plane Shear	None	RTD	•	ı	2 h	•
ီ၀	in-Plane Shear	None	260°F	•	•	7 3	
0	In-Flane Shear	None	350°₽		1	3.4	ı
Ĵ	Int. Shear	Nont	RID	,	1	12 9	i
0	Int. Shear	None	260°F	1	•	œ œ	;
0,	Int. Shear	None	350°F	ı	ı	5.1	٠
[<u>36</u> /0/981/35/5]	Int. Shear	Sone	RTD		•	5°.	•
[C '45' 135/0/ 9 C]	Int. Shear	None	260°F	ı	•	9.9	•
[5/45/135/0/ 9 6]	Int. Shear	None	350⁴₽	•	•	9.4	ŧ
°	Flexinal	None	CLD	•	•	188	•
Ô	Flexural	None	260°F	•	•	151	•
0	Flexiral	None	350°F	•	•	97	•
.05	Flexinal	None	RTD	•	i	1.7	•
,06	Flexiral	None	260°F	•	•	9 9	•
.06	Flexural	None	350°F	•	•	3.0	ı
<u>104/0/381/39/00</u>	Flexural	None	KTD	1	1	104	•
1 26/0/581 55/0	Flemaral	None	260°F	•	•	87	ı
[0/43/135/0/9E]	Flexoral	None	350°F	1	•	21	•

Sardwich Beam butt

Colectation	Type Load		T st Temp.	E (pol x 10 ⁶)	, (11/n;)	Jule ksf)	est (in. 'te.)
3 C	To sive			33.5	C.25	173	7,220
ئر	401842	84 105 HT 186	3.0 ·			173	•
٠,	Tension	98 3H/500 Hrs		,	•	166	•
a C	Tension	98° 34'1000 Hrs	KTD	-3.9	ú 25	182	7,350
ວູ	Tension	98' 3H 100 Hrs		23.3	0.44	163	6,710
.0	Tension	98] RH/1000 Hrs		6.7.	۰ 48	148	5,250
0 ء	Tension	Thermo-Humidity Cyrie	RTD	22.0	0.2-	163	7,230
٥,	_ension	Thermo-Sumidity Cyale	260 · F	•	•	175	,
٠, ٥	Tension	Thermo-Humidity Cvule	350°F	•	•	151	•
° 0	Tension	Acc. Withing.	RTD	6.12	0.28	179	7,870
, ,	Tension	Acc. wthrng.	260°F	0.	0.25	146	5, 520
ွင်	Tersion	84 300 Hrs	RID	1.35	,	4.0	3,010
,05	Tension	987 PH 500 Hrs	260°F		•	3.7	•
, Ot	Tension	987 RH 500 Hrs	350°F	•	,	1.6	,
. 05	Tension	98 . 3H 10C3 Hrs	RTD	1.17	00 O	2.8	2,350
, 0 5	uoisual	98% RH 1000 Hrs	260°F	0.35	0.0	1.9	4,500
°C S	Tension	98% RH 1000 Hrs	350°F	0.28	0.00	1.2	5,690
- 0 ú	Tension	Thermo-Humidity Cycle	RTD	1.13	0.00	3.2	2,830
, Op	_ension	Thermo-Humidity Cycle	260°F		•	1.5	•
, 050	Tension	Thermo-Humidity Cycle	350°F	•	•	1.0	

TABLE XV STATIC PROPERTIES SUPHARY - NARMCO 5206/MODHOR II GRAPHITE COMPOSITES

^c ult (μ-fn./ in.)	2,360	2,870	Ş	0*			90	01	3	29			2	3	ş
13 - H	2,3	2,8	5,0	7,1	· •	•	6,830	6,7	6,160	9.	•	•	7,630	0,940	5
oult (kat)	2.9	1.7	1.3	1	18	3	5	11	65	2	18	75	82	22	3
, (16/12)	0.0	0.0	8 .0	0.39	•	•	0.40	0.48	0.38	0.42	•	•	0.45	0.42	57 0
E (psi x 10 ⁶)	1.25	0.71	0.36	11.7	*	•	11.7	11.3	11.4	11.5	•	•	10.9	11.4	11.1
Test Temp. (*?)	RTD	260°F	350°F	RTD	260°F	350°F	ECT.	260°F	350°F	8	260°F	350*F	₽	260°F	350°F
Prior Conditioning	Acc. Wthrag.	Acc. Wthrng.	Acc. Wthrng.	98% RR/500 Hrs	98% RH/500 Hrs	98% RH/500 Hrs	987 KN/1000 Hrs	98% RH/1000 Nrs	98% RH/1000 Hrs	Thermo-Bueldity Cycle	Thermo-Bumidity Cycle	Thermo-Rumidity Cycle	Acc. Wthrng.	Acc. Wthrng.	Acc. Wthrng.
Type Load	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tension	Tenston	Tension
Crientifon	06	<u>.</u>	: 06	[0/45/155/c/90]	[0/45/155/c/ 90]	[0/45/1:5/0/90]	[0/45/1[5/0/90]	[0/45/125/0/90]	[0/45/155/0/90]	[0/45/1[5/0/ <u>90</u>]	[0/45/155/0/ 90]	[0/45/115/0/90]	[0/45/115/0/90]	[0/45/115/0/ 90]	[0/45/113/0/90]

Orientation	Type L'ad	Prior Conditioning	Test femp.	E (psi x 10 ⁶)	, (1n/1a)	oult (kei)	^e ult (μ-in./in.)
,0	Compression	98% RH/500 Nrs	QD	,	•	141	4
•0	Compression	98% RH/500 Nrs	260°F		•	143	
•0	Compine 8 Lon	98% RH/500 Hrs	350°F	•	,	134	•
•0	Compression	98% RH/1000 Nrs	e ta	21.2	0.26	134	9,450
• 0	Compre: s ton	987 RH/1000 Hrs	260°F	16.8	0.28	127	009'9
• 0	Compression	98% RH/1000 Hrs	350°F	17.1	4.0	125	7,420
• 0	Compression	Thermo-Humidity Cycle	RTD	18.4	0.22	136	8,000
• 0	Compression	Thermo-Hamidity Cycle	260 ℃	•	•	138	•
•	Compression	Thermo-Humidity Cycle	350°F	•	•	132	ı
• 0	Compression	Acc. Wthrng.	RTD	19.8	•	8 21	099.9
* 0	Compre: ston	Acc. Wthmg.	260°F	20.6	16.0	140	8,850
• 0	Compression	Acc. Wthrng.	350°₽	19.6	0.¥	721	7,070
•06	Compre: sion	98% RH /500 Hrs	£	1.12	0.00	23.1	21,630
.06	Compre: sion	98% RH /500 Mrs	260°₽	•		13.8	•
•06	Compression	98% RM/503 Mrs	350°₽	•	•	7.6	•
• 06	Compre: # fon	98% RH /1000 Ers	œ.	1.30	0.00	22.4	24,600
• 06	Compression	98% RH /1000 Rrs	260°₽	1.28	0.01	15.2	20,260
°06	Compression	98% RH /1000 Hrs	350°F	#	‡	:	‡
•06	Compre. sion	Thermo-Humidity Cycle	e e	1.38	0.00	23.5	19,260
.06	Compression	Thermo-Humidity Cycle	260⁴₽	•	1	14.6	•
•		Thomas Beatle (at the Can le	3500	•	•	1. e	•

**Specimen Eroken During Hendling

TABLE NV STATIC PROPERTIES SUPPARY - NARMOD 5206/MODMOR II SRAPHIC COMPETTES

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Orientation	Type Load	Prior Conditioning	Test Temp. (*F)	E (psi x 10 ⁶)	, (in/in)	dult (ks1)	e.it (μ-in./in.)
.06	Compression	Acc. Wthrng.	RTD	1.25	00.00	24.1	21, 320
•06	Compression	Acc. Wthong.	260°F	1.63	0.00	18.0	21,040
•06	Compression	Acc. Wthong.	350°F	99.0	0.00	9.2	28,420
[0/45/135/0/ <u>90</u>]	Compression	982 RH/500 Hrs	RTD	\$.£	0.42	88	10,100
[0/45/135/0/90]	Compression	98% RH/530 Hrs	260 °F	•	,	87	•
[0/45/135/0/90]	Compression	98% RH/500 Brs	350°F		ı	2	•
[0/45/135/0/90]	Compression	987 RH/1000 Hrs	RTD	8.86	0.37	88	10, 380
[0/45/135/0/90]	Compression	98% RH/1000 Hrs	260°F	10.32	97.0	2	9,590
[0/45/135/0/ <u>90</u>]	Compression	981 RH/1000 Hrs	350°F	9.62	0.45	82	6,880
[0/45/135/0/ 90]	Compression	Thermo-Humidity Cycle	RTD	9.60	0.43	85	11,030
[0/45/135/0/90]	Campression	Thermo-Numidity Cycle	260°F	,	•	3	•
[0/45/135/0/ <u>90</u>]	Compression	Lacran-Humidity Cycle	350°F	•	•	87	•
[0/45/135/0/90]	Compression	Ase. Wthrng.	RTD	9.86	97.0	06	11,100
[0/45/135/0/90]	Compression	Acc. Wtheng.	260°F	9.55	0.39	98	10,170
[0/45/135/0/90]	Compression	Acc. Wthrng.	350°F	9.01	0.37	80	10, 200
•0	In-Plane Shear	98% RH/500 Hrs	RTD	0.72	•	8.6	18,330
•0	In-Plane Shear	987 KH/SCO Hrs	260°F	ı	•	7.3	•
•0	In-Plane Shear	987 RH/500 Hrs	350°₽		•	6.0	•
•0	In-Plane Shear	98% RH/1000 Hrs	£	0.74	ŧ	4.8	23,000
•0	In-Plane Shear	987 RH/1000 Hrs	260°F	0.17	•	5.6	> 30,060
•0	In-Plane Shear	987 RH/1000 Hrs	3.20€	90.0	•	3.8	> 30,000

TABLE AV STALLE PROPERTIES STAMARY - TENNES \$20K PROBOK IL STALE TANANS FES

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Orientation	Tyne Load	or ioning	Test Temp.	G (ps1 x 10 ⁶)	Tult (ks1)	^ε ult (μ-in./in.)
30	In-Plane Stear	Thermo-Humidity Cycle	RID	99,0	4.8	26,730
0.0	In-Plane Shear	Thermo-Humidity Cycle	3.00°E	•	4.9	•
,0	In-Plane Shear	Thermo-Humidaty Cycle	350°F	1	4.4	•
0 0	In-Plane Shear	Acc. Wthrng.	RTD	97.0	8.8	23,870
	In-Plane Shear	Acc. Wthrmg.	260 °F	0.31	9.9	> 30,000
٥٥	In-Plane Shear	Acc. WELLIS.	350°F	0.14	4.5	> 30,000
.0	Int. Shear	987 RH/500 Hrs	RTD	•	10.4	•
۰۵	Int. Shear	85, 786 Hrs	50°F	1	6.4	•
0,	Int. Shear	98% RH/500 Hrs	350°F	ı	5.6	,
.0	Int. Shear	98% RH/1000 Hrs	KTD	•	9.3	,
Ô	Int, Shear	98% RH/1000 Hrs	260°F	í	7.4	ı
ů.	Int. Shear	98% RH/1000 Hrs	350°F	·	5.5	ı
٥	Int. Shear	Thermo-Humidity Cycle	RTD	•	9.8	•
٥	Int, Shear	Thermo-Humidity Cycle	260°F	•	6.5	•
•0	Int. Shear	Thermo-Humidity Cycle	350°F	•	4.3	•
. 0	Int. Shear	Acc. Wthrng.	RTD	•	11.8	•
° 0	Int. Shear	Acc. Dthrng.	260 ₽	•	9.2	•
.0	Int. Shear	Acc. Wthrng.	350 ⁴F	ı	5.8	•

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FASTE XV STATIC PROPERTIES SUMMARY - NARMOO 5206/MOUMOR II GRAPHITE COMPOSITIES

Orientacion	Tvpc Load	Prior Conditioning	Test Temp, (°F)	E (psi x 10 ⁶)	, (in/in)	dult (ksf)	e _{ut} (μ-fn./fn.)
.0	Tenston	260°F/100 Hrs	KTJ	22.5	0.25	171	7,500
٥	Tension	260°F/100 Hrs	260°F	23.0	0.32	170	7,340
٥	Tension	260°F/500 Hrs	RTD	22.4	0.28	170	7,330
°C	Tension	200°F/500 Hrs	260°F	22.6	0.27	163	7,120
,0	Tension	350°F/100 Hrs	кТD	22.2	0.28	167	7,180
,0	Tension	350°F/100 Hrs	260°F	22.0	0.28	146	6,570
,0	Tenston	350°F/100 Hrs	350°F	23.5	0.24	160	6,630
٥٠	Tension	350°F/500 Hrs	Ę	23.0	0.26	173	7,410
°0	Tension	350°E/500 Hrs	350°F	23.4	0.21	159	6,450
₌ 06	Tenston	250°F/100 Hrs	RTO	1.12	0.00	4.3	3,770
06،	Tension	250°F/100 Hrs	260°F	1.02	0.02	3.6	3,360
۰06	Tension	260°F/500 Hrs	RTD	1.15	0.01	4.0	3,590
06ء	Tension	260°F/500 Hrs	260°₽	1,09	0.02	3,4	3,280
06،	Tension	35G*F/100 Hrs	Œ.	1,14	0.0	3.8	3,370
° 0€	Tension	350°F/100 Hrs	260°F	0.95	0.01	2.0	2,190
°06	Tenston	350"F/100 Hrs	350°F	0.92	0.02	2.0	2,120

TABLE AV STALLC PROPERTIES SUSMARY - NARMO 1 // NOBYOR II GRAPH EL JEOSTIES

Orientation	Type Load	Prior Conditioning	Test Temp. (*F)	E (ps1 x 10 ⁶)	, (In/In)	dult (keij)	^e ult (μ-tn./1α.)
•06	Tension	350°F/500 Hrs	rg.	1,14	0.01	2.7	2,310
• 06	Tension	350 F/500 Hrs	350*	1.05	0.02	1.8	1.740
[0/45/135/0/90]	Tension	260 F/100 Hrs	E E	10.9	0.36	82	27.300
[0/45/135/0/90]	Tension	260°F/100 Nrs	260°F	11.8	6.41	. 5	7 %
[0/45/135/0/ <u>90</u>]	Tension	260*F/500 Brs	RTD	11.7	0.40	. . .	7.330
[0/45/135/0/90]	Tension	260*F/500 Hrs	260°F	11.4	0.37	9	7, 130
[0/45/135/0/90]	Tension	350°F/100 Hrs	ET.	11.3	0.41	82	7,210
[0/45/135/0/90]	Tension	3.50°F/100 Hrs	260°F	11.3	0.43	\$	27.7
[6'45/135/0/ <u>90</u>]	Tension	350°F/100 Hrm	350⁴₽	,	•	- 	<u>;</u>
[0/45/135/0/ <u>90]</u>	Tension	350"F/500 Hrs	£	11.8	3 .0	2 02	Ore ?
[0/45/135/1/90]	Tension	350*F/500 Hrs	350⁴₽	12.4	0.51	78	90.9
• 0	Compression	250°F/100 Hrs	CD)	•	•	%	
•0	Compression	260°F/100 Hrs	260 ⁴F	ı	•	148	,
•0	Compression	260 P /500 Hrs	ē	19.3	0.32	134	7.950
• 0	Compression	260°F/500 Hrs	260 ° ₽	14.0	0.29	121	9,900
• 0	Compression	350°F/100 Hrs	E	16.2	0.31	115	7,740
•0	Compression	350*F /100 Hrs	3≥0 •₽	•	•	ž	1
•0	Compression	350°F/500 HE	ē	19.2	0.30	124	7.290
• 0	Compression	350*F/500 Rrs	320.1	17.5	0.27	123	7,390

COLE N STATIC PROPERTES SUPEARY - NARMO 5706/MCNOR II GRAPHIC CONFOSTITS

O rl cat ation	Type Load	Perior Conditioning	,	E (ps1 x 10 ⁶)	, (fa/la)	oult (ks!)	enle (1-in./in.)
.06	Compression	260°F/100 Hrs	ATD	•		25.6	4
, 0 6	Compression	260°F/100 Hrs	3.00°F	•	ı	21.8	•
93.	Compression	260°F/500 Hrs	ET.	1.17	0.01	24.7	22,920
•06	Compression	260°F/500 Hrs	26 6°F	0.90	0.00	19.1	26,330
90.	Compression	350°F/10C Hrs	KT	•	•	26.0	•
ي ئن ئ	Сощртенвіоп	350°F/100 Hrs	350°	•	1	16.5	•
90°	Compression	350°F /500 HT8	RTD	1.07	0.01	22.5	16,610
. 06	Compression	350°F /503 Hrs	350°F	1.01	0.01	6.71	25,970
[0/65/135/0/ <u>90</u>]	Compression	260°F/100 Hrs	Œ.	•	•	56	•
[<u>06</u> /0/\$£1/5 7 /0]	Compression	260°F/100 Hrs	260°F	•	•	93	•
[<u>06</u> /0/\$£1/\$ 9 /0]	Compression	260*F/500 Hrs	KTD	9.07	97.0	*8	10,480
[06/0/22/132/0]	Compression	260*F/500 Hrs	260°F	7.84	0.40	11	11,500
[0/6/0/32/0/ <u>00</u>]	Comp ?ession	350*F/100 Hr3	Ę	•	ı	93	,
[06/0](T / E7/0]	Compression	350°F/100 Rrs	350°F				
[0/45/135/0/90]	Compression	350 F /500 Hrs	Ę	9.38	0.42	Z	10,610
[0/45/135/0/90]	Compression	350*F/500 Hrs	350.1				

LARLE XV STATIC PROPERTIES SUMMARY - NARRY S 106 /MOUNOR IL LARPITC COMPOSITES

Ortentation	Type Load	Prior Conditioning	Test Temp. (°F)	G (psi x 19 ⁶)	¹ ule (3:1)	eult (μ-in./in.)
c)	In-Plane Shea:	260°F/1.00 Hrs	RTD	•	4.8	•
ر)	In-Plane Sheam	260°F/109 Hrs	260°F	•	8.1	1
ر،	In-Plane Shear	260°F/500 Hrs	RTD	0.70	9.6	22,030
٠,)	In-Plane Shear	260°F/500 Hrs	260°F	9.51	7.6	> 30, 000
ن	In-Plane Shear	350°F/100 Hrs	RTD	•	7.8	1
ຳ	In-Plane Shear	350°F/100 Hrs	260°F	•	7.4	ı
ပိ	In-Plane Shear	350°F/500 Hrs	€	0.70	8.3	25, 500
ຳ	In-Plane Shear	350°F/500 Hra	350°F	0.47	5.5	> 30° 000
ບໍ	Int. Shear	260°F/100 Hrs	RTO		11.1	i
ပံ	Int. Shear	250°F/100 Hrs	260°F	•	9.6	ı
ຳ	Int. Shear	260°F/500 Hrs	Ē	•	11.9	r
ໍ້ບ	Int. Shear	260°F/500 Hrs	260°F	•	8.7	ŧ
ů	Int, Shear	350°F/100 Hrs	£ £	•	12.4	ù
ບໍ	Inc. Shear	350*F/100 Hrs	260°F	•	7.6	ē
ໍບ	Int. Shear	350°F/500 Hrm	£	,	11.9	í
ů	Int. Shear	350°F/500 Hrs	350°F	•	4.5	Ç

TABLE XV STATIC PROPERTIES SEWARY - NARMO SZEKZWOWOR TI GRAPHIC CHYPN ITES

Oriencetion	Type Load	Prior Conditioning	Test Temp. (°F)	E (ps1 x 13 ⁶)	(a1/a1)	oult (kai)	^ε ult (μ-1n./1n.)
.0	Tension	260°F/500 Cy.	Œ	•	•	145	•
٥	Tenaton	260°F/500 Cy.	260°F		•	153	ı
٥	Tenston	260°F/4000 Cy.	Ē	21.2	0.30	165	7,440
. 0	Tension	260°F/1000 Cy.	260°F	23.7	0.19	156	6,470
ô	Tension	350°F/500 Cy.	Ę	•	•	158	•
	Tension	350°F/500 Cy.	260°F	ı	•	165	•
• 0	Tension	350°F/500 Cy.	350°F	•	•	158	•
٥٠	Tension	350°F/1000 Cy.	ę	21.3	0.27	15.	7,150
٥	Tension	350°F/1000 Cy.	350°F	23.6	0.24	071	7,160
.06	Tension	260*F/500 Cy.	Ę	,	•	4.2	•
့ ၁ ₆	Tension	260°F/500 Cy.	260°F	•	ı	3.9	1
.06	Tension	260°F/1000 Cy.	ę	1.22	0.00	9.4	3,680
.06	Tension	260°F/1600 Cy.	260°F	1.12	0.00	4.3	3,830
06،	T. ston	350°F/500 Cy.	£	•	•	3.6	•
.06	Tension	350°F/500 Cy.	260°F	•	•	3.0	•
.06	Tension	350°F/500 Cy.	350°F	•	•	2.2	•
•06	Tension	350*F/1000 Cy.	£	1.20	0.0	3.0	2,636
: 0	Toneton	350°F/1000 Cv.	350 %	3.0	0.0	2.7	2,913

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TABLE XV STATIG PROPERTIES SCHARP SARANGE II GRAPHIC CUMPSSITES

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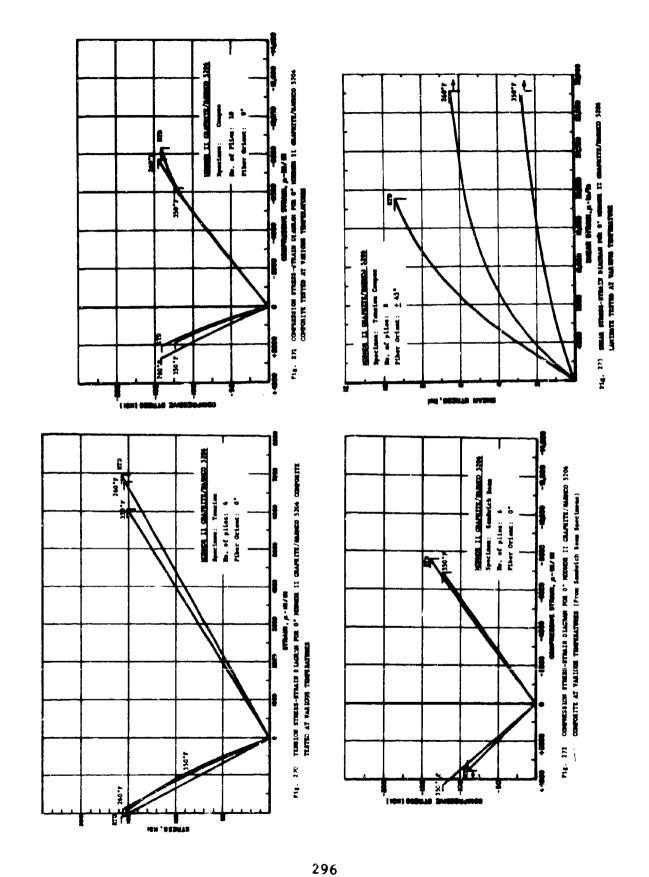
Orientation	Type Load	Prior Conditioning	Test Temp.	ε (⁹ υι × 1να)	, (in/in)	^c ult (ks1)	ente 'tn./in.)
<u>[06</u> , 0. \$51, £7, 0]	Tension	260°F 500 Cy.	RTD	•	•	86	•
[06/0/921 25/0]	Tension	260°F/500 Cy.	3e0.F	ı	•	85	•
[0/45/135/0/ <u>90]</u>	Tension	_6 F71000 Cy.	RTD	11.4	0.42	80	7,290
[0/43/125/0/90]	Tenston	260 F /1000 Cy.	260°F	11.9	0.42	81	6,830
[06/0]\$\$[/\$7/0]	Tension	350°F/500 Cy.	RTD	1	ī	11	,
[0/45/135/0/90]	Tension	350°F /500 Cy.	260°F	•	•	%	•
[0.45, 135/0/90]	Tension	350°F/500 Cy.	350°F	•	•	83	4
[0/05/135/0/0]	Tension	350°F/1000 Cy.	RTD	10.7	0.47	78	7,380
[0/45/125/0/90]	Tension	350°F/1000 Cy.	350%	9.11	97.0	90	6,640
, ,	Compression	260°F/500 Cy.	RTD	•	•	146	·
, 0	Compression	260°F/500 Cy.	260°F	•	•	142	ſ
,0	Cimpression	260°F/1050 Cy.	č.	18.3	0.32	671	9,260
ຶ່ດ	Compression	260 F /1000 Cy.	260°F	19.5	0.39	145	8,350
٥ ۽	Compression	350°F /500 Cy.	E	•	•	153	•
,0	Compression	350°F /500 Cy.	3\$0.2	•	•	133	ı
_? 0	Compression	350°F /1000 Cy.	CE S	17.9	0.20	135	9,280
.0	Compression	350°F /1000 Cy.	350°F	16.6	0.28	103	5,750

TABLE W STALIC PROPERCIES SUPMARY - GARAGO 5206 (MDMOR II GRAPHIC COMPOSITES

Orientation	Type Load	Frior Conditioning	Test Temp. (°F)	5 (pst x 19 ⁶)	, (in/in)	σult (ks1)	e _{υl} τ (μ-in./in.)
، 46	Compression	260'F 500 Cy.	RT	1		26.4	•
<u></u> 06	Compression	260°F/500 Cy.	260°F	ı	1	22.2	,
066	Compression	260'F 1000 Cy.	RTD	11.1	0.01	25.9	26,160
>06	Compression	260°F'1000 Cy.	260°F	1.18	0.00	23.6	23,450
₂ 0 6	Compression	350°F'500 Cy.	Ç	ŧ	•	23.1	ı
,06	Conpression	350°F /500 CV.	350°F	•	•	17.4	٠
,06	Compression	350°F'1000 Cy.	CT.	60.1	0.01	23.3	22,520
³06	Compression	350°F/1000 Cy.	350·F	1.02	0.00	17.9	24,170
[0/45/135/0 90]	Compression	260°F'500 Cy.	Œ.	•	•	100	ı
[0/45/135/6/90]	Compression	260°F/500 Cy.	260°F	ı	•	6	•
[0,45,135/0/90]	Compression	260°F/1000 Cy.	RID	10.7°8	0.42	82	9,930
[0/55/135/0/ <u>90</u>]	Compression	260°F/1000 Cy.	260°F	8.83	0.42	62	10,910
[0/45/135/0/90]	Compression	350°F/500 CV.	RTD	•	•	93	•
[0/25/135'0/ <u>90</u>]	Compression	350°F/500 Cy.	350°F	•	•	16	•
[0//5/135/0/90]	Compression	350°F/1000 Cy.	RTD	8. ₆	0.42	85	12,030
8[<u>06</u> /0 []. \$-,0]	Compression	350 °F /1000 Cy.	350°F	9.26	77.0	78	12,370

TABLE XV STATIC PROPERTIES SUPPARY - NARMOD 5206 (MODMOR II GRAPHIC COMPOSITES

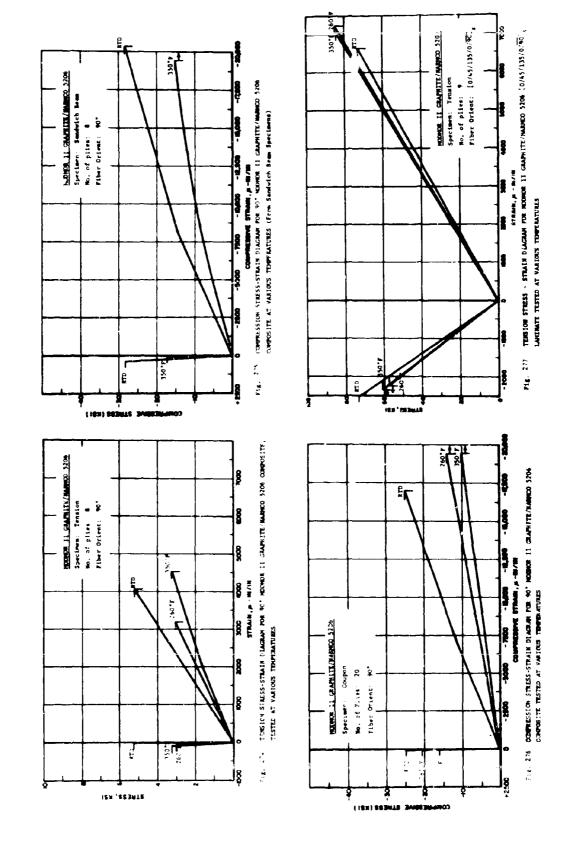
				The state of the s		
Orientatica	Tyr. Load	Prior Conditioning	Test Temp. ('F)	6 (ps1 x 10 ⁶)	ult (keil)	^e υ1ε (μ-in./in.)
0.	In-Flane Shear	260°F/500 Cy.	CTA		8.0	
o (In-Plane Shear	260°F/500 Cy.	260°F	•	6.9	•
.0	In-Plane Shear	260°F/1000 Cy.	RTD	0.71	7.9	18,070
.0	In Plane Shear	260°F/1000 Cy.	2 60 °F	09.0	6.8	> 30,000
90	In-Plane Shear	350*F/500 Cy.	RTD	,	7.9	•
. 0	In-Plane Shear	350°F/500 Cy.	260 ℃	·	7.1	•
₂ 0	In-Plane Shear	350°F/1000 Cy.	CT 2	69*0	7.9	17,300
٥	In-Plane Shear	350°F/1000 Cy.	350°F	0,40	6.1	> 30,000
- 0	Inc. Shear	260°F/500 Cy.	RTD	•	11.6	,
. 0	In:, Shear	260°F/500 Cy.	2 60°F	•	8.6	1
°O	In: Shear	260°F/1000 Cy.	æ	•	10.7	1
. 0	Inc. Shear	260°F/1000 Cy.	260°F	•	8.0	ı
.0	In., Shear	350°F/500 Cy.	Ē	ı	11.5	ì
.0	Inc. Shear	350°F/500 Cy.	260°F	•	5.5	ı
٥	Inc. Shear	350°F/1000 Cy.	£	•	10.6	•
• 0	In:. Shear	350°F/1000 Cy.	350°F	•	8.4	•

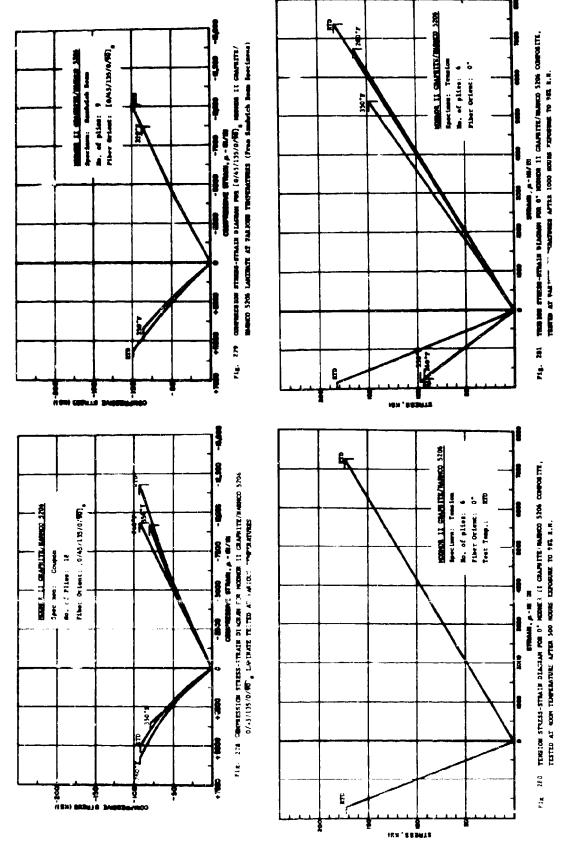


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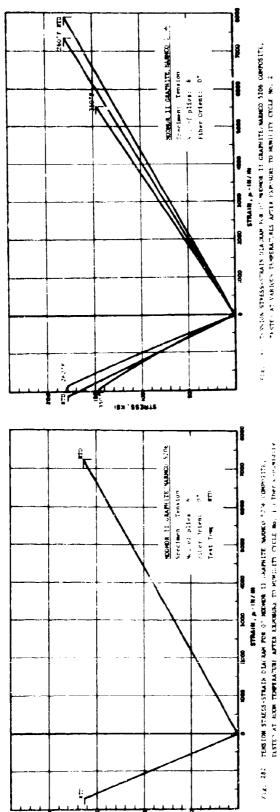


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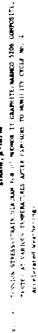
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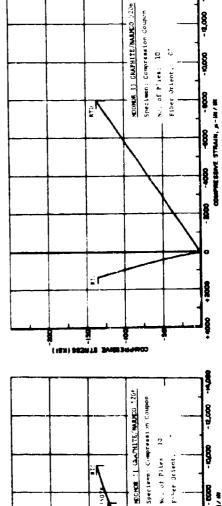


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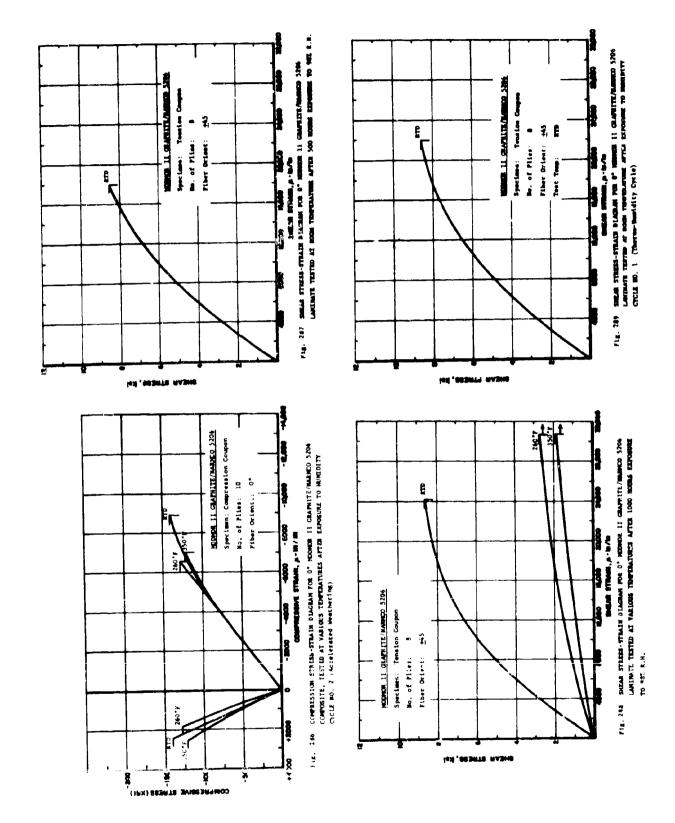
(1831) \$63ELS 3AISS384

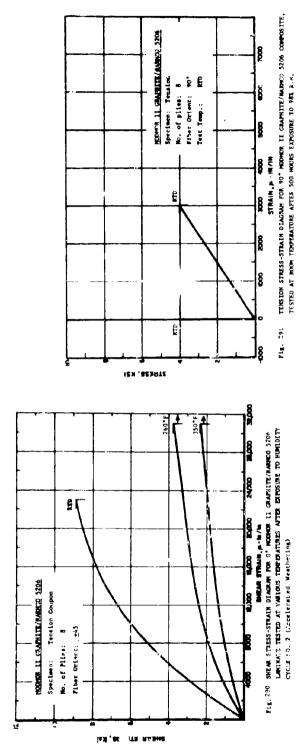
FIG. 24- COMPRESSION STRESS-CTAIN DIAGRAM FOR C'HOMME II HAMPHITT MARY COPYSITE, TESTER AT VARIOUS TEMPERATHES AFTER I'M HANTS ECHOSIBE TO 92E R.

COMPRESSOR STRAIN, p. 10./ 49

FIR. 284 COMPRESSION STRESS-STRAIN DIAMAMEND "MOMENTE CRAPHITE NAMED NOOL COMPOSITY, TENNED AT ROOM TEMPERATER MATER EXPECTED HOMINITY CYCLE NO. 1 (Therm-Humality Cycle)

3,c1e)

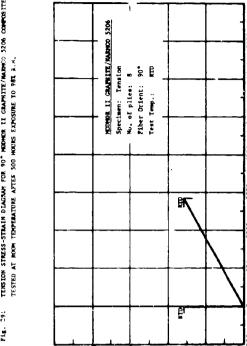




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MODMOR II GRAFHITE/MARKOD \$206 Speciern: Tension No. of plies: 8 Fiber Orient: 90*

15X 1353915

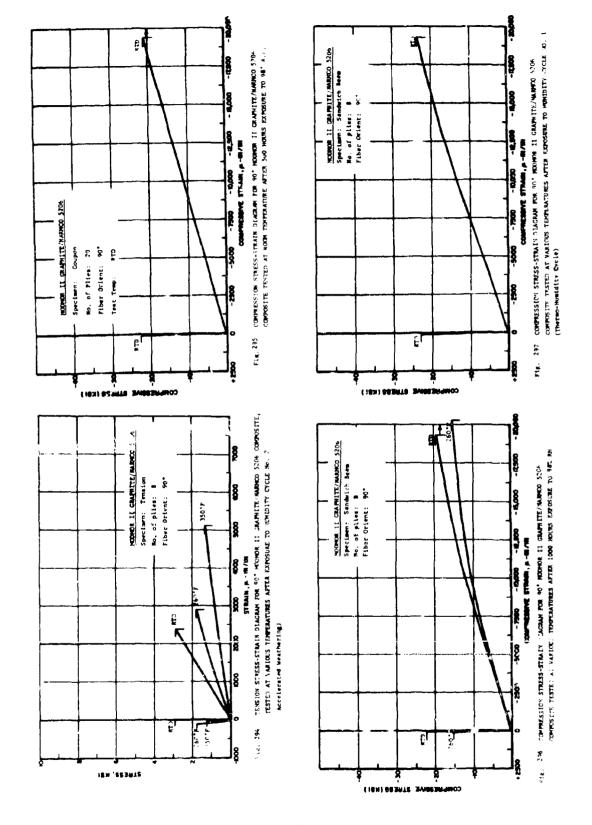
*14. 292 TERRICK STRESS-STRAIN DIAGRAN FOR 90" MONDR II CRANNITE/NAUMCO 5206 CUNFOSITE, TESTED AT VARICES TEOPERATURES AFTER 1000 NOURS EXPOSURE TO 96% R.H.

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STRAIM, p. -m./m.
Fig. 293 IEESION STRESS-SITAIN DIAGNAR FOR 90' HODGER II CHARHITE/NAMBOO 3206 CONGROSITE,
TESTED AT NOOM TOPERATURE AFTER EXPOSURE TO HUMIDITY CYCLE No. I (Thermo-Sweldity
Cycle)



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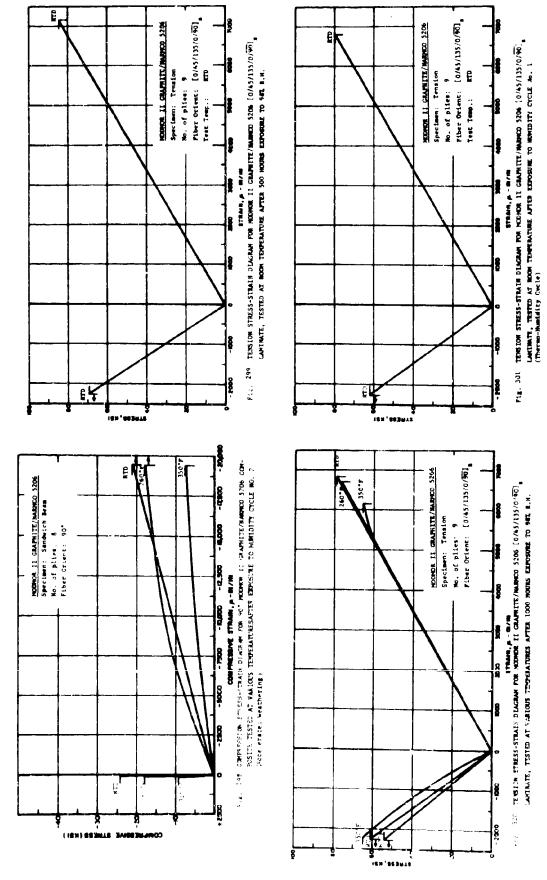
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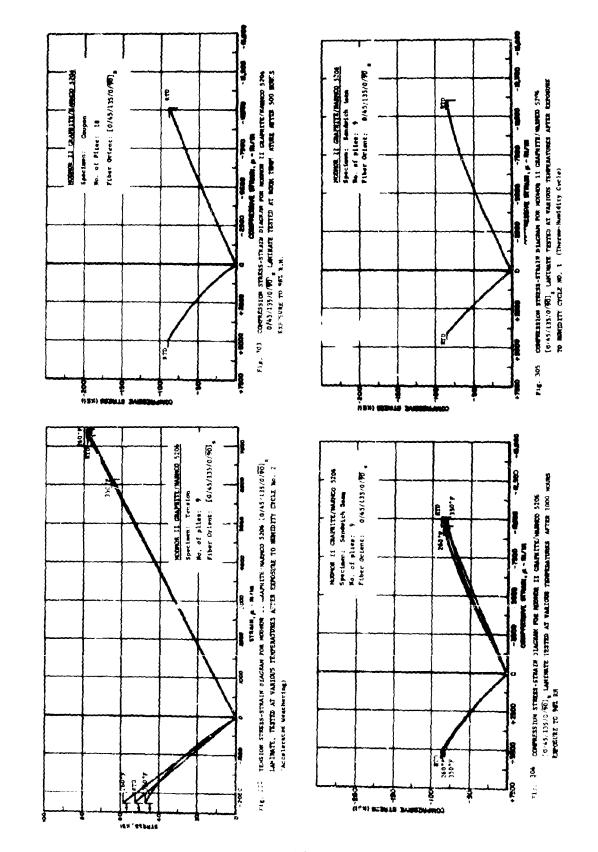


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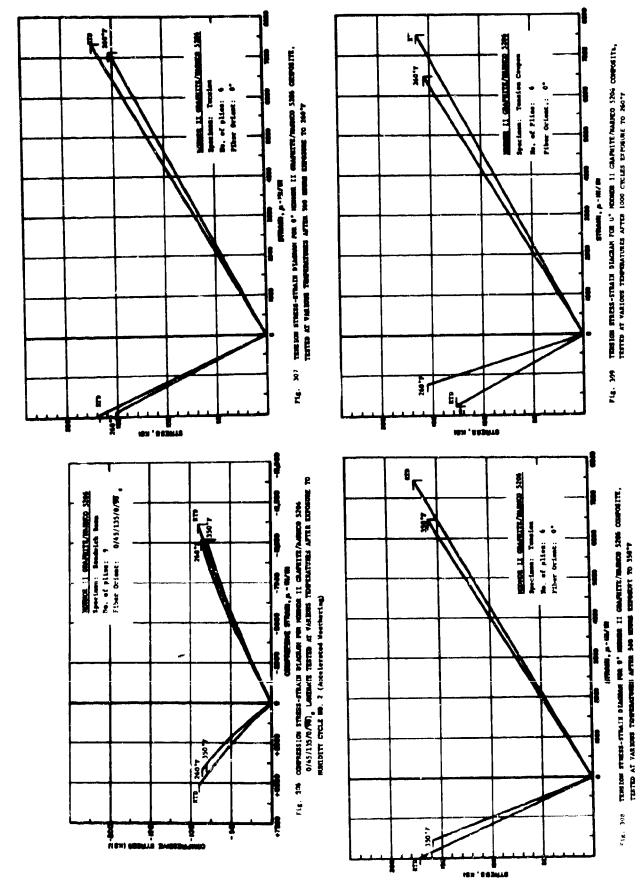
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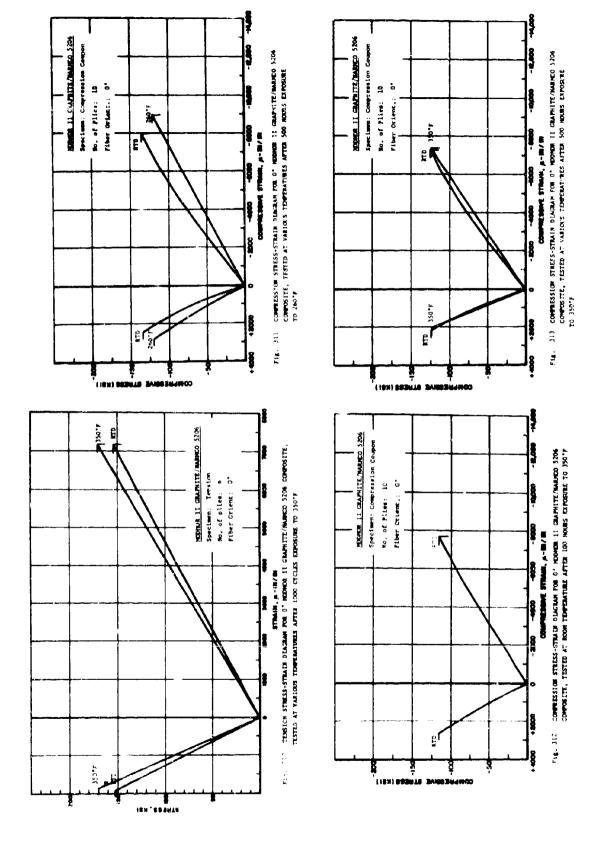
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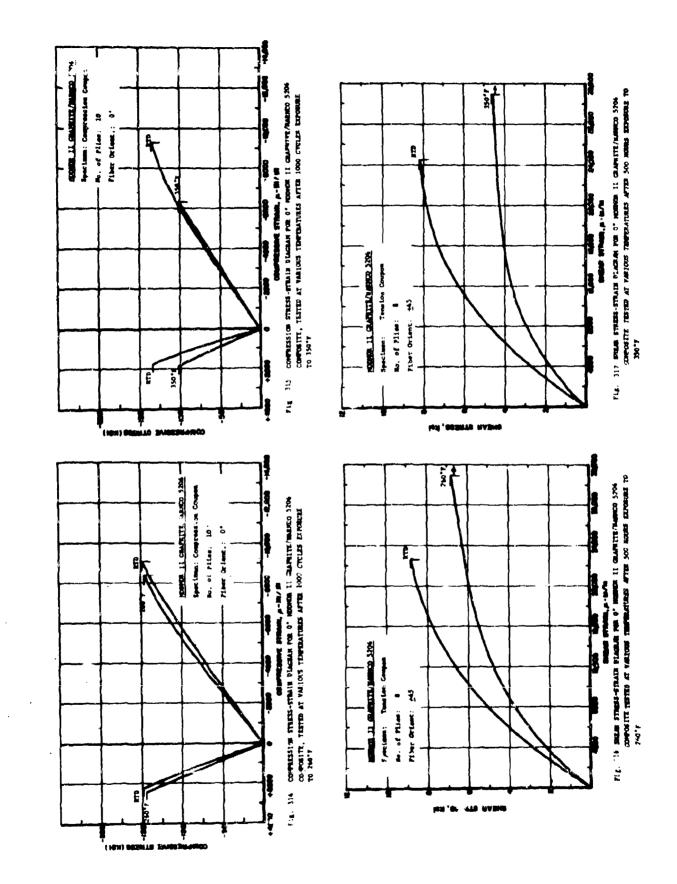
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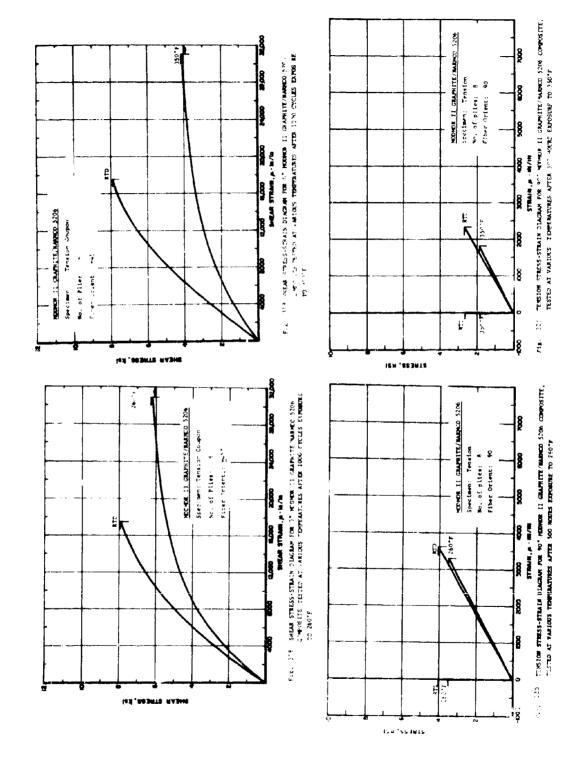
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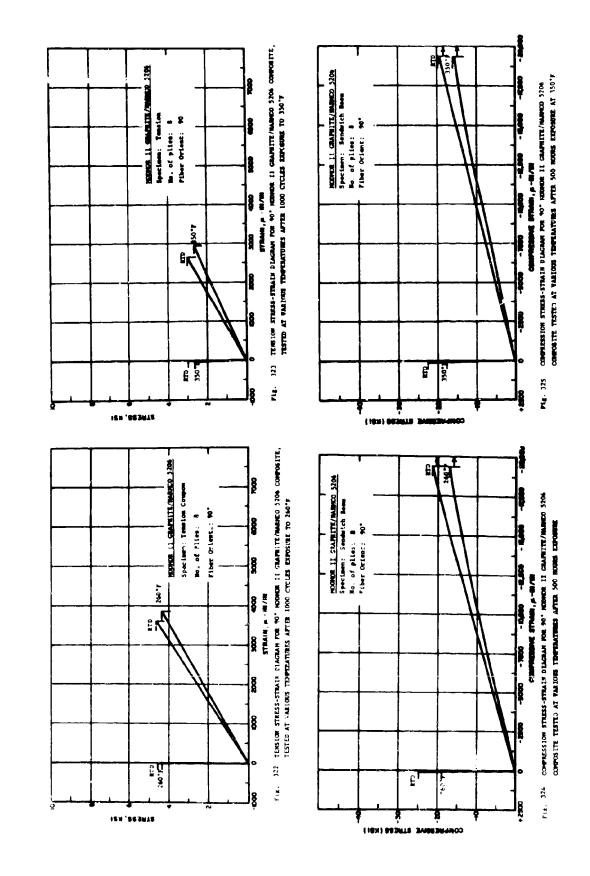
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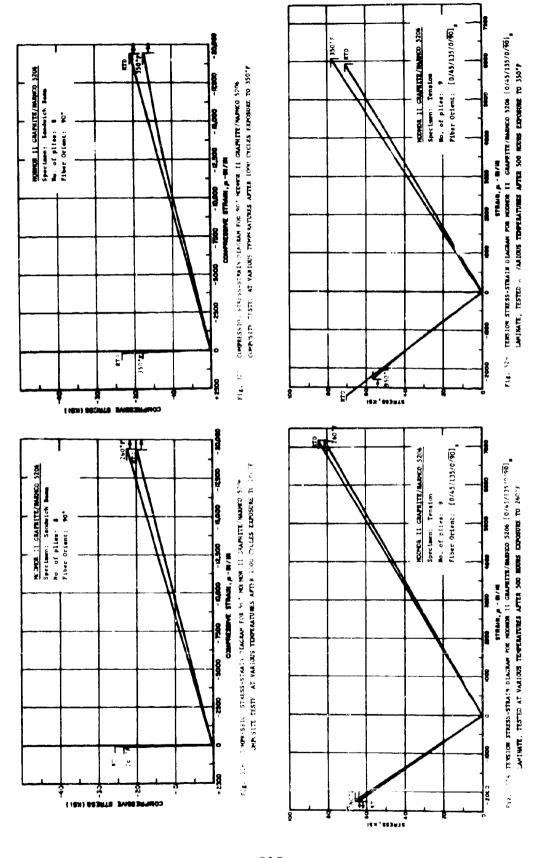
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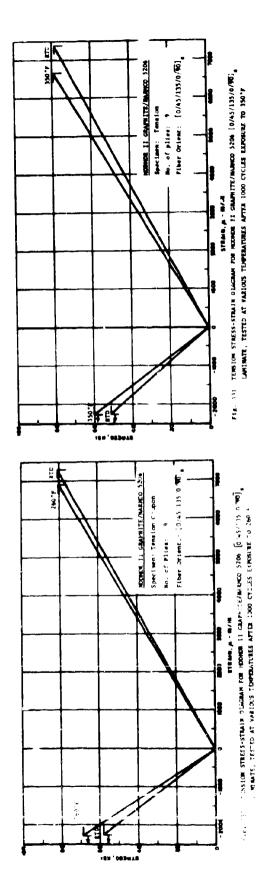
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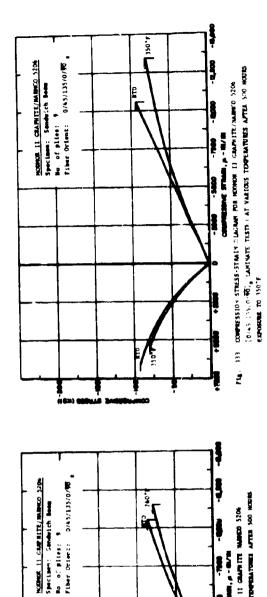
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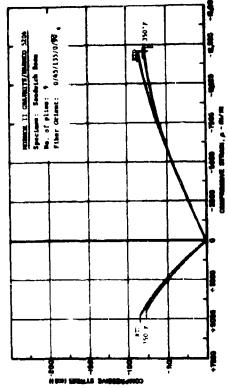
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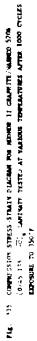


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NODECE II GAATHIE MARKS 2200 Specimen: Sendelch Beem Ho of plies: 9 Fiber Orlent: 0/45/135 0.FG



COMPLESSES STREETS - STREETS - STREETS IN DIGITAL TOW HODING IT IN METER WANTON 5704 (1.5 s.) 1/5 (1.50) (METANTE TESTE) AT VARIOUS TURNISMENTALINES AFTER 1000 CYCLES EMPOSINE TO 150 F

-2500 -3600 -7500 -2500 -3600 -7500 -1000 - 1000

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TAPLE AVI AT GULLER PRACTOS COMARCO

Specimen Pumber M1105A-6 M1105A-7 M1105A-8	i				, 0			Cycles	Applied	900	
M1105A-6 M1105A-7 M1105A-8	Inickness (Plies) (In.)	Orientation	Type	PRIOR CONDITIONING Type Duration	Temp.	Stress Level (7, ult) (ksi)	evel (ks1)	Fallure (cycles)	Fallure (cycles)	Strength (ks1)	Comment
M1105A-7 W1105A-8	5 - 0.032	,0	Mone	•	KTX	780	1	1.000		,	•
X1105A-8	1	ó	Nune		. J.	ac I	125	901	•	,	Tab Area Failure
6-850. IK	6 - 0.033	Ç	Mone	i	ي ر	ł	135	2,000		•	•
	٠	ě	None	ı	άĽ	.t r	135	105,000	•	•	,
W1105A-10	١	.0	Minne		ÇŢ.	33	130	00u*,		•	Tab Area Failure
H11054-11	•	0	No.1e	•	KT:	ĩ	30	848,000		•	
M1:05A-12	- 1	, 0	Methe	•	GT.	r. ©	•	1,000	•		Tab Area Failure
M1105A-13	٠	.0	None	•	٠. کا	8.0	.33	13,000	•	•	Tab Area Failure
M1105A-14	- 1	•0	None		RTD	.co	٠,	2,000	,	,	Tab Area Failure
M1:05A-15	6 - 0.032	0.	No 16		RT.	۵٫	1.25	5,678,000	•	1	Tab Area Failure
¥1102-6	8 - 0.042	°.06	N. De	,	Ę	: 	7	ı	•	•	Immediate Tab
											Failure
M1102-7	5,00,0	.06	None		RT5	ج. ن	,	1,000	•	•	Tab Area Failure
M1102-8	ı	,06	None	,	RTD	52		2,000	•		Tab Area Failure
M1102-9	٠	° 0 6	None	1	CTA	39	,	6,569,000	•	•	Tab Area Failure
M1102-10	1	, 0 6	None	ı	£,	20	- 1	881,000	•	1	•
M1102-11	8 - 0.043	.06	None	•	CT3	00 √7	3.7			•	1
M1102-12	٠	. 06	None	•	RTD	67	3.8			•	•
M1102-13	8 - 0.043	.06	None	٠	RTD	97	3,5		•	•	
Mi 102-14	٠	.06	None	•	RTD	47	3.6		ı	•	Tab Area Failure
M1102-15	8 - 0.042	. 06	None		RTD	r,	3.6	6,818,000	•	•	•
M1127A-6	9 - 0.047	(0/45/135/6/90.	None	,	RTD	Z	09	•	10 × 106	83.2	·
M1127A-7	9 - 0.047	[0/45/135/C/ <u>90</u>]	None	t	K	86	70	324,000	ı	•	Immediate Tab Failur
M1127A-8	9 - 0.046	0/45/135/(/90)	None		KTD	112	9	2,000	•	•	•
M1127A-9	9 - 0.046	0/45/135/C/90	None	•	E S	105	75	•	•	1	Immediate isb Failur
M1127A-10	870.0 - 6	[0/45/135/C/90]	None	ŧ	£	100.5	72	1	ι	•	Immediate Tab Failur
W1127A-11	950.0 - 6	0/45/135/6/90]	None	•	£	91	65	10,200,000	•	•	4
M11278-1	870.0 - 6	[06/0/561/57/0]	None	•	6	86	70		,	ı	Immediate Tab Failur
M11278-2	6 - 0.047	[06/0/32/6/0]	None		KLD	102	73	3,000	•		Tab Area Failure
M11278-3	9±0 0 = 6	[0/45/135/0/90]	None	1	£	103	7,6	1	•	ı	Inmediate Tab Failur
411279-4	870 0 - 0	[0/65/135/0/90]	Notice	•	Ę	100.5	72	1,366,000		1	•

TABLE XVI FALIGIE PROPESTIES SUBMINY - NAMINO 2206/NOMBIN IT GRAPHITE COMPOSITES

Thickness (Piles) (In.)	Orientation	PRIOR C	PRIOR COMDITIONING Type Duration	ijĖ	Stress Level (E ^g ult) (ksi)	Level (kai)	Cycles to Failure (cycles)	Cycles Applied without Failure (cycles)	Residual Strength (ksi)	Comment
- 0.038	ů	None	•	260°F	8	135	1.000			
- C.038	0	None	•	2 60 °F	47	130				Immediate Failure
	o é	None	•	260°F	83.5	125	6,000	•		
	٠ ټ (None		260	2	120	•	2,113 x 10°	179.0	
0.039	່ເ	e e		260°F	8 5	123	3,000			Tab Failure
•	٥٠	E 1	•	J. 097	1	121	8			Tab Feilure
٠	5 6	None		260°F	0	120	•			Immediate Tab Pailure
•	;	eon:		260°F	.	121	•			
6 - 0.039	, °	e de	1 ,	7.092 2.092	9 9	123	• (Ismediate Tab Failure
	•	3100	•	3	63.3	5 71	•			immediate Tab Failure
9 - 0.044	• &	None		260 °F	65	2	70.000			
- 0.043	\$	Kone	,	260°F	96	m				Pailed under static
										load while coming
- 0.041	8	, and	,	3.090	8	2 6	5			up to temperature
770				35	4	; .	3 8			I SO VIES FEITHER
20	•			3,5		, r	36.5			
170	· •			5	; ;		36			
0,043	.06	None	,	260.7	, F.	2.3	200			
270 0 - 8	. 06	S.	•	. 92	5	. 6				The same of the same of the same
	•					?				load while coming
	•			;	,			•		up to temperature
E 0	, , , , , , , , , , , , , , , , , , ,	None		260	6	1.5	•	2.1 × 10	3.4	Tab Area Failure
6.0°	, S	Kone	•	1,092	90 37	m				Failed under static
										up to temperature
870.0 -	[0/45/135/0/90]	None	•	260°F	2	2	14,000			
670.0 -	[06/0/381/138/0]	None	•	1.092	83.5	73	,			Immediate Pailure
- 0.048	[0/42/132/0/90]	None	•	4.092	2	2	1,900			Tab Pailure
- 0.048	[0/45/135/0/90]	None	,	Z60.F	74.5	65	000.4			
840.0 -	[0/45/135/0/90]	Mone		7.092	74.5	65	10,000			
- 0 248	[06/0/561/57/0]	Kone	•	260.7	68.5	3	9			Tab Failure
670.0 -	[0/45/135/0/90]	Kone	•	260°F	63	55		2.362 x 10 ⁶	78.4	
- 0.04	[0/45/135/0/90]	Mone	•	260.7	68.5	3	8.000			
- 0,048	[0/45/135/0/90]	None	,	260°F	56.5	2		2 327 = 106	7. 7.	
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TABLE XVI FATIGIE PROPERCIES STUMMEN -

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								Cycles	Applied		
Spectmen	Tifokness		PRIUR	PRIUR COMDITIONING	Test	Stress Level	Level	to Fat lury	without	Residual	
Amber	(Plies) (In.)	Orientation	Type	Duration	(°F)	(Z ^d ult) (ks1)	(ks1)	cycles)	(cycles)	Strengtn (ks1)	Comment
11078-1	6 - C.039	0	None	•	350°F	<u>,</u>	001	2.000			Tak Failure
1107B-2	6 - 0.039	° 0	None		350°F	63.5	95	•	•		Immediate Fallure
1107B-3	•	•0	None		350°"	9	Q.		2.619×10^{6}	139.2	Tab Failure
1107B-4	20.03 - 9	۰,0	None	t	356	1.9	100	•	2.421 x 10 ⁶	140.5	,
1107B-5	€ - 0.038	٥,	None	•	350°F	70	105	17,000			Tab Fcilure
1107B-6		• 0	None	į	350°F	ĸ	105	2,000			
11078-7	•	,0	None		350°F	73.5	110	1,000			
.107B-6	ı	٥٠	None	•	1.05s	6.8.5	103	2,000			
1078-9	6 - 0.034	.0	None	1	350°F	63.5	95	1,930,000			
0T-8/0TT	e = 0.037	, 0	None	•	350°F	67.5	101	1,000			Defective Tab
E-71154	E - 0.041	36	None		350°F	30	06.0	23,000			
1114-4	4	o o	None	1	350°F	86	3.0	. 1			Failed under static
											load while confrg
1111-5	8 ~ 0.043	້06	None	•	380°F	õ	06.0	2,000			up to temperature
1114-5	ŧ	.06	None	•	350°F	67	2.0	٠,			Failed under static
											load while coming
F_114-7	8 - 9.044	.06	None		3.056	1.6	07.0	000 7			up to remperature
P_114-8	٠	06،	None		350 %	16.5	0,50		7.0 × 10 ⁶	3.7	
114-9	470.0 - 8	•06	None	٠	350 F	23	0.70	8,000	•		
₩.114-10	ł	06،	None		350°F	21	0.65		7.0 x 10"	3.7	
F1114-11	1	• 26	None	•	350°F	33	1.00	3,000			
M:114-12	•	06	None	•	350°F	33	1.00	1,000			
X.1134A-1	9 - 0.550	[0/45/135/0/90]	None	,	350°F	82	65	•			Stec. slipped in grips
M.134A-2	670°0 - 6	[0/45/135/0/90]	None	•	350°F	82	65	2,000			
×.134A-3	6 - 0.049	[0/45/135/0/90]	None	•	350°F	63	20		2.332 x 10 ⁶	0.09	
M:134A-4	9 - 0.050	[0/45/135/0/90]	-kone	•	350 °F	69.5	55	,	2.033 x 106	72.2	
M.134A-5	9 - 0.050	[0/45/135/0/90]	A cre	•	350°F	69.5	55	ŀ	2,fle x 10 ⁶	73.3	
M.134A-6	6 - 0.049	[0/45/13 5 /0/ <u>90</u>]	None	•	350°F	92	Ç.	2,000			
M.134A-7	9 - 3.049	[0/45/135/0/90]	Mone	•	350°F	92	09	000*9			
M.134A-8	9 - 0.048	[0/25/135/0/90]	Mone		350°F	7.2	13	2,000			
M. 134A-9	6 - 0.049	[0/45/135/0/90]	None	•	350°F	73.5	6 0	20,000			
	0		,			1	;				

TARGE XVI - COLLUE PROPERS SCHARKY - COMPOSITES CARRESTED SCHEDINGS TO GRAPHITE COMPOSITES

Specimen Number	Thickness (Pites) (In.))rientation	EFIOR C	FFIOR COMDITIONING Type Duration	Test Temp. (*F)	Stress Level (1 ⁰ ult) (ksi)	evel (ksf.)	Cycles to Fatlure (cycles)	Cycles Applied without Failure (cycles)	Residual Strength (ksi)	Consent
X11058-7	G -	90,		/ 500 Hrs.	ECT D					'	-(Specimen broke
M1105B-8	,	ပ်	.86 86	/ 500 Hrs.	1					•	-(during fabrication
M11058-9	6 - 0.032	Ö		/ 500 Hrs.	Ę	28	135	2,000			Immediate failure
Y11058-10	•	0	-	/ 500 Hrs.	E	۲. ۲.	£ :	,			immediate tallure
¥11058-11	F - 0.032	,0	#2 80	/ 500 Hrs.	E	70.5	777	•			
411088-15	6 - 0.038	Č	H.3 86	500 Hrs.	260°F	69.5	120	1,000			
411068-16	1	Ö			250°F	66.5	115	18,000			
Y11088-17	•	°		500 Hrs.	7.097	68	11)	2,000			
X1108B-18	6 - 0.036	°	₩2 80°		3.09Z	63.5	011	11,900	90.	•	
41108B-19	1	ت.		500 Hrs.	1.092	61	Se	•	2.033 x 10°	186.7	Tab Failure
41109A-15	6 - 0.033	ຸບ	98 RH	/ 500 Hrs.	350 F	63,5	105	2,000			
41159A-15	ن	ن د		/ 500 Hrs.	350°F	9	100	7,000	•		
41109A-17	,	0 ،		/ 500 Hrs.	350°F	57	95	•	2.34 × 10	2.3	
41109A-18	0	0 و			350°F	59	60	2,000			Tab Failure
£109A-19	6 - C.032	0	18 RH	/ 500 Hrs.	350°F	58	\$	13,000			
V11058-12	6 - 0,032	ပ်	98' RH	/ 1000 Hrs.	KTD	96	120	2,000			Tab Failure
.11C5B-13	0	£ ()	₩ 86		Ē	63	115	•			Immediate 14b Failure
H1058-14	•	0	152 262 7		£	60.5	110	2,00			Tab Failure
411058-15	6 - 0.032	ů	98, 38,		E	80	105	631,000			
M11C58-16	٠	ů	98° RH	/ 1000 Hrs.	6	88	101	51,000			
**11088-20	6 - 0.033	0.	HN .86	/ 1000 Hrs.	260°F	70.5	115	3,000			Tab Failure
M1109A-1	•	.0	98, KH	/ 1000 Hrs.	7.097	67.5	110	0 0			lab Fallure
F1109A-2	6 - 0,035	င်	HE ,85	/ 1000 Hrs.	260°F	61.5	8	11,000	9".	į	
11109A-3	6 - 0.033	0		/ 1000 Hrs.	260°F	25	8		7.20 × 10,	107	18b reliure
M1109A-4	6 - 0.033	0،	98° RH	1000 Hrs.	260°F	58.5	4 5		7.83 x 10	o.	
06 1001174	6 6 032	Ċ	H& .60	/ 1000 Hrs.	350*P	3	95	1	4,766,000		
7.1109A-20	,	်			350°F	11	105		1,000		Tab Failure
0.11008		Ĉ		/ 1000 Brs.	350*F	67.5	8		14,000		Tab Failure
7-100TT	•	. 0	18. 18.		350*F	\$	8		•		Immediate Tab Failure
7-86013	•	0	98°. RH	/ 1000 Hrs.	350*F	99	98		•		Failed under static
!											to temperature

TABLE XVI : FAIT DE PEDE ROUS SONMARY - NAVOUS 2206 ROUNDE : CAMPUSITUS

100 100	Specimen	Thiches (File:) (in.)	Orientation	PRIOR COMDITIONING Type Duracton	1110NTNC Duration	Test Tesp. (*F)	Stress Level (I'ult) (ksi)	.ve.1 (ka1)	Cycles to Failure (cycles)	Cycles Applied without Feilure (cycles)	Residual Strength (ksi)	Coment
1.00	M11058-17	1 1	ָּט בֿ	Thermo-Hundd.		6	73.5	120	1,000			Tab Failure
### 10.033 0° Thermo-Humidity Cycle RTD 64.5 105 3,000 2 x 106 141.7 ### 10.033 0° Thermo-Humidity Cycle RTD 64.5 105 3,000 2 x 106 141.7 ### 10.033 0° Thermo-Humidity Cycle 260°F 66.5 115 4,000 ### 10.032 0° Thermo-Humidity Cycle 260°F 66.5 116 12,000 ### 10.032 0° Thermo-Humidity Cycle 260°F 66.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 260°F 66.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 86.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 86.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 86.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 86.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 86.5 116 12,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 87.5 120 17.00 ### 10.033 0° Thermo-Humidity Cycle 350°F 87.5 120 2.22 x 106 188.4 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 110 - 2.22 x 106 188.4 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 113 27,000 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 115 27.000 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 115 27.000 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 115 27.000 ### 10.033 0° Thermo-Humidity Cycle 350°F 64.5 115 27.000 ### 10.033 0° Thermo-Humidity Cycle 350°F 67.5 100	M1105B-19	•	່ວ	Thermo-hunid		£	70.5	115	1,000			Tab Failure
Fig. 10.033 O' Thermo-Hunddity Cycle NT 63 102 2 x 10° 141.7	M11058-20	1	o°0	Thermo-Hunid:		KT3	64.5	105	3,000	7		
6 - 0.033 0° Thermo-Hunddity Cycle 260°F 66 113 - 0.0246 x 10° 155.2 6 - 0.033 0° Thermo-Hunddity Cycle 260°F 68.5 116 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 260°F 68.5 116 12,000 6 - 0.033 0° Thermo-Hunddity Cycle 260°F 68.5 116 12,000 6 - 0.033 0° Thermo-Hunddity Cycle 260°F 66.5 116 12,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 89.5 130 - 2.339 x 10° 170.0 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 86 130 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 86 130 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 81.5 123 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 81.5 123 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 81.5 123 1,000 6 - 0.033 0° Thermo-Hunddity Cycle 350°F 81.5 123 1,000 6 - 0.033 0° Acc. Withrug. RTD 61.5 110 - 2.23 x 10° 188.4 6 - 0.031 0° Acc. Withrug. RTD 64.5 115 27,000 6 - 0.033 0° Acc. Withrug. 260°F 61.5 110 - 2.42 x 10° 6 - 0.033 0° Acc. Withrug. 260°F 61.5 110 1,000 6 - 0.033 0° Acc. Withrug. 260°F 63 113 1,357,000 6 - 0.033 0° Acc. Withrug. 260°F 65.5 115 2,000 6 - 0.033 0° Acc. Withrug. 260°F 65.5 117 2,000 6 - 0.033 0° Acc. Withrug. 260°F 65.5 117 2,000 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100 6 - 0.033 0° Acc. Withrug. 350°F 87.5 100	M11058-21		0,	Thermo-Hundd		Ē	63	102		2 × 10°	141.7	
6 - 0.033 0° The race-Hunddity Cycle 260°F 71.5 123 4,000 5 - 0.032 0° The race-Hunddity Cycle 260°F 66.5 118 4,000 6 - 0.033 0° The race-Hunddity Cycle 260°F 66.5 118 12,000 6 - 0.033 0° The race-Hunddity Cycle 260°F 66.5 118 12,000 6 - 0.033 0° The race-Hunddity Cycle 350°F 86 130 17,000 6 - 0.033 0° The race-Hunddity Cycle 350°F 86 130 17,000 6 - 0.033 0° The race-Hunddity Cycle 350°F 86 130 17,000 6 - 0.033 0° The race-Hunddity Cycle 350°F 86 130 17,000 6 - 0.033 0° Acc. Withrag. RTD 67.5 120 41,000 6 - 0.031 0° Acc. Withrag. RTD 67.5 115 27,000 6 - 0.031 0° Acc. Withrag. RTD 67.5 110 - 2.28 x 10 ⁶ 188.4 6 - 0.031 0° Acc. Withrag. RTD 67.5 110 - 2.22 x 10 ⁶ 188.4 6 - 0.031 0° Acc. Withrag. 260°F 67 113 1,357,000 6 - 0.033 0° Acc. Withrag. 260°F 67.5 110 1,357,000 6 - 0.033 0° Acc. Withrag. 260°F 64.5 113 1,357,000 6 - 0.033 0° Acc. Withrag. 260°F 65.5 117 2,300 6 - 0.033 0° Acc. Withrag. 260°F 65.5 117 2,300 6 - 0.033 0° Acc. Withrag. 260°F 65.5 117 2,300 6 - 0.033 0° Acc. Withrag. 350°F 87.5 120 2,000 6 - 0.033 0° Acc. Withrag. 350°F 87.5 127 1,000 6 - 0.033 0° Acc. Withrag. 350°F 87.5 127 1,000 6 - 0.033 0° Acc. Withrag. 350°F 87.5 127 1,000 6 - 0.033 0° Acc. Withrag. 350°F 87.5 127 1,000	S-4:01.7	F	0	Thermo-Humid:		260°F	99	115	٠	2.246 × 10 ⁶	155.2	
6 - 0.032 0° Thermo-Humidity Cycle 260°F 66.5 1120 1,000 6 - 0.033 0° Thermo-Humidity Cycle 260°F 66.5 118 4,000 7 Thermo-Humidity Cycle 260°F 66.5 118 12,000 7 Thermo-Humidity Cycle 350°F 89.5 135 7 7 Thermo-Humidity Cycle 350°F 89.5 135 7 7 Thermo-Humidity Cycle 350°F 89.5 135 1,000 7 Thermo-Humidity Cycle 350°F 89.5 135 1,000 7 Thermo-Humidity Cycle 350°F 81.5 123 1,000 7 Acc. Wthrmg. RTD 64.5 110 - 2.28 x 10 ⁶ 188.4 7 C - 0.031 0° Acc. Wthrmg. RTD 64.5 113 27,000 7 Acc. Wthrmg. 260°F 64.5 113 27,000 7 Acc. Wthrmg. 260°F 64.5 113 1,357,000 7 Acc. Wthrmg. 260°F 64.5 113 1,357,000 7 Acc. Wthrmg. 260°F 64.5 113 1,357,000 7 Acc. Wthrmg. 260°F 65.5 110 - 2.42x10 ⁶ 171.8 7 C - 0.033 0° Acc. Wthrmg. 260°F 65.5 110 - 2.42x10 ⁶ 171.8 7 C - 0.033 0° Acc. Wthrmg. 260°F 65.5 110 - 2.42x10 ⁶ 171.8 7 C - 0.033 0° Acc. Wthrmg. 260°F 65.5 112 27,000 7 C - 0.033 0° Acc. Wthrmg. 350°F 85.5 125 27,000 7 C - 0.033 0° Acc. Wthrmg. 350°F 87.5 120 1,000 7 C - 0.033 0° Acc. Wthrmg. 350°F 87.5 120 1,000 7 C - 0.033 0° Acc. Wthrmg. 350°F 87.5 120 1,000 7 C - 0.033 0° Acc. Wthrmg. 350°F 87.5 127 1,000	M_103A-6	٠	0,	The rme-Hundd.		260°F	71.5	125	7,000			
6 - 0.033 97 Thermo-Humidity Gycle 260°F 67.5 118 4,000 6 - 0.033 0° Thermo-Humidity Gycle 260°F 66.5 116 12,000 7 Thermo-Humidity Gycle 350°F 78.5 120 - 2.329 x 10° 170.0 6 - 0.033 0° Thermo-Humidity Cycle 350°F 88.5 135 - 2.28 x 10° 170.0 6 - 0.032 0° Thermo-Humidity Cycle 350°F 88.5 135 - 1.000 6 - 0.032 0° Thermo-Humidity Gycle 350°F 88.5 125 1,000 6 - 0.033 0° Acc. Whrmag. RTD 67.1 120 41,000 6 - 0.031 0° Acc. Whrmag. RTD 65.5 110 - 2.28 x 10° 188.4 6 - 0.031 0° Acc. Whrmag. RTD 65.5 110 - 2.442 x 10° 188.4 6 - 0.031 0° Acc. Whrmag. 260°F 65.5 115 27,000 6 - 0.033 0° Acc. Whrmag. 260°F 65.5 117 35,000 6 - 0.033 0° Acc. Whrmag. 260°F 65.5 117 35,000 6 - 0.033 0° Acc. Whrmag. 260°F 65.5 117 35,000 6 - 0.033 0° Acc. Whrmag. 260°F 65.5 117 35,000 6 - 0.033 0° Acc. Whrmag. 350°F 89.5 128 272,000 6 - 0.033 0° Acc. Whrmag. 350°F 89.5 128 272,000 6 - 0.033 0° Acc. Whrmag. 350°F 89.5 128 272,000 6 - 0.033 0° Acc. Whrmag. 350°F 87.5 120 1,000	H1109A-7	•	Ö	The rate-Humid.		260°F	68.5	120	1,000			Tab Failure
6 - 0.033	M1109A-8	1 1	က် ငံ	Thermo-Hunid.		260°F	67.5	116	4, ⁷			Tab Failure
6 - 0.033			,			-				•		
te - 0.033 0' Thermo-Numidity Cycle 150°F 89.5 135 1-0 6 - 0.032 0° Thermo-Humidity Cycle 150°F 86 130 17,000 6 - 0.032 0° Thermo-Humidity Cycle 150°F 81.5 123 1,000 6 - 0.033 0° Thermo-Humidity Cycle 150°F 81.5 123 1,000 6 - 0.031 0° Acc. Withrig. RTD 61.5 110 - 2.335×10 ⁶ 169.4 6 - 0.031 0° Acc. Withrig. RTD 64.5 115 27,000 6 - 0.031 0° Acc. Withrig. RTD 64.5 115 27,000 6 - 0.031 0° Acc. Withrig. 260°F 61.5 110 - 2.442×10 ⁶ 111.8 6 - 0.033 0° Acc. Withrig. 260°F 64.5 115 21,000 6 - 0.033 0° Acc. Withrig. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withrig. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withrig. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withrig. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withrig. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withrig. 350°F 87.5 100 6 - 0.033 0° Acc. Withrig. 350°F 87.5 100 6 - 0.033 0° Acc. Withrig. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Withrig. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Withrig. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Withrig. 350°F 87.5 128 5,000	111098-5	- 0	Ö	The rmo-Hunid		350'F	79.5	120		2.329 x 10°	170.0	
E - 0.032 0° Thermo-Humidity Cycle 350°F 86 130 17,000 6 - 0.033 0° Thermo-Humidity Cycle 350°F 83 125 1,000 2.28 x 10 ⁶ 155.2 6 - 0.033 0° Thermo-Humidity Cycle 350°F 81.5 125 1,000 155.2 6 - 0.031 0° Acc. Wehring. RTD 61.5 110 - 2.335 x 10 ⁶ 169.4 6 - 0.031 0° Acc. Wehring. RTD 64.5 115 27,000 2.22 x 10 ⁶ 188.4 6 - 0.031 0° Acc. Wehring. 260°F 61.5 110 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Wehring. 260°F 61.5 110 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Wehring. 260°F 61.5 110 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Wehring. 260°F 63.5 113 1,357,000 - 6 - 0.033 0° Acc. Wehring. 350°F 85.5 112 <td< td=""><td>11098-6</td><td>ŀ</td><td>·, o</td><td>Thermo-Reald:</td><td>ity Cycle</td><td>350°F</td><td>89.5</td><td>135</td><td>•</td><td></td><td></td><td>Immediate Failure</td></td<>	11098-6	ŀ	·, o	Thermo-Reald:	ity Cycle	350°F	89.5	135	•			Immediate Failure
6 - 0.032 0° Thermo-Numidity Cycle 350°F 813 125 1,000 2.28 x 106 155.2 1,000 155.2 1,000 155.2 1,000 155.2 1,000 155.2 1,000 1 1,000 1,00	1,1098-7	•	Û	Thermo-Humid:		350°F	26	3	17,000			
6 - 0.033	111098-8	1	°O	The rate-Numid		350°F	83	125	1,000	•		
6 - 0.031 0° Acc. Webring. RTD 61.5 110 - 2.335×10 ⁶ 169.4 6 - 0.032 0° Acc. Webring. RTD 64.5 115 27,000 6 - 0.031 0° Acc. Webring. RTD 64.5 115 27,000 6 - 0.033 0° Acc. Webring. RTD 64.5 113 27,000 6 - 0.033 0° Acc. Webring. 260°F 67 120 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Webring. 260°F 67 120 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Webring. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Webring. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Webring. 350°F 65.5 117 5,000 6 - 0.033 0° Acc. Webring. 350°F 67 120 - 2,000 6 - 0.033 0° Acc. Webring. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Webring. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Webring. 350°F 87.5 127 1,000	111098-9	•	0,	Thermo-Hundd		350'F	81.5	123	•	2.28 x 10°	155.2	Tab Failure
6 - 0.032 0° Acc. Webring. RTD 67 120 41,000 6 - 0.031 0° Acc. Webring. RTD 70 125 6,000 6 - 0.031 0° Acc. Webring. RTD 64.5 115 27,000 6 - 0.033 0° Acc. Webring. 260°F 61.5 110 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Webring. 260°F 67 120 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Webring. 260°F 65.5 117 1.357,000 6 - 0.033 0° Acc. Webring. 260°F 65.5 117 1.357,000 6 - 0.033 0° Acc. Webring. 350°F 85.5 128 27,000 6 - 0.033 0° Acc. Webring. 350°F 87.5 128 5.000 6 - 0.033 0° Acc. Webring. 350°F 87.5 128 5.000 6 - 0.033 0° Acc. Webring. 350°F 87.5 120 5.000 6 - 0.033 0° Acc. Webring. 350°F 87.5 120 5.000	111050-1		0 0	And Weber	18.	Q Q	61.5	110	•	2.335 × 10 ⁶	7.691	
6 - 0.031 0° Acc. Wthrng. RTD 64.5 115 27,000 2.22 x 10 ⁶ 188.4 (c. 0.031 0° Acc. Wthrng. RTD 64.5 115 27,000 2.22 x 10 ⁶ 188.4 (c. 0.031 0° Acc. Wthrng. 260°F 67 120 — 2.442x10 ⁶ 171.8 (c. 0.033 0° Acc. Wthrng. 260°F 67 120 — 2.442x10 ⁶ 171.8 (c. 0.033 0° Acc. Wthrng. 260°F 65.5 117 5.000 (c. 0.033 0° Acc. Wthrng. 260°F 65.5 117 5.000 (c. 0.033 0° Acc. Wthrng. 350°F 85.5 127 5.000 (c. 0.033 0° Acc. Wthrng. 350°F 87.5 128 5.000 (c. 0.033 0° Acc. Wthrng. 350°F 87.5 128 5.000 (c. 0.033 0° Acc. Wthrng. 350°F 87.5 127 1.000	11030-2		°c	Acc. Lithin	2	£	67	120	41.000			
6 - 0.031 0° Acc. Wthrng. RTD 64.5 115 27,000 2.22 x 10 ⁶ 188.4 c - 0.031 0° Acc. Wthrng. RTD 63 113 - 2.42x10 ⁶ 171.8 c - 0.033 0° Acc. Wthrng. 260°F 67 120 - 2.442x10 ⁶ 171.8 c - 0.033 0° Acc. Wthrng. 260°F 65.5 117 5.000 c - 0.033 0° Acc. Wthrng. 260°F 65.5 117 5.000 c - 0.033 0° Acc. Wthrng. 350°F 89 130 2.000 c - 0.033 0° Acc. Wthrng. 350°F 87.5 128 5.000 c - 0.033 0° Acc. Wthrng. 350°F 87.5 128 5.000 c - 0.033 0° Acc. Wthrng. 350°F 87.5 128 5.000 c - 0.033 0° Acc. Wthrng. 350°F 87.5 127 1.000	11050-3	t	o '0	Acc. Wthen		£	20	125	6,000			Teb Failure
6 - 0.031 0° Acc. Withing. RTD 63 113 - 2.22 x 10° 1884. 6 - 0.033 0° Acc. Withing. 260°F 61.5 110 - 2.442x10 ⁶ 171.8 6 - 0.033 0° Acc. Withing. 260°F 64.5 115 21,000 6 - 0.033 0° Acc. Withing. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withing. 350°F 65.5 120 2,000 6 - 0.033 0° Acc. Withing. 350°F 89 130 2,000 6 - 0.033 0° Acc. Withing. 350°F 87 128 5,000 6 - 0.033 0° Acc. Withing. 350°F 87 127 1,000	11050-4	•	Ö			C C	\$ \$	115	27,000	•		
6 - 0.033 0 0 Acc. Withing. 260°F 61.5 110 - 2.442m10 ⁶ 171.8 6 - 0.033 0° Acc. Withing. 260°F 64.5 115 21,000 Acc. Withing. 260°F 63 113 1,357,000 6 - 0.033 0° Acc. Withing. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Withing. 350°F 89 130 2,000 6 - 0.033 0° Acc. Withing. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Withing. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Withing. 350°F 87.5 127 1,000	11050-5	- 1	• 0			£	63	113	. 1	2.22 × 10 ⁶	188.4	
6 - 0.033	01-46011		٥	Acc. Uthra	ğ	260°F	61.5	110	•	2.442x106	171.8	
6 - 0.033 0° Acc. Wehrng. 260°F 64.5 115 21,000 6 - 0.032 0° Acc. Wehrng. 260°F 63 113 1,357,000 6 - 0.033 0° Acc. Wehrng. 260°F 65.5 117 5,000 6 - 0.033 0° Acc. Wehrng. 350°F 85.5 125 272,000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5,000	11094-11		စ်	Acc. Wthrn	, j	260°F	67	120	,) ; ;	Immediate Failure
6 - 0.032	11094-12	0	٥		· je	260°F	64.5		21,000			
6 - 0.033 0° Acc. Wehrng. 260°F 65.5 117 5.000 6 - 0.033 0° Acc. Wehrng. 350°F 85.5 125 272,000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5.000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5.000 6 - 0.033 0° Acc. Wehrng. 350°F 87.5 1.000	11094-13	0	٥,		اوا ا	260°F	63		1.357,000			
6 - 0.033 0° Acc. Wehring. 350°F 89 1.30 2.000 6 - 0.033 0° Acc. Wehring. 350°F 85.5 1.25 272,000 6 - 0.033 0° Acc. Wehring. 350°F 87.5 1.28 5,000 6 - 0.033 0° Acc. Wehring. 350°F 87.5 1.000	41-V6011	0	•		, pj	260°F	65.5		5,000			
6 - 0.033 0° Acc. Wehrig. 350°F 85.5 125 272,000 6 - 0.033 0° Acc. Wehring. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Wehring. 350°F 87 127 1,000	111098-10	1	• 0	Acc. Wthra	ž.	350°F	68	130	2,000			
6 - 0.033 0° Acc. Wehrng. 350°F 87.5 128 5,000 6 - 0.033 0° Acc. Wehrng. 350°F 87 127 1,000	11098-11	٠	°	Acc. Wehr.		350°F	85.5	125	272,000			
6 - 0.033 0° Acc. Wthrng. 350°F 87 127 1,000	11098-12	- 1	.0		, ,	350°F	87.5	128	s, 900			
	11098-13	٠	°	Acc. Wehrn	÷	350 F	87	127	90,	9		

TABLE AVI TO THE STATE OF THE S

Stress Lave Cycles Applied Cycles Applied Cycles Applied Cycles Applied Cycles C	PRIOR CONDITIONING								Selve		
Type Duration (T) (Touit) (ks1) (cycles) s 98" RH / SOO Hrs. RTD	Type Duration (*F) (T_ult) (kst) (cycles) 9g* NH 500 Hrs. RTD R4.5 73 28.306 9g* NH 500 Hrs. RTD R4.5 73 28.306 9g* NH 500 Hrs. RTD R4.5 73 28.306 9g* NH 500 Hrs. RTD 85 72 11.56 × 106 9g* NH 500 Hrs. 260°F 87 60 7.000 9g* NH 500 Hrs. 260°F 84 68 198,000 9g* NH 500 Hrs. 260°F 84 66 1,000 9g* NH 500 Hrs. 350°F 84 57 1,000 9g* NH </th <th></th> <th>PRIOR CC</th> <th>NO IT TONING</th> <th>Test Temp.</th> <th>Stress V</th> <th></th> <th>Cycles to Failure</th> <th>Applied without Failure</th> <th></th> <th></th>		PRIOR CC	NO IT TONING	Test Temp.	Stress V		Cycles to Failure	Applied without Failure		
987 RH 500 Hrs. RTD 84 75 28,306 987 RH 500 Hrs. RTD 64.5 73 28,306 987 RH 500 Hrs. RTD 64.5 73 28,306 987 RH 500 Hrs. 260°F 74.5 60 7,000 987 RH 500 Hrs. 260°F 83 65 191,000 987 RH 500 Hrs. 260°F 83 65 191,000 987 RH 500 Hrs. 260°F 83 66 1,000 987 RH 500 Hrs. 350°F 84 68 388,000 987 RH 500 Hrs. 350°F 89 5 1,000 987 RH 500 Hrs. 350°F 89 5 1,000 987 RH 500 Hrs. 350°F 89 5 1,000 987 RH 1000 Hrs. 87 5 60 1,000 987 RH 1000 Hrs. 86 5 1,000 987 RH 1000	98° RH / 500 Hrs. RTD 84 75 28,006 98° RH / 500 Hrs. RTD 84 73 28,006 98° RH / 500 Hrs. RTD 85 72 11.05 × 10 ⁶ 98° RH / 500 Hrs. RTD 85 72 11.05 × 10 ⁶ 98° RH / 500 Hrs. 260°F 74.5 60 191,000 98° RH / 500 Hrs. 260°F 83 65 191,000 98° RH / 500 Hrs. 260°F 84 68 1900 98° RH / 500 Hrs. 2500 Hrs. 350°F 84 66 1,000 98° RH / 500 Hrs. 350°F 84 66 1,000 98° RH / 500 Hrs. 350°F 84 66 1,000 98° RH / 500 Hrs. 350°F 84 66 1,000 98° RH / 500 Hrs. 350°F 84 66 1,000 98° RH / 100 Hrs. 350°F 86 55 100 98° RH / 100 Hrs. 260°F 94 60 119,000 98° RH / 1000 Hrs. <th>ਨ ।</th> <th></th> <th>Duration</th> <th>(.F)</th> <th>(Toult)</th> <th>(ks1)</th> <th>(cycles)</th> <th>(cycles)</th> <th>1</th> <th>(ks1)</th>	ਨ ।		Duration	(.F)	(Toult)	(ks1)	(cycles)	(cycles)	1	(ks1)
\$ 98" RH 500 Hrs. RTD 66.5 73 28,306 98" RH 500 Hrs. RTD 66.5 73 11.06 x 10 ⁶ 98" RH 500 Hrs. RTD 66.5 72 11.06 x 10 ⁶ 98" RH 500 Hrs. 260°F 80.5 60 191,000 98" RH 500 Hrs. 260°F 80.5 65 191,000 98" RH 500 Hrs. 260°F 80.5 65 11,174,000 98" RH 500 Hrs. 260°F 80.5 65 11,174,000 98" RH 500 Hrs. 260°F 80.5 65 11,000 98" RH 500 Hrs. 350°F 80.5 57 9,000 98" RH 1000 Hrs. RTD 94.5 75 11,000 98" RH 1000 Hrs. RTD 94.5 75 11,000 98" RH 1000 Hrs. RTD 75.5 60 119,000 98" RH 1000 Hrs. 260°F 94.5 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F 94.5 75 75 11,000 98" RH 1000 Hrs. 260°F	\$ 98° NH 500 HFS. RTD	19	o o	500 Hrs.	Œ	or ac.	25	,			
RH / 500 Hrs. RT)	98° RH / 500 Hrs. RTD		984	500 Hrs.	RT.	\$.5	73	28,006			
RH / 500 HFS.	987 RH / 500 Hrs. RTD			500 Hrs.	Ê	84.5	73				
HH	987 RH / 500 Hrs. 260°F 74,5 60672 x 10 6 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9			500 Hrs.	Êi	iÇ i. AÇ d			•		
KH / 500 Hrs. 260°F	967 RH / 500 Hrs. 260°F 87, 70 3,000 3,000 9,000				^ ¥	í, o	.1	O1 × 60.			
RH 500 Hrs. 260°F 80°. 5° 191,000 RH 500 Hrs. 260°F 80°. 5° 191,000 RH 500 Hrs. 260°F 84 66 1,000 RH 500 Hrs. 350°F 94 66 1,000 RH 500 Hrs. 350°F 89°. 5° 9,000 RH 500 Hrs. 350°F 89°. 5° 9,000 RH 500 Hrs. 350°F 89°. 5° 9,000 RH 500 Hrs. 350°F 86 5° 9,000 RH 500 Hrs. 350°F 86 5° 1,000 RH 1000 Rrs. RTD 94°. 5° 1,000 RH 1000 Rrs. RTD 94°. 5° 1,000 RH 1000 Hrs. RTD 86°. 5° 1,000 RH 1000 Hrs. RTD 94°. 5° 1,000 RH 1000 Rrs. RTD 94°. 5° 1,000 RH 1000 Rrs. 260°F 98 7° 5° RH 1000 Rrs. 350°F 90 7° 5° RH 1000 Rrs. 350°F 90 2° RH 1000 Rrs. 350°F 97 60 2° RH 1000 Rrs. 350°F 84°. 5° RH 1000 Rrs. 350°F 84°°F 86°F 8	9F. RH / 500 Hrs. 260°F 80.5 65 191,000 9F. RH / 500 Hrs. 260°F 83.6 65 191,000 9F. RH / 500 Hrs. 260°F 84.6 68 1,174,000 9F. RH / 500 Hrs. 350°F 94.6 1,000 9F. RH / 500 Hrs. 350°F 89.5 57 9,000 9F. RH / 500 Hrs. 350°F 89.5 57 9,000 9F. RH / 500 Hrs. 350°F 89.5 57 9,000 9F. RH / 1000 Hrs. RTD 94.5 57 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 94.5 75 1,000 9F. RH / 1000 Hrs. RTD 75.5 60 119,000 9F. RH / 1000 Hrs. 260°F 98 75 5,000 9F. RH / 1000 Hrs. 260°F 98 75 5,000 9F. RH / 1000 Hrs. 260°F 98 75 5,000 9F. RH / 1000 Hrs. 350°F 97.8 70 1,000 9F. RH / 1000 Hrs. 350°F 97.8 70 2,000 9F. RH / 1000 Hrs. 350°F 84.5 55 00 2,000 9F. RH / 1000 Hrs. 350°F 84.5 55 00 2,000 9F. RH / 1000 Hrs. 350°F 84.5 53 47,000 2.0 x 106 9F. RH / 1000 Hrs. 350°F 84.5 53 47,000 2.0 x 106				260°F	74.5	99	•	632 x 10°	82	7
RH	987 RH / 500 Hrs. 260°F 83 67 1,174,000 987 RH / 500 Hrs. 260°F 83 67 1,174,000 987 RH / 500 Hrs. 350°F 89, 66 1,000 987 RH / 500 Hrs. 350°F 89, 57 9,000 987 RH / 500 Hrs. 150°F 89, 57 9,000 987 RH / 1000 Hrs. RTD 84, 5 67 1,000 987 RH / 1000 Hrs. RTD 94, 5 75 1,000 987 RH / 1000 Hrs. RTD 84, 5 67 1,000 987 RH / 1000 Hrs. RTD 84, 5 67 1,000 987 RH / 1000 Hrs. RTD 84, 5 67 1,000 987 RH / 1000 Hrs. RTD 75.5 60 119,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 350°F 107 70 1,000 987 RH / 1000 Hrs. 350°F 107 70 1,000 987 RH / 1000 Hrs. 350°F 107 50 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000 987 RH / 1000 Hrs. 350°F 84, 5 55 60 2,000				3-097	7.60	٦٢.	3,000			
RH 500 Hrs. 260°F 83 67 1,174,000 RH 500 Hrs. 260°F 84 68 388,000 RH 500 Hrs. 350°F 94 66 1,000 RH 500 Hrs. 350°F 89.5 57 9,000 RH 500 Hrs. 350°F 89.5 57 9,000 RH 7 500 Hrs. 350°F 89.5 57 9,000 RH 7 1000 Hrs. RTJ 101 80 1,000 RH 7 1000 Hrs. RTJ 101 80 1,000 RH 7 1000 Hrs. RTJ 101 80 1,000 RH 7 1000 Hrs. RTJ 260°F 75 60 119,000 RH 7 1000 Hrs. 260°F 98 75 5,000 RH 7 1000 Rrs. 260°F 94 75 5,000 RH 7 1000 Rrs. 350°F 92.5 71 7,000 RH 7 1000 Rrs. 350°F 92.5 71 7,000 RH 7 1000 Rrs. 350°F 92.5 55 47,000 RH 7 1000 Rrs. 350°F 84.5 55 47,000	98 RH 500 Hrs. 260°F 83 67 1,114,000 98 RH 500 Hrs. 260°F 84 68 388,000 98 RH 7 500 Hrs. 350°F 102 65 1,000 98 RH 7 500 Hrs. 350°F 89.5 57 9,000 98 RH 7 500 Hrs. 350°F 86 55 1,000 98 RH 7 1000 Hrs. RTJ 101 80 1,000 98 RH 7 1000 Hrs. RTJ 101 80 1,000 98 RH 7 1000 Hrs. RTJ 260°F 75 60 119,000 98 RH 7 1000 Hrs. RTJ 88.5 75 60 119,000 98 RH 7 1000 Hrs. RTD 75.5 60 119,000 98 RH 7 1000 Hrs. RTD 75.5 60 119,000 98 RH 7 1000 Hrs. 260°F 98 75 - 1 x 100 98 RH 7 1000 Hrs. 260°F 98 75 - 1 x 100 98 RH 7 1000 Hrs. 260°F 98 75 - 1 x 100 98 RH 7 1000 Hrs. 260°F 98 75 - 1 x 100 98 RH 7 1000 Hrs. 350°F 92 50 2,000 98 RH 7 1000 Hrs. 350°F 92 60 2,000 98 RH 7 1000 Hrs. 350°F 84.5 55 47,000 98 RH 7 1000 Hrs. 350°F 84.5 55 47,000 98 RH 7 1000 Hrs. 350°F 84.5 55 47,000 98 RH 7 1000 Hrs. 350°F 84.5 55 47,000 98 RH 7 1000 Hrs. 350°F 84.5 55 47,000				260°F	80.3	63	191,000			
RH / 500 Hrs. 260°F 84 68 388,000 RH / 500 Hrs. 350°F 94 60 1,000 RH / 500 Hrs. 350°F 94 60 1,000 RH / 500 Hrs. 350°F 89.5 57 9,000 RH / 500 Hrs. 350°F 89.5 57 9,000 RH / 500 Hrs. 350°F 89.5 57 9,000 RH / 1000 Hrs. RTJ 101 80 1,000 RH / 1000 Hrs. RTD 94.5 75 1,000 RH / 1000 Hrs. RTD 84.5 67 119,000 RH / 1000 Hrs. 260°F 78 60 119,000 RH / 1000 Hrs. 260°F 96 75 1 1 x 10° RH / 1000 Hrs. 260°F 96 75 1 1 x 10° RH / 1000 Hrs. 260°F 96 75 1 2,000 RH / 1000 Hrs. 350°F 96 75 1 2,000 RH / 1000 Hrs. 350°F 96 75 1 2,000 RH / 1000 Hrs. 350°F 96 2,000 RH / 1000 Hrs. 350°F 97 5 1,000	98° RH / 500 Hrs. 260°F 84 66 388,000 98° RH / 500 Hrs. 350°F 94 60 1,000 98° RH / 500 Hrs. 350°F 89.5 57 9,000 98° RH / 500 Hrs. 350°F 89.5 57 9,000 98° RH / 500 Hrs. 350°F 89.5 57 9,000 98° RH / 500 Hrs. 350°F 89.5 57 9,000 98° RH / 500 Hrs. 870 100 1,000 98° RH / 1000 Hrs. 870 100 1,000 98° RH / 1000 Hrs. 870 1000 98° RH / 1000 Hrs. 260°F 98 75 1000 98° RH / 1000 Hrs. 260°F 98 75 1000 98° RH / 1000 Hrs. 350°F 107 70 1,000 98° RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 R 106 98° RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 R 106 98° RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 R 106				260°F	83	ı,	1,174,000			
RH / 500 Hrs. 350°F 94, 60 1,000 RH / 500 Hrs. 350°F 94, 60 1,000 RH / 500 Hrs. 350°F 89.5 57 9,000 RH / 500 Hrs. 350°F 89.5 57 9,000 RH / 500 Hrs. 350°F 86 55 1°C,000 RH / 1000 Rrs. RTD 94.5 75 1,000 RH / 1000 Rrs. RTD 94.5 75 1,000 RH / 1000 Rrs. RTD 75.5 60 119,000 RH / 1000 Rrs. RTD 75.5 60 119,000 RH / 1000 Rrs. 260°F 98 75 1 1 x 10°C RH / 1000 Rrs. 260°F 98 75 1 1 x 10°C RH / 1000 Rrs. 260°F 98 75 5,000 RH / 1000 Rrs. 350°F 94.5 77 7,000 RH / 1000 Rrs. 350°F 92.5 71 7,000 RH / 1000 Rrs. 350°F 84.5 55 47,000	98° RH / 500 Hrs.				260°F	ž	9	388,000			
RH / 500 Hzs. 350°F 94 60 1,000 RH / 500 Hzs. 350°F 89.5 57 9,000 RH / 500 Hzs. 350°F 89.5 57 9,000 RH / 500 Hzs. 350°F 86 55 1 000 RH / 1000 Hzs. RTD 94.5 75 1,000 RH / 1000 Hzs. RTD 94.5 75 1,000 RH / 1000 Hzs. RTD 75.5 60 119,000 RH / 1000 Hzs. RTD 75.5 60 119,000 RH / 1000 Hzs. 260°F 78 60 - 1 x 10° RH / 1000 Hzs. 260°F 98 75 - 1 x 10° RH / 1000 Rzs. 260°F 98 75 - 1 x 10° RH / 1000 Rzs. 260°F 98 75 - 1 x 10° RH / 1000 Rzs. 350°F 92.5 71 7,000 RH / 1000 Rzs. 350°F 92.5 71 7,000 RH / 1000 Rzs. 350°F 92.5 71 7,000 RH / 1000 Rzs. 350°F 92.5 55 47,000 RH / 1000 Rzs. 350°F 84.5 55 47,000	967 RH / 500 Hrs. 350°F 89.5 57 9,000 1,000 967 RH / 500 Hrs. 350°F 89.5 57 9,000 2,225 x 10 ⁶ 98.7 81 7 900 2,225 x 10 ⁶ 98.7 RH / 500 Hrs. 350°F 86 55 1 000 800 2,225 x 10 ⁶ 987 RH / 1000 Hrs. RTJ 88.5 75 1,000 800 987 RH / 1000 Hrs. RTJ 88.5 75 60 119,000 987 RH / 1000 Hrs. RTJ 84.5 67 1,000 987 RH / 1000 Hrs. RTJ 84.5 60 119,000 987 RH / 1000 Hrs. RTD 75.5 60 119,000 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Rrs. 350°F 92.5 71 7,000 997 RH / 1000 Rrs. 350°F 92.5 71 7,000 2,000 997 RH / 1000 Rrs. 350°F 84.5 55 47,000 2,000 997 RH / 1000 Rrs. 350°F 84.5 55 47,000 2,000 2,000 997 RH / 1000 Rrs. 350°F 84.5 55 47,000 2.0 x 10 ⁶ 997 RH / 1000 Rrs. 350°F 84.5 55 47,000 2.0 x 10 ⁶ 997 RH / 1000 Rrs. 350°F 84.5 55 47,000 2.0 x 10 ⁶			500 Hrs.	350 F	102	e E	1,000			
RH / 500 Hre. 350°F 89.5 57 9,000 2.25 x 10 8	987 RH / 500 Hrs. 350°F 89.5 57 9,000 2.25 x 10 ⁶ 987 RH / 500 Hrs. 350°F 83 53 - 2.25 x 10 ⁶ 987 RH / 500 Hrs. 350°F 83 53 - 2.25 x 10 ⁶ 987 RH / 1000 Hrs. RTD 94.5 75 1,000 98° RH / 1000 Hrs. RTD 94.5 75 1,000 98° RH / 1000 Hrs. RTD 84.5 67 6.000 98° RH / 1000 Hrs. RTD 75.5 60 119,000 987 RH / 1000 Hrs. 260°F 78 60 119,000 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Hrs. 260°F 98 75 - 1 x 10 ⁶ 987 RH / 1000 Hrs. 260°F 98 75 - 2,000 987 RH / 1000 Hrs. 260°F 94 75 5,000 987 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2,000 2,000 47.00 Hrs. 350°F 84.5 55 47,000 2.00 x 10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.00 x 10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.00 x 10 ⁶			500 Hrs.	350°F	\$	9	1,000			
RH / 500 Hrs. 350°F 83 5322 x 10 RH / 500 Hrs. 350°F 86 55 1°C00 C22 x 10 RH / 1000 Hrs. RT	96° RH / 500 Hrs. 150°F 83 5323 x 10 68°Z RH / 500 Hrs. 150°F 86 55 1°C 90023 x 10 98°Z RH / 1000 Hrs. RTD 94.5 75 1,000 98°Z RH / 1000 Hrs. RTD 84.5 67 1,000 98°Z RH / 1000 Hrs. RTD 84.5 67 1000 Hrs. RTD 75.5 60 119,000 98°Z RH / 1000 Hrs. 260°F 78 60 119,000 - 1 x 10°C 98°Z RH / 1000 Hrs. 260°F 98 75 - 1 x 10°C 98°Z RH / 1000 Hrs. 260°F 98 75 - 1 x 10°C 98°Z RH / 1000 Hrs. 260°F 98 75 - 1 x 10°C 98°Z RH / 1000 Hrs. 260°F 98 75 - 1 x 10°C 98°Z RH / 1000 Hrs. 260°F 98 75 - 1,000 98°Z RH / 1000 Hrs. 260°F 98°Z RH / 1000 Hrs. 350°F 10°Z 80°Z 80°Z 80°Z 80°Z 80°Z 80°Z 80°Z 8			500 Hrs.	350°F	89.5	57	000.6	9.	;	
RH / 500 Hrs. RT: 101 80 1,000 Rrs. RT: 100 88.5 17 900 Rrs. RT: 101 80 1,000 Rrs. RT: 100 88.5 75 1,000 Rrs. RT: 100 88.5 75 1,000 Rrs. RT: 100 88.5 70 6,000 Rrs. RT: 1000 Rrs. RT: 1000 Rrs. RT: 1000 Rrs. 260°F 78 60 119,000 Rrs. 260°F 91 70 - 1 x 100 Rrs. 260°F 94 75 5,000 Rrs. 350°F 107 70 1,000 Rrs. 350°F 107 70 1,000 Rrs. 350°F 95 60 2,000 Rrs. 350°F 95 65 2,000 Rrs. 350°F 94,5 55 47,000 Rrs. 350°F 94,5 55 47,000	987 RH / 500 Hrs. KT3 101 80 1,000 987 RH / 1000 Hrs. KT3 101 80 1,000 987 RH / 1000 Hrs. KT3 86.5 70 6,000 987 RH / 1000 Hrs. KT3 86.5 67 119,000 987 RH / 1000 Hrs. 260°F 78 60 119,000 987 RH / 1000 Hrs. 260°F 98 75 1 10,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 98 75 5,000 987 RH / 1000 Hrs. 260°F 94 72 5,000 987 RH / 1000 Hrs. 260°F 94 72 5,000 987 RH / 1000 Hrs. 350°F 107 70 1,000 987 RH / 1000 Hrs. 350°F 97 500 2,000 987 RH / 1000 Hrs. 350°F 97 500 2,000 987 RH / 1000 Hrs. 350°F 84.5 55 47,000 987 RH / 1000 Hrs. 350°F 84.5 55 47,000 987 RH / 1000 Hrs. 350°F 84.5 53 47,000 987 RH / 1000 Hrs. 350°F 84.5 53 47,000 987 RH / 1000 Hrs. 350°F 84.5 53 47,000				350'F	63	53	•	7.25 \$ 10	<u>.</u>	
RH / 1000 Hrs.	967 RH / 1000 Hrs. KT 1 101 80 1,000 967 RH / 1000 Hrs. KT 1 86.5 75 1,000 967 RH / 1000 Hrs. KT 1 84.5 75 1,000 967 RH / 1000 Hrs. KT 1 84.5 67 6.0 119,000 967 RH / 1000 Hrs. 260°F 78 60 119,000 967 RH / 1000 Hrs. 260°F 98 75 1 1000 Hrs. 260°F 98 75 5,000 967 RH / 1000 Hrs. 260°F 94. 75 5,000 967 RH / 1000 Hrs. 260°F 94. 75 5,000 967 RH / 1000 Hrs. 260°F 94. 75 5,000 967 RH / 1000 Hrs. 260°F 94. 75 5,000 967 RH / 1000 Hrs. 350°F 97 8 60 2,000 967 RH / 1000 Hrs. 350°F 97 80 2,000 2,000 967 RH / 1000 Hrs. 350°F 97 84.5 55 60 2,000 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 60 2,000 2,000 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 60 2,000 2,000 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 60 2,000 2,000 2,000 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 60 2,00				3507F	%	25	000 J.			
RH / 1000 RTS. RTD 94.5 75 1,000 RH / 1000 RTS. RTD 88.5 70 6,000 RH / 1000 RTS. RTD 84.5 67 - RH / 1000 RTS. RTD 75.5 60 119,000 RH / 1000 RTS. 260°F 78 60 - RH / 1000 RTS. 260°F 98 75 - RH / 1000 RTS. 260°F 98 75 - RH / 1000 RTS. 260°F 98 75 - RH / 1000 RTS. 260°F 99 75 - RH / 1000 RTS. 260°F 96 75 - RH / 1000 RTS. 260°F 96 75 - RH / 1000 RTS. 260°F 96.5 71 7,000 RH / 1000 RTS. 350°F 107 70 1,000 RH / 1000 RTS. 350°F 95 60 2,000 RH / 1000 RTS. 350°F 95 60 2,000 RH / 1000 RTS. 350°F 84.5 55 47,000	96° RH / 1000 HTW. RTD 94.5 75 1,000 98° RH / 100° "FE. RTD 86.5 70 6,000 98° RH / 100° HTF. RTD 86.5 67 - 98° RH / 1000 HTF. RTD 75.5 60 119,000 98° RH / 1000 HTF. 260°F 78 60 - 98° RH / 1000 HTW. 260°F 98 75 - 98° RH / 1000 HTW. 260°F 98 75 - 98° RH / 1000 HTW. 260°F 99 75 - 98° RH / 1000 HTW. 260°F 99 75 - 98° RH / 1000 HTW. 350°F 107 70 1,000 98° RH / 1000 HTW. 350°F 97 50 2,000 98° RH / 1000 HTW. 350°F 84.5 55 47,000 98° RH / 1000 HTW. 350°F 84.5 55 47,000 98° RH / 1000 HTW. 350°F 84.5 55 - 98° RH / 1000 HTW. 350°F 84.5 55 - 2.0 R 106				C 13	101	80	1,000			
RH / 10°C "-s. RT) 88.5 70 6,000 RH / 10°CO HFF. RTD 84.5 67 - RH / 10°CO HFF. 260°F 78 60 119,000 RH / 10°CO HFF. 260°F 98 75 - RH / 10°CO RFF. 260°F 94 75 5,000 RH / 10°CO RFF. 260°F 94 75 5,000 RH / 10°CO RFF. 260°F 94. 75 5,000 RH / 10°CO RFF. 260°F 94. 75 5,000 RH / 10°CO RFF. 350°F 10°7 70 1,000 RH / 10°CO RFF. 350°F 10°7 70 1,000 RH / 10°CO RFF. 350°F 95 60 2,000 RH / 10°CO RFF. 350°F 95 60 2,000 RH / 10°CO RFF. 350°F 95 55 47,000	98° RH / 10° """. RT7 88.5 70 6,000 98° RH / 10° """. RT7 84.5 70 6,000 98° RH / 10° Hrs. RT7 84.5 67 119,000 98° RH / 10° Hrs. 260°F 78 60 1 x 10° 98° RH / 10° Rrs. 260°F 98 75 1 x 10° 98° RH / 10° Rrs. 260°F 98 75 1 x 10° 98° RH / 10° Rrs. 260°F 98.5 71 7,000 98° RH / 10° Rrs. 350°F 10°7 70 1,000 98° RH / 10° Rrs. 350°F 92 60 2,000 98° RH / 10° Rrs. 350°F 84.5 55 47,000 2.0 x 10° 98° RH / 10° Rrs. 350°F 84.5 53 47,000 2.0 x 10° 98° RH / 10° Rrs. 350°F 84.5 53 47,000 2.0 x 10°			1000 Hrs.	KTD	94.5	7.5	1,000			
RH / 1000 HFF. RTD 64.5 0/ 119,000 RH / 1000 HFF. 260°F 78 60 119,000 RH / 1000 HFF. 260°F 78 60 - 119,000 RH / 1000 HFF. 260°F 94 75 - 1 x 100 RH / 1000 RFF. 260°F 94 75 5,000 RH / 1000 RFF. 260°F 92.5 71 7,000 RH / 1000 HFF. 350°F 107 70 11,000 RH / 1000 HFF. 350°F 92 60 2,000 RH / 1000 HFF. 350°F 84.5 55 47,000	987 RH / 1000 Hrs. RTD 75.5 60 119,000 987 RH / 1000 Hrs. 260°F 78 60 - 2.01×10 ⁶ 987 RH / 1000 Rrs. 260°F 98 75 - 1×10° 987 RH / 1000 Rrs. 260°F 94 72 5,000 987 RH / 1000 Rrs. 260°F 92.5 71 7,000 987 RH / 1000 Hrs. 350°F 107 70 1,000 987 RH / 1000 Hrs. 350°F 92 60 2,000 987 RH / 1000 Hrs. 350°F 92 60 2,000 987 RH / 1000 Hrs. 350°F 92 60 2,000 987 RH / 1000 Hrs. 350°F 92 60 2,000 2.01×10 ⁶ 987 RH / 1000 Hrs. 350°F 91.5 53 47,000			100	E	288.5	6:	9,000			
RH / 1000 Hrs. 260°F 78 60 - 2.01x10 ⁶ RH / 1000 Hrs. 260°F 78 60 - 1 x 10 ⁶ RH / 1000 Hrs. 260°F 94 75 - 1 x 10 ⁶ RH / 1000 Hrs. 260°F 94 75 5,000 RH / 1000 Hrs. 260°F 92.5 71 7,000 RH / 1000 Hrs. 350°F 107 70 1,000 RH / 1000 Hrs. 350°F 77 50 2,000	967 RH / 1000 Hrs. 260°F 78 60 - 2.01×10 ⁶ 967 RH / 1000 Hrs. 260°F 98 75 - 1 × 10 ⁶ 967 RH / 1000 Hrs. 260°F 94 75 - 1 × 10 ⁶ 967 RH / 1000 Hrs. 260°F 94 75 5,000 967 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 107 70 2,000 967 RH / 1000 Hrs. 350°F 92 60 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 × 10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 20°C 2.0 × 10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 20°C 2.0 × 10 ⁶			1050 Hrs.	Ê	3	6				
RH / 1000 Hrs. 260°F 78 60 - 2.01×10 ⁶ RH / 1000 Hrs. 260°F 98 75 - 1×10 ⁶ RH / 1000 Hrs. 260°F 94 75 5,000 RH / 1000 Hrs. 260°F 94. 72 5,000 RH / 1000 Hrs. 260°F 92.5 71 7,000 RH / 1000 Hrs. 350°F 107 70 1,000 RH / 1000 Hrs. 350°F 92 60 2,000 RH / 1000 Hrs. 350°F 84.5 55 47,000	967 RH / 1000 Hrs. 260°F 78 60 - 2.01×10 ⁶ 967 RH / 1000 Hrs. 260°F 98 75 - 1 × 10° 967 RH / 1000 Rrs. 260°F 94 75 5,000 967 RH / 1000 Rrs. 260°F 94. 72 5,000 967 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 92 60 2,000 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 × 10° 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 × 10° 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 × 10° 2				RTD	(3.3	3	773	,		
RH / 1000 Hrs.	967. RH / 1000 Hrs. 3°F 91 70 - 1 x 10° 957. RH / 1000 Hrs. 260°F 98 75 - 9,000 967. RH / 1000 Hrs. 260°F 92.5 71 7,000 967. RH / 1000 Hrs. 350°F 107 70 1,000 967. RH / 1000 Hrs. 350°F 92 60 2,000 2,000 967. RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10° 967. RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10° 967. RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10°				260°F	78	9	•	2.01 x 10°	81.7	
RH / 1000 Hrs. 260°F 98 75 - RH / 1000 Hrs. 260°F 94, 72 5,000 RH / 1000 Hrs. 260°F 92.5 71 7,000 RH / 1000 Hrs. 350°F 107 70 1,000 RH / 1000 Hrs. 350°F 92 60 2,000 RH / 1000 Hrs. 350°F 84.5 55 47,000	967 RH / 1000 Hrs. 260°F 98 75 - 967 RH / 1000 Hrs. 260°F 94, 72 5,000 967 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 92 60 2,000 967 RH / 1000 Hrs. 350°F 97 50 - 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 2.26 x 10 ⁶				ጋ * F	91	2	1	1 x 10°	19.	_
RH / 1000 RFE. 260°F 94, 72 5,000 RH / 1000 RFE. 260°F 92.5 71 7,000 RH / 1000 RFE. 350°F 107 70 1,000 RH / 1000 RFE. 350°F 92 60 2,000 RH / 1000 RFE. 350°F 94.5 55 47,000	967 RH / 1000 RTS. 260°F 94, 72 5,000 967 RH / 1000 RTS. 260°F 92.5 71 7,000 967 RH / 1000 RTS. 350°F 107 70 1,000 967 RH / 1000 RTS. 350°F 92 60 2,000 967 RH / 1000 RTS. 350°F 84.5 55 47,000 967 RH / 1000 RTS. 350°F 84.5 55 47,000				260°F	86	7.5	•			
RH / 1000 HTW. 260°F 92.5 71 7,000 EH / 1000 HTW. 350°F 107 70 1,000 EH / 1000 HTW. 350°F 92 60 2,000 EH / 1000 HTW. 350°F 94.5 55 47,000 EH / 1000 HTW. 350°F 84.5 55 47,000	967 RH / 1000 Hrs. 260°F 92.5 71 7,000 967 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 92 60 2,000 967 RH / 1000 Hrs. 350°F 77 50 - 2,261×10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 967 RH / 1000 Hrs. 350°F 81.5 53 - 2.0 × 10 ⁶				260°F	đ	72	°,000			
ER / 1000 Hrs. 350°F 107 70 1,000 RR / 1000 Hrs. 350°F 92 60 2,000 RR / 1000 Hrs. 350°F 77 50 . 2,261x10 RR / 1000 Hrs. 350°F 84.5 55 47,000	967 RH / 1000 Hrs. 350°F 107 70 1,000 967 RH / 1000 Hrs. 350°F 92 60 2,000 967 RH / 1000 Hrs. 350°F 77 50 - 2,261x10 ⁶ 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10 ⁶ 967 RH / 1000 Hrs. 350°F 81.5 53 47,000 2.0 x 10 ⁶				260°F	92.5	71	7,000			
RH / 1000 Hrs. 350°F 92 bd 2,000 2,000 RH / 1000 Hrs. 350°F 77 50 2,261x10 Hr / 1000 Hrs. 350°F 84.5 55 47,000	967 RH / 1000 Hrs. 350°F 92 bd 4,000 2,000 967 RH / 1000 Hrs. 350°F 17 50 - 2,261x10 967 RH / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10 96° RH / 1000 Hrs. 350°F 81.5 53 - 2.0 x 10			1000 Hrs.	350°F	107	0.5	1,000			
NA 1000 HER. 350°F 84.5 55 47,000	967 RM / 1000 Hrs. 350°F 84.5 55 47,000 2.0 x 10 6 96° RM / 1000 Hrs. 350°F 81.5 53			1900 Hrs.	350-1	7.6	2 5	7,000	2.261 x 106	78.	و.
	96" RH / 1000 Hrs. 350"F 81.5 53 - 2.0 x 10			1000	350°F	3	55	47,000			

TABLE WI FALLOUF PROPOSITIS SUPERNY - SARGEO SIGN CONTOUR TEACHER OF ROSIFES

地震 多川川 明 田本 昭 1860年 1860年 1863年 1868年 1

Spi : faen	I :tckness		PALOR CONDITIONING	1 2 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	Stress Level	Cycles to Fatlure	Cycles Applad without Pallure	Resident	
# Ber	(Plies) (In.	Orientat. on	Type Juration	£	(1ºulr) (ksi)	(cycles)	(cycles)	(jes t)	Comment
\$-38E-1	90) 1 1 1 1 1 1	3 67 89 0	Themo-Hunfdite Cecle	9		1.000			Tab Fatlure
41.28E-6	900 C	, ; :		GIA	88 70				Smediate Tab Failure
460,18	4	:		RT.)		3,000	•		Tab Failure
MIN. 88 - 8	9 TC	:		RID			2.515 × 100	7.2	Teb Failure
6-18: TX	9-2:	=	Thermo-Humidity Cycle	€		•	2.421 x 10	8.9	Tab Failure
1-196.50	,	Ξ	elas (dire feels	260°F		1 999	,		
×11.364-2		=		260°F	3		2.5 x 10 ⁶	81.6	
VI.364-3	•	:	The rac Hunddity Ovele	260 °F		•	2.226 × 10	82.7	
W1136A-4	5-2-2 m	=		260°F	84.5 72	7,300			
Y1.364-5	•	=	Thermo-Humidity Cycle	360°F		s, 90 0			
41.174-1	•	:	The Table (see)	35038		1,000			
7.77-5	•	:		350 F	104.5 65	3,000	•		
5-7,6	(C)	:		350 F	88.5 55	•	2.361 x 10°	83.0	
1-7,5318	•	:	Thermo-Manddity Cycle	350°F	%;s 60	900			
M137.k-5	٠	:	Thermo-Hamidity Cycle	350°F		2,000			
X11288-10	670°C - 6	=	Acc. Wthrng.	£		2.000			Tab Failure
M1128E-11	c	**	Acc. wthrug.	G		179,060			
M11294-1	0.0	:	Acc. Wthrng.	e E		2,000			
M11294-2	و ټ	E :	Acc. Wthrng.	G	17.5 72	1,000	90.		Tab Pailure
M1129A~3	3		Acc. Sthrag.	1		•	01 x co.7	63.5	
M136A-6	6 - 0.048	Ξ	Acc. Wthrng.	7.09Z		3,000			
M136A-7	٠	Ξ	Acc. Wthrng.	760°F		9,000	•		
V.136A-8	2 + 0.047	:	Acc. Wthrng.	260 °F	3	•	2.186 x 10	78.3	Teb Failure
M1136A-9	٠	Ξ	Acc. Wth:ng.	260°F		•	$2.667 \times 10_{\rm g}$	81.7	Tab Failure
¥1136A-10	8	Ξ	Acc. Wthrmg.	3e0 aL		•	2.398 x 10"	78.2	
	875	=	Acc. Ethros.	350°F	91.5 60	34,000	•		
M1337A-7	1975 T 55	:		350 'F		•	2.178 x 10"	102.4	Tab Failure
M1137A-8	(ac)	÷		350 'F		2,000			
K1137A-9	73.0	= 1	Acc. Wthrng.	350'F	8 8	88.			*** F-4 1
X2.37A-1	3.5.0	06/03132 0/30	Acc. Wthrng.	₹. 2.2.		3,5			IND FALIDITE

TABLE XVI - SAFT COLOR CONCESS - CON

VIII VIII		Orientation	Tupe Duration	Ė	Stress Level (T ² ult)	En Fallure (cycles)	without Faliare (cycles)	Residual Strength (ksi)	Comment
2212 <u>3</u> 2	÷ - (.03 ⁻	-	2007 / 500 Brs.	RTD		ı	2.012 x 10 ⁶	174.8	
210 25	6.03	κ		F	82.5 140	75,000			
ile ar	•	~,	Š.	2		2°000			
5 gray	. C.037	• 6		Ê	67 730 750 750 750 750 750 750 750 750 750 75	7,000 7,000			Tab Area Failure Teb Area Failure
जुमा . संस्थ	•		260 F / 300 Hrs.	· .		11,000			I TO VIEW COLLUS
E);	E - 0.033	7.		c T		17,000	4		
	£ . 3.03	Ċ.		Ē			10.09 x 10°	175.9	
	€ - 0.03	c		<u>.</u>		345,000			
¥1104A-17	6 - 2,037	С.	200	E C	62.5 160	1,000			Tab Failure
5 .7	e - [.03]	ċ	350 F / 500 Hrs.	Ē		2,000			Tab Failure
6T-V9017	-£.'.' 3	, c	267 F / 500 Cyc.	C 13		•	2.5 × 10 ⁶	181.2	Tab Failure
W1105A-20	t - 0.03	c	260°F / 500 Cyc.	Ę		•			Immediate Failure
411068-1	é - 0.03 ⁻	C.,.	200	เม	96.5 140	2,000			Tab Failure
V1106B-2	6 - 0.03°	· · · · ·	, 20 20	Û		•			Immediate Tab Failure
3	6 - 0.033	ċ.	260°F / 500 Cyc.	KT.		3,000			Tab Failure
,	6 - 0.33	Ċ	240°F / 1000 Cyc.	C L		•	5.253x106	194.3	
5-290EEX	6 - 0, 33		260°F / 1000 fyc.	C Dy	85 140	2,000	•		
	5	O	260°F / 1000 Cyc.	12		•	7.607x10°	168.3	
411(6E-3	6 - 2,336	C.	260"F / 1000 Cyc.	Ę		3,000	·c		Tab Failure
6-890EE	ı	 C	260°F / 1000 Cyc.	RT.)		•	2.436×10	174.9	Tab Failure
6-89012	6 - 0.030	ن.	350"F / 500 Cyc.	C L	60.5 110	9,000	7		
C.	6 - 5.03	:0	350°F / 500 Cyc.	612	63.5 100	•	2.653 x 10 ^E	177	Tab Failure
ij	1	ر د	350*F / 500 Cyc.	Ē		95,000	•		
	6 - 2.33	; o	350'F / 500 Cyc.	£;			2.061 x 10°	177	
Ç	F	့်ဝ	350'F / 500 Cyc.	CT3		•	•		Immediate Tab Failure
71-	6 - 0.03-	ن۔	350°F / 1000 Cyc.	RTD		•	2.93 x 10 ⁶	172.6	
2.5	6 - 0.036	ů	/ 1000	e E	81 125	900,4			Tab Failure
ا تو ا د	6 - 0,03	د د	350°F / 1000 Cyc.	e i		88			Tob Endlines
ts o	•	ບ ເ	30°F / 1000 Cyc.	T.		8			Tab Failure

TABLE XVI FALIGIE PROPERTIES SUNMARY - VANCOUS TE PROMOTE COMPOSITES

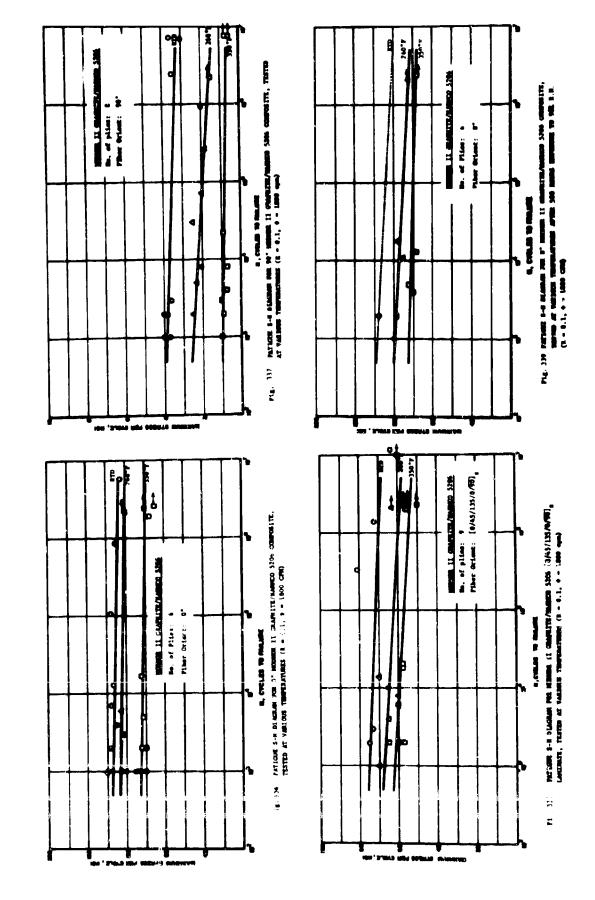
75 115 62 125 88.5 135 88.5 135 91.5 140 70.5 110 80 140 86.5 135 47 75 69.5 110 69.5 110	260°F / 1000 Cyc. 350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Hrs.		ာ ပီဝီစီ
69.5 110 1,000 63 100 64,000 66 105 66,000 67.5 107 8,000	350°F 350°F 350°F 350°F	500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs.	350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Cyc.
92 145 2,006 88.5 140 - 85.5 135 15,000 84 133 7,000	350 °F 350 °F 350 °F	350*F / 500 Cyc. 350*F 350*F / 500 Cyc. 350*F 350*F / 500 Cyc. 350*F 350*F / 500 Cyc. 350*F	/ 500 Cyc. / 500 Cyc. / 500 Cyc. / 500 Cyc.
82.5 140 3,000 79.5 135 209,000 81 136 1,000 78.5 133 59,000	350 T 350 T 350 T 350 T		350°F / 1000 Cyc. 350°F / 1000 Cyc. 350°F / 1000 Cyc. 350°F / 1000 Cyc. 350°F / 1000 Cyc.

TABLE XV: FAFIG: PROPERTIES STOCKER: - IVE SLEEK

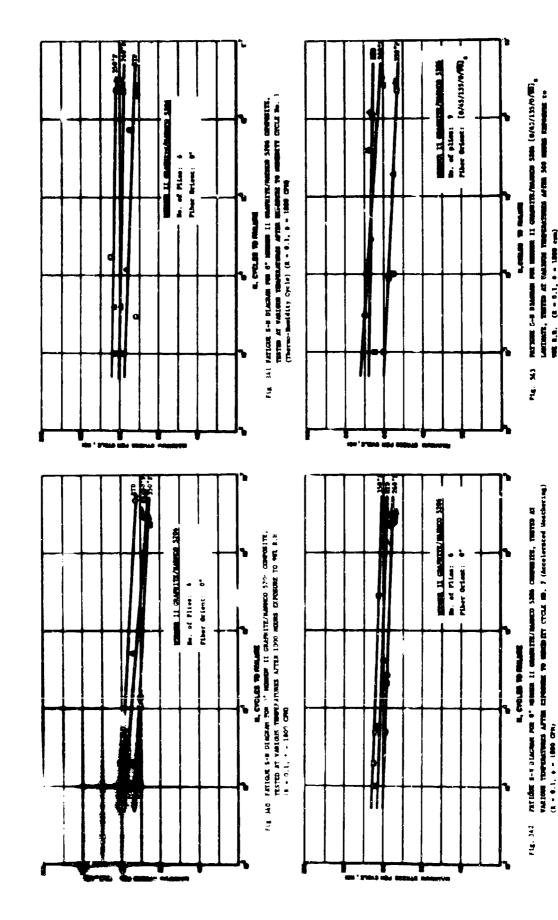
Spictmen Namber	Thichness (71ies) (In.)	Orientation	Falos conditioning	17 T T T T T T T T T T T T T T T T T T T	Stress Level (I'ult) (ksi)	Cycles to Failure (cycles)	Cycles Applied without Failure (cycles)	Residual Strength (ks1)	Comment
M. 308-1	6 - 0.047	[0//5/135/0/90]	88	E	87.5 75	1,000	901 - 300 -	8	
M1.308-2	670		260°F / 500 Hrs.	8 6	_		2.15 x 206	81.1	
M1.308-4	9	ž.	8	£					Immediate Tab Failure
M. 308-5	8 - 0.048	Ē.	260°F ' 500 Hrs.	€		2,0001			Tab Failure
M1 208-6	9 - 0.048	å		£	128 90	•			Immediate Failure
H1.308-7	840.0 - 6	į	28	£		1,000	9	;	
M11308-6	670.0 - 6	٤	× 3	£		. ;	2.17 x 19 ⁻	9 8	
M11308-9	2 - 0.047	- :	8	£.	107 75	3 800	901 - 117 6	7	
M11308-10	6 - 0.049	-	350"F / 500 Hrs.	g II		•	01 X 619.7	•	
M1131A-1	9 - 0.348	÷	250°F / 500 Cyc.	0. 11		000 . 6			
M1131A-2	840.0 - 6	-	250*F / 500 Cyc.	d D		65,000			Tab Failure
M11314-3	6 0.047	<u> </u>	260°F / 500 Cyc.	Ē		117,000	•	;	:
K1131A-4	870.0 - 6	_	8	e E	73.5 648		2.495 x 10	50.5	Tab Failure
M1314-5	9 - 0.048	٤	8	e			2.626 x 10	2.9	
A-4171174	8 0.048		250 °F / 1000 Cyc.	£	87.5 70	21,000,			Tab Failure
W11311.7	170	è	260°F / 1000 CVC	E		2,282,x10°			Tab Failure
M1111A-8				E		18,000			Tab Fallure
X11314-9	800	113	200 F / 1000 Cyc.	£	82.5 66				Tab Failure
M1131A-10	9 - 0.049	E	260*F / 1000 Cyc.	£		2,000			Teb Failure
W11304-9	570.0	\$	350°F / 500 Cvc.	0.0	116		•		Immediate Failure
W11101-9		2		E	_		7.58 × 10°	87.4	
M133A-10	9.0.0	=		£	97 75	164,000			
K1112	370°0 - %	÷		e		1,000			
M1313-2	540.0 - 6	=	350*F / 500 Cyc.	e		27,000	•		
M1118-1	70.0	85	350*F / 1000 Cyc.	£	76.5 60	•	2.076 x 10 ⁰	81.8	
4-81014	60.0	¥	900	E		20,000	•		
Z-11.11	870.0	=	_	E	83 65	•	2.412×10^{2}	, Se	Tab Failure
M11313-6	0.048	=	0001	E		3,000			
H1131B-7	970.0	=	900	E		239,000			

TAK F. XVI. - OVE GER POLITIKE OF STOWARY - CORPOSITES

Specimen Rumber	Thickness (Plies) (In.)	Orientation	PRICE COMDITIONING Type Euration	T E	Stress Level (Toult) (ksi)	.e.1	Cycles to Failure (cycles)	Cyc es App.led without Fai.ure (cycles)	Residual Strength (kal)	Summer
M11-0A-1	78 G	s (06/0/581.35/0)	260°F / 500 Hrs.	280.7	*6 8	02	7,000	931 - 106		Tab Failure
M11-04-3	6500 1 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6		8 8 8	260 1	83.5 81	. 	7, 200 1, 000 1, 000 1, 000	01 × 167.7	0	Tab Failure
H11-0A-6 H11-0A-6			260*F / 500 Cyc. 260*F / 500 Cyc.	260°F 260°F	32.5	8 2 5 5	000	90		Tab Fallure Tab Area Failure
M.1.0A-9	10 10 10 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	. = 1		260°F 260°F 760°F	72.5	6.3	7.000	2.5x10 2.6x10	81.3	
1 - 68 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 -			260°F / 1000 Cyc. 260°F / 1000 Cyc. 260°F / 1000 Cyc. 260°F / 1000 Cyc.	260°F 260°F 260°F 260°F	72 87 80.5	58 70 65 63	1,021,000			Tab Failure
X	() () () () () () ()		000	350°F 350°F 350°F	75.5 83.5 77. 74.5		2, 6, 600 10, 10, 10, 10, 10, 10, 10, 10, 10, 10,	2.015 x 10 ⁶	74.5	
01- 8 -11			350°F / 500 kfs. 350°F / 500 Cyc. 350°F / 500 Cyc. 350°F / 500 Cyc. 350°F / 500 Cyc.	350°F 350°F 350°F	20.5 20.5 20.5 20.5 20.5		2,000	4.427 x 10 ⁶	<u>`</u> .	Immediate Tab Failure
		*(<u>96</u> /0/\$\$1)'\$7/0]	330°F / 1000 Cyc. 330°F / 1000 Cyc. 350°F / 1000 Cyc. 350°F / 1000 Cyc.	350.F 350.F 350.F 350.F	3 8 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3	6 7 6 5 5 6 5 6 5 6 5 6 5 6 5 6 5 6 5 6	1,000 2,000 10,000	2.337 x 10 ⁶	85.5	Tab Failure Tab Failure Tab Failure Immediate Tab Failur



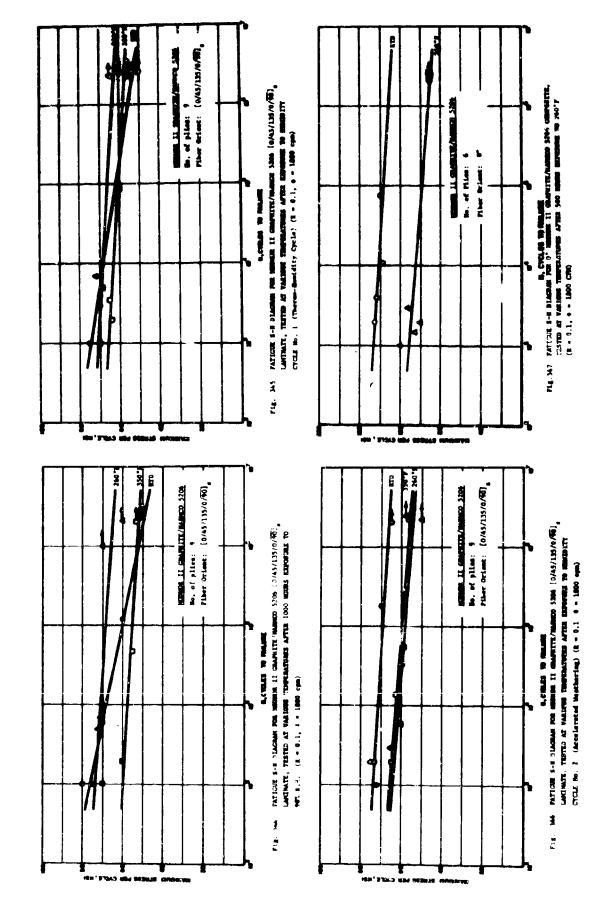
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MR B.R. (R - 0.1, 0 - 1500 cym.)

観光神道が振ぶら 生間の水動産業の自合を含みないます。半点の1900年まで、1000年に

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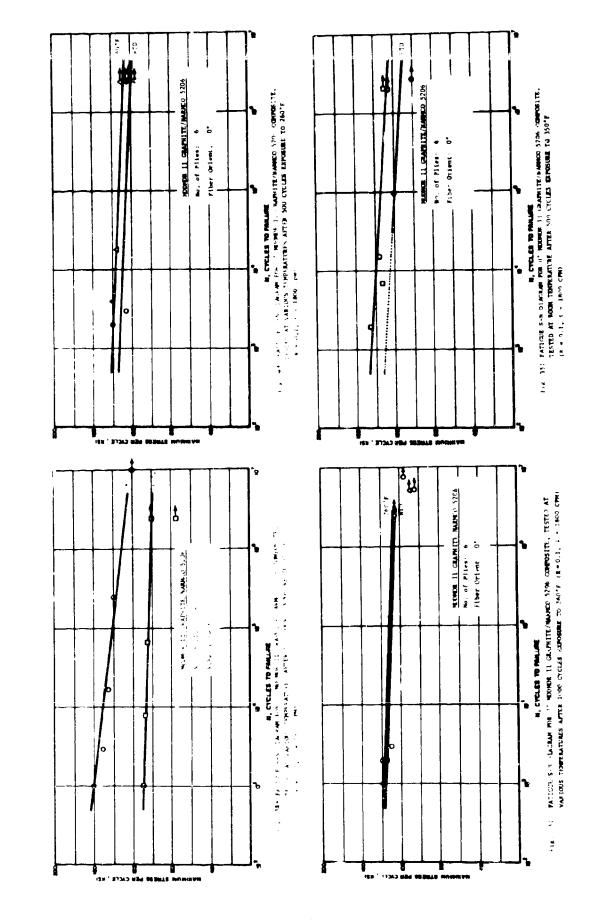


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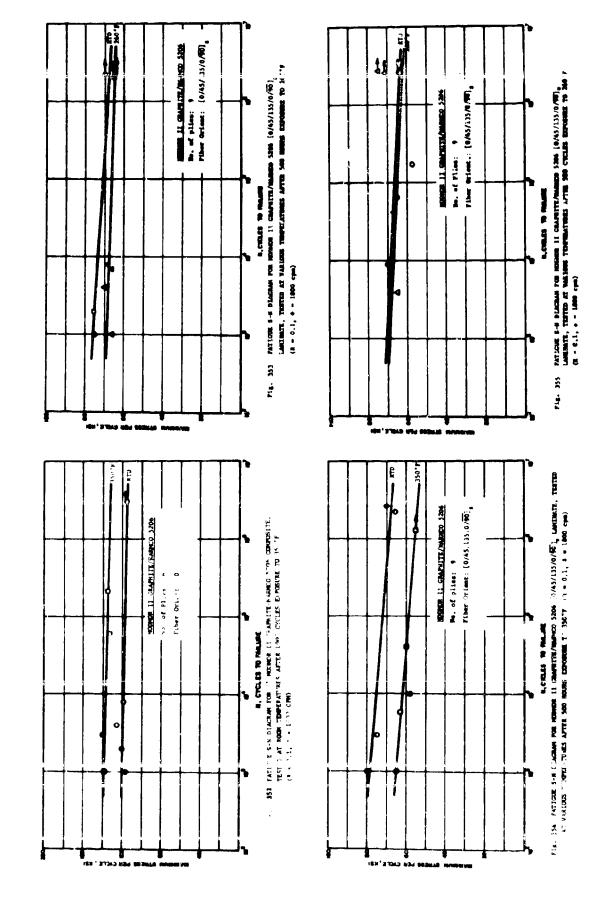
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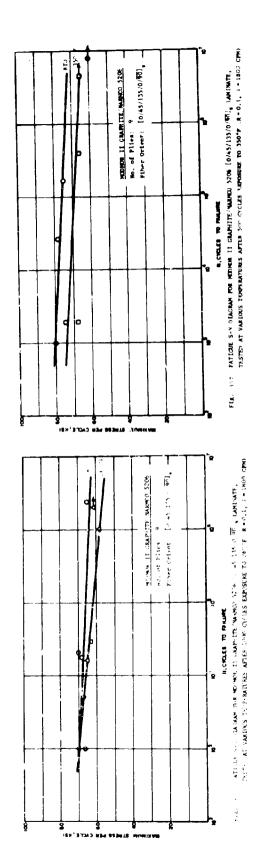
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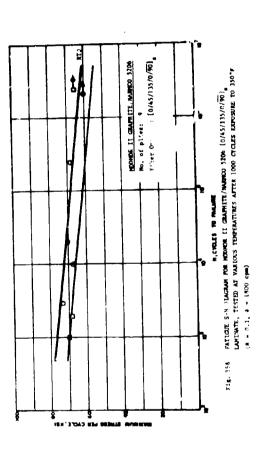


TABLE XV.1.1 CREEP AND STRESS RUFTURE PROPERTIES STANDARY - NARKO 5206/JRD/40R [1 CRAPLITE COMPASTTES

Stress Leve: Failure Failure (7°ult) (keil) (Hours) (H							1			Time	Applied	
111 6 0.037	Specimen	Trick (Plies)	kness) (In.)	Orientation	Tipe	ONDITIONING Duration	Temp.	Stress (X ⁰ ult	level (KB1)	Failure (Hours)	Feilure Feilure (Hours)	Comment
112 6 0,032 3° None - 260°F 85 128 - 1002 1.4 6 0,032 3° None - 260°F 87 112 - 1002 1.5 6 0,032 3° None - 260°F 80 130 - 1002 1.6 6 0,033 0° None - 260°F 90 135 - 1007 1.7 6 0,033 0° None - 260°F 90 135 - 1003 1.8 6 0,033 0° None - 260°F 90 135 - 1003 1.9 6 0,033 0° None - 260°F 90 135 - 1003 1.0 6 0,033 0° None - 260°F 90 135 - 1003 1.0 7	K11078-11	9	0.037	<u>د</u> .	None	,	260°F	æ	120		1010	
1002 1002 1002 1002 1002 1002 1002 1003	F11078-12	9	0,032	نۍ	None		260°F	85	128	•	1001	
-14 6 0.032 2° None - 260°F 80 120 - 1007 -15 6 0.034 0° None - 260°F 90 135 - 1007 -16 6 0.037 0° None - 260°F 90 135 - 1003 -17 6 0.037 0° None - 260°F 90 135 - 1003 -18 6 0.037 0° None - 260°F 90 135 - 1003 -20 6 0.033 0° None - 260°F 90 135 - 1011 -21 6 0.033 0° None - 260°F 90 135 - 1011 -22 6 0.033 0° None - 260°F 90 135 - 1011 -23 6 0.033 0° None - 260°F 90 135	M11078-13	ص	0.032	, C	Mene	1	260°F	7.5	112	•	1002	
-15 6 0.034 3° None - 260°F 90 135 - 1007 -16 6 0.033 0° None - 260°F 90 135 - 1003 -17 6 0.033 0° None - 260°F 90 135 - 1003 -18 6 0.034 0° None - 260°F 90 135 - 1003 -20 6 0.033 0° None - 260°F 90 135 - 1011 -21 6 0.033 0° None - 260°F 90 135 - 1011 -22 6 0.033 0° None - 260°F 90 135 - 1011 -23 6 0.033 0° None - 260°F 90 137 1011 -24 6 0.033 0° None - 260°F 90 135 1011 -25 6 0.033 0° None - 260°F 90 135 1011 -26 6 0.033 0° None - 260°F 90 135 1011 -27 6 0.033 0° None - 260°F 90 135 1011 -28 6 0.034 0° None - 260°F 90 135 1011 -29 6 0.034 0° None - 260°F 90 135 1011 -29 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7	F11078-14	•	0,032	500	Mene	,	260°F	၁	120	•	1012	
10 10 10 10 10 10 10 10	M1107B-15	ø	0.034	C;	None	1	260°F	06	135	•	1007	
-17 6 0.037 0° NGDC - 260°F 90 135 - 1003 -18 6 0.032 0° NGDC - 260°F 95 143 - 1001 -19 0 0.034 0° NGDC - 260°F 90 135 - 1001 -21 6 0.033 0° NGDC - 260°F 90 135 - 1011 -22 6 0.033 0° NGDC - 260°F 92 127 - 1011 -23 6 0.035 0° NGDC - 260°F 90 120 - 120 -24 6 0.035 0° NGDC - 260°F 90 120 - 130 -25 6 0.035 0° NGDC - 260°F 90 135 - 130 -26 0 0.035 0° NGDC - 260°F 90 135 - 130 -30 0 0.035 0° NGDC - 260°F 90 135 - 130 -31 5 0.034 0° NGDC - 260°F 94 141 - 131 -32 5 0.034 0° NGDC - 260°F 96 144 0.17 - 131 -33 5 0.034 0° NGDC - 260°F 96 144 0.17 - 131 -34 6 0.033 0° NGDC - 260°F 96 144 0.17	×11078-16	w	0.033	0,	None	•	260°F	6	135	ı	7001	
-18 6 C.032 0° NGDE - 260°F 95 143 - 1003 -19 C.034 0° NGDE - 260°F 90 135 - 1.4 -20 C.033 0° NGDE - 260°F 90 135 21.4 -21 6 C.033 0° NGDE - 260°F 98 147 - 1011 -22 6 C.033 0° NGDE - 260°F 80 127 - 123 -24 6 C.035 0° NGDE - 260°F 80 117 .033 - 124 -25 6 C.035 0° NGDE - 260°F 99 117 .033 - 124 -25 6 C.035 0° NGDE - 260°F 99 125 - 124 -26 6 C.035 0° NGDE - 260°F 99 125 - 124 -27 6 C.035 0° NGDE - 260°F 90 125 - 124 -28 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 125 - 124 -29 6 C.035 0° NGDE - 260°F 99 141 - 125 -29 6 C.035 0° NGDE - 260°F 99 144 0.17	71.076.17	vo	0.037	.0	Nene	•	260 F	06	135		1003	
-19	K11078-18	ve	0.032	ů	¥ C∃e	•	260°F	95	143		1003	
-20 6 C.033	M11078-19	1	\$ 03¢	ل ه	NCJe	•	260°F	06	135	•	1011	
-21 6 C.033 0° NCTC - 260°F 98 147	M11078-20	9	C.038	•ປ	Mene	ı	360°F	06	135	21.4	•	
-22 6 C.033 C° NG-P - 260°F 55 127		JD	0.933	. 0	MCJE	,	260°F	86	147	ı	1	Broke during Loading
-23 6 C.033 C° NGTE - 260°F 92 138		v.	C.033	ت و	MC.3e		260°F	ė, č	127	,		Tab Failure, broke during loading
-24 6 C.035 C° NGR6 - 260°F 80 120 - 260°F 78 117 .033 - 250°F 0.035 C° NGR6 - 260°F 95 143 - 250°F 90 135 C° NGR6 - 260°F 90 141 C° NGR6 - 260°F 90 141 C° C° NGR6 - 260°F 90 144 C° C° NGR6 - 260°F 90 144 C°		• •	0.033	ໍ່ບໍ	Nc∷e	,	2007F	35	138		•	Broke during landing
-25 6 C.035 C° None - 260°F 78 117 .03329 6 C.035 C° None - 260°F 95 14330 0.034 C° None - 260°F 90 13532 5 0.034 C° None - 260°F 92 13834 5 0.034 C° None - 260°F 94 14135 5 0.034 C° None - 260°F 95 14336 0.033 0° None - 260°F 96 144 0.17		9	0.035	• •	NC:0e		260°F	0 8 0	120		•	Broke during loading
-29 6 C.035 C° NGDe - 260°F 95 143		9	C.035	ຳ	Mcne	ı	260°F	78	117	.033	•	Broke at tabs
-30 0.035 C° No.e - 260°F 90 135 260°F 90 0.035 C° No.e - 260°F 90 135 260°F 90 135 260°F 90 135 260°F 90 135 260°F 92 138 260°F 94 141 260°F 95 143 260°F 96 144 0.17		ve	0.035	°	9 2	•	260 °F	95	143		•	Broke during loading
-31 6 0.034 C° None - 260°F 90 125 260°F 92 138 260°F 94 141 260°F 95 143 260°F 95 143 260°F 95 144 0.17 260°F 96 144 0.17) (5	0.035	. ບໍ	# 016		260°F	06	135		•	Broke dur loading
-32 6 0.031 C° None - 260°F 92 138		· vta	0.034	ٺ	X 0.70		260°F	90	135	•	•	Broke during loading
-34 6 C.034 C° None - 260°F 94 141		· Æ	0.031	ప	Mone		260°F	92	138			Broke during loading
-35 5 0.034 C° None - 260°F 95 143		140	0.034	ໍ່ບ	Morrie	,	260°F	75	141	•	•	Broke during loading
-33 6 0.033 0° None - 260°F 96 144 0.17 -	26 - 871 -W	u	75.0	e C	a C	,	260 °F	95	143	ı	•	Broke during loading
	148 -33	ı ıţ.	0.033	ိုင်	None		260°F	*	144	0.17	•	Broke during loading

TABLE XVII CREEP AND STRESS RUFFURE PROFESTIES SUPPLIE - WANDO 52/6/YOHOR II GAAFIIIE CONTOSITES

THE REPORT OF THE PERSON BELOW

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	i] !			*	App 1: 4	
	(Pites) (in.)	(ie.)	Ortentation	Type	Type Darecton	ĮįE,	Stress Level (R ⁰ ult) (ksi)	(12 E)	failure (Sours)	of thout Failure (Bours)	3
1-4-0104	•	6.037	è	1	,	3.92	§	Š	69		
7-18	•	8		į	•	5	8 6	301	697	•	
10 X 10 41 - 3	*	0	•			3	5 8		100		
1 1 1 1	•	7)	100	8 8	2	÷.	. !	
2-41-01-91	•				•	200	2 8	<u> </u>	, ;		
	•		,	ľ	•	320.1	3	120	Š		
4 A 18 14 4	•				•	4.05%	d	151		ı	1
7-Y-	•	20.0			•	5	6	3 5			Surface during house to
4-4-4-4-4-4-4-4-4-4-4-4-4-4-4-4-4-4-4-	•	0.037		1	•	5	: 8	3.5		;	
TENT	•	5			•	5	<u> </u>	1	234.5	1	
	•		.	į	•	380	2	501	97	•	
17- WE	•	9.60	5	1	•	350*7	98	121	•	•	Broke during loading
A- ST	•		n Q	į		350	28	120	•	•	Broke during loading
	•			Ĭ	•	350.7	78	127	0.02	•	
A THE	•	o. 63 5	t	Į	1	150	75	11.		•	Broke derive landing
MM27	*	9.63	ະ	Į	•	350°F	:	77	e.:		
*-	•		i	1	•	350°5	ł	145	,	•	1
	•	0.633	£,	į	•	7.030	92	91	•		¥
7	•	3	ະ	l	•	1.051	*	152	•	•	Broke at Lond
N- STEE	•	Č	ະ	į		1501	æ	ភ	•	•	¥
T T	•			į	•	1.00	×	351	0-01	•	

TABLE AVII CREEP AND STRESS REPORTED STREET
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Thic	Thickness		PRICR O	PRICE CONDITIONING	Test Tono.	Stresa	Stress Level	1 ime to Failure	Applied vithout Failure	
(Plies,	(Files) (It.)	Orientation	Type	Durstion	(. E)	(7 ³ ult	(7 ³ ult) (ks1)	(Hours)	(Hours)	Comment
-	0.043	06	None	ı	260°F	0,	2.14	•	101	
80	0.(44	, 0 6	None		260°F	9	2,40	.067	•	
9 0	770.0	₌ 06	None	•	260°F	,	ı	٠	•	Broke in Handling
œ	0.044	₌ 06	None	•	260°F	•		•	•	Broke in Handling
90	770.0	906	None		260°₽	75	2.29	077	•	
20	770	. 06	Non	,	260°F	980	2.40	776		
•	777.0	. 06	None	•	360°F	,	•	,	•	Broke in Rondling
۰	0.043	, &	None		260°F	85	2.60	139.0	•	1
œ 0	0.044	₂ 0 6	None	•	260°F	80	2.40			Broke during loading
æ	0.(44	³ 0 6	None	•	260°F	ş	2.75	43.1	,	
æ	0.044	, 06	None	,	350°F	080	2.43	.25	1	Strain gauge failed
*	0.043	₌ 0 6	None	•	350°F	7.7	, K	•		Failed in loading
•0	0.043	. D6	None	•	350°F	73	2.21		•	Failed in loading
*	0.043	, 26	Hone		350°F	2	2.13	.167		
e c	0.043	8	None	,	350°F	89	5.09	٠		Failed in loading
•0	0.044	₀ 06	None		350°F	0,	1.22	,	•	Failed in loading
ao	0.043	&	X CDE	¢	350°F	6 2	1.88	г.		
0	0.043	· 06	None	,	350°F	69	2.71	•	•	Sailed in loading
,	0.043	- 06	None	•	350°F	58	1.76	•		Failed in loading
•	0.044	<u>.</u>	None	•	350°F	2 6	1.69	6.05	1	
œ	0.0.50	[0/45/135/0/90]	Kone		260°F	70	61.3	,	1010	
. 0	0.051	-	None		260°F	70	61.3		1011	
o	0.050	:	None	•	260°F	980	92	•	1009	
•	0.050	=	Hone	1	260°F	8	2	೫		
•	0.051	[0/45/135/0/90]	None	•	260°F	8	78.8	16	1	
•	050	[0/45/135/0/90]	Kone		260°F	8	78.8	•	•	[mmediate failure
	0.050		None		260°F	85	74.4	674	•	
• •	0.030	=	None	•	260°F	82	74.4	766	•	
o	0.050	=	None	•	260°F	75	65.5	725		
c	0	[00/0/201/27/0]	į	,	36.095	S		ı		action destroy adone

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TABLE XVII CREFP AND STRESS SCITTRE FROPERTIES SUMMARY - NARMOD 5206 NODWOR II GRAPHITE, COMPOSITES

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Specimen	Thickness	Desa		PRIOR CC	PRIOR CONDITIONING	Test Test	Stress Level	Levei	Time to Failure	Applied without Failure	
J ACTION	(rure) (mr.)	(m)	Orlencarion	1ype	Duration	(#.)	(1 ult) (ks1)	(K61)	(Hours)	(Hours)	Comment
M1149-24	σ	0.050	[0/45/135/0/90]	None	1	3.09?	85	7,7		,	Broke during loading
M1149-25	o	0.050			•	260°F	ş	7.8	•	•	durine
16-9-11W	· o	840	:		•	260°F	Ş		•	•	7.14.10
FE-671 DM	·or	0 70		, and	•	3.090	26	4 2 6	50		200
M1149-12	· σ	2	Ξ	Mone of	•	260 %	8 %	20.00	· ·		4114
M1149-22	on.	0.050	:	Mone	•	260°F	8	74.5	111.4	•	
M1149-25	O-	670	Ξ	e II C	•	260 ° F	65	77.9	0.05	•	•
M1140-22	· a	7.70	:			260.5	. 0	. 44			United Assessment Conference
M1140-23	ı. O	3	:	anow.		1007	. 6	7.0) 1		Broke duting loading
M1149-54	n d	3	:			20.70	2 6				decke immediately
75 6717	^ C		=		1	1007	7.0		0		
M1149-35	ra	5.5	(05/0/581/57/0)	Mone	, ,	260°F	6	2.89	0.4	, ,	•
		:	100	1			· i		•		
#1149-28	6	0.050	$(0/45/135.0/\overline{90})$	None	•	350°F	06	71.2	•	ı	Broke during loading
M1149-29	6	0.050	ī.	None		350°F	85	67.3		•	Broke during loading
01149-30	ď	0.050	Ξ	None	•	350 °F	80	63.3	•	•	during
M1149-36	Ω,	0.050		None	•	350°F	20	55.7		•	
M1149-19	6	0.050	z	None	,	350°F	88	9.69	,	1	during
M1149-37	6	0.050	Ξ	Mone	•	350°F	92	76.8			Broke during loading
M1149-39	5	0.049	=	None	•	350°F	95	79.3		•	
M1149-46	O.	0.051	=	None	•	350°F	4	80.9			
M1149-38	6	0,049	[0/45/135/0/ <u>90</u>]	None	,	320 °F	93	77.6	2.01	ı)
M11354-1	σ	0,40	[0/45/135/2/90]	None e	•	350°F	70	55.4	309.4	•	
M1135A-2	ď	0,0		Mone		350°F	75	59.4	214.1	•	
M11354-3	· o	200	=	a more	•	150°F	80	63.3	769	•	
AC117	· o	0,0	•	No.		350°F	- C	63.3	657	•	
M1135A-5	. 0	0,049	=	None	1	350°F	88	67.3	800	•	
M1135A-6	•	0,049	=	Mone	•	350°F	83.5	99	w.	•	
W1135A-7	•	0,049	:	Mone		350.5	2	55.4	125		
M1135A-8	ď	0,00		None	•	350 F	81.6	65.6		•	Immediate failure
H1135A-9	·o·	0,049	:	None	•	350 F	02	55.4	۵.	•	
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TANGE AVIT GREEP AND STRESS RUPTURE PROPERTIES STREAM - NARMOGE 5206/HIDDMOR II GANDRIFF GOMENSTIES

									Applied	
S a magn	Thickness		PRIOR Q	PRICE CONDITIONING	i de i	Stress Level	Cevel (bed)	Failure (Hours)	Fatture	Comment
N Ser	(Files) (in.)	a.) Orientation	1 yp e	Maracton		וא חזר)	(20)	(2.50))	
1.8.1	vc	ပိ	.86 .88	500 Hrs.	260°F					
2 - 8 - 1 - 8 - 2 - 8 - 1 - 8	o ve	• 0		500 Hrs.	3€0,₽					
2.026	ve	°ບ	H2 786	500 Hrs.	260°F					
90	w	0,		/ 500 Hrs.	260°F					
k 148-5	9	•0		500 Hrs.	2 60°F					
GI ST	920 c	*C	₽5 39	SMO Hrs.	350°F	65	108	,	4	Broke during loading
D 0 7 1 2				500 Hrs.	350°F	75	125	•	,	
/ = 0 17 T				SOC HES.	350 F	080	133	•		
0 0 0			## #86		350°F	85	1+1		•	
×.148-10	6 0.034	03%		, 500 Hrs.	3≥0 3€	6	149	•		Broke during loading
				7 1000 HTS	36038	8	145		1000	
C1-9601.2				1000 H	3,090	200	142	,	0001	Strain gage failed after 485 brs.
67-9601-E			1 20		260°F	8	32	12.4	•))
01-0601.7			186	1000 Hrs.	260°F	83	135	8. 44		
61-8601 K	6 0.033	033 0.			200°F	81	132	•	1000	
			H: 680	7 1000 Hrs.	35008	90	130	50.7	,	
A-1104-5				/ 1000 Hrs.	350°F	*	127		1000	Lost strain gage at .067 hrs.
1104-7			1286 1886	/ 1000 Hrs.	350 F	*	124	,	1000	Strain gage failed after 568 hrs.
0 TOT				/ 1000 Hrs.	350 °F	82	121		1000	
110A-9	6 5.033	033 0.	987 FH	/ 1006 Hrs.	350°F	98 0	118	•	1000	Strain gage failed
O S			The rate -Hull	idity Cycle	260 ⋅ F	80	140	.033		
14.0-12			The Tar	idity Cycle	260°F	78	137	•		
14.8-13			Der et et	idity Cycle	260°F	8	158	•		broke during
21-051 M			The Table	idity Cycle	260°F	3	154	•	•	broke during
M. 148-15	6 0.033	033 0*	Thermo-Humidity C	idity Cycle	260°F	92	149	•	•	Tab failure, broke during loading
			High Carried T	idiry Cycle	250°F	ð	166.5	,	•	Broke during loading
17-95T.M			The rate - Huga	1diry Cycle	260°F	92	191	•		
57-041-W	2000 A	000	Thereno-Jumidity C	Idity Cycle	260°F	80	72	•	•	
07-017-1										•

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TABLE NOTE ORELP A 30 STREAS REPONDE ESCHARTLES STRAMES - NARKO 5236 A OPHOREIT CRAPHITE CORPOSITES

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	louding louding louding	78	4 *4 £	loading loading loading loading	loading loading loading loading
Comment	Broke during loss Borke during loss Broke during loss	Strain gage failed	Strain gage Eailed Failed in loading Strain gage Eailed Strain gage Eailed	Broke during loading Broke during loading Broke during loading	Broke during loss Broke during loss Broke during loss Broke during loss Broke during loss
Tim. Applied without Fail.re (Woure)	000		1990 1070		• • • • •
Time to Failure (Hours)	0.25	.033 .05 1.0 6.2 6.2	926	0.02 0.01 2.5	
Level (kst)	132.6 137.4 140.4 143.4 141.9	150 163 157 154 154	133 130 127 124 121	76.6 65.3 66.1 64.5 72.6	62.5 61.2 59.3 57.4 56.1
Stress Level (10ult) (ksi)	98 93 95 84	48 88 8	99 99 99 99 99 99 99 99 99 99 99 99 99	8 6 6 6 6 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8	# % & & &
Test (*)	350°F 350°F 350°F 350°F	260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F	260°F 260°F 260°F 260°P	350°F 350°F 350°P 350°P
PRIOR COMDITIONING Type Duration	idity Cycle idity Cycle idity Cycle idity Cycle	Wthing. Wthing. Wthing. Wthing.	Wthrng. Wthrng. Wthrng. Wthrng.	500 H/s. 500 H/s. 500 H/s. 500 H/s. 500 H/s.	500 Brs. 500 Brs. 500 Brs. 500 Brs. 500 Hrs.
PRIOR CC Type	Thermo-Humidity of Thermo-Humidity of Thermo-Humidity of Thermo-Humidity of Thermo-Humidity of Thermo-Humidity of	Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng. Acc. Wthrng.	Acc. VI	25 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5	25.25.25 28.88.8
Ortente: ton			••••	[0/45/135/0/30]	
Dees (1s.)	0.031 0.033 0.034 0.034	0.032 0.033 0.033 0.031	0.032 0.031 0.032 0.032	0.050 0.050 0.049 0.049	0.045 0.059 0.050 0.050
Thickness (Flies) (Im.)	00000	****	သလလာလာလ	ው ው ው ው	ው ድ ው ድ ው
Specimen	M1148-16 M1148-17 M1148-18 M1148-19 M1148-20	M11098-20 M1110A-2 M1110A-2 M1110A-3 M1110A-4	MILLOA-10 MILLOA-11 MILLOA-17 MILLOA-15 MILLOA-14	M1169-1 M1169-4 M1169-3 M1169-2 M1.49-5	M1149-6 M1149-7 M1149-8 M1149-9 M1149-10

	Comment		- - - Rroke during loading	Broke during loading Tab failure, broke during loading	Overload Broke on Loading Broke on Loading	Immediate failure Oven overheated No strain readings	Broke during loading
1 1 1	Applied without Failure (Hours)	1000 1000 1000 1000 1000	1000 1000 1000 1000				1000
	to Eatlure (Hours)		1 (1)		7 - 7 - 7 - 7 - 7 - 7 - 7 - 7 - 7 - 7 -	140.9	
	Tiress truel Tult (Keiz	2 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	1 4 7 110 4 3 4 6 7 5 5 6 6 8	TWO THE	2.00 2.00 2.00 2.00 2.00 3.00 4.00 4.00 4.00 5.00 5.00 5.00 5.00 5	77.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7	62.9 59 61 67.7 59.6
	<i>₹</i> .	! :	मामामामामा इ.स.च. १९	per tar tar tar tar	ir irr ger ber ber au ch ch ch ch ch ch au ch	######################################	######################################
	Trst Tebp.	100 100 100 100 100 100 100 100 100 100	in the factor to		35000000000000000000000000000000000000	250 PE	350°F 350°F 350°F 350°F
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	Specinen Munder	X X X X X X X X X X X X X X X X X X X	100 - 100 -	21129-11 VIL49-12 VIL49-13 VIL49-13	30 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	M11378-6 V11378-7 V11378-8 W11378-9 H11378-1	M1138A-6 M1138A-7 M1138A-9 M1138A-9

TAST XV.I (FIED AV) STUSS HEPTRE PROPERTIES SPONGAY - VARING FOR OUR DR. II (RAPHITE OLD PROPERTIES

			fter .5 hrs , temp too
Comment		Strein Gage Pailed	Strain gage felled after .5 hrs Heater malfunctioned, temp two high
Time Applied without Failure (Hours)	1055 1061 1061	1018 1015 1004 1006 1000 1000 1000	1012 1012 1015 1006 1007 1006 1007 1007 1007 1007 1007
Tine to Failure (Hours)	- + - 627		97.7 685 918 950 186 186 186 186 186 186 186 186 186 186
Level (kai)	114 122 130 114	1121	138 127 1121 1121 1127 1127 127 127 127 127
Stress Level (T ^d ult) (ket)	96 5.	50555 B558	£ 986788 90888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888 50888
Test Temp. ('F)	260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F 260°F 260°F	250°F 350°F 350°F 350°F 350°F 350°F 350°F 350°F 350°F 350°F 350°F
FRICE COMDITIONING Type Duration	260°F ' 500 hrs. 260°F ' 500 hrs. 260°F ' 500 hrs. 260°F ' 500 hrs.	260°F / 500 HTG. 260°F / 500 HTG. 260°F / 500 HTG. 260°F / 500 HTG. 260°F / 500 HTG. 350°F / 500 HTG. 350°F / 500 HTG. 350°F / 500 HTG.	350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Hrs. 350°F / 500 Hrs. 260°F / 500 Hrs.
	% % % %	สีสีคลัล ครีคคร	n nnan aaaaa aaaaa
Orientation	0000		[04/0/86] [04/0/86] [04/0/86] [04/0/86]
Th chaess (Pitzs) (in.)	0.032 0.032 0.032 0.032	0.032 0.032 0.032 0.033 0.033 0.033	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0
1 c (7 (7 (7 (7 (7 (7 (7 (7 (7 (7 (7 (7 (7			ନ ବ୍ୟବନ୍ଦ ବ୍ୟବଶ୍ୟ ବ୍ୟବବ୍ୟ
Specimen Number	MIIIB-16 MIIIB-17 MIIIB-18 MIIIB-19 MIIIIB-20	MILIZA-1 MILIZA-3 MILIZA-4 MILIZA-4 MILIZA-16 MILIZA-16 MILIZA-17 MILIZA-18	######################################

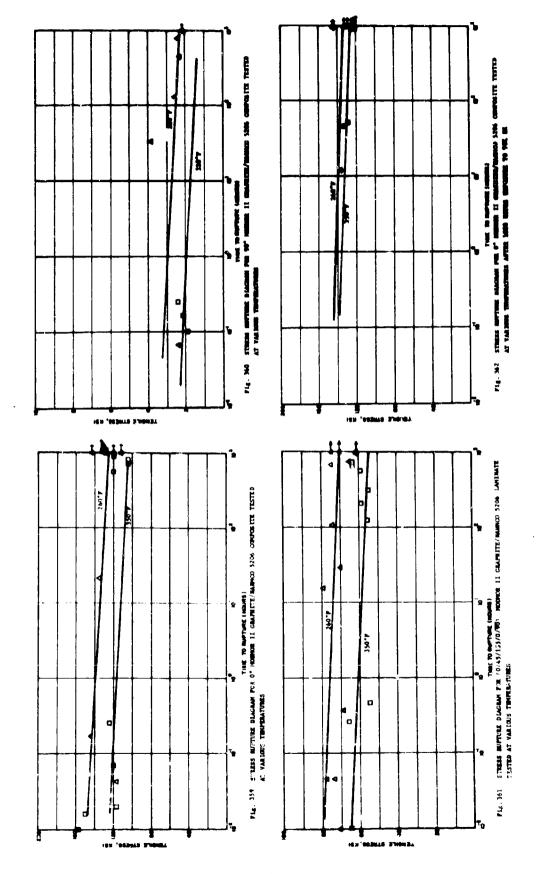
OF IT CAPER AND STRESS REPT AND SCHOOL SCHOOL OF THE SCHOOL STRESS ASSETS AS FIRST COMPOSITES

Comment		Broke	rheated		Strain gage failed Strain gage failed	
			Oven overheated		Strain g	
Applied without Failure (Hours)	1008 1004 1004	111.	- 1000 1000	1000 1000 1000 1000		1000
Ti-c to Fatlure (Hours)		309 309 10.1	39 50 508	89	9.8 2.4 167 168 918	861 1.5 0.167
Siress Level	0 9 80 40 40 0 10 10 10 10 10 10 10 10 10 10 10 10 10	0 1 2 4 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	104 123 133	118 110 104 126 133	83 i30 138 130 121	119 128 136 144 153
Stress (A ^c ult	\$ -1 - 1.36 \$ -2 - 1.35	15 U	ტ ყო დ ყო ბ ტ 10 00 00 00	75 75 70 90 90	83 79 79 73	70 75 80 885 90
Test Temp. (°F)	19 19 19 19 19 19 19 19 19 19 19 19 19 1	350 PE S S S S S S S S S S S S S S S S S S	260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F	260°F 260°F 260°F 260°F	350°F 350°F 350°F 350°F
OR CONDITIONING e Duration	\$00 hrs. \$00 hrs. \$00 hrs. \$00 hrs.	500 hrs. 500 hrs. 500 hrs. 500 hrs.	1000 cyc. 1000 cyc. 1000 cyc. 1000 cyc.	/ 1000 eye. / 1000 eye. / 1000 eye. / 1000 eye. / 1000 eye.	/ 1000 cyc. / 1000 cyc. / 1000 cyc. / 1000 cyc. / 1000 cyc.	1000 cyc. 1000 cyc. 1000 cyc. 1000 cyc.
			0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0	250 0 1 1 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	350°E/ 350°E/ 350°E/ 350°E/ 350°E/	350° E'
Orientation	\$ <u>10e</u> v \$81.57 ú,	: :06 0 561'54 5	00000	.	00000	, ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° °
	0.051 0.051 0.051 0.050	870.0 870.0 870.0	0.033 0.033 0.033 0.033	0.000 0.000 0.000 0.000 0.000 0.000 0.000	0.0000 0.0000 0.0000 0.0000 0.0000	0.033 0.033 0.033 0.033
Th.kness (Plics) (In.)	00000	ው ው ው ው ው	καφφω	σινσυ	σφφφφ	φφφφφ
Specimen	M11428-1 M11428-1 M1428-1 M1428-1 M1428-1 M1428-1	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	H1112A-6 H1112A-7 H1112A-8 H1112A-9	M1112A-11 M112A-12 M112A-13 W112A-14	M11128-6 C1112B-7 M1112B-8 M1112B-9 M1112B-10	M.1128-12 M.1128-12 M.1128-12 M.1128-14 M.1128-14

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TARLE NVIE (REFE AND STRESS REPTURE PROPERTIES STMMARY - NARMO 5706/1000KMF II GRAPHITE (OMPUSITES

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Special	.hickness		PRIOR COMDITIONING	Test Test	Stress Level	[eae]	Time ro Fallure	Applied erithout	
* mber	(Piies) (In.)	.) Ordentation	Type Duration	£	(Toult) (kat)	(ks1)	(Hours)	(Hours)	Counent
M 142A-1	050-0	50 (545/1135/0/95)	/1000	260°F	8	72.5	,	0001	1
M1142A-2	0.050		1000	260°F	2	6.07	•		Tab failure, broke during loading
H1142A-3	670.0	: 6.9	/1000	260°F	93	6.47	•		Broke during loading
M1142A-4	670.0	. 65	0001	260°F	92	74.1	1	1000	•
H1142A-5	0.048	:	260°F / 1000 cyc.	260°F	95	9.92	•	1	Oven overheated
¥1142A-6	ن	05.	/ 1000	350°F	80	71.2	910.	•	
W1142A-7	670.0		/1000	350.1	7	68	65.5		
8 - 4C-7L-3	0.050	05	/ 1000	350°F	26	8.79	.039		
91142A-9	670.0	- 6.	/1000	350°F	2	9.69	312.9	•	
H1142A-10	0.051		260 F . 1000 cyc.	350°F	82	7.99	\$2.	•	Tab failure
M1142C-1	870.0	<u>[36, 0/\$21,132/6]</u>	/1000	260°F	70	55.8	•	1001	
M1142C-2	870		71000	260°F	75	59.7	•	100	
M1142C-3	870.0	: 00,4	/1000	260°F	8	63.7	•	1060	
H1142C-4	870.0	. 95	1000/	260°F	20	55.8		9101	
¥1142C-5	8700'0 6	89:	350°F /1000 cyc.	260°F	2	63.7	•	1005	
81142C-6	9,0,0	:	/1000	350°F	2	3.3	,	0001	
M1142C-7	870.0		/1000	350°F	75	58.0	•	1000	
M1142C-8	6	: 20,	350 *F / 1000 cyc.	350°F	2	62.3	•	9001	
¥1142C-9	9.0		/ 1000	350	\$	66. 2	345	•	,
M1142C-10	870.0	48 [0/45/135/0/ <u>90</u>]		350.8	ş	٦. ج		•	Broke in loading



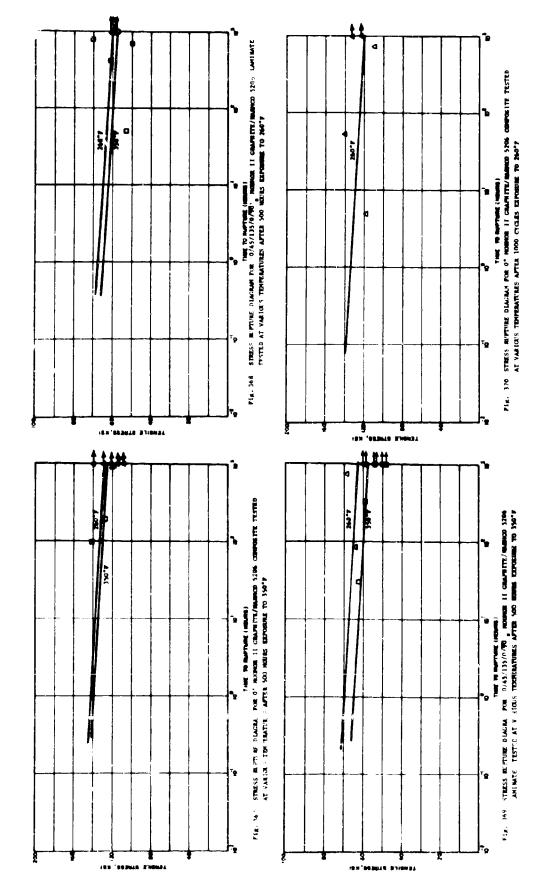
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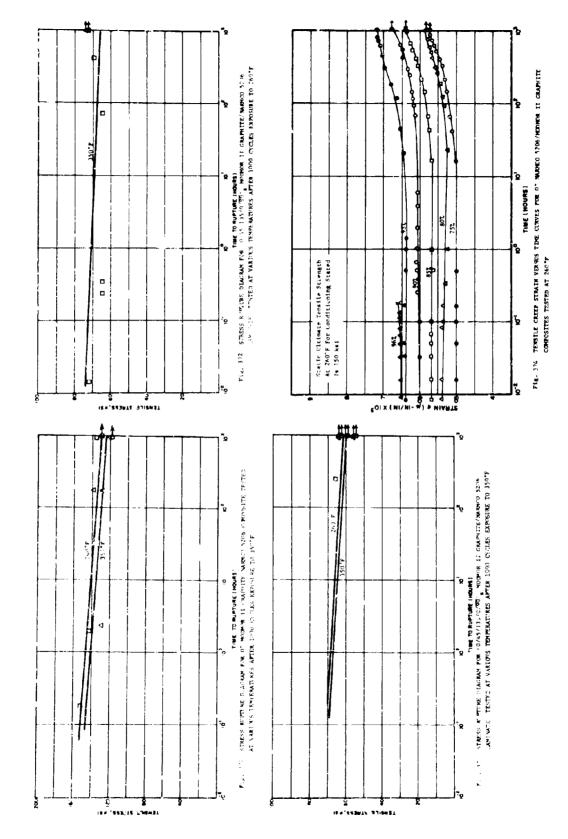
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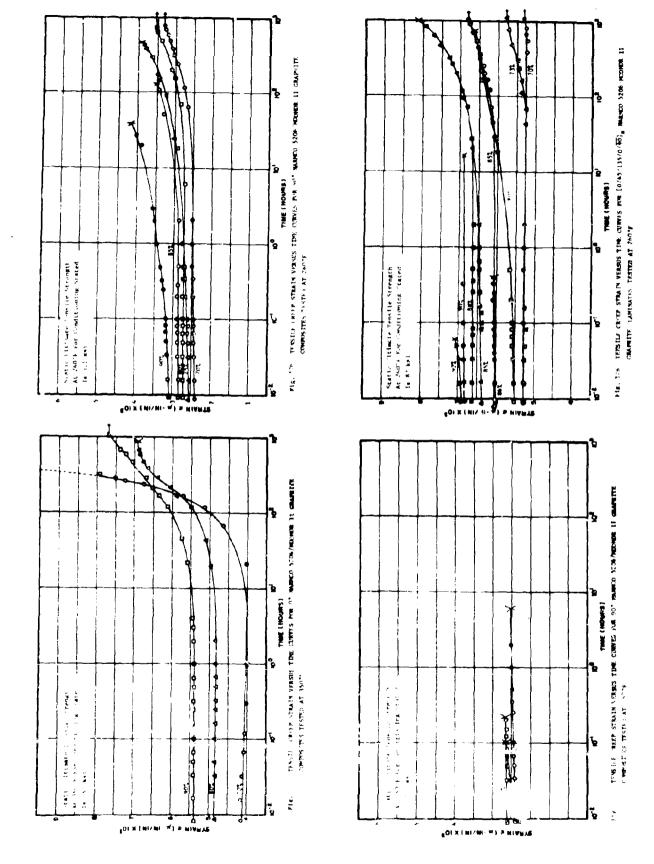
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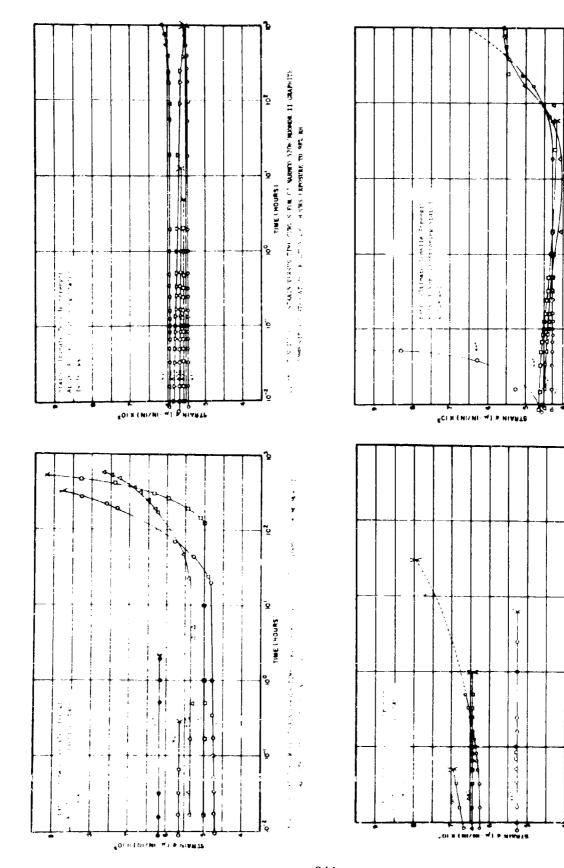
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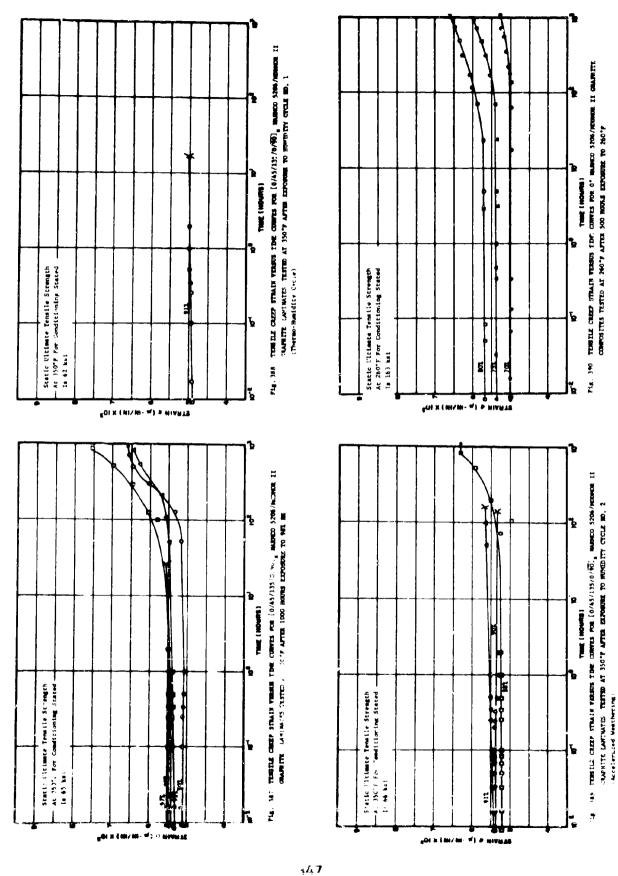
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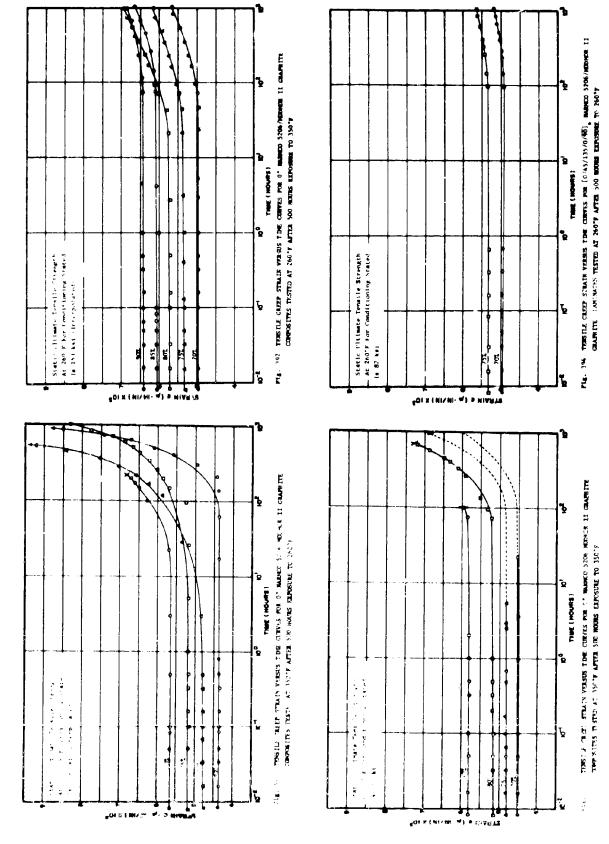
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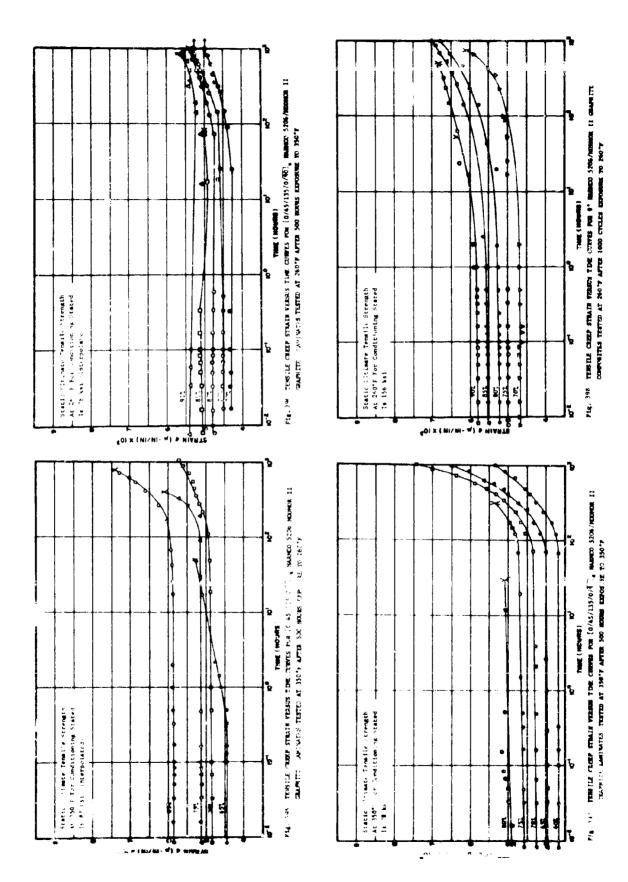


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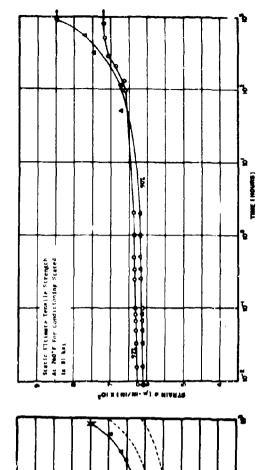


FIG. 402 TENSILF CREEF STRAIN VERSUS TIME CONVER FOR [04/45/135/0/96], MARIOD 2506/NEWORD TE CALMINES : ANTINESS TESTED AT 260.°F AFTER 1800 CYCLES EXPONENTE TO 240.°F

TOTAL LES CHEMBER THAIN VERSUS TITRE CHEMBER 0° MARKET STELLENGE TOTALS EXPRESSED TO NO. "AN UNITE CHEMBER 19 TISSET STELLENGE OFFICES EXPRESSED TO NO. "AN UNITE CHEMBER 19 TISSET STELLENGE OFFICES EXPRESSED TO NO. "AN UNITE CHEMBER 19 TISSET STELLENGE OFFICES EXPRESSED TO NO. "AN UNITE CHEMBER 19 TISSET STELLENGE OFFICES EXPRESSED TO NO. "AN UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICES TO NO. "AND UNITE CHEMBER 19 TISSET STELLENGE OFFICE STELLE

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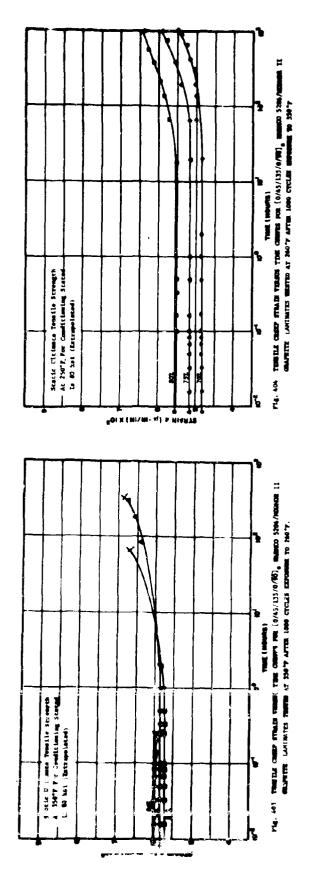
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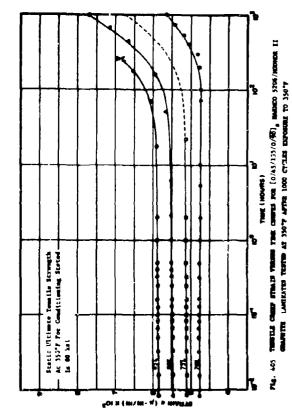
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APPENDIX III

DATA SUMMARY FOR HERCULES 3002M/COURTAULDS HMS GRAPHITE COMPOSITES

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APPENDIX III

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7	Figs. 507 to 528 Creep Strain Versus Time Curves	414-420

FABIT AVI () FATIC FRUPERTIES SPEAKY FACTOR 3002PC OFFICIALS HIS FAMILY FOR FEET

Orientatis:	Type Load	Price Condict ming	fest Temp.	E x 10°)	(u)/u)	ult (kei)	^e ult (a-in./in.)
.0	Tension	Kone	KTD	26.9	0.20	86	3,360
•	Teneion	X: 114	260°F	28.3	0.23	119	4,100
•	Teneton		350°F	29.8	0.20	115	3,850
%	Teneton	Kone	Ē	86°U	0.00	2.3	2,260
.04	Traston	Kone	3,092	0.45	0.00	4.7	5,000
• 0	Tension	Kone	350°F	0.89	0.03	4.1	4,570
[04/07811/87/0]	i. aston	Kone	Ę	13.9	0.45	3	3,500
0/45/135/0 90.	Tention	Kone	24C-F	17.2	0.47	*	3,530
[048/138/0/90]	Tension	Kme	3°)\$E	16.2	17.0	23	3,500
.0	Compression	Mone	reta Teta	24.6	0.68	700	4,300
.0	Compression	Mone	****	22.5	0.20	46	4,110
.0	Compression	ì	2 6 0'F	25.8	0.30	*	3,870
, O	Compression	None	350*	24.6	0.55	•3	3,540
•0	Compression	lone	350°F	24.9	0.23	92	3,420
\$	Compression	Mon	Ě	1.25	0.01	21.4	19,670
. 8	Compression	Mone	, cr	1.21	0.00	32.9	29,000
2	Compression	Mon	360°F	11.11	0.01	7.62	> 30,000
\$	Compression	Hone	350 %	1.36	0.01	16.7	20, 530
\$		j	350,035	-	5	11 1	900 000

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Or: entation					Tr., s.	(kat) (a.in 'in')	"ule (,17 'in.)
(0 -> 135 (90)	Compression				į	60	9.9.
0.45/13576.90.	Capresion	N. 13¢	41 ()*	11.6		59	4,720
0/45 135/0/90]	Complex of Con	¥.ne	4,79,	7.6		ĸ	6,410
[<u>98</u> /8/511 5+/0]	. of was do.	Kone	4 υς	1.2.5		55	069.7
0/.5.115/0/40]	Compression.	Жэле	34048	, o		53	5,750
ί	in-Plan Shear	Your.	Ê	0 .8 >		10.4	•
• • •	n-Plus Shear	You	1 09/	3. 0	•	ø.	30 ,000
0.	n-Flanc Shear	Mone	150°F	15.5	,	6.3	» 30° 000
.0	Int. Shear	None	E TI	,	ı	13.7	
, 0	Inc. Shear	Nune	1 04	•	P	4.7	•
. 0	Int. Shear	Mon	350 F	•	•	7.5	
[0/4 1/135 /0/90]	Int. Shear	Kone	ē	•	•	10.2	
(5/45/135/0/90)	Int. Shear	None	2 6 C F	•	•	7.4	•
109/0.411/17/	Irt. Steat	Kone	1.051	•	•	5.2	•
ò	Fierment	Kone	E	•	•	130	•
, 0	Fierman	section	260.7	•	•	110	•
. 0	YI smare!	Kina	1.050	ı	,	107	•
* O *	Plemeal	Scott	£	•	•	5'01	•
• • •	Flemmel	Mone	260⁴₽	•	,		•
	*) .mira)	Ever	1.052		•		•

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Oriun ation	Orien ation Type Load	Prior	. 8 (T. ∰.) (F)	r G (10 x 10°)	, (in/in)	Jult or Tult (kal)	*ult (u-fn./in.)
(18 18: 0 Get	lenor.	, n	•		•	83	•
20 138 1 St 12	T T KIND OF THE	N state	3,0%	,	•	Z.	•

ACTED STAFFG INDE CO. TANKS
HERETTES TO COMPUTE TO SERVICE STAFF

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Orientation	Type Load	Brior Conditioning	, et l'en . ('F)	, ps: x 10 ⁵ ,	in/in)	ult (ks1)	1 de (1877)
•0	Tension	987, RH /500 Hrs	ÇL.	6,85	0.24	117	3,910
•0	Tension	987 RH/500 hrs		•	,	601	
•0	Tension	987 RH/500 Hrs	350°F	•	,	109	•
•0	Tension	980 RH /1000 Hrs		26.6	0.20	601	3,930
•0	Tension	987 RH/1000 Hrs		28.1	77.0	124	4,280
• 0	Tension	98°, 84/1000 Hrs		25.6	19	101	3,830
•	Tension	Thermo-Humidity Cycle	ct	17.1	0.29	105	3, 790
• 0	Tension	Thermo-Humidity Cycle	2 60°F	•	•	126	;
•0	Tension	Thermo-Humidity Cycle		ı	•	115	
• 0	Tension	Acc. Wthrng.		26.9	0.29	101	3,681
• 0	Tension	Acc. Wthrng.	260°F	28.3	0.39	133	069.4
• 0	Tension	Acc. Wthrng.	350°F	27.5	0.45	114	4,140
• 06	Tension	967 RH/500 HTS		0.95	0.00	1.7	1,810
• 06	Tenston	987 RH/500 hrs	260°F	•	•	2.4	•
. 06	Tension	98% RH/500 Hrs	350°F	1	ı	3.5	•
• 28	Tension	98% RH/1000 Hrs	e e	1.07	0,00	1.5	1,360
	Tension	967 RH/1000 Hrs	260°F	1.13	0.00	1.7	1,620
.	Tension	98% RH/1000 Hrs	350°F	0.89	0.01	ŋ. 4	4,780
` Z	Yension	Thermo-Humidity Cycle	e ra	1.08	0.00	1.7	1,660
• 06	Tension	Thermo-Rumidity Cycle	260°F	•	•	1.1	
°06	lension	Thermo-Humidity Cycle	350*F	•	•	6.0	

TABLE XVIII STATIC PROPERTED SCHWARY - OF CORRESPONDED ON CORRESPONDED CONTRACTORS

Ortentation	Type Load	Prior Conditioning	Test Tesp.	L (pst x 10 ⁶)	, (10/1n)	ult (kst)	⁶ ult (μ-fn./fn.)
,06	Tension	Acc. Wthrng.	£	1.22	0.01	5.5	099'7
.06	Tension	sec. Whiring.	260°F	0.87	0.0	2.0	4,240
,06	Tension	Acc. Winting.	350°F	0.27	0.0	1.1	7,940
[0]/d/5E1/57/0]	Tension	98% RH/530 Hrs	6	14.4	67.0	53	3,600
[0/25/105/0/[0]	Tension	987 RM/530 Hrs	260 ₹	·	•	23	•
[0/45/135.0/[0]	Tenston	96% RH/530 Hrs	350*F	•	•	15	•
[0/45/135/0/[3]	Tension	98% AH/1300 Nrs	Ę	14.0	0.43	*	3,990
[0]/0/251/57/0]	Tension	98% RH/1 00 Rrs	260°F	13.5	0.43	55	4,020
[0/45/135/0/{0]	Tension	96% RH/1300 Hrs	350°F	12.6	0.59	4 3	3,470
[0/45/135/0/50]	Tenston	Thermo-Hum dity Cycle	e	14.7	0.45	32	3,500
[0/45/145/0]	Tension	Thermo-Hum dity Cycle	260°F	•	•	88	•
[0/45/135/0/93]	Tension	Thermo-Hum.dity Cycle	356°F	•	ı	20	•
[0/45/135/0/95]	Tens	Asc. Wthrng.	E	15.6	0.46	8	3,530
[0/45/135/0/9]	Tension	Acc. Wthrng.	260°F	14.2	0.42	2	4.040
[0/45/135/0/9]]	Tenston	Acc. Wthrng.	350*7	14.1	0,40	51	4,190

(Add. AVLUE State Belling RTUS SPACINY)
(1997) Seat MOUNTANDS
(1997) SAPPLED OF BRIDGE

eute /4-ta./ta.)	4.370	•	•	4,290	3,900	3,640	ن ب	,		3,670	3,510	3,070	> 30,000	•	•	> 3(,000	> 36, 000	> 3('000	209, 2	•	•
(n) (ket)	8	108	47	*	3	82	700	105	108	16	7.8	65	30.0	18.2	12.1	27.3	15.4	9.1	25.7	17.5	12.3
(in in)	0.22		•	0.26	0.22	0.30	°, 38	•	•	0.45	0.26	0.24	0.0	ı	•	0.01	0.0	0.0	0.00		•
that x ice,	31.6	,	1	23.1	20.9	23.5	24.3		•	22.6	21.0	23.2	1.09	•	•	1.23	0.78	0.45	1.19	•	•
'test Temp.	HTD	3.09T	350°F	RTD	3.09C	320°F	KTD	3.097	350 °F	ET.	260°F	350°F	RTD	₹90%	3.50 ₽	OT.	260°F	350°F	RTD	260°F	350⁴₽
Prior tent Temp. L Conditioning ('F)	987. Riv 300 Hrs	98% RH/500 Hrs	987 RH/500 Hrs	98". RH/1000 Hrs	987 RH/1000 Nrs	987 RH/1000 Hrs	Thermo-Humidity Cycle	Thermo-Humidity Cycle	Thermo-Runddity Cycle	Ace, Wthrng.	Acc. wthrng.	Acc. Wthrng.	98% RH/500 Hrs	987, RH. 500 Hrs	987 RH/500 Hrs	987 RH/1000 Hrs	9E7 RH. 1000 Hrs	987 RH 1000 Hrs	Thermo-Humidity Cycle	Thermo-Humidity Cycle	Thermo-Humidity Cycle
Orientation Type Load	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	Compression	(ompression	Capression
Orientation	1																				

n de la companya de La companya de la co							
Orientation	Typ. Lead	Prior Conditioing	fest femp. (°F)	E E M Jack	(11, 11)	fult (kst)	*ult (4-11./1n.)
÷06	Compression	Acc. Wthang.	발길	61 -	(10,10)	29.0	900° UE <
, 96	Compression	Acc. Webring.		6, 0	00,0	15.9	> 30,000
°06	Compression	Acc. Wtheng.		6.17	00. u	9.5	> 36,000
0,45/135/0/90	Compression	981 RH/520 Hrs		10.6	0.32	52	5, 180
0.75 /135 /0/90.	Compression	981 RH/520 Hrs				53	•
0.45/135/0/90]	Compression	987 RH. 570 Hrs	350	,	•	53	•
06/0,501/57/0]	Cumpression	98" RH 15:00 Hrs	RTD	f : 10	1L U	55	6,010
0.45,135,0/90.	Compression	98] RH/1000 Hrs	260	i d	0.37	2	5,430
0/45/135 0/90]	Compression	98% RH/1300 Hrs	350	10.5	0.K	61	4,420
[0/45/135/0/96]	Compression	Thermo-Humidity Cycle	RTD	11.5	•	x	3,810
0/45/135/0/901	Compression	Thermo-Humidity Cycle	260	•	•	51	ė
06/02:135:0/90]	Compression	Thermo-Hunidity Cycle	350	•	•	8	
0/45/135/0/90]	Compression	Acc. Wthing.	R T T	5.6	97.0	*	5,170
0/45/135 '0/90]	Compression	Acc. Wthang.	266	10.1	77.0	555	5, 050
0.45/135.0/90]	Compression	Acc. Wthong.	350	8.2	0.27	\$	5,700
20	In Plane Shr	98% RH/530 Hrs	RTD	78.0	•	11.7	> 30,000
ູ້ນ	In Plane Shr	98% RH/500 Hrs	260	•	•	5.5	•
.	In Plane Shr	98% RH/530 Hrc	350	06.0	•		> 30,000
.0	In Plene Shr	98% RH/1300 Hrs	RTD	06.0	•	11.1	> 30,000
. 0	In Plane Shr	98% RH/1100 Hrs	260	0.71	•	0.0	> 30,000
ę ()	Ir Plane Shr	98% RH/1 00 Hrs	350	0.10	•	5.3	> 30,000

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CONTRACTOR OF THE PARTY OF THE CONTRACTOR OF THE

Crim cation	Type Load	Prior Comifeioning	Total (emp.	- 7	(10/40)	9 (3 (8 (8 (8 (8 (8 (8 (8 (8 (8 (8 (8 (8 (8	alt Lastra 'sn
	in Plane Shr	Thermo-Humidity Cycle	KTD			, ' < -	90°.,0€ <
ص 0	In Plane Shr	Thermo-Runidity Cycle	، 4ر	·		2 4	
о г.	ir Piane Shr	Thermo-Humidity Cycle	150			\$	1
e m	In Flane Shr	Acc. Wthrmg.	KTD	69,68		11.3	29, 700
и <i>Г</i> 1	In Plane Shr	Acc. Whiring.	760			9 .1	30,006
a C.	ads andid of	Acc. Wthrng.	350			5.2	30, 100
G , ,	interl or shr	981 RH 500 Brs	RTD	,		12.5	,
753	Inter! or Shr	287 RH/500 Hrs	260	•		15.2	
r: 1	Interlour Shr	987 RH > 500 Hrs	350			7.3	
<u>, , , , , , , , , , , , , , , , , , , </u>	Interlim: Shr	98% RH/1000 Hrs	KTD	,		¢.\$	i
٥.	Interl'mr Shr	98% RH/! 000 Hrs	260			4.5	ı
G T*.	Interl ar Shr	98% RH/1000 Hrs	350	,		5.4	,
to for	Interlug Shr	Thermo-Humidity Cycle	Q.			10.4	•
ά (*,	Interl par Shr	Thermo-Humidity Cycle	097	,		2 2	•
8	Interl mr Shr	Thermo-Humidity Cycle	350	•		o 1	,
(interl mr Shr	Acc. Wthmg.	ET	•		1 11	ı
či L	Interl'mr Shr	Acc. Wthrng.	260			9.6	1
o t	Interlar Shr	Acc. Wthrng.	350	•		7.1	,

TABLE NATURE TRANSPORTER OF THE PERCENTIAN AND MACHINES SHOOT MACHINES COMPANIES COMPA

Orientation	Page 1	Petuc Codinicates	Total Temp.	ε Cost # 10 ^E :	. in/fai	, : ik (kn f.)	() () (u)
.0	tosten	260°F/103 Hrs	KTD			140	
.0	Tension	260 F/100 Hrs	760	•	•	135	٠
°o	Tension	260°F/500 Hrs	RTD	28 8	0.27	•	3,310
.0	Tension	260°F/500 Hrs	260	28.9	96 0	129	4,460
0	Tension	350*F/100 Hrs	KTD		•	111	•
•	Tension	350°F/100 Hrs	260	•	•	130	•
•0	Tengica	350*F/100 Hrs	350	•	•	128	•
.0	Tensica	350*F/560 Hrs	et.	27.1	0 25	101	3,860
ູ້ດ	Tension	350°F/500 Nrs	350	30.7	9.36	121	006'5
.06	Tension	260°F/100 Hrs	KTD	•	•	7.0	•
206	Tension	260°F/109 Hrs	260	•	•	4 .1	•
90ء	Tension	260°F/500 Hra	RTD	1.00	0.01	2 2	2,180
. 06	Tension	260°F/500 Hrs	240	•	•	•	•
,06	Tension	350°F/100 Hrs	CE2	⁴.		1.3	•
.06	Tension	350°F/100 Hrs	260	•	•	0.2	•
°0°	Tenefor	350°F /100 Hr.	5	•	•	-	•

** Specimens Broke During Conditioning Cycle

TABLE OF THE SOUTH PROPERTIES SCHOOLS THANK THE BEST STOCKNES OF CANADA STOCKNESS OF C

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Orientation	Type Load	Prior Conditioning	Test Temp. (pri x 10°, (pri x 10°,	(pri x 10°)	(u ₁ , u ₁ ,	^J ult (kai)	1.3 r (2.3n./1n
•06	Tension	350°F/50 Nrs	EF.	% 0	2 c	0.7	ok 1 ° t
•06	Tension	350"F'570 Nrs	350	2870	0.0	0 7	1,040
[0/45/135/0/ 9 5]	Tension	260°F/130 Nrs	RTD	•	•	57	٠
[0/45/135/0/∯(1)]	Tension	260*F/130 Nrm	260	•	5	3	•
[0/45/135/0/95]	Tension	260°F/530 Hrm	Ę	13.9	0 39	4.7	3,280
[0/45/135/0/93]	Tension	260°F/500 Hrs	260	3 12	0, 6	6.5	1,960
[0/45/135/0/95]	Tersion	350°F/100 Nrs	Œ.	•		4.3	•
[0/45/135/0/ 9]]	Tension	350°F/100 Hrs	260	•		63	•
[0/45/135/0/ <u>9</u>]]	Tension	350°F/130 Hrs	350	ı	,	99	•
[0/45/135/0/9]]	Tension	350*F/500 Nrs	£	14.4	0 43	21	3.430
[0/45/135/6/9]]	Tension	350°F/500 Nrs	350	15.5	95.0	53	3, 390
,	Compression	260°F/150 Nrs	£	•	•	44	•
•0	Compression	260°F/100 Nrs	260	•	•	103	•
0.0	Compression	260°F/500 Hrs	e E	21.6	97.0	į	9,190
Û	Compression	260°F/530 Hrs	340	17.5	3, 31	ë	4,620
0.0	Compression	350°F/100 thre	£	20.5	0.29	6	7,010
0.0	Compression	350°F/139 Nrs	350	•	,	10;	•
ိုင	Compression	350°F/500 Ers	Ē	22.8	0.20	45	7,110
200		CONTRACTOR OF THE CONTRACTOR O	Ş	73.0	ç	8	4.00 A

TABLE XVIII STOTIC : STORY BESTELLE STORY GROOT

					,		
Sol wattin	באַרָר בּייִלאָרָ	Prior Comitioning	(() () () () () () () () () (1. 4 2. 4	(fn/fa)	Tule (Nest)	(ln./in.)
93,	Cumpression	260 F /100 hrs	GIS				
.06	Compression	260°F/100 Hrs	56 0	•	•		•
9).	Compression	260°F/500 Hrs	RID	1.10	0.0		> 30, 000
406	Compression	260°F/500 Hrs	366	1 17	0 0	30.9	> 30,000
°C6	Compression	350°F /100 Nrs	RID		,	ě,	•
£0.6	Compression	350°F /100 Hrs	350	•		21.5	•
•06	Compression	350°F/500 Hrs	RID	C 32		27.6	904 03 <
° 06	Copression	350°F /500 Hrs	350	37.0		22 6	28, 000
[0/45/135/0/90]	Compression	260°F/100 Hrs	RID	•	•	57	•
[0/45/135/0/90]	Compression	260°F/100 Hrs	200	•	,	9	•
[0/45/135/0/ <u>90</u>]	Compression	260°F/500 Hrs	RTD	8.31	ø. %	X.	6, 580
[0/45/135/0/90]	Compression	260°F/500 Hrs	260	\$7.86	0 28	\$	5,350
[0/45/135/0/90]	Cempress ton	350"F/100 Hrs	KTD	•		9	•
[0/45/135/0/90]	Compression	350°F/100 Hrs	350	•	,	\$\$	•
[0/45/135/0/90]	Compression	350 F /500 Hrs	£	10.50	0.33	Ł	7,470
[0,43/135/0/90]	Compression	350°F/500 Hrs	350	10.02	0.32	23	6,470

			1	! ! ! ! ! ! ! ! ! ! ! ! ! ! ! ! ! ! ! !			1
Ortentation	Type lead	Prior Calitivains		E	. In/In)	Cult (ks1)	cult (u-in./in.)
°°	In Plane Shr	260°F 100 Hrs	KTD			9 6	٠
°0	In Plane Shr	260 F /1 Hrs	26 0	•	•	8.7	
• D	In Plane Shr	260°E /303 Hrs	KTD	\$£.0		10.0	17,196
0.	In Plane Shr	260°F/500 Hrs	26.0	64.0	•	10	9 30, 000
.0	In Plane Shr	350°F/100 Hrs	RTD	1	,	9 01	
0	In Plane Shr	350'F/100 Hrs	340	•	÷	0.0	
•0	In Plane Shr	350°F/500 Hrs	KTD	87.0	•	6.6	19, 300
°o	in Plene Shr	350"F/500 Hrs	350	97.0	•	6.7	> 30,000
, 0	Int Shear	260°F/100 Hrs	RTD		,	13.5	•
.0	Int Shear	260'F/100 Hrs	260	•	1	9.5	•
, 0	Int Shear	260°F/500 Hrs	Ę	•	1	13.4	•
.0	Int Shear	260°F/500 Hrs	260			• •	
0	Int Shear	350°F/100 Hrs	e E		ı	1.1	•
0	Int Shear	350°F/100 Hrs	260	•	•	10.0	•
0	Int Shear	350'F/500 Hrs	Ē	•	ŧ	13.6	•
ືບ	Int Shear	350°F/500 Nrs	350	•	•	5.2	•

Orthitalion	Typinad	C. nell'anding	2		= :		(- - - - - - - - - -
.0	Tension	260'F'500 evelus	a:	·		0.17	
0,0	Tension	Selono 200 Sector	260	,	,	-71	•
٥.	Tenstor	200 F'1000 eveles	KTD	7.	\$6.0	113	ं ह
0.	Tension	260°F/1000 cv.les	240	£ 5 c.	5 6.	116	3,439
0.0	Tersion	350°F'500 cv. les	RIC	,		104	
• 0	Tension	3+0.1 2005/1.05£	260		•	118	
0.	Tension	350°F 500 cvcles	ů.			124	
0,0	Tension	350°F/1000 cvcles	RTD	- w	1	114	± 100
٥٠	Tension	350°F/1000 cycles	150	ş.	or	129	04.
06ء	Tension	260°F/500 cycles	RTD	,		1.6	•
•06	Tension	260°F/500 cycles	260			1.6	•
.0.	Tenston	560°F/1000 cycles	RID	11 11	000 %	2.7	2,420
.06	Tension	260"F/100G cycles	260	1.64	0.00	1.7	1.590
•06	Tension	350°F/500 cycles	KID			2.3	•
.05	Tension	350"F/500 cycles	260	•		1.6	,
,05	Tension	350°F/500 cycles	05،	•		1.2	•
°0.5	Te sion	350'F/1000 cycles	кTD	‡,	•	•	•
.05	Tension	350°F/1000 cycles	350	0.83	00.00	• . o	1,190

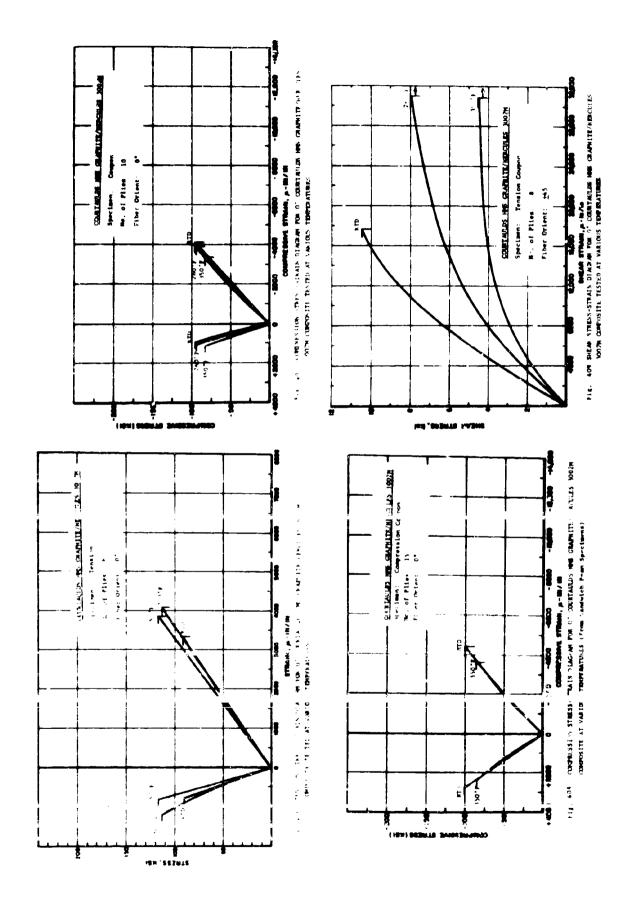
** Broken During Conditioning

Orientation	Type :: ad	Section (C.F.	, paf m 106,	, (10/fn)	uit (ka1)	(city (u) - r
[3/45/135/0/90]	Tenston	26 FF/500 Cales	£,			(7)	
[./45/135/0/ <u>90</u>]	Tension	26 F/500 creles	260			09	ı
\$ [06/0, SE1/55/C]	Tensian	26 °F/1000 eyeles	£	13.7	0.43	23	3, 530
[.06/0, 5£1, 55 ·]	Tension	26(°F/1000 cycles	260	15.1	69.0	\$	3,880
3/06/0, \$21/39/()	Tensi n	35/°F/500 c; cles	Ę		•	1	•
[] /45 /135 /0/90]	Tension	35(*F/500 cycles	260			\$	•
[06/0/51/37/]	Tension	350 °F/500 cycles	350			3	•
[3/45/135/0/90]	Tension	35 °F/1000 eveles	E	1.	9.3 6	14	2, 760
[06/0/321/57/:]	Tension	35(°F/1000 cycles	350	15.1	69.0	*	3,760
	Compression	261 7F/500 cycles	91		•	10	•
• 0	Compression	260°F/500 cycles	260	•	•	109	ı
.0	Compression	26f 2F /1090 cycles	E	19.2	0.28	*	096.4
Ū,	Compression	26(°F/1000 cycles	260	24.0	0.23	102	0%,4
.0	Compression	35(°F /500 cvcles	RTD	•	•	101	,
0,	Compression	35(°F/500 c) cles	350	•	•	001	•
0.	Compression	35(°F/1000 cycles	MTD	20.8	0.19	106	5,270
°,	Compression	35(F/1000 cvcles	350	20.6	0.23	66	7.900

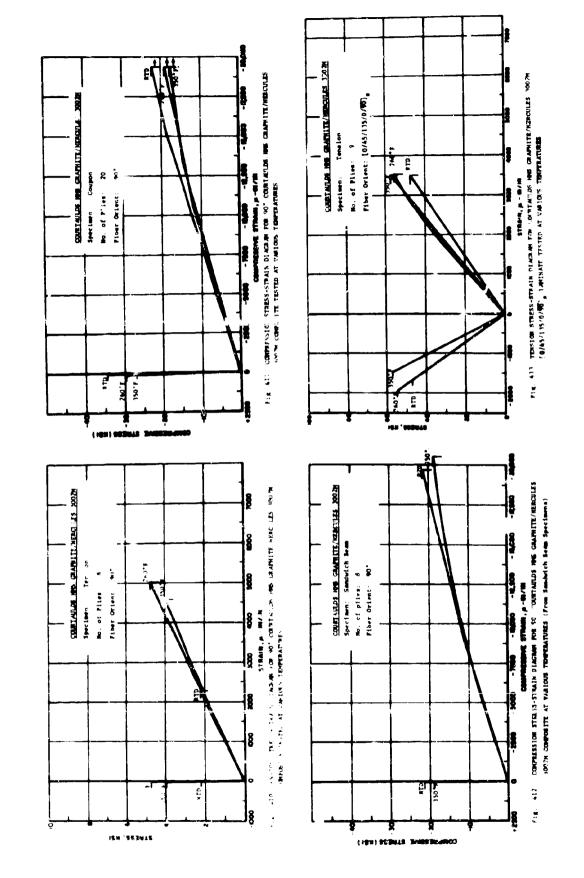
Orfecta: ton	7yr - Joac	Cand Cioning			cal s	ult #41,	-
, O.D.	Compression	260 F. 500 cv. Tes	n).				i ,
90,	Compression	256787 1 1 1 1 25	047	•	,	2B.£	•
• O	Compression	2607F, 1005 colles	RTD		0.70	32.6	J00' 18 <
000	Compression	26) P'inon eveles	260	1	0 0 0	29.8	300' % <
•05	Compression	350°F/500 eveles	KTD	•	•	29.9	•
, C 5	Compression	350°F/500 cycles	350	ı	i	22	•
و ۵ م	Compression	350°F/1000 cvcles	RTD	· -	69	35 6	200° 8 8
ر ع	Compression	320°F/1003 coules	350	7.77	(G) (B)	2) w. 00
[0.45/131/0/90]	Compression	260°F/300 evcles	N.T.	,	•	65	•
[0/45,135/0/ <u>90</u>]	Compression	200°F/;00 cycles	260	,	,	58	•
[0,45/135/0/ <u>80]</u>	Complession	260°F/1000 eveles	CIN	0.01	E#10	62	6.100
[C/45/135/0 90]	Compression	160°F/1000 eveles	260	10.2	0.33	59	τ, 26
[C:/25/135/0/90]	Compression	350°F/500 cvcles	RID	r	•	59	•
[C'45/135/0/90]	Compression	350°F, 500 cycles	350	•	•	55	٠
[C/45/135/0/90]	Compression	350°F/1000 cycles	RTD	us. o	0.38	59	6,410
[0/45/135/0/90]	Compression	350°F/1000 cycles	350	3.6	0.29	X.	5,56(

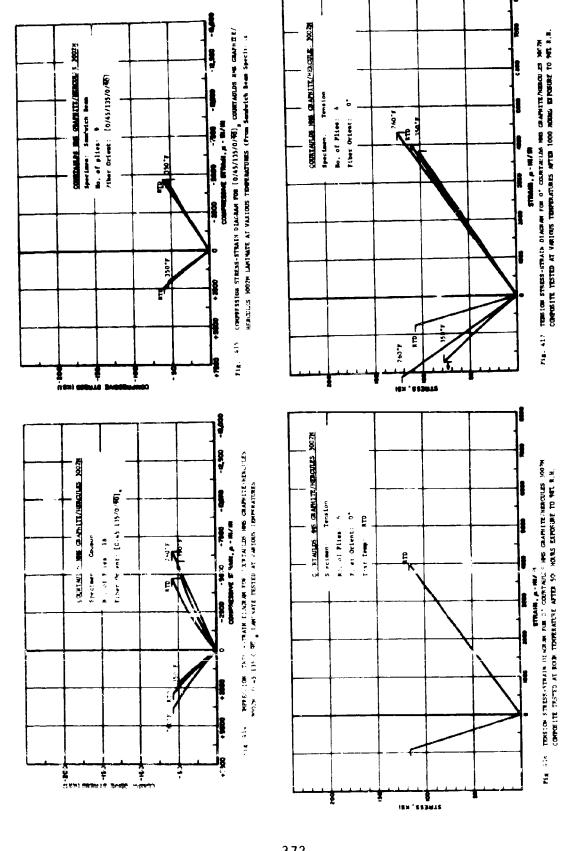
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tonfacton	Type Load	Str. Ford Charles	(i.e. t.g. (f) (e) (e)		, (11/11)	^c ul. (ks1)	"ult (u-fn./fn.)
, C	In Plane Shr	2507Finon weles	кТD			10.8	•
с. •	In Flane Shr	260°F'SOO cycles	260	1		7.0	
a	In Flane Shr	260°F/1000 cycles	RTD	0.79		10.7	23,500
· 1	In Plane Shr	260°T/1000 cyrles	260	0.65		7.4	25,000
'n	In Plane Shr	350°F/590 cycles	RTD	•		7.6	•
0,,	In Plane Shr	350°F/500 cycles	260	ı		•	•
J.	in Plane Shr	350'F'1000 eveles	RTD	0 76		9.6	17,100
<u>ر</u>	I- Plane Shr	153:E/1030 cycles	350	97.0		6.0	> 30,000
0	Int Shear	260:97500 cycles	KTD	•		12.7	•
0 -	Int Shear	26077'500 cycles	260	•		9.5	•
0.	Int Shear	260°7'1000 cvcles	KTD	•		14.2	ı
0,0	Int Shear	260°7/1000 cycles	260	,		11.6	•
ί.	Int Shear	350°F/500 cvcles	RTD	•		12.9	•
٥	Int Shear	350°F/500 cycles	260	•		10.7	•
٥٥	Int Shear	350°=/1000 cycles	RTD			13.0	•
.0	Int Shear	350°F/1000 cycles	350	1		7.9	•



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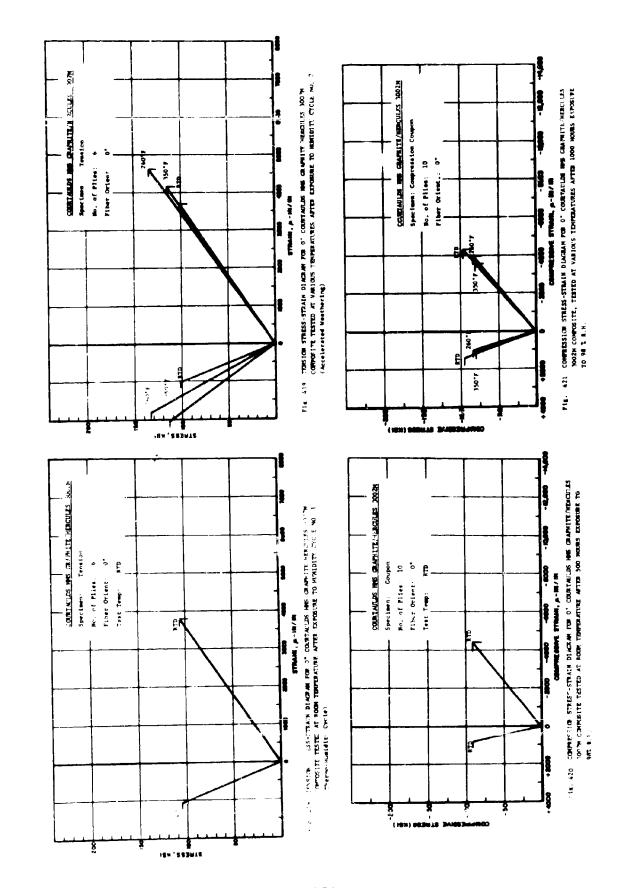




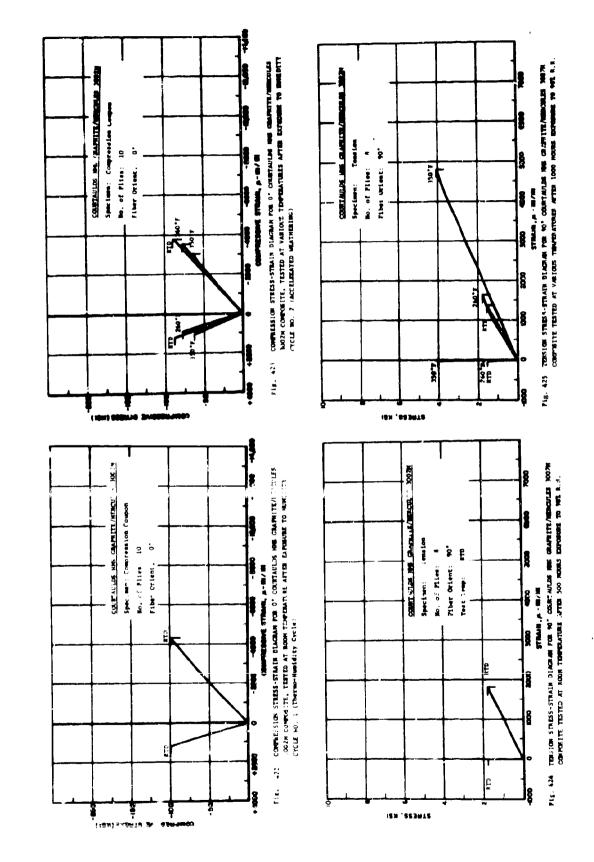
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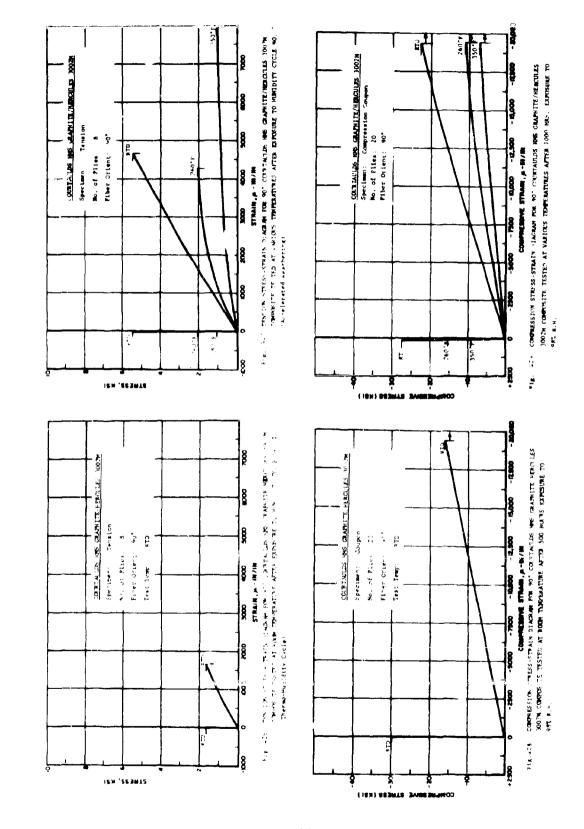
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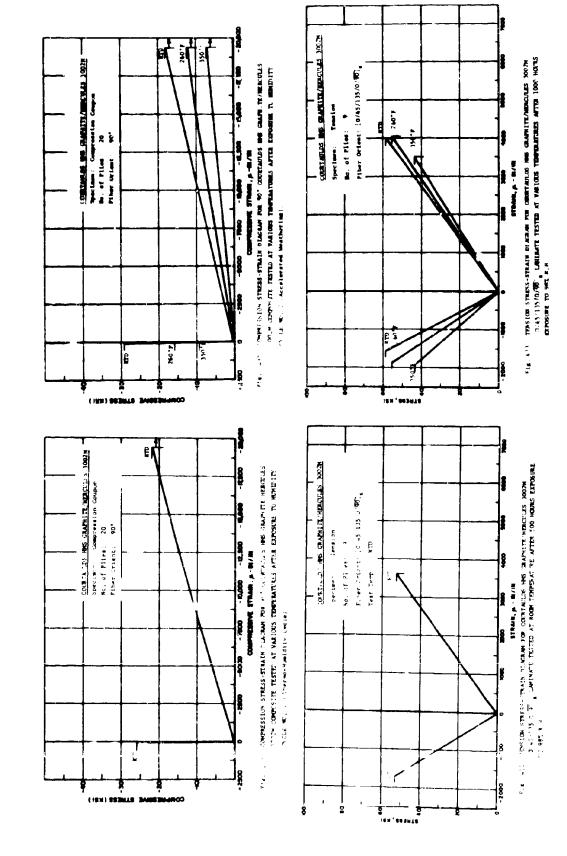
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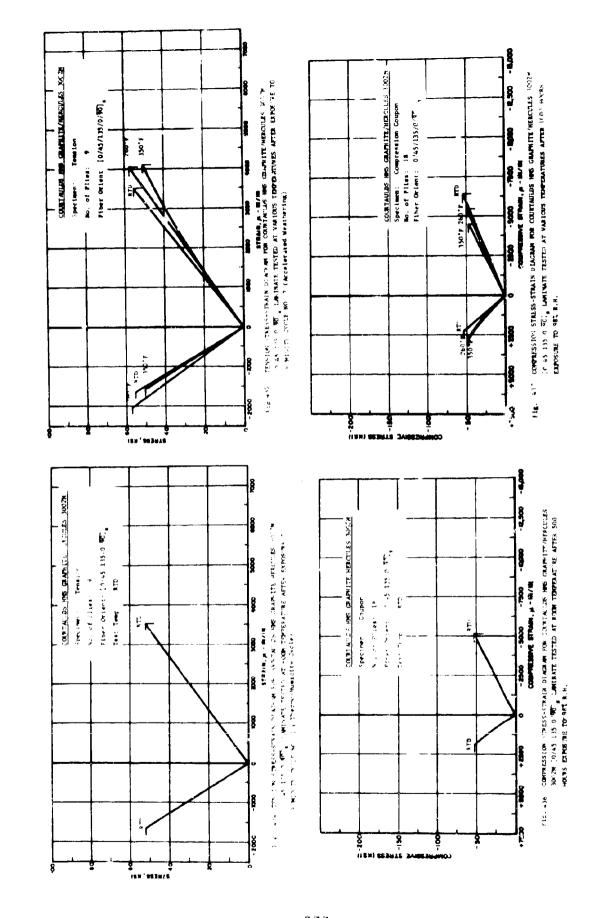
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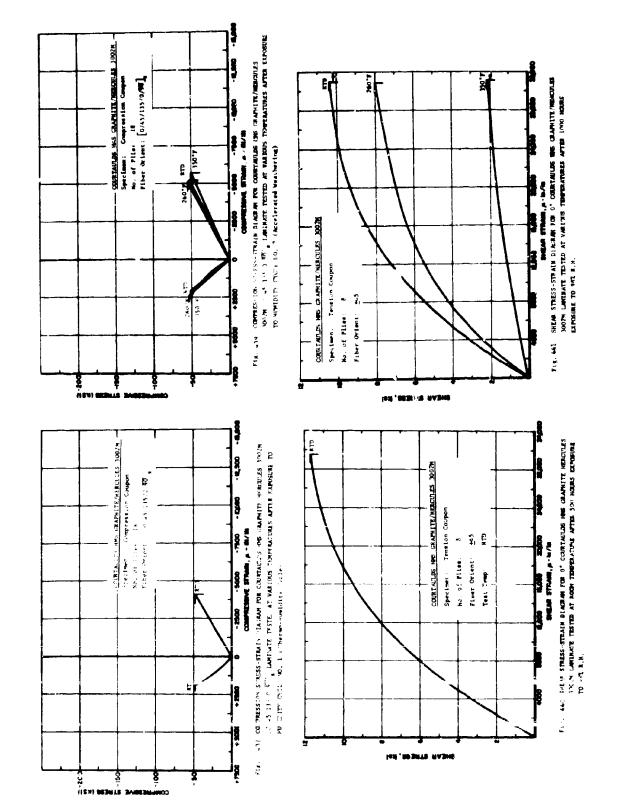
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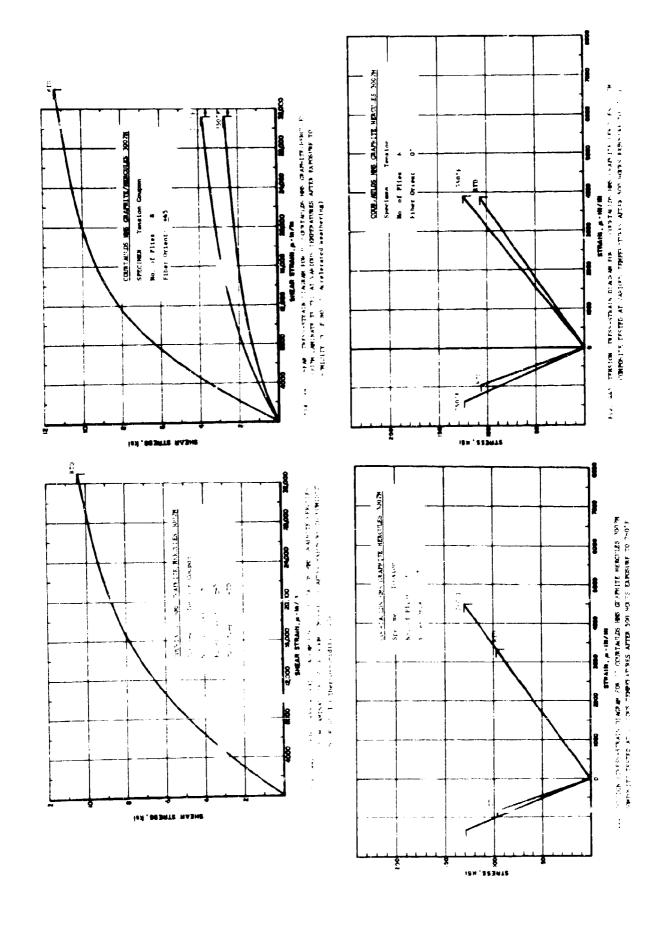
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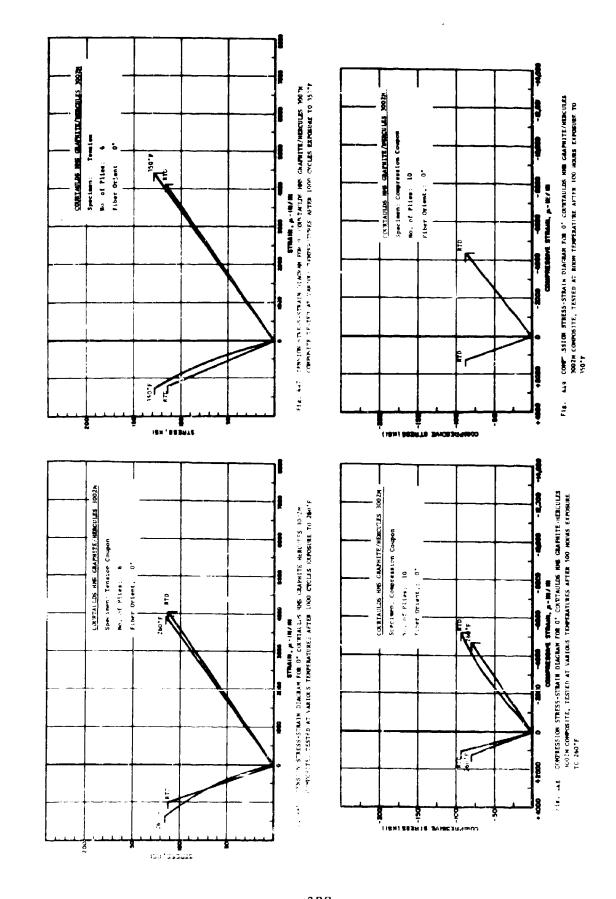
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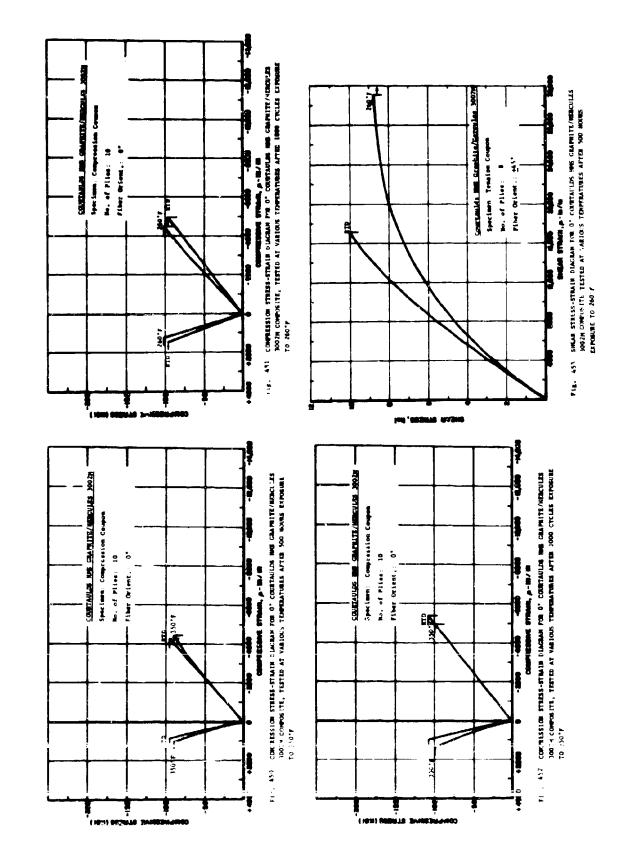
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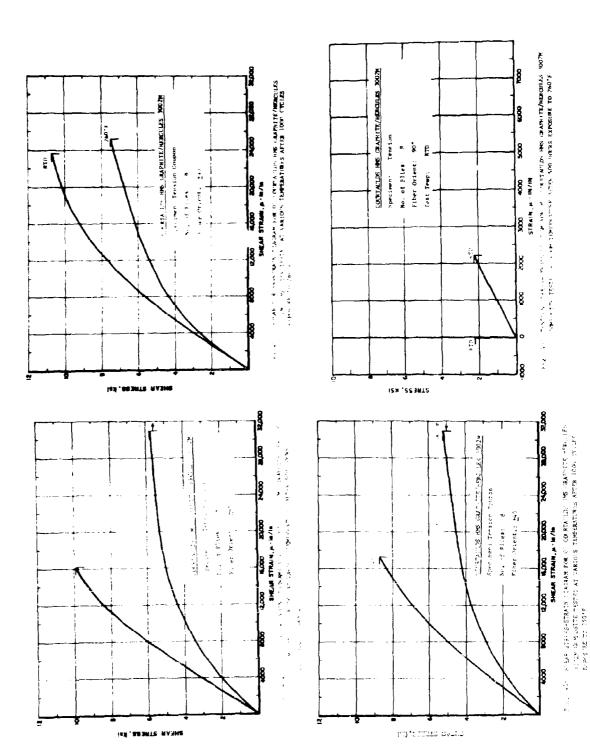
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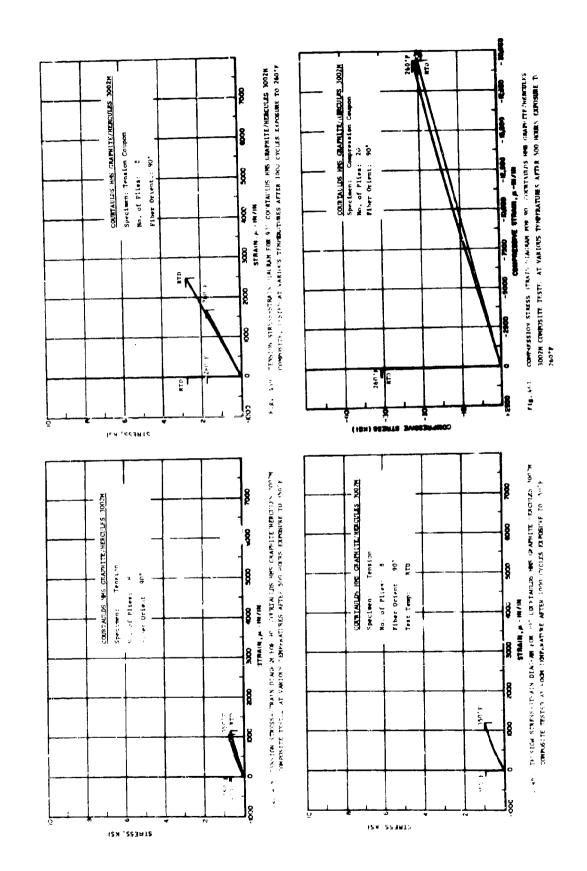


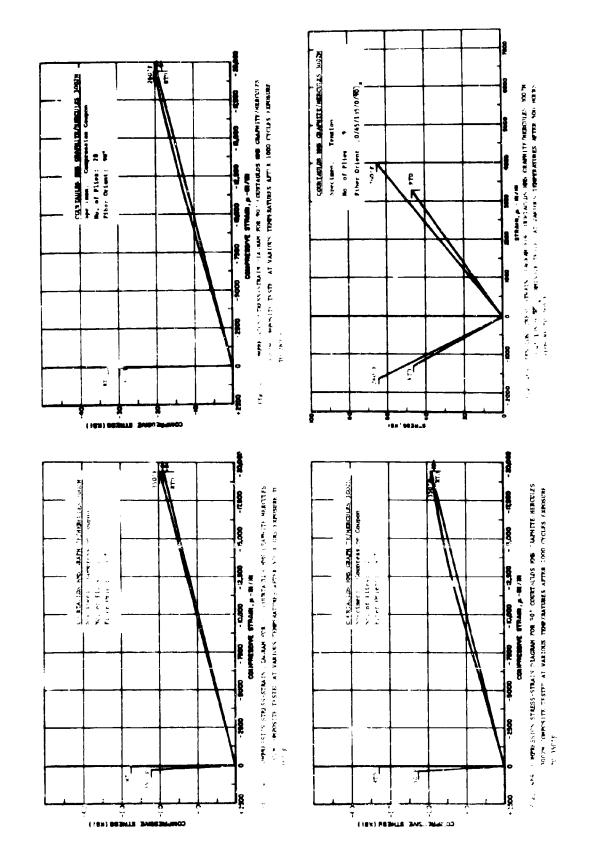
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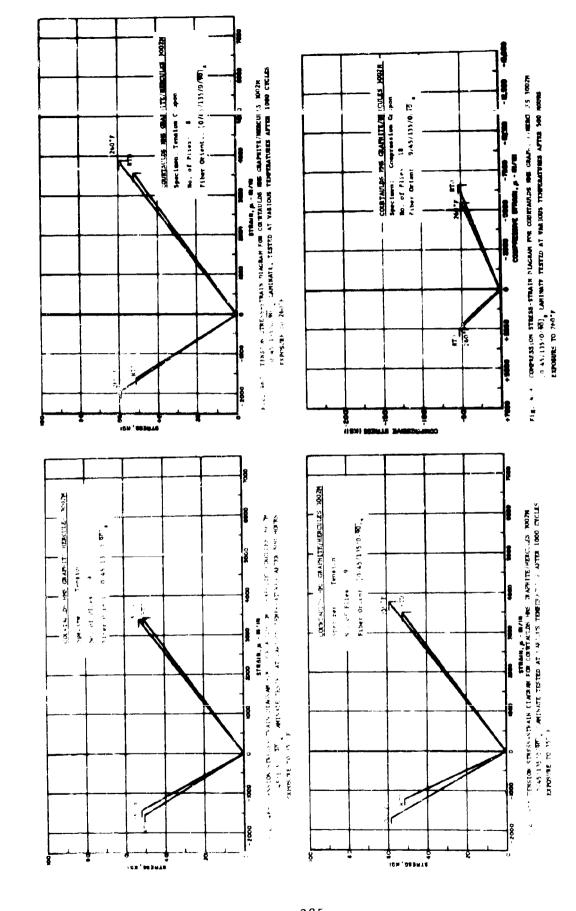


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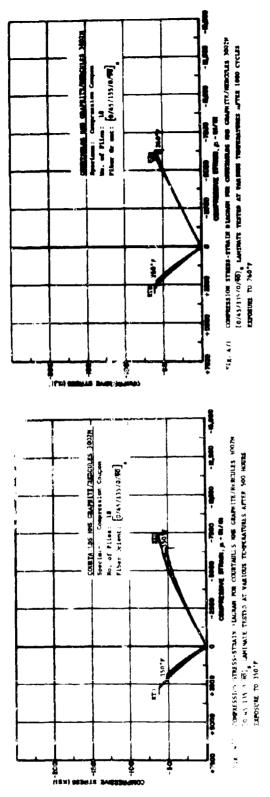
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THE PERSON NAMED IN COLUMN 1

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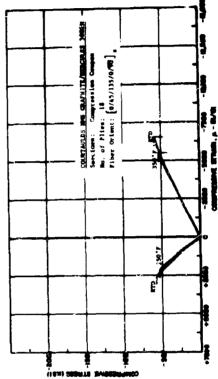


FIG. 472 CONFIESTON STREET-STREET SIZENA TO CONTRACTO ON CONFITTINGCIES 300 MG (0/45/135/0/PG), LANIMAN TESTED AS VALOUS TESTEMBRIES AFTER 1000 CHILES EXPOSED TO 330'F

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ABIT NIX FALLS PROFERITOS STATES

or as	Thickness		PR TON O	PRION CONDITIONING	1	Stress W.	1	1	Applied without Failure	Residue!	
N amber	(Plies) (In.)	Orlensacion	Type	Duration	(°F)	(T ^o ult) iksti	ikst.	(cycles)	(cycles)	(ks1)	Comment
C1205A-t	70°° - 3		3 <u>4</u> .7	,	: *	· · ·	01				Failed under static
1	-				į	4	-				load at tab
C1205A-7	•		3000	,	: 4 :	-	# /i		7 10 . z 10 t	. (11	Timediate 180 Fallois
(120:A-8	·		Seo.	,	i k	· · · ·	- 1		7. IA2 X 11:	7 . 4 .	Toward it. Tak Endlines
9- 4 <0213		-			x a	r. Od	ê ä				Teh Feilure under
C1 202 A-10	•		WW.	•	e K		ć				statte lead
.: "\$00,10			Kana	,	ŝ	· .	, "				Immediate Tab Failure
C120213			J. Co.	,	É	76.5	:				Immediate Tab Failure
C1205A-13	,	c	Kone		RT	,r zo	ž		,		Immediate Tab Failure
41-450210	•		Mone		يَلَ		<i>*</i> .		, o r	112.6	Tab Failure
C1:05A-15	F	C	¥rin.	1	် ည	76.5	ï	363,000			
9 .00.10	, C	í,	2	,	2 : 1 to 1		c.	000.7			
C1.02.78	G : .	ć			É	e/	ء .				Immediate Tab Failure
(A) (C) (C) (C) (C) (C) (C) (C) (C) (C) (C	e C	; \$, Lo	•	KTD	41	- 1	545,0(4			
7-80870	\$ 6 5 5 4 5 4 5 4 5 4 5 4 5 4 5 4 5 4 5 4	ç	None		RTD	*,	ŗ	9.000			
- L - L U	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1		Mone	•	8	٠,	,				immediate Tab Failure
C1303-11	, C 00	Ot.	Monc	•	Ê	:-	ر. •	000.1			(
01307-13	E - P.(157	Q.b	Mone	ŀ	Ê	≉.	. ت				Immediate Tab Failure
C1 102-13	•	ê	Mone		E L	7, 1		000			
C1192-1-	8 - 4.055	, 06	None		Ē	٠ : ٢		900			
C1.32-15	.8 - 0.05 ⁷	100	Ę			?·	^	7,000			
C1027A-s	9 - 0.962	0/45/135/0/90	None		a La	125	60	•			Immediate Tab Failure
C1127A-7	6 - 0.063	0/45/:23/0/93	Kone	1	E	105	9,0	1,000	,		ab Failure
C1227A-3	₹ - 0.063	0/45/135/0/90	None	•	E	2	9	•	10,	59.1	
517.278-3	9 - 0.058	0/45/135/0/90	None	•	Ē	105	s	ı	,		Jumediate Tab Failure
C1127A-10	9 - 0,062	(0/45/135/0/90)	None	•	E	z	57	Ą	, ₀ 1	66.2	
C1 - 27A- 11	090-0 - 6	[0/45/135/0/90]	NO.	,	€	48.5	14	3,000			
C17278-	190'0 - 6	(06/0/5):1/5 * /0]	None	ı	É	961	60	1,000			
- 22	9 - 0.062	06/0/501/55/01	None	•	E	\$	97	,	3.09×10^{6}	3.7	Tab Failure
1 5 1 3 1 4 2 2 1 3	•	0/45/105/0/90	Rone	٠	Ē	901	87				Immediate Tab Failure
		\$ (00/0/00/00/00/00/00/00/00/00/00/00/00/0	į	,	Ē	sr eg Ø	7.4				Immediate Tab Failure
C12273-4	790.0 - 6	10/45/155/0/a					;				

AND ALK CALIFOR AGENCIES SCOME TO SERVE TO STATE TO SERVE THE CONTRACT OF SERVE TO SERVE THE SERVE TO SERVE THE SERV

101.1 100.2 100.	Specimen Marber	Thickness (Flies) (In.)	Orien, actor	FRIOR C	RIOR COMBITIONING	Test Temp. ("F)	(387) (10)	ا آها په چ	Cycles to Failure cycles)	Cycles Applied without Fallure (cycles)	Residual Strength (ksi)	Competit
	C1207A-11	£ - 1,013		# · ·		-		,		3 168=106	101.1	Tab Failure
6 - 0.00 1.00	C.207A-12	C. C+-		1		.				2 500.106	8	Tab Failure
	C1207A-13			ř.		- F,		-				Isb Failed under
	C1297A-1-	0 - 0 PL			•	ع		£	yU12**			Tab Failure
	C1207A-15	O		÷.		· .	:	-				Immediate Isb Failure
	17.18-24 C	, i			•		1	(03	•			Impediate fab Failure
	V:0.		-	1.1	•	2		:03	•			Immediate 1sb Failure
	(120/A-18	<u> </u>	**	Ē	•	Ž			٠, ار			
	C1207A-19	6 - 6.0.		ŭ,				·	ت. نارین	,		Teb Failure
F = Color	C:20_A-70	ı		ć ~						DE 187	27.0	14b Fallure
	C1213-11	1	÷	ří.		1.47			300° 12			
F = 7.01.2 F F F F F F F F F	C1213-12	6.43.3 - ×	5			Ł,	÷	~	•			Failed under Static Load
6 - C. 1042 90	C1213-13	, , , , , , , , , , , , , , , , , , ,	٠	٠٦٠.		i.	4		· .			Tab Area Failure
F = C C C C C C C C C C	C1213-14	ı	¥	70					<u>غ</u> ئە			
6 - (f. c.	C1213-15	3-0-3	ð	31.7			4					
K = 1,045	C1213-16	(1)) - p	ð	Mone.		٠	? ,		# å .			
No.	C1213-17	, 10°, 1	ð	٠ ٢	•	Ξ.		-:	(H)			And I shall be maked to the first
8 - C.04.7 97 8 72.7 260°F 95 7 7 131810°F 94.1 260°F 95 7 7 131810°F 94.1 260°F 95 7 7 7 131810°F 94.1 260°F 95 7 7 7 131810°F 95 7 7 7 131810°F 95 7 9 7 9 10.00 9 - C.06.7 10.457135°0/90° 8 80°F 95 7 7 7 131810°F 95 7 9 10.00 9 - C.06.3 10.457135°0/90° 8 80°F 95 7 9 10.00 9 10.00 9 10.00 10.	C1213-18	3	ੱ ਰ) () 4		ر بنا ر بغ	J	٠.		906	7	Failed Under Static Logs
9 - C.0c. 0.45/135'0/90's Nanc 260°F 91 57 - 7.131x10° 44.1 9 - C.0c. 0.45/135'0/90's Nanc 260°F 91 57 - 7.131x10° 44.1 9 - C.0c. 0.45/135'0/90's Nanc 260°F 91 50 9.000 9 - C.0c. 0.45/135'0/90's Nanc 260°F 91 50 9.000 9 - C.0c. 0.45/135'0/90's Nanc 260°F 86.5 48 - 7.301x10° 53.8 9 - C.0c. 0.45/135'0/90's Nanc 260°F 100 55 - 7.301x10° 53.8 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 - 7.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 - 7.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100° 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100° 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100° 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100° 55 2.000 9 - C.0c. 0.45/135'0/90's Nanc - 260°F 100° 55 2.000	C1214	ı	<u>:</u> حَدَ			E .	1 0	: .	•	11 × 1	• .	Father under Static Load
9 - C.0e1	C1214-	•	÷	ŗ	t		;					
9 - C.066	C12338-2	1	0.670, 311,55.0	# Sine	•	3.09;		5.	•			Immediate Tab Failure
9 - C.06C	C12738-3	- 1	06/0.38(1/57/0]	ארייא	,	560'5		: ₇	•	7.131×106	Ţ.	
9 - C.0e3	C12338-4	•	06/9,521/54/0.	Mine	•	260°F	160	~ ;	0,00,1			Tab Failure
9 - 0.062 (0/45/135'0/90) Nane - 260°F 86.5 48 - 28.100 9 - 0.062 (0/45/135'0/90) Nane - 260°F 86.5 48 - 2.301x106 9 - 0.061 (0/45/135'0/90) Nane - 260°F 100 55 - 2.301x106 9 - 0.061 (0/45/135'0/90) Nane - 260°F 95.5 53 186,00°C 9 - 0.062 (0/45/135'0/90) Nane - 260°F 10°C 55 2,00°C	C12335-5	ì	06/01/01/05/01	100	•	260°F	σ.	50	9,000			Tab Failure
9 - 0.062 (0/45/135'0/90) Nanc - 260'F 86.5 48 - 2.301x106 9 - 0.061 (0/45/135'0/90) Nanc - 260'F 100 5: - 2.301x106 9 - 0.061 (0/45/135'0/90) Nanc - 260'F 95.5 53 186,00°C 9 - 0.062 (0/45/135'0/90) Nanc - 260"F 10°C 55 2,00°C 9 - 0.062 (0/45/135'0/90) Nanc - 260"F 10°C 55 2,00°C	C12338-6		06/0,361,57/0	None	ı	360°F	80.5	Ę	48,100			
9 - 0.061 (0/45/135'0/ 90 's Nanc - 260'F 100 53 - 260'F 0/45/135'0/ 90's Nanc - 260'F 100 53 186,00C 9 - 0.061 (0/45/135'0/90's Nanc - 260'F 100' 53 186,00C 9 - 0.062 (0/45/135'0/90's Nanc - 260'F 100' 53 2,000 30731306	C12138-7	•	06/0,561/57/0	ř	•	260°F	86.5	87		2.301×106	53.E	
9 - 0.061 30/45/135 0/90.5 Mane - 260°F 95.5 53 186,000 9 - 0.062 30/45/135 0/90 Mane - 260°F 100 55 2,000 9 - 0.062 30/45/135 0/90 Mane - 260°F 100 55 2,000	C12335-8	ı	(0/45/135 /0/ 9 0.	Suc.	•	3.09Z	100	ç;				Falled under Static
9 - 0.062 (0/45/135.9/90 s Name - 260*F 100 55 2,000	ST ME	,	(0/45/135 0/90]	None	,	260 °F	45.5	53	1 86. 000			
901-1101	01-348-10	•	06/0.381/59/0	<u>#</u> .πe	•	260 € F	100	55	2,300			Tab Failure
	בו יוננינט			1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	,	260'F	, . 60	, ,		2.673x106	51.8	Tab Failure

HING SHOPPING WINDS

Specimen	Thickness (Plies) (In.)	Ortentation	FLIOE O	COMPLITIONING Duration	Test Temp. (*F)	Stress Levei (T ^o ult) (ks1)	eve i (ks1)	Cycles to Failure (cycles)	Cycles Applied without Failure (cycles)	Residual Strength (ks1)	Coment
1-670-17	E 30	C	, di		::	F. 5	10,	1.29b.0%			
C12078-		ی د	2		4500E	3.	ž		2.142×10^6	7 98	Tab Estlure
C12078-1	100	. c	# NO.	,	0.7		, D	2. ng		3	Tab Mailure
C12075-4	770	, c	juo j	•	1.00	£.	1				Tab Failure
21070-1	ر ان ان	. 2		•	1.051	91.5	501	;			Immediate Tab Failure
C12078-6) o		Evole .		120.F	90.5	3	•	2.474 × 10 ⁶	114.5	
C1297E-	5,000	٠.	None S	,	3,0°F	£:,5	105	25,000			Tab Failure
C12C7B-8	6 - 2 043	5	Kope	•	350	٠ ت	.01	79,000			Tab Failure
C12078-9	770 0 - 9	5	*COD	•	320.1	26	š	•	-4		Ispediate Tab Failure
C1207B-10	e - C. Je.	9	None	•	350-F	, 68 ,	103	•	5.071 x 10°	106.2	
C1214-3	8 - 0.043	- 36	None	•	350°F	37	1.5	•			Failed Under Static
C1214-4	8 - 0.044	3	NOCA	•	350 F	24.5		ı			\ Load while coming up
C1214-5	8 - 0.043	, Co	Hone	•	350 F	5.61	ĸ,				to Temperature
C1214-6	6 - 7.043	•06	None	•	350°F	17	. 7	10,000			
C1716-7	26 - C. 04.3	• \$	Kon	,	350 F	17	۲.	,			Immediate Failure
C1214-8	8 - 0.0r.3	ŝ	None	•	350°F	, ,	9,	1,000			
C1214-9	8 . 0.043	• \$	Mone	•	350°F	S :	ξ.				Failed Under Static Ld
C1514-1C	3.0°C - 80	8	Cope	•	350.4	<u>.</u>	Ţ	24,000			
01214-11	4	. 3.	je j	ı	350 F	ĵ.,	٠, ۲	200.7			
71-71717	50.0	2		•	R	ن	?	3	4		
C1234A-1	9 - 0.451	06/0/58/1/55/0	Hone	•	350°F	83	50	•	2.504 x 10°		
C1234A-2	9 - 0.053	(0/45/135/0/90) J	•	350°F	\$.	09	1,000			Tab Fallure
C1234A-3	9 - 6.053	[0/45/135/0/90]	<u>.</u>	1	3507	88	53	132,000			
C1234A-4	9 - 0.052	06/0/51/32/0/30		•	350°F	2	53				Immediate Failure
C1234A-5	9 - 0.050	(0/45/135/0/90)		•	350°F	8	75	,	2.068 x 106	52.5	
C12344-6	9 - 0.054	[0/45/135/0/90]		•	3.050	93	£	11,000			Tab Failure
C1234A-7	9 - 0.054	[0/45/135/0/90]	100	ı	350°F	93	*	12,000			
C1234A-8	9 - 0.053	[0/45/135/0/90]	į	•	350°F	99.5	3	22,000			Tab Failure
C1234A-9	450.0 - 6	[0/45/135/0/90]	į	•	350°F	91.5	\$5	ı			Immediate Tab Pailure
							:				: : : : : :

TABLE KIN FALLOTE PROPERTIES SURVAPOLEM BIORUMS 100 CONTRACTOR (COMPONING COMPONING CO

Cycles Appl: ed Mesidual re Failure Strength (cyc.es) (kal) Comment	Tab Failure Tab Failure Tab Area Failure	2,23 x 10 ⁶ 71.9 Failed while coming 2,163 x 10 ⁶ 91.3 up to temperature Immediate Failure	Falled coming up to temp. Imacdiate Tab Falluro 2.07 × 10 ⁶ 79.1 Tab Fallure Falled under static load 2.25 × 10 ⁶ 89.2 Tab Fallure		2.29 x 10 ⁶ 62.6
Cycles Exp. Fatlure (cycles)	90° -) t +) t		1,000 33,000 21,000 1,000	1,000
Stress Level	85.5 100 84 98 79.5 93	64 70 50.5 55 59.5 65 55 60 59.5 60	64 45 41,5 44,5 44,44 44		40.5 48.5 60 47 47 88 88
Test Test ('F)	55.55	260 ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° °	350 m 350 m 350 m 350 m	MTD MTD MTD MTD MTD	260°F
PRIOR COMDITIONING Type Duration	500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs.	500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs.	500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs. 500 Hrs.		1000 Hrs. 1000 Hrs. 1000 Hrs. 1000 Hrs.
Pator CO	96 RM 196	H2 136	HM 196		
Orlentation	65,000	လိုင ် ဝေ	ဝင်္က ဝ		
Thickness (Piles) (In.)	6 - 0.043 6 - 0.043 6 - 0.045 6 - 0.045	6 - 0.0-3 6 - 0.044 6 - 0.044 6 - 0.044	0.042 0.042 0.043 0.043 0.043	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	35000 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1
, pecimen Murber	112058-7 112058-8 112058-9 112058-10 012058-11	C12(98-15 C1208:-16 C1208:-17 C1208:-18 C1208:-19	(1209A-15 (1209A-16 (1209A-17 (1209A-18 (1209A-19	C12058-12 C12058-13 C12058-14 C12058-14 C12058-16 C12058-16	C1209A-1 C1209A-2 C.209A-3 C1209A-4

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TALLE VIX POTTOR PROPERTIES SCHWAS - GASLLES 3000 FOR TRITISHINS CRAPPORTE CARBOSCHES

, and the second	Thickness		300		Stress Level	7	Cycles to Failure	Applied without Failure	Residual Strength	
	(LTTER) (TU.)	OLI CURRENCI CO	Type Duration	(.)	(Z'ult) (ksi	KB1.	(cycles)	(cycles)	(ksi)	Comment
C12058-17	6 - 6.041	٠.	Therac-Humidity Cycle	3	36	2	,			January Sat Lat lives
C1 205B-18	6 - 0.043		ThermHunidity Cycle	RTD	66.5	5		•		Immediate Tab Failure
C12048 -19	270.0 - 9	۲.,	Therms-Humidity Cycle	RTD	2.5	0		2.45 × 10°	7.78	
C1:038-20	6 . C.042	ų.	Thermo-Humidity Cycle	Ĺ	29	د ک	00.7.40			Tab Patline
C17028-71	6 - (.043		Thermo-Hunidity Cycle	C.	65	89	77,00€			
C1.209A-5	6 - 0.0-2		Thermo-Humidity Cycle	260°F	-	Ş	,	2 087 - 106	3 (4)	Fat 0.41
C1 209A-6	6 - 0.042	6	Thermo-Bunddity Cycle	3,09€	55.5	32	•	2.194 × 10	0.000	
C1209A-7	6 - 0.04?	ن	rac-Humidity	260°F	79.5	100	,			Parediate Teb Pailings
C1209A-8	170.0 - 9	Ç.	rao-Humidity	260°F	71.5	۶	200°			Teb Faflure
C1209A-9	C + 0,042	_	Thermo-Humidity Cycle	260°F	75.5	45	000 788			Tab Pailure
C12096 -5	6 - 0.043	5	Thermo-Humidity Cycle	3504	65	75	1	•		Failed coming to temp.
C12098-6	6 - 0.043		Thermo-Humidity Cycle	350°F	56.5	65	•	2.472 x 10 ⁶	91.6	
C12098-7	0.042	c	Thermo-Hunidity Cycle	350°F	61	50	•	2.345 x 10°	0.8	Tab Failure
C12098-8	6 - 0.043	ن:	Thermo-Humidity Cycle	350°F	9	75	1			Peiled coming to temp.
C12038-9	9 - 0.04:	ċ	Thermo-Humidity Cycle	350°6	63.5	73	67,000			Tab Pailure
C1205C-1	9.0.0 - 9	ŋ	Acc. Wthrng.	e	.	87				Immediate Tab Pailure
C120\$C-2	6 - 0.043	つ	Acc. Wining.	E	C 20	\$	165,000			Tab Pailure
C1205C-1	6 - 0.043	,0	Acc. Wthrng.	Q.	885					Immediate Tab Pailure
40X0310	6 - 0.043	ပ	Acc. Wthrng.	Ē	76	2	•			Immediate Tab Pailure
C1205C-5	6 - 0.043	,0	Acc. Wthrmg.	Q L	75	75	•			Immediate Tab Failure
C1209A-10	6 - 0.043	•3	Acc. Wthrng.	260° F	5.2	75	•	9012417 6	105.0	
C1209A-11	6 - 0 043	•0	Acc. Uthrug.	3 60	3	85	•	7.685x106	4.5	Tab Patlure
C1209A-12	6 - 0,045	•0	Acc. Wthrng.	260°F	75	001	1	!		Immediate Teb Pailure
C1209A-13	§ - 0.04§	່ດ	Acc. Wthmg.	260°F	67.5		1.040,000			
C1209A-14	6 - 0.043	ů	Acc. Wthrag,	260'F	71.5	95	000			Tab Failure
C12098-10	- 0.0e- 9	•6	Acc. Wchrng.	350.5	\$	75	. ,	2.085 x 10 ⁶	98.6	
C1209B-11	. 0.0k	,0	Acc. Wthrmg.	350	74.5	S	•			Teb Failed under Static
C12098-12	6 - 0.04.	ູ້ວ		350"#	2	8	•	4		Tab Failed under Static
C110013	6 - 0.043	•	Acc. Wthrng.	350°7	¥7.5	77	•	2.458 x 10°	101.9	
C120978-14	6 - 0.043	•0	Acc. Wehrng.	350°F	£ 0.5	2				Tab Failed under

,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们也没有一个时间,我们的一个时间,这种时间,也是这个时间,也 一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们就是我们的一个时间,我们也是

COLUMN TERROR PHONE BY THE COLUMN TO THE SECOND SECONDARY SECONDAR

							; !	Cyc.ies	Cycles		
Seectmen	Thickness		PRIOR (PRIOR CONDITIONING	,	Stress Level	401	Failure	without Failure	Residue!	
Jack .	(Piter) (In.)	Orientation	Type	Duration	E	(Lait) (kst)	ks1)	(cycles)	(cycles)	(kst)	Comment
C12384-6	9000-0	06 1.55 1.59/0	-86	500 Hrs.	g g	1	2	•			Immediate Tab Failure
C1228A-7	90.0		126	/ 500 Hrs.	Ê	95	ž		•		Immediate Tab Faliure
C12284-8	790.0 - 6	:		/ 500 Hrs.	ţ	76	ټ		2.121 :: 10°	я	Tab Area Peilure
C12284-9			187 RH	/ 500 Hrs.	KT.)	85.5	Ş	1,300	•		
C17284-10	990.0 - 6	Ŧ		. 500 Hrs.	KTD	81.5	7	•	2.755 × 10°	53.6	Teb Area Failure
C12354-1	590 0 - 6	•	196. EM	/ 500 Hrs.	260*	96	53	•	•		immediate tab failure
233		:		500 Hrs.	240°F	70	9	,	2.475 x 10°	48.9	Tab Area Fallura
1217	0.0	•	2 .56	/ 500 Hrs.	260 F	78.3	45				Fellod coming to two.
5.00	9 - 0.065	:		500 Hrs.	3.097	ز.۱.	7	•			Semediate Tab Paibure
ر. و	940.0 - 6	÷	98. RH	/ S00 Hrs.	200'F	22	43	ı			immediate tab Failure
1 3 4 6 1 7	340.0	s	987 RH	, 500 Hrs.	350°F	5	\$				Tab Patlure to Tame.
-12380-	590.0	:			350 F	118	3	,			lemediate teb Patluce
712 KB - 1	0.063	•	200		350 'F	3. 5	3	•			immediate 1sb Pailure
7.348.4	800	:	. B.	500 Brs.	3.00.	S-86	3	•	4		Tob Felled under static
C12368-5	9 - 0.043	:		/ 500 Brs.	350°F	88.5	* 2		2.0 × 10	51.6	loss cenist to temp.
C12364-11	\$ 90 ℃ * e	*		/ 1000 Hrs.	ŧ	3	65	•			Immediate 1sb Pailure
(12721)	\$ 60 C	:	£		£	103	3	•			Insedicte Teb Fallure
C1324R-7	593.	٤		/ 1000 Hrs.	£	2	Z				Pailed mader static load
C12202-1	\$ 6 C F 2	:			£	6.8.5	3	•	•	;	Empediate Tab Failure
C12288-4	9 - 0,065	Ξ	186 188		ē	51.5	2		2 z 10-	41.3	Tab Failure
712151.4	1900	£	226	/ 1000 Hrs.	1.092	\$	2	•	•		Immediate Failure
C1715E-7	190	:			7.09Z	ş	35	•	2.06 x 10,	20.4	Tab Padlure
C12358-R	290	:		/ 1000 Brs.	250°F	72.5	9	•	2.323 x 10°		Tab Fallure
5-85273	190	:	98". 84	/ 1000 Nrs.	260°F	81.5	45	•			Falled under static 1d.
C12355-10	9 - 0.063	:	86. RH	/ 100C Hrs.	260 F	81.5	• •2	•			Immediate Tab Failure
617348-6	790	=	HW 186	/ 1000 Hrs.	350 FF	108	14		•		Impediate Failure
C12.208-0	790		987 EH		350°F	43	3		2.377 a 10°	o. 1	
CEC 300-7	590	:	796 HM 796	/ 1000 Hrs.	350	103.5	53		4.6 x 106,	55.7	
0-00071	290	=	₩ 196	/ 1000 Brs.	350 F	113	Z	•	2, 666 x 10°	 S	Tab Failure
C12368-10	9063	(0/45/135/0/90,	957 EE	/ 1000 Hrs.	350 F	126	55				Immediate Fallure

"AB B XIX FALLAGY NEW KLITS SCHEAKS - HERCILLS BOTO CONTROL DO A CONTROL DO A CONTROL
įį	Thickness (Pitec) (in.	Ortemeation	PRIOR COMPITIONING Type Duracton	ijĖ	Stress Level	(kei)	Cycles to Failure (cycles)	Cycles Applied without Failure (cycles)	Residual Strongth (kat)	1
C122fa-5	9 - 0.065	05/0/SE1/S\$/0]		£	\$°	50	1,000			The Patlure
C12248-6	9 - 0.(63	, = =	idity.	e	67.5	<u>۾</u>	. 1	2.23 x 10 ^b	50.1	Tab Failure
C12288-8	990.0	:	Thermo-Hamidity Cycle	2	73.5) e		2 267 - 106	6	Emmediate Failure
C12288-9	9 - 0.666	=	-	E	81	4.2	1,000	21 4 21:1		Teb Failure
C12364-1	6 - 0.662	z ;	midity.	3,092	78	4.5	•	2.037 × 10.6	40 40 7	Tab Fedlur
C12344-2	0.063	. :	Thermo-Bundity Cycle	260°F	65.	5.5	1	2.603 × 106) ; ;	y 151111
C1236A-4	9 - 0.062	Ξ		260 7	36.5	y 4		2.469 x 10°	57.8	Tab Failure
C1236A-5	9 - 0.663	Ξ	addey.	260°F	3	9	900			Isb Failure
C1237A-1	9 - 0.051	=		350°F	110	55	• •			Immediate Teb Zailure
C1237A-2	9 - 0.651	= :	ddity	350°F	80	07	•			(Tab failed under static
C1237A-3	1.5.0 · 6	: ;	dairy	350*	ာ	07		**		(load coming to temp.
C1237A-4	C - 0 - 6	: 1	Thermo-buildity Cycle	7.05.	2 %	35	•	2.25 x 10°	۳. ۲.	Tab Pailure
		:	Ì .		2	2	•	01 X 70.7		
C12288-10	3000 - 6		Acc. Vol. m.;	6	200	5 5	160,000			Tat " . Ilure
C1229A-1	400	r		3 6	77 5	9 9	, 000 ,	901 - 10 6		Tab Failure
C1229A-2	9 - 0.055	z		E	28.	7	000	01 × 10.7	21.0	Tes Failure
C1229A-3	9 - 0.0°5	£		£	91	. S	3			immediate Tab Pailure
C1236A-6	6	2	Acc. Wehring.	260°F		20	3.000	•	,	
C1236A-7		2		260°F		45	900		•	
C1236A-8	•	z :	_	260°F			•			Immediate Tab Failure
C1236A-9	i i i i ch gr		Acc. Without.	 		% 5	1,020,000	• •	•	
	0.00		-		;		33,11	•	•	
C1237A-6		: 2		25	\$ 5	ጸ :	30.0			;
		•	-	1,055	3 2	i :	36,			Tab Pailure
C1237A-9	9 - 0.0' 1	•		350.7	5.5	3	•	2.163 x 106	55.4	Served makes pressed in .
C1237A-10	5 - 0.0 - 6	[0/45/135/0/90]	Acc. Williams.		707	25	•			Teb Pailed under Static
										Load While Condrg up to Tomostature

A Section of the sect

c Applied Residual Festidual Contaent (cycles) (ks) Contaent	140 Teb Fallure 100.6 Teb Fallure 100.6 Teb Fallure 100.0 Teb Fallure 100.0	13.0 × 10 ⁶ 116.6 Tab Failure 000 7.056 × 10 ⁶ 99.4 Tab Failure	J, (UG Tab Failure	Tab Tab Failure
*c1e *	1,330,00m 1,330,00m 1,00	"	, -	%
Stress level	2 x 3 5 x	*\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\\	发生	88.7 10.7 2.5 10.5 10.5 10.5 10.5 10.5 10.5 10.5 10
Total Total		E E E E E E	######################################	######################################
Polow CONDITIONING	260 F 7 300 60 c. 2 360 F 7 300 80 c. 2 260 F 7 300 F 7 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	***	260 F 7 500 Cvc. 260 F 7 500 Cvc. 260 F 7 50 Cvc. 260 F 70 Cvc. 260 F 70 Cvc.	260°F / 1000 (W., 260°F / 1000 (W., 260°F / 1000 (W., 260°F / 1000 (W., 260°F / 300°C (W., 350°F / 300°C (W., 350°C
in the contract of the contrac		Sec.	<u>.</u> • .	ည်း ကား ကုလ်လို့လို့ပြု
Thickness (Plest the				
Sp. timen R. Sber				68-10 01.088-10 01.088-10 01.088-10 01.088-11 01.088-11 01.088-11

TABLE XIN FATIOT PROPERTIES SINCHARY RPPOTIES \$002R/CONTACIDS NOS
GRAPHITE CONPOSITES

C12114-5 C12114-6 C12114-7 C12114-9 C12114-10 C12114-10 C12114-11 C12114-15 C12114-15 C12114-16 C12114-16 C12114-16 C12114-16 C12114-16 C12114-16 C12114-16 C12114-16	(Pites) (In.)	Ordens et fra	7996 260 F 260 F 260 F 260 F 260 F 260 F	Suration Sur hra.				,,,,			
Omegae 500000.	50.000 50.0000 50.000 50.000 50.000 50.000 50.000 50.000 50.000 50.00		260 F 260 F 260 F 260 F 260 F 260 F	Sin hir.	()	i viti (481,	K51,	(£3()(8)	(cyc.ee)	(201)	
Omages 50000 10000 1000	20000 00000 00000 20000 00000 00000 20000 00000 000000		260 F 260 F 260 F 260 F 260 F 260 F	SOC HEE	é	3	à	1	,		Pailed under Static La
ഠാലയിയുടെ വിക്കുക്കും കൊരുന്ന പ്രത്യേക ഉപ്തുക്ക് നുകൾ 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	2000 000 000 000 000 000 000 000 000 00	1. 2	260 F 260 F 260 F 260 F 260 F 260 F		, Q	<u>, ,</u>	7.	ì	2.26 # 10°	5.5	Teb Fallure
Omerica cista (187).	200 0000000000000000000000000000000000	1.5	260°F 260°F 260°F 260°F		7.74	•	¥				Failed under Static Ld
	99999999999999999999999999999999999999	1. <u>2</u>	260°F 260°F 260°F 260°F		 6. 4	. 4	; ;;	63,650			Teb Pailters
, , , , , , , , , , , , , , , , , , ,	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	1. .	260°F 260°F 260°F	E	4) sa T				Immediate Tab Failure
	4 5 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4 4	Nu s	260°E 260°E 260°E	MO ETS.		5	:				
	18 4 4 5 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	N. S. C. A.	260°F	50 0 Cvc.	# 0 9 0	76.3	95	2,000	9"	•	
	1000 0000 1000 0000 1000 0000	Now 1	260 +		14. 147	65.5	٠. و٠		2.04t x 10	ŗ.	
, , , , , , , , , , , , , , , , , , ,	20 00000 20 00000 20 000000000000000000	<u>.</u> .		20	3.09Z	;;	¥	•	7. 363 x 10	7.5	
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	9 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6		3 9 0.1	اور د د د	7.0.7		. 0	. (immediate Tab Follower
	600		7.095	100 CV:	9-	7	0	,			Total Petilogram
	5 6 6 6		£.097		36.	, . •	ζ:	200	9 me 1 - 106	1	
			F. 097	100 CA		× .	Ç;	•	901 - 100 - 1	8	Tab Patlane
		,	3.09E	1000 Cyc.	260°F	, . 1	()				
1 1 1 4 40 4			5	344 39 /	350 F	*	7.	•	2.04 m 10.	1.8	
i i	٠ ا ا	. (3 5	2	350°F	į	ĭ	261,000			\$ 41
•	100	7- €	25	500 Hrs.	355 ·F	65.5	S.	•			Intelligie 140 February
) •	200	÷	5	, 500 Hrs.	1.056	ع'	83	•	4	•	TOTAL MENT STORY
٠		č	1.55	. See Brs.	350°F	ę.	6	•	2.392 x 10	ī	
Cizilb-5 6 -		, ;		700	3.056	\$ 79	9	•	7.482 m 10	4.7	
C12118-6 6 -		. :	1,000			7. 7	8	968,000			1
9	1	ا ت	1000		350°F	. C.	200	900			
•	0.040		1000	3 2	1.03	76.5	56				Immediate To Palies
•	c.045	ت :	1,016		1,050	eri 13	93	•			Pation unter Park in.
. 9 0	٥. دع	ت	1 DC1			;	ï	•			lamedists Tob Pailter
7	440	ئر	1.05.	. 1000 CAC	330-1	C. (n e	97.			Tab Petlure
CIZIID-II	4	ن.	350°F	1000 Cyc	33C F	: · ·	3 8	3			Tab Patlure
C12115-12 6 -	0.0		356.1	1000 Cyc	1000	2 4	2 =	000	•		Tab Failure
- 9	70.0		2	1000 CVC	200	9	. £		2.126 x 10°	K .2	
C12118-15 6 -	0.046	ີ.	386	, 1000 Cyc.	100°C	5	:				

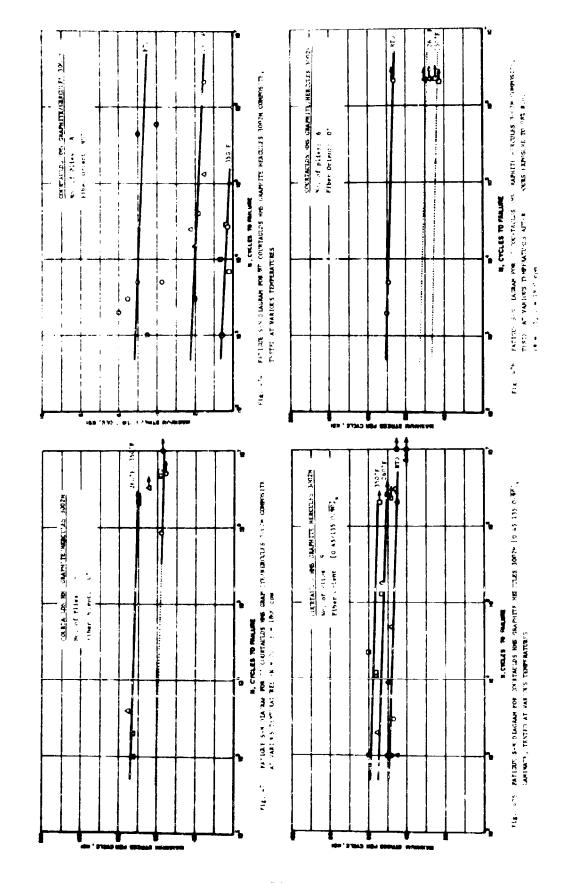
A TIMEN FACE FROM ALD STUDBERS FOR THE PROPERTY OF THE PROPERT

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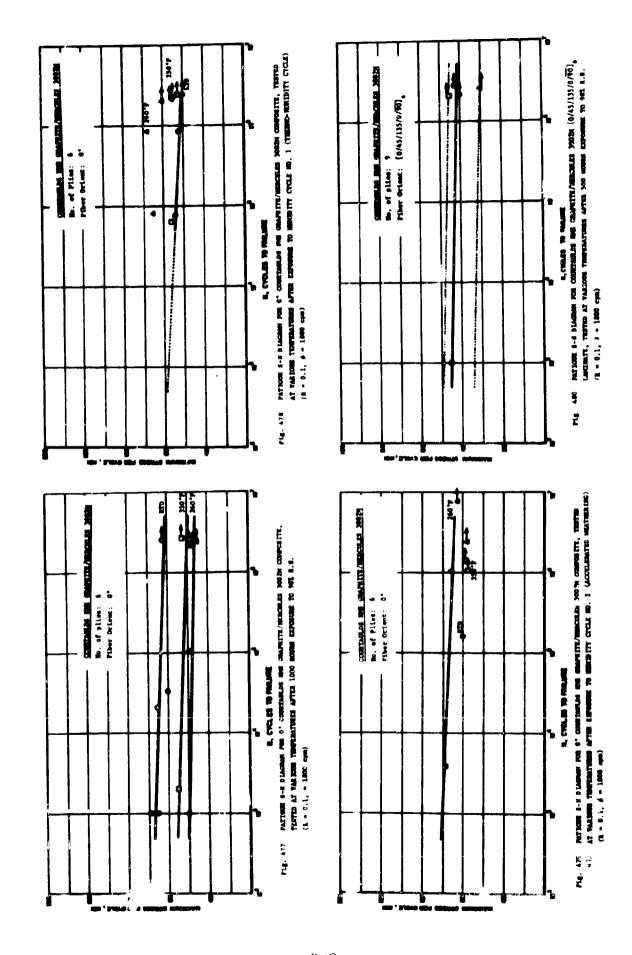
Spicinen Fimber	Thickness (Plies) (In.)	Orientation	PRION C	PRION CONDITIONING	₽ ₽ 3 31 0 1 1 1 1 1	Stress level	evel (ksi)	Cycles to Failure	Cycles Applied without Failure [cycles]	Residus! Strength (ks1)	Comment
01,130		1.	4). 	1	1	۶				Land to the Control of the Control o
(1736-1)	un	•	, i.	į į	: 'c		2 5				Impediate Tab Failure
8-306-3 1 - 306-3	0 . 0.057		; 9 ,		×	<u>(</u>)	50				Immediate Tob Failure
14円のしつこう	50.		ė	-	ox [ć	৽		•		Immediate Tat Failure
C1.30E-5			0	Ģ.	ž	:	S	•	4.651 x 10°	43.0	Tab Fat.ure
C1330E-5	.0.0 - 0	-	350			ż	30	•			Immediate failure
C1.1308-7	20000		35.		-E	Œ.	. 1	2,000	,	•	Possible Tab Failure
71.230E-8	160 U - 3		350 :		ĸ	3 ¥	3,		2.256 x 10 ⁰	17.1	Tab Failure
6-30E71J	100 - 1 2		350 ₽		ų.	7.5	38	000*611	•	•	Tab Failure
C1336F-1f	1277 - 3	-	35018	'F 500 Hrs.	a.	¥8.5	7		•	•	Immediate Tab Failure
010314-1	, (C. C		26		'n		13	•			Immediate Tab Failure
C1231A-2	1	:	3: J97	•	æ	15 4	ŭ,		•		Immediate Tab Failure
011314-3	550.0 - 6	1	260.07		œ	63.	Ā	,	5.066 x 107	48.2	Tab Fatlure
C1131A-	150°0 - 5	-	360		S.	.1	¥.,	335,000			Tab Failure
C1.31A-5	1667	-	260 F		!z:	i.	ž	74,060			Tab Failure
C1231A-F	USC 0 - 6	:	3€€ ₹		RIJ	115		1			Tab Fallure Under
C1-31A-			26€ · F	1000	R	96		•			Immediate Tab Failure
C1.316-	-50°C - 5		7-007	1000	R.T.	86.5		,	y		Immediate tab Failure
C1.31A-6	130 C + 0 C	٠.	260°F	7. 1000 Cyc.		57.5	e 9		2.202×10°	47.0	Tab Failure Tempediate Tab Estima
C1C3TA-10	200 - A		00.		č			•			
C1030A-8	5 + 0.051	٤	350°F	°F / 500 Cyc.	RIC	114	Ö,				Immediate Tab Failure
C1230A-9	670 0 - 5	•	350 - F	-F '500 Cyc.	E.	103		1,000			The statement with market
C1030A-10	6 - 0.048	-	350°F /	F / 500 Cyc.	RI	41.5		' '			THE CLASS FAILURE
C12318-1	9 - 0.052	:	350	F / 500 Cyc.	E	200		44,000	901 - 46	1 31	Tel Pallure
010318-0	9 - 0.05ê	=	350°F		KT :	68.5		•	01 × C.2	1.07	140 FALIUTE
C11313-3	•	ž	350 ℃		RIC	110	4.5	2,000	9		: :
C12318-4	6-0.7 - 6	-	350 °F /	1000	. \ \	97.5	07	ı	2,092 x 10°	52.4	Tab Fallure
C12313-5	1	=	350°F	1000	RTD	105		- 0			insectate lab rallure
C12318-6	670.0 - 6	Ē	350°F	000	CIX	707	71 ,	22,000			Ten serious
C12316-7	570.0 - 6	Ē	350°F	1000	XI.	200	 3	1,000			ied ratiure

TABLE XIX FATTULE FROPERTIES SUMMARY +
HPERTIES 3702M/COURTALLDS HMS
GRAPHITE COMPOSITES

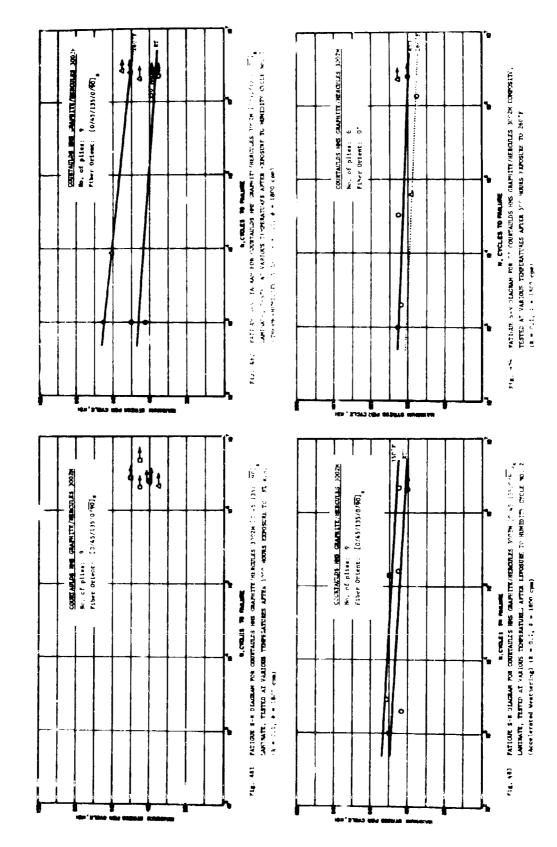
	Thickness (Piles) (In.) 9 - 0.053 9 - 0.054 9 - 0.054 9 - 0.054 9 - 0.054 9 - 0.054 9 - 0.054 9 - 0.053 9 - 0.053	Orientation [0/45/135/0/90]	FRIOR CONDITIONING Type Duration 260°F / 500 Hrs.	Ceat					
0	9 - 0.053 9 - 0.054 9 - 0.054 9 - 0.054 9 - 0.051 9 - 0.053 9 - 0.053 9 - 0.053 9 - 0.053	1 (Durat 160°F / 500	CHE.	Stress Level	Fat lure	without Pailure	Straneth	
C1240A-1 C1240A-2 C1240A-3 C1240A-4 C1240A-6 C1240A-6 C1240A-8 C1240A-10 C1240A-10 C1240B-1 C1240B-1 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-7			200	°F)	(10ult) (ks.	(cycles)	(cycles)	(E e1)	Comment
C1240A-2 C1240A-3 C1240A-4 C1240A-5 C1240A-6 C1240A-9 C1240A-9 C1240A-9 C1240A-9 C1240B-1 C1240B-1 C1240B-6 C1240B-7	$(1, 1, 1, 1, 1, 1, 1, 1, 1, 1, \dots, r)$			260°F	77 50				ignoffate Tab Pailure
C12404-5 C12404-6 C12404-6 C12404-6 C12404-8 C12404-8 C12404-10 C12404-10 C12408-1 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-7 C12408-8	(1 , 1	117 17 5 1	250°F / 500 Hrs.	260°F	61.5 40	•	2.173x106	5	
C1240A-5 C1240A-6 C1240A-6 C1240A-8 C1240A-8 C1240A-10 C1240A-10 C1240B-1 C1240B-1 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-7	$(\mathbf{I}_{-},$.	200	260 F				;	Tab Pailure
C12404-5 C12404-8 C12404-8 C12404-9 C12404-9 C12404-10 C12408-1 C12408-4 C12408-6 C12408-6 C12408-6 C12408-6 C12408-7 C12408-7 C12408-9 C12408-9 C12408-9 C12408-9 C12408-9 C12408-10 C12408-10 C12408-10 C12408-10 C12408-10	$(-1,-1,-1,-1,-1,-\epsilon)$	r 1 (5 1	/ 500	260°F	30 52	2.000			Tab Failure
C12404-6 C12404-9 C12404-9 C12404-10 C12408-1 C12408-1 C12408-4 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-6 C12408-8	9 - 0.054 9 - 0.054 9 - 0.053 9 - 0.054 9 - 0.053	1 7 5 1	/ 500	260°F	Specimen	ă	Pabrication		
C1240A-5 C1240A-6 C1240A-10 C1240A-10 C1240B-1 C1240B-4 C1240B-5 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-6 C1240B-1 C1241A-1 C1241A-1	9 - 0.054 9 - 0.053 9 - 0.054 9 - 0.053	: : :	260'F / 500 Cyc.	260°F		•	•		lemediate Jab Failure
C1240x-8 C1240x-10 C1240x-10 C1240x-10 C1240x-10 C1240x-1 C1240x-1 C1240x-1 C1240x-1 C1240x-1 C1240x-1 C1240x-1 C1241x-1 C1241x-1	9 - 0.053 9 - 0.054 9 - 0.053 9 - 0.053	: :		260°F	75 45	•	7.462 x 10	46.1	
C12404-9 C12404-10 C12404-10 C12408-1 C12408-4 C12408-6 C12408-6 C12408-6 C12408-8 C12408-9 C12408-9 C12408-9 C12408-9 C12408-9 C12408-9 C12408-10 C12414-1	9 - 0.054 9 - 0.053 9 - 0.053	=	260°F / 500 Cyc.	260°F		2,000	1		Tab Paillers
C12404-10 C12408-1 C12408-1 C12408-5 C12408-5 C12408-6 C12408-6 C12408-9 C12408-9 C12408-9 C12408-10 C12414-1	9 - 0.053			260°F		115,000			Teb Pailure
C12408-1 C12408-2 C12408-5 C12408-6 C12408-6 C12408-9 C12408-9 C12408-9 C12408-9 C12408-9 C12408-1 C12408-1 C12414-1	9 - 0.053	r	260 F / 500 Cyc.	260°F	81.5 49	1,000			ich Failure
C12408-2 C12408-3 C12408-4 C12408-6 C12408-7 C12408-9 C12408-9 C12408-9 C12408-10 C12414-1		Ξ	260°F / 1000 Cyc.	260°₽	38	1.000	•		Tab Patlure
C12408-1 C12408-5 C12408-5 C12408-8 C12408-8 C12408-1 C12414-1 C12414-1 C12414-1 C12414-1	650 0 - 0	r	/ 1000	260 °F	76 45		2.293 x 10 ^b	52.8	Tab Pailure
C12408-4 C12408-5 C12408-6 C12408-8 C12408-9 C12408-10 C12414-1 C12414-1	9.00 - 6	z		260°F		•	2.3% x 106	53.7	
C12408-6 C12408-6 C12408-6 C12408-7 C12408-9 C12414-1 C12414-1 C12414-3 C12414-3	750 0 - 6	r	900	266. F	٠,	1,000			Tob Paillure .
C12408-6 C12408-7 C12408-8 C12408-9 C12408-10 C12414-1 C12414-2 C12414-3	9 - 0.053	ı	1000	260°F		. 1			leardists Inb fullers
C12408-7 C12408-8 C12408-9 C12408-10 C12414-1 C12414-1 C12414-3 C12414-4	4 0 0 - 6	E		350°F	94.5 50	•			4
C1269-8 C1268-9 C1268-9 C12614-1 C12614-2 C12614-3 C12614-3	0.053	٤	200	350°F	85 45		4		Tab Pailing under Statie.
C12408-10 C12414-1 C12414-1 C12414-2 C12414-4	750 0 - 6	£	•	350°F	75.5 60	•	2.234 x 10	%	Load coming to Temp.
C12406-10 C12414-2 C12414-2 C12414-3 C12414-4	*0.0	=	<u>8</u>	350°F	81 43	•	2. X5 x 10°	4.7.4	Teb Pailure
C12614-1 C12614-2 C12614-3 C12614-3	- 6		~	320.6	Specialen	n Broke During	Pahrication		
C12414-2 C12414-3 C12614-4	. 0.03		350'F / 50' Cre.	350*	106.5 65				Pailed und Trette 1d.
C1241A-3	950.0	t	8 /	350*F		2. 2.			
CL241A-4	90.0		350 *# / 500 Cyc.	350°F		•			Pailed under Makic 14.
	960		8 ×	350°F	80.5	•			Descripte Teb Failure
C12414-5	9 - 0.053	£	350"F / 500 Cyc.	350°F	75 45	•			Immediate Teb Failure
77786	460.0	•	350°F / 1000 Cyc.	350°F	103	•			
	100	ε	1000	3.00		•			٠_
	200	ı	100	350°F	•	•	7	:	Demotiate Tab Pailure
	0.056	z	350°F / 1000 Cyc.	350'F	6 0 35		2.459 x 10°	47.2	Tet Paflure
01.WID	9 - 0.034	[0/45/135/0/30]	100	350°F	•	•			Baseflete Teb Failure

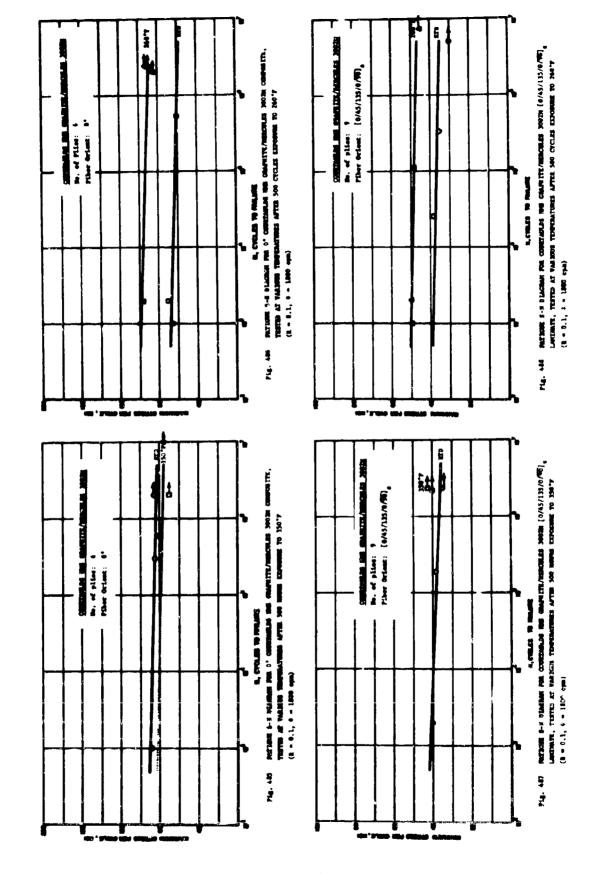


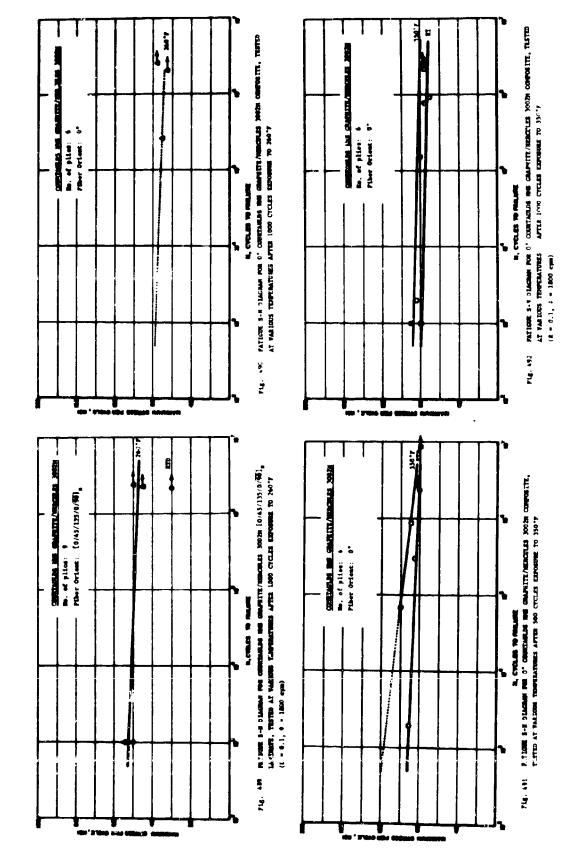
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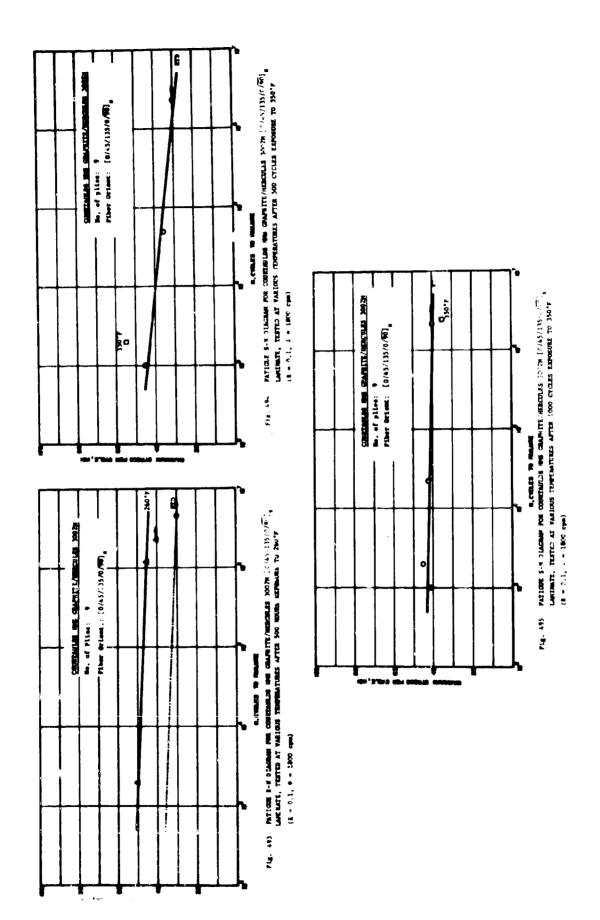


TABLE XX CREEP AND STRESS FUTTURE PROPERTIES SUPMAR. - HERCHES BOYCH COURT'T DS HES - GRAPHIE CONTNETS.

P. 11.	() Comment	Fat led at tal			fedland at tak	failed at tab	4 6 70 70 70			Tab Area Pailure	Teb Ares Failure								Strain Gage	Strain	Strain gage Failed			Strain gage failed	Failed during loading	•			
Tine Applied without Failure	(Hours)		•	•	•	1		•				•		•	,	•	•	•	1000	1000	•	•	•	•	٠		•		•
fisc to Failure	(Hours)	800.	.016	6	800	.016	800	8 ~	. 6.	.033	.00R	60	1.3	5.5	80	32	7.	926	,	•	165	.167	900	.016		8	8	i	Š
Level	(kst)	116	116	114	114	112	113	5	8	5.76	95.2	108	011	113	8	š	101	98.9	9.9	;; ;;	92.0	1.65	2.8	2.13	7.76	2.8	2.2.03		77.7
Stress Level	(I dult) (ksi)	85	86	*	8	ま	j	76	. 3	50	6	*	\$	96	65	2	3	2	1	23	2	£	S	\$	8	S	.	:	ì
12 T	(.t)	260°F	26C°F	260°F	3.90°	260°F	340.	260°F	260°F	260°F	260°F	350°F	350°F	350°F	350 F	350°F	350°F	350°F	1.056	350 %	4. OKE	260.7	260°F	10 · E	260°F	1.092	260°F	2.00	
PRIOR COMPITIONING	Duration																												
FEIOR	Type	None	None	None	None	Mone	NO.	None	None	Mone	None	None	Mone	None	Mone	Mone	Mone	Mone	Morse	ec :	ago de	Mone	Mone	Mone	Mone	None	Mone	į	
	Orientation		رز	ت	ڼ	٠,	6	, 0	ئن	č:	: o	.0	: 0	ຸ່ ບ	ţ,	Ö	,0	ပီ	°o	• •	5	• 06	\$	• \$	• &	8	E	\$	2
	(B ,	0,045	0.045	0.045	0.9	0.0 4 3	10.0	0.045	0.045		0.04	0.04	0.04	7000	0.04-		0.04	90.0	3	9 9	3	0.046	0.045	, 8 ,	5	0.00 0.00		5	
Thicmess	(Plice, (In.)			4,					•					۵						· ·						œ (œ	
u u u	70.0	C12378-11	C12778-12	C1273-13	C12:78-14	C123 18-15	C12378-16	C12:78-17	C12:78-18		C11 78-20	C12)8A-1	C12 384-2	C12)84-3	C12 384-4	C12:8A-5	C12)8A-6	C12 84-7	C12 :64-8	C12 184-9	07-W0: 777	C12.7-13		C12 7-15	012.7-16	C12_7-17	C12.7-18	C12: 3-1	

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l	1					96	9 54					ire.	•															116 10	ore nre.		
Comment			Strain game failed	Failed during loading		Failed at tak during loading	Failed at tab during loading	Failed during loading	Impediate fediuse	Fatled during loading	Strain gage failed	Oven overheated after 666 hrs.	Strain gage failed						Oven overheated	Failed during loading	Failed during loading							Committee Pathod Africa 014 Day	Stiell Dege Felled Altel	Fallad during loading	Failed during loading
Time Applied without Failure (Hours)				•	•	,	,	ı	,		ı	,			1000	000.	0001	1000	•		+	,	•	1000		1	,)			1 1
Time to Failure (Mours)		× E.	€ \$:•		, E.J.		•	1			<u>-</u> :		17	.37	•	1	,	•		•		070	800.		070.	.020				: .	
Stress Level (2 ^d ult) (ksi)				,		•	ž.	uç E	7		3.45	54.3	53.2	52.1	7.67	5.07	5.1	53.2	52.1	53.2	 Y:	58.7	57.7	55.3	7.	56.5	6,5		505	57.7	58.7
Stress Level (2 ^d ult) (ksi	; 	ĉ	· .	į	F	J,	£	30			.g	ź	5 6	ð	6	Ġ.	6	36	ð	8	86	86	*	92	6	7		8 2	3 2	; ¥	8 8
Test Temp. ('F)	1 1 1 1	 	130 F	1.000	3.51	33.1 F	÷ :	1.1358	350 F	350.5	350°F	260 'F	H (197	3,092	3600	J. 097	240°F	3.092	260°F	260 'F	260°F	350°F	350°F	350°F	350°F	350°F	350°F	350.5	350°F	350.5	350°F
CONDITIONING Duration																															
PRIOR G	3	o E	Acn	Non	None	None	No.10	None	None	None	None	None	Non-	None	None	None	None	None	None	None	None	None	None	None	None	None	9	, and	Kon	900	None
Orientation	Co	íá	<u>.</u>	ã	÷	3	()	و	ĩ	. 05	ă	00 0 11 5+ 3		1	=	:	ε	:	:	-	.0/45/135/0/90 s	[0/45/135/0/90]	.	=	=	=	:	:	:	:	. 0/45/135/0/90;
Th. 2kmess 11.3) (In.)		1	1	· []	6-2-5	6.0.0	¥- j.'∂	1.0.3	٠ ١٠ ١	4.0.0	' ;	1.21	1.75	ر. د د د	6.000	٠.٥٥٠	0.053	0.054	0.053	0.033	0,053	0.055	7,055	0.055	0.055	0.055	5500	0.05	0.055	0.05	0.055
Th. (P14	g:	5 4	L	r	x ∪	uc	x .	oz.	œ	,	3	j	ţ	,	j	7	g.	ъ.	3	5	J.	•	5	ø.	σ	σ	ci	· a	J	œ	• •
Specimen Number	5-8:515	7 0 . 6 . 7	0-0"717		C12.8-5	C1218+6	01-9777	C12.8-73	(11.8-12	3.12.3-13	C10.8-1-	11:10:10	.1235	7 - a - 7 - 1 - 1	01:3-b	S - 84 T C T T	C12348-6	C11348-7	19 - HT 11 7	C12748-9	C12348-10	C1235A-1	C1235A-2	C1235A-3	C1235A-1	C1235A-5	4-45-517	C1215A-7	C1235A-8	C1235A-9	C1235A-10

TABLE XX CRIEP AND STRESS REPTURE PROPERTIES SUFFARY - HEMCULES JOOPH CHREGATION HYS - GRAPHITE CHMPASITES

	1	Londing	in in its second	1		100	losding	in the second																				
Comment	Tak Gadinaa	failure - broke during	Tob failure - broke during Tab failure - broke during		fellure a broke desired	failure - broke during	failure - broke during	failure - broke during	Broken during Canditioning	der i e	duribe	durie	Broken during Conditioning	Broken durine Conditionine	in the	during	During	Broken During Conditioning	Broken in Conditioning	=	5	5	5	Brokes to Confittoning	4	5	5	•
Time Applied with our Pailare (Rours)	0001	•	. ,	•				•		•	•		•	•	•	•	•	1	•	•	•	•	ŧ	•	•	•	•	
Time to Patlure (Nours)	Ş	}.		•		•		•	•	•	•			•		•		•	•			•	•	·				
Stress Level (I [©] ult) (ksi)	98.1	2	10.7	9 50		97.6	7.06	88.3		•	1	•	•	•				•		•	•		•	•		•		1
Stree (L ^o u)	23	35	¥ \$	88	2	2	\$	~	•	•	•	•	•	•	•	•	•	,	•	•	٠	•	•	•	•	•	•	,
Test Samp. (*)	260°F 260°F	2.09E	700.E	1 05L	350'r	350°F	350°F	350°F	260 € F	260°F	200°F	260°F	260°F	350 'F	350	350.7	350°F	350.7	7.09Z	760.7	7.04Z	1.092	760°F	350°F	320.6	350°F	350°F	160.0
PAIOR COMPITIONING Type Derecton	500 Hrs. 500 Hrs.		500 Hrs.	500 Hrs.				500 Nrs.	1000 Hrs.		1000 Hrs.	1000 Hrs.	1000 Hrs.	1000 Hrs.		1000 Hrs.	1000 Hrs.	1000 Hrs.	dity Cycle	dity Cycle	dity Cycle	dity Cycle	dity Cycle	dity Cycle	dity Cycle	dity cycle	dity Cycle	Athen Carollo
Paton o	12 12 12 12 12 12 12 12 12 12 12 12 12 1						\ \tag{2}					- M 15		/ HE 196			/ HE 196		Thermo-Shari	The Carried	Thermo-line	Chermo-Hund	Thermo-Humidity	Thermo-Bunidity (Thermo-Hund	Thermo-Head	Thermo-Hunt	The same of
Orientation	ခ် ဝ	<u>.</u>	: c	·	c	Ė		, C	•	ć	•0	0	, C	ۍ	,0	c	0	•	• 0			•					•	
Thickness (Flies) (Im.)	0,047	0.045		6 (.047			6.047	5 40.0	6 (.043			6 (0.043		5.00.0			0.043	5 0. 043	0.045	710.0			770.0	0.045	9,0,0	0.045	0.04	200
Specimen T Mumber (P)		01246-3		C1248-6				C1248-10 (C12098-19 6	C1210A-5 6				C12104-3 6	C1248-1: 6	C1248-1.: 6	C1248-1; 6	C1248-14 6	C1248-15 6	C1248-16 6	CL246-17 6	C1246-16 6	C1248-19 6	C1248-70 6

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TABLE XN CREEP AND STRESS RUPTURE PROPERTIES SUPPARY - HERCLES 3002h CO'RTAULDS HWS - GRAPHITE COMPOSITES

Specimen Marie	Thác (P1108	Thickness (Flies) (In.)	Orientetion	PRIOR	PRIOR COMPITIONING Type Duration	Test Tap. (*F)	Stress Level (T ⁰ ult) (ksi)	Leve l (kst)	Time to Failure (Hours)	Time Applied without Failure (Houre)	Coment
C12C98-20	.م	0.0 ⊒.0.0	, 0		licheng.	260°F	80	117		•	Oven overheated
01710 01710 01710		રી તે કે લે કે લ	= <	VCC.	Wining.	7.00°F	8 5	<u> </u>	80 (0001	
01210	.		ی د		Tithing.	7 097 F 096	g S	100		201	
C1210A-4	, æ	۰.0.			Wthrng.	260°F	3	111.5	29,6		
C12:0A=10	£	ر ا	Ċ	ACC.	Withrag.	350 F		•	•	,	Broken during Conditioning
C12104-11	.	0.0	c		Vchrag.	32035	٠	•	•	1	
C1210A-12	¥	0.0	. 0	Acc.	Vthrng.	350 F	•			•	
C1230A-13	vc	0.053	¢		With Trans.	350°F			•	•	during
C12.0A-14	•	6.7.9	0	Acc. 1	Wthrng.	350°F			•	,	
01249-1	J.	0.067	07/25/135 0/90	48.7 RH	/ 500 Hrs.	260 F	•		ı	•	Broken during Conditioning
C1249-2	٠.	0.066	v	12.96	/ 500 Hrs.	260°F	•		•	•	during
C1249-3	ı ur-	0.066				260°F	•	•		•	during
C1249-4	ŋ,	0.065	.		, 500 Hrs.	260°F	•		•	•	
C1249-5	Jr.	0.066	÷	98°. 24	500 Hrs.	260 °F	•		•	•	Broken during Conditioning
61349-6	و	ر ا ا	1	48°	500 Hrs.	350°F	•			•	Broken during Conditioning
C1249 - 7	U	0.756	÷			350 °F		,		•	
C1249-8	U	0.066	=		/ 500 Hrs.	350°F	•		,		Broken during Conditioning
C12-5-9	ij.	0.067	=	987. RH	/ 500 Hrs.	350°F	•	•		•	Broken during Conditioning
C1249-10	J	990.0	.0/45/135/0/90.	987. RH	/ 500 Hrs.	Дц. С С	ı	,	ŧ	•	Broken during Conditioning
012378-1	J	0.062	06/0/381/57/0	967 RH	, 1000 Hrs.	260'F	96	53.1	•	1000	
C12378+2	J	0.063	10	987, EH		260 F	ጀ	8.67	•	1000	
C12378-3	٠	090	:		/ 1000 Hrs.	260°F	86	52	•	•	Teb feilure - broke during loading
4 K C C C	J	0.063	:			260 F	46	53.7		1000	
C1237B-5	0	0.063	2	98. RH	/ 1000 Hrs.	260 °F	66	% %	,	1000	
C1238A-1	J	0.065	=		/ 1000 Hrs.	350°F	8	39.0	•	1000	
C1238A-2	J	0.065	:	987 EH	/ 1000 Hrs.	350°F	93	40.3	,	1000	
C1238A-3	U	790.0	=		1000 Hrs.	350°F	9 6	42.5	•	1000	
C1238A-4	J	990.0	Ξ		/ 1000 Hrs.	350°F	*	41.6	•	1000	
C1238A-5	ن	790.0	0/45/135/0/90	.86 .86	1000 Hrs.	350°F	95	41.2	•	1000	

IABLE XX CREEP AND STRESS RUPTURE PROPERTIES SUPPARY - RERCULES 3002M COURTAULDS HMS - GRAPHITE COMPOSITES

Coment		during conditioning			during conditioning		_	-			during conditioning		failure - broke during loading										during loading and split	furing loading and split	leading at	T split			Broke during locating and split	•	broke during		failure - broke during 10001Mg
	Broke di	Broke d	Broke 4	Broke di	Broke di		_		•	•	Broke 4		12										Broke d	Broke	Broke d				Broke			15	
Applied without Failure (Houre)	•		•	•			•	•	•	•			•	• •		8		•	0001	9001	1000	1000	•	•	•	•	•	,	•	900	•	•	•
Time to Failure (Hours)	•	•	•		•		•		•		•		• !	259.7	•	•	8	+30.2		•	•	ı	•	•	•	•	, ,	•	•	•	•	~	•
Level (kat)	1		•				•	•		•			55.3	51.9	52.9	7 .5	55.9	6.8	49.0	46.5	45.1	2 0.1	126.4	123.8	121.2	114 6	0.011	1.011	101.1	92.3	95.2	o. 1	95.2
Stress Level (1,0ult) (ksi)	,	•	,	•	•		•	•	•	•	•		*	\$	95	ኔ	6	*	2	93	8	9	ā	8 3	2 3	t S	2 6	3	S	2	8	2	\$
15 15 E	260°F	260.5	260.1	240.7	7403		150°F	350°F	150°F	350 %	350 F		260°F	260 F	260°F	260°F	260°F	350°F	3.00	350	350°F	350°F	40000	207	3.090	1007	1.007	26C'F	1.05	95	200	1.051	350*F
PRIOR COMPITIONING Type Duration	Thermo-Humidiev Cycle		The state of the s				Thermo-Hamidity Cycle						Acc. Wthrng.			_	_		_		Acc urbine							260 F/500 Hrs.	140 003 / 40 03C				
Orientation	00 0/361/3/10		: :	: :	: 3		r		Ξ	:	00 3. Sell 2000	B 06 0 001.04/0	0,6,5,1135 C 90		-	•	=	-	=	: #	:	[0/45/135/()90]		• 0	•0	• 0	•	•0	•	. 0		5 6	. •
Thick ses (Plies) (In.)		95	8 :	8	28.	8.5	,	980.0	95.5	8	8.	990.	. 963	3	3	28	88	,	198		9.0	6.0		0.045	3.648	3.044	0.047	0.045		0.00°	970.0	0.046	9 90. 0
Special Specia Specia Specia Specia Specia Specia Specia Specia Specia		C1249-11	C1249-12	C1249-13	C1249-14	C1249-15		C1249-16	C1549-17	C1249-18	C124.9-19	C1249-20	9 86 66 70	0-8/5713	C123/8-/	C123/8-0	C12378-9		C12384-6	C123 EA-7	C1238A-8	C12384-9	01-10-5110	C1211B-16	C12113-17	C12118-18	014110	C12138-20		C1212A-1	C1212A-2	C1212A-3	C1212A-4 C1212A-5

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TABLE MX CREEP ANY STRESS BUTIFURE PROPERTIES S. "ARE A PERTUES 300, A CONTINUE S NO. A CONTINUE TILLS S NATHERY CONTINUE TILLS

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	Thic	Thichness (*1ies) (in.)	Orienten. Con	PLOS CONDITIONING Type Durecton	i i î	Stress Level (X ^d ult) (kei	Stress Level (1 ⁰ ult) (ksi)	Time to Failure (Nours)	Applied without Fatlure (Hours)	Comment
C12124-6 C12124-7	υ w	0.06	66	260°F/1000 cyc. 260°F/1000 cyc.	260°F 260°F	853	26.4 26.4 26.6	135.4	, , ,	Job fallure - broke during lecting Broke - demile tab failure Broke during leading 1/4" from tab
C1212A-0 C1212A-9 C1212A-10	en en en	0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0	သမာ	260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc.	260'F	388	\$ \$ \$.5 \$.5	.012	1000	Broke - domble tab falluro
C1212A-11	۰۰	300	င်င်	260°F/1000 cyc. 260°F/1000 cyc.	350 F 350 F	£ \$	100 1		1000	broke during leading - double fail.
11-421213 612124-12 612124-14	s 😻 🗴	566	ລ້ວໍ່ຕໍ່ເ	260°F/1000 eve. 260°F/1000 eye. 260°F/1000 eye.	350°F 350°F	222	100 101. 100.1	7.91	8	Tob failure - broke during leading
C12126-13 C12126-6 C12126-7	• ••	33	, , ,		260 F 260 F 260 F	238	98.2			Tab failure - broke during leading broke during leading - dumble field. Broke during leading - demble field.
C12128-9 C12128-9	•••	333	, . .	350°F/1000 cvc.	260°F	3 8	101.7		,	tab failure - broke during leading
C12126-11 C12126-12 C12126-13 C12126-14		9.00 0 9.00 0 1.00 0 1.	0000	350°F/1000 eye. 350°F/1000 eye. 350°F/1000 eye. 350°F/1000 eye.	350'F 350'F 350'F	28485	109.6 116 103 103 109.6	26.	, , , , 991	Tab failure Tab failure - bruke during landing Tab area failure Tab failure - bruke during loading

TARLE XX CREEF AND STRESS REPTURE IN DEPRITIES STREAM - HERCHLES BROAD OF MARCHINE HERCHLES TONIONSTES

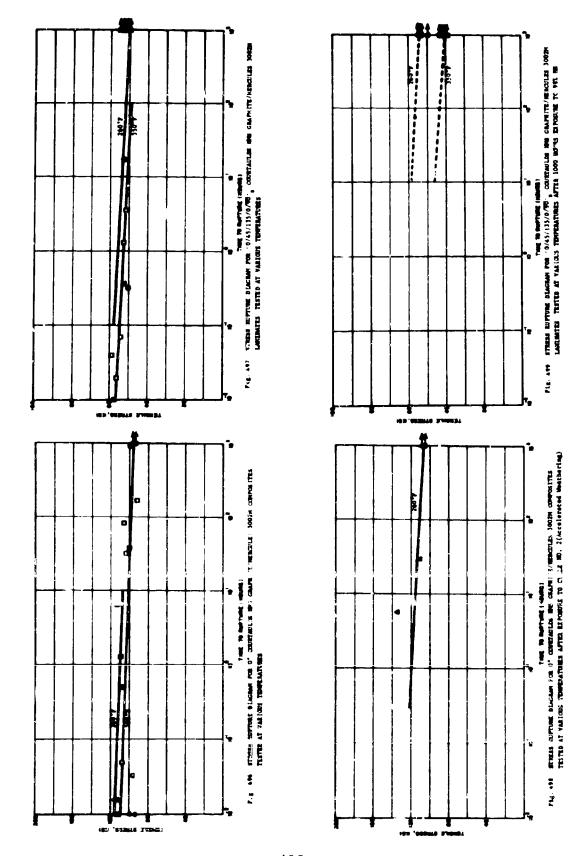
1112.4.16 6 0 0.06 6 0.0	Section 1	Thic (Pites	Thickness (Pites) (In.)	Orientation	PRICE CUMDITIONING Type Duration	Test Trap	Serves Control	Stress Love)	fil to radlure (Rours)	Time Applied without Failure (hours)	Cognent
6 0.048 0° 330 F'500 Hrs. 140° 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	12124-16	9	970 0	,0	350°F'300 HF			,		,	5
6 6 0.048	12124-17	φ	670.0	٥,	350 F 50° HE	.,		,			S
6 6 0.046 0° 330 F*500 rec, 66 ° F · · · · · · · · · · · · · · · · · ·	1212A-16	•	0.748	6.7	350 F 500 FF	•	٠			,	
6 0.046 0.0 1 300 F 500 F 70	1212A-19	•	690.0	.0	350 F '500 HE					•	
F C.046 O	31212A-20	æ	0.048	ċ	350 F 500 · r				1	•	Delaminated in Conditioning
6 0.046 0.0	112128-1	4	970.0	, O			-		5,4	•	
## 0.0046	12128-2	c	5	, ~			,	7.7		•	
6 0.046 0.0 9 0.057 C −5.135.0/90 2 2.0 F 5.0 Hrs. 350 F 40 81. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.8 1. 4.3 4.3 4.3 4.3 4.3 4.3 4.3 4.3 4.3 4.	1212B-3	ط ا	970 0	.53	5		. 4.		: .	1001	Strain east failed after 1 hour
6 0.046 0.7 C ±5:35 0/90	12121	Ų	90	. C	and Green Tables		Š		36 4	· : •	
9 0.057	312125-5	w	0.046	D	350°F 50°F fir		5	; ;	.014	•	
9 0.036 9 0.045/135/0/97,8 130/p/500 Hrs. 150/p 10 0.037 10 0.	~1241B-1	œ	0.057	t.		06	Ī	7.5,	(· · · · · · · · · · · · · · · · · · ·	•	
9 0.056	12418-2	יכי	0.056	•		260	3	-	8	•	Tab Fatlure
9 0.056	124.18-3	ď	0.056	:	260°F/500 Hr		3	1,	r. ci		Tab Failure
9 0.056 " 260°F'500 Hrs. 350°F 90° 42.5 "	12418-	o n	0.059	:	200°F 500 Hr		\$	1.61	ξ.	•	Tab Tailure
9 C.056	112418-5	σ	0.056	÷	240,2,200 Hz		\$	· · · · · · · · · · · · · · · · · · ·			Tab fallure - broke during loading
9 C.059 " 260°F'500 Hrs. 150°F ' 2.5.5 140.4" 9 C.066 "0.45/135/0.90.5 260°F'500 Hrs. 150°F 97 45.8	312418-6	σ·	0.056	:			ક	42.5	•	•	Broke during loading
9 C.060	112418-7	o r	0.059	:			3	43.5	1.04		Broke in middle
9 C.066 [0.45/135/0.97], 260°F/500 Hrs. 330°F 94 44.4 .33 1000 9 C.066 [0.45/135/0.97], 330°F/500 Hrs. 260°F	C12413-8	U i	C-8.5	:			88	9:14	289	•	Strain Gage Failed
9 C.056 [0.45.135/0.97], 260 Fr.500 Hrs. 150 F 97 45.8 - 1000 9 C.066 [0.45.135/0.97], 350 F/500 Hrs. 160 F	C12618-9	or.	0.060	:			76	7.17	.33		Tab failure
9 C.066 '0.45/135/0.50's 350'F/500 Hrs. 160'F	C12418-10	or .	C.058	. 0.45.135 '0 '97. _{\$}			47	45.B		1000	
9 C. Ueb	1747R-1	Or	990	05.0713870.00	350 F / 500 HE		1	,	1	1	
9 C.066 " 350°F/500 Hrs. 260°F	12428-0	יני	990	U.	350°F '500 Hr		1				
9 C.066 " 350°F/500 Hrs. 260°F	12428-3	. 0	95	ī	350°F/500 Hr		1		1	•	Eroke during conditioning
9 C.066 " 350°F/500 Hrs. 260°F	1242B-4		990,	:	350*F/500 HE			•		٠	Broke during conditioning
9 C.066 " 350°F/500 Hrs. 350°F 90 44.6 - 1000 9 C.066 " 350°F/500 Hrs. 350°F 93 49.2 - 1000 9 C.067 " 350°F/500 Hrs. 350°F 97 51.3 202.4 - 1000 9 C.063 " 350°F/500 Hrs. 350°F 95 50.2 - 1000 0 C.063 [0.45/135/0/90] ₅ 350°F/500 Hrs. 350°F 99 52.3 - 1000	312428-5	6	990.0	=			•	į	ì	•	Broke during conditioning
9 C.066 " 350°F/500 Hzs. 350°F 93 49.2 - 1000 9 C.067 " 350°F/500 Hzs. 350°F 97 51.3 202.4 - 1000 9 C.063 " 350°F/500 Hzs. 350°F 95 50.2 - 1000 0 C.065 [0:45/135/0/90] ₅ 350°F/500 Hzs. 350°F 99 52.3 - 1000	C12428-6	o,	0.068	Ξ	35C F/500 Hr		ò	9.74	1	1000	
9 C.067 " 350°F/500 Hrs. 350°F 97 51.3 202.4 - 1000 C.063 " 350°F/500 Hrs. 350°F 95 50.2 - 1000 C.065 [0:45/135/0/90] ₅ 350°F/500 Hrs. 350°F 99 52.3 - 1000	-12428-7	o	990	=	350°F/500 Hr		63	7.65	•	1000	
9 C.063 " 350°F/500 Hrs. 350°F 95 50.2 - 50.5	012428-8	0	790.	ī	350°F/500 Hr		46	51.3	202.4	•	Broke in middle
9 C.065 [0:45/135/0/96] _s 350°P/500 Hrs. 350°F 99 52.3 -	01242B-9	•	063	=	350 PF / 500 HE		95	50.2	•	0001	
	31242B-10	. on	C.065	[<u>05/0/32//55;0]</u>	350°F/500 Hr		66	52.3		1000	

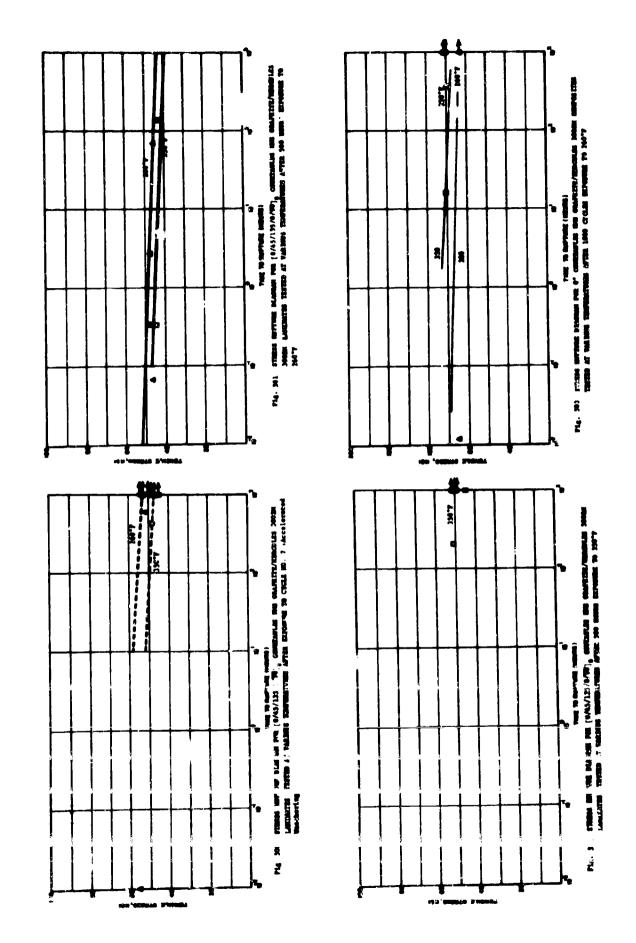
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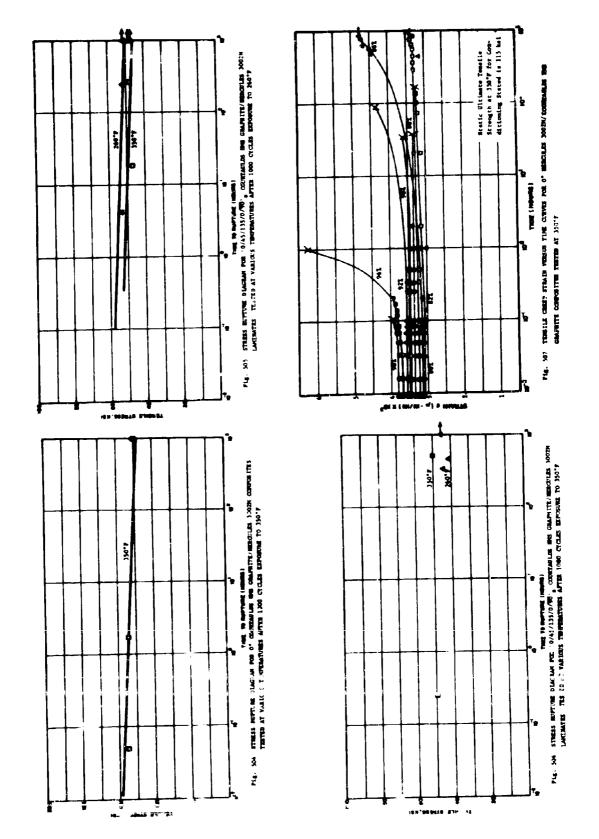
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TABLE XX CREEP AND STRESS RUPTURE PROPERTIES SUPPLY. - MERCULES 3002M COURTAINES

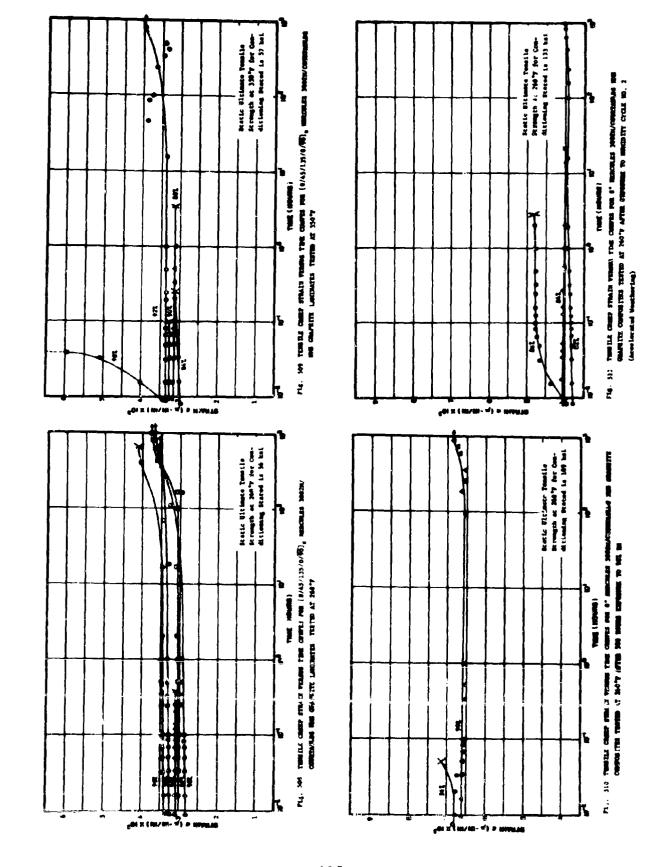
Comment	Broke in middle Tab feilure - broke during loading Int failure - broke during loading	Broke during loading	lab failure - broke during lossing Oven overheaded	Tab failure Tab failure - broke during losding
Applied Applied Atthor Faiure (Houre)	. Tab . Tab 1000	1000 1000 1000	1 ab	
Time to Failure (Hours)	5.2 - - 246	18.5 256	336	. 25
(kat)	53.5 54.0 52.9 52.2	5.74 4.74 4.74 4.74	44.3 41.0 45.3 45.3	50.1
Stress Level (T ^o ult) (kai)	90 90 90 90 3 4 9 1 8	90 90 90 90 90 90	58 888	88
i.c	260'F 260'F 260'F 260'F	330°F 350°F 350°F 350°F	260°F 260°F 260°F 260°F	350*F
PRIOR COMPITICATING Type Duration	260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc.	260°F-1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc. 260°F/1000 cyc.	350*F/1000 cyc. 350*F/1000 cyc. 350*F/1000 cyc. 150*F/1000 cyc.	350°F/1000 eyc.
PEIOE C	*******		HMMRM	6 00
Culentation	10.75/1337/0/90 s		8.048/135/0/90.	r :
Thickness (Plica) (In.)	0.066 0.067 0.067 0.065	\$ 3 4 4 9 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	0.000 000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.	0.065
Thich (Pites)	் கைக்கைக்	. நாருகுக	თტაა თ	ar a
Specimen	(12424-1 (12424-1 (12424-3 (12424-3 (12422-4	01242A-6 01242A-7 01242A-7 01242A-9 01242A-10	C12420-1 C1C-20-2 C-2420-3 C12420-4	C1242C-6



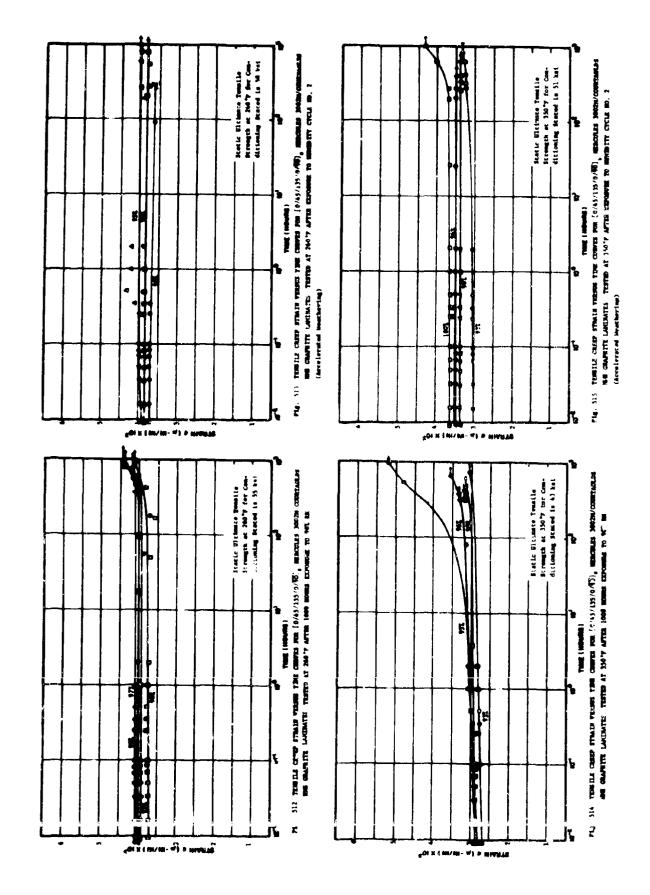




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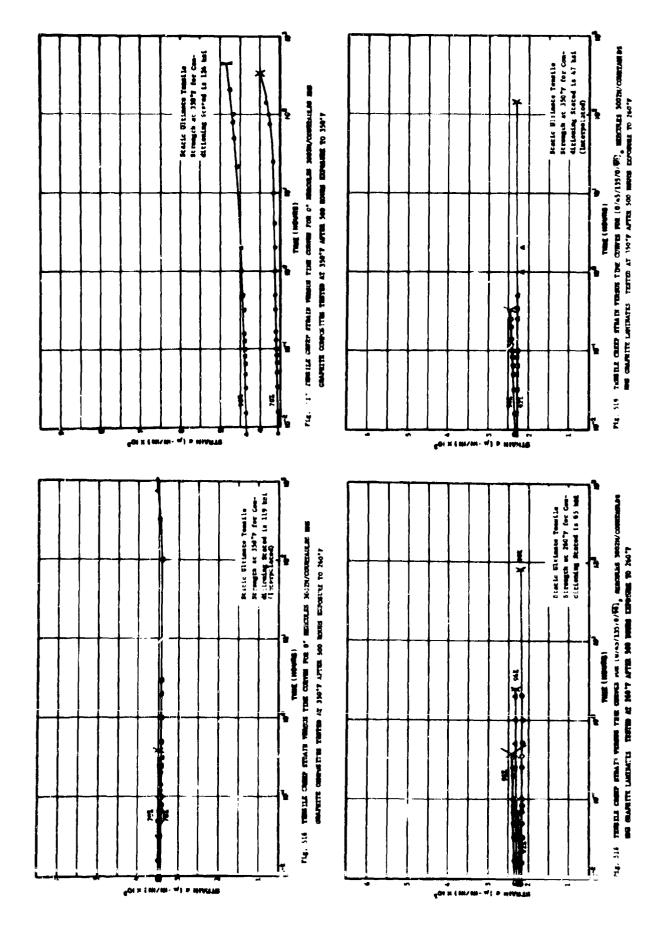
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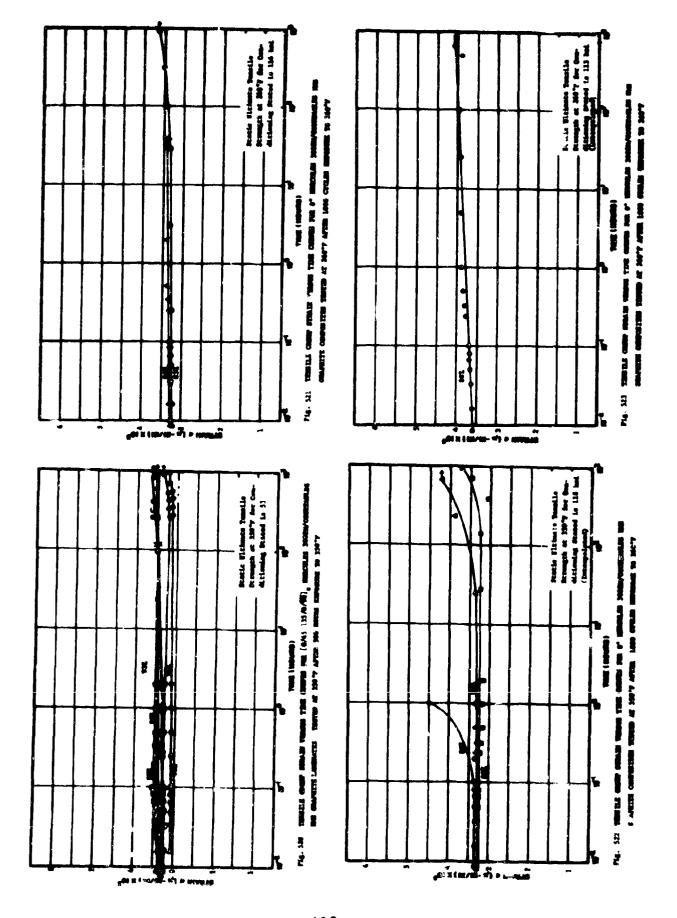


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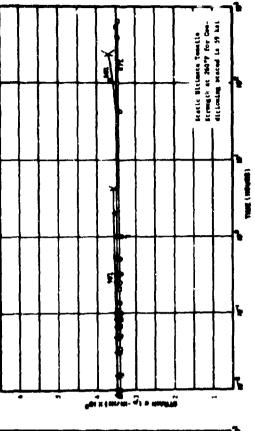
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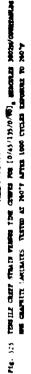
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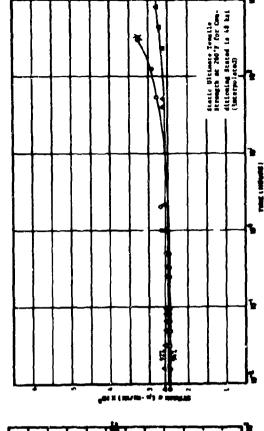
Static Ultimate Temaile Strength at 350°F for Conditioning Stated in 379 hal

HENCELES NOOTH/CONTENENS

TH CONCRETES THEFTO AT 1907 AFTER 1000 CFGLES EXPOSURE TO 350*T

TENE LE CREAT STEMEN PORCE TENE COUNTS POP

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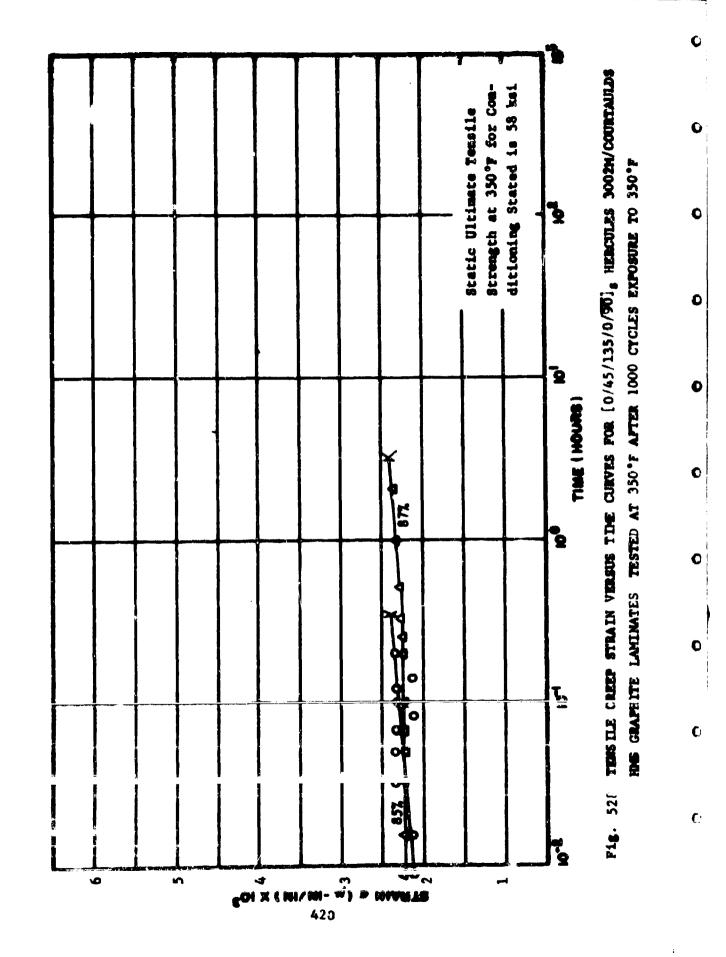
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FIG. 536 TERRILL CHREY STRAIN VIRGING THAIR CHRISTA FOR 10/45/135/0/FOL; INDICAGE 500EN/CHREADANA

State Utimaca Temaile Strength at 190°F for Conditioning Stated in 50 km (Interpolation)

Pig. 327 TERRIA CREEP STRAIN VERSON THE CONVEN STRE [0.657155/07/80], MELCHAR 30026/CONSTANTOS MIN GRANNETE LANDARIZE TANDED AT 250°P APTER 1000 CRCAIN EXPONENT TO 330°P.



APPENDIX IV

DATA SUMMARY FOR 6061 ALUMINUM/BORON COMPOSITES

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APPENDIX IV

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FABLE NX: STATIC PROPERTIES SUBGARY - 6061 AT N. N. BORON COMPOSITES

е _{иле} (н-ба./ба.)	7,150	6,920	078'9	6, 560	3,780	4,000	4,650	2,370	11,550	11,900	10,930	9,610	11,450	12, 510	14,640	4,630	•	•	•	•
"alk (Bet)	205	¥	180	\$1	23.5	25.9	4 .2	20.0	18	35	324	72	42.4	43.7	67.6	31.1	ž	ž	ã	82
, (1a/1a)	0.27	0.79	0.21	0.21	0.12	0.15	0.14	•	0.26	9 .0	0.14	0.28	9.0	8.0	8.0	8.0	•	•	•	•
E (paf x 10 ⁶)	29.0	30.0	33.8	29.7	19.9	18.6	13.2	14.9	33.7	31.9	30.€	32.5	17.9	18.2	15.0	12.6	•	•	•	•
Test Temp. (*P)	ב	160	9	89	£	091	99	9	ᄕ	160	99	909	t	91	004	9	Ħ	91	004	909
Frior Conditioning	None																			
Type Loss:	Tension								Compares \$ 1 on								Florurei			
Orientation	,0				, 06				0				26				 O			

TABLE XXI STATIC PROPERTIES SUMMARY - 6061 ALTHIN M BORON COMPOSITES

Orientation	Type Load	Prior Conditioning	Test Temp. (*F)	E (ps1 x 10 ⁶)	, (11/11)	dult (ksi)	*ult (p-im./in.)
.06	Flexural	None	ĸŢ	•		52.5	
			160	•	•	51.3	•
			907	•	•	1.47	•
			099	•	•	36.9	•
0.	Intl'mar Sh'r		RT	•	,	20.1	,
			160	•	•	19.9	•
			700	•	•	17.2	•
			009	,	•	14.2	ı
.06			Ł	•	•	7.5	•
			160	•	•	7.2	•
			007	ŧ	•	5.1	,
			909	•	•	3.2	•

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VILVINAL STATIC PROPERTIES SUMMRY - 6061 ALUMINUM BORON COMPOSITES

Orientation	Type Load	efor Consistentag	Test Temp. (*F)	E (psf x 10 ⁶)	, (11/11)	^g ult (ksi)	*ult (u-in./in.)
ن،	uo r sua j	500°F/109 hrs	73	27.1	0.27	177	6,800
			160	28.2	0.19	179	7,740
			007	28.2	0.21	159	6,190
			909	31.7	0.18	173	7,360
		500 F/500 hrs	RŢ	28.5	0.26	183	6, 370
			160	27.6	0.19	165	6, 780
			007	31.2	0.18	153	5, 140
			009	31.7	0.24	163	5, 990
		603°F/1000 hrs	er t-	27.8	0.28	162	6,070
			091	24.6	3.27	169	6, 780
			004	29.7	0.20	170	6,50
			9	28.7	0.20	991	5,726
.06		600°7/100 hrs	¥	15.8	0.03	16.7	5,020
			160	10.6	0.19	16.2	8,170
			700	14.4	0.14	15.8	4, 510
			9	12.8	9.0	10.9	6, 890
		600°F/500 hrs	TH.	1.81	0.0	17.3	7,750
			160	17.7	9.0	16.6	7,130
			004	12.9	9.0	14.9	5,200
			009	13.7	9.0	12.5	9,810

TABLE XXI STATIC PROPERTIES SUMMARY - 6061 ALIMINIM BORCE COMPOSITES

Orientation	Type Load	Prior Conditioning	Test Temn. (*F)	E (psi x 10 ⁶)	(in/in)	ult (ksi)	ult (u-fn,/fn,)
, 26	Tension	600°F/1000 hrs	RT	17.6	0.00	17.6	026'7
			160	19.5	0.00	17.1	5,100
			004	12.6	0.00	15.2	5,640
			909	6.01	0.0	12.6	3, 530
.	Compression	600°F/100 hrs	E	29.7	0.29	311	10,160
			160	29.5	0.27	23	9,910
			007	29.0	0.25	288	9,940
			009	27.5	0.25	246	9,230
		600*F/50G hrs	RT	30.1	0.29	275	8,890
			160	31.2	0.23	285	9,240
			00*	30.1	0.38	279	10, 100
			609	30.7	0.26	72	9, 380
، کو		600°F/100 hrs	RI	11.9	0°.0	29.7	13, 700
			160	11.8	8.0	30.8	14, 900
			004	9.7	8.0	29.8	13, 600
			Ç	1,1	3	27.8	20, 500

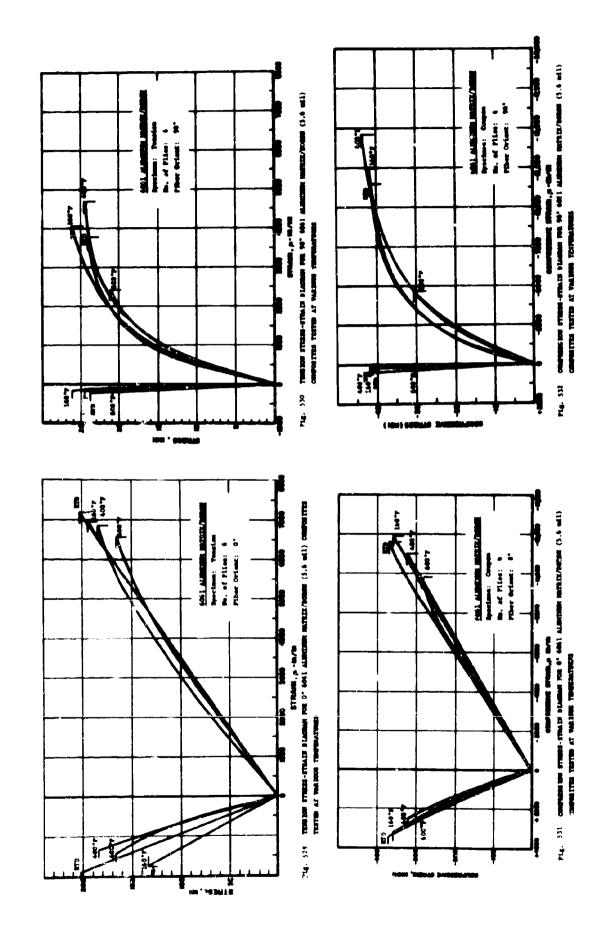
TABLE XXI STATIC PROPERTIES SUMMARY - 6061 ALUMINUM RORON COMPOSITES

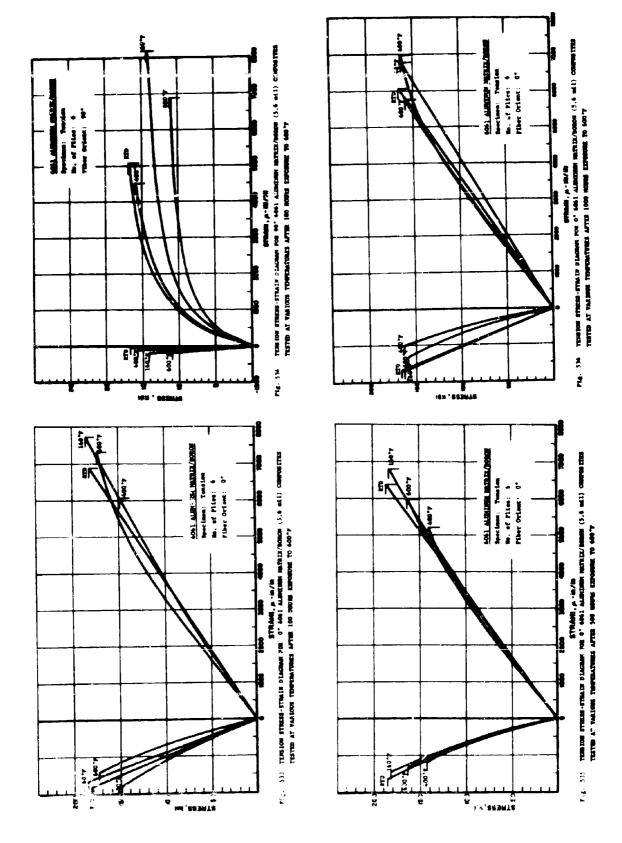
Orientatica	Type Load	Prior Conditioning	Test Temp.	E (pet x 10 ⁶)	, (in/in)	ğř	"ult (s-in./in.)
•06	Coap re 1 a f.on	600'F/500 hrs	ħ	11.8	0.0	30.9	> 30,000
			160	0'6	9.0	30.5	> 30,000
			904	9.1	9.0	27.8	> 30,000
			009	9.0	8.6	28.1	> 36,000
• 0	Tens lon	600'F/100 cycles	Ľ	27.6	0.27	205	7,488
			160	28.2	0.29	196	7,220
			400	23.1	0.22	173	6, 846
			009	28.9	0.19	107	7,300
		600°F/500 cycles	Ħ	26.9	0.28	13	7,020
			160	25.5	6. X	2	6, 76
			00+	27.9	9,30	*1	€, 01€
			009	28.3	0.19	31	6, 830
		600*F/1000 cycles	E	4 .1	0.2	191	6,78
			160	3. ¥	0.25	181	
			904	28.0	•	147	
			9	26.6	0.Z	1	8,970
• 06		600*F/100 cycles		17.7	8.	10.9	5,060
			91	16.1	8.6	17.6	5, 720
			9	11.9	9.0	15.2	3,960

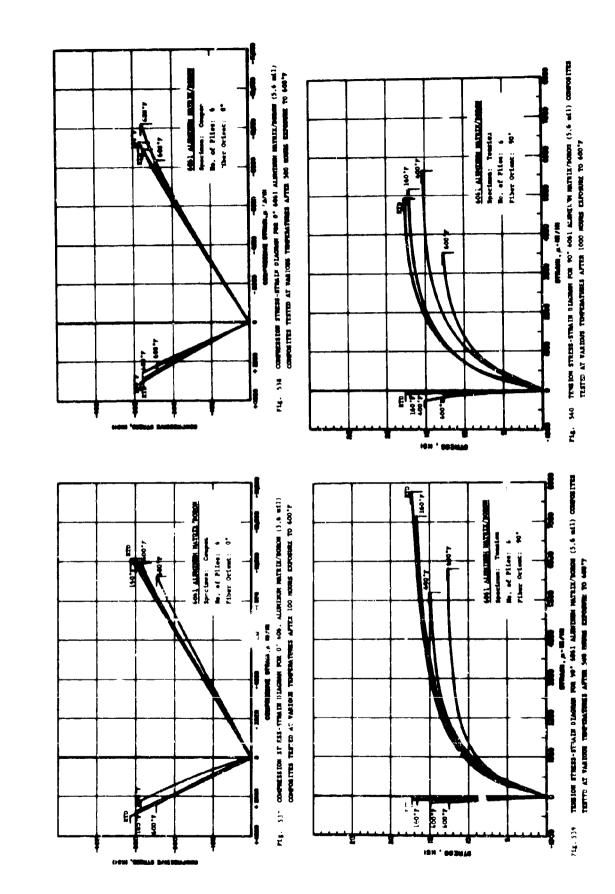
TABLE XXI STATIC PROPERTIES SUPPARY - 6061 ALIMINIM BORON COMPRAIR

eult (14-fn./fn.)	6, 550	5,640	6, 580	5,440	019.4	6,690	960'9	9,820	10, 330	9, 730	10,150	9,050	30,000	30,000	30,000	30,000
dult (kel)	15.9	15.5	13.2	10.1	14.9	15.2	12.8	9.1	230	50	292	242	29.6	31.4	30.1	27.7
, (11/11)	0.0	0.00	0.0	0.00	0.00	0.00	0.0	0.00	0.23	0.23	0.27	0.26	9.0	0.0	0.60	0.0
E (psi x 10 ⁶)	17.2	17.4	10.5	9.2	16.5	4.91	11.11	9.5	29.0	29.1	28.0	28.9	6.9	0.6	9.0	0.8
Test Temp. (*F)	RŢ	091	007	009	ㅂ	091	700	009	Ħ	091	004	909	Ľ	160	004	009
Prior Conditioning	600*F/500 cycles				600°F/1000 cycles				600"F/500 cycles				600°F/500 cycles			
Type Load	Tension								Compression							
Orientation	а С U								•0				. 05			

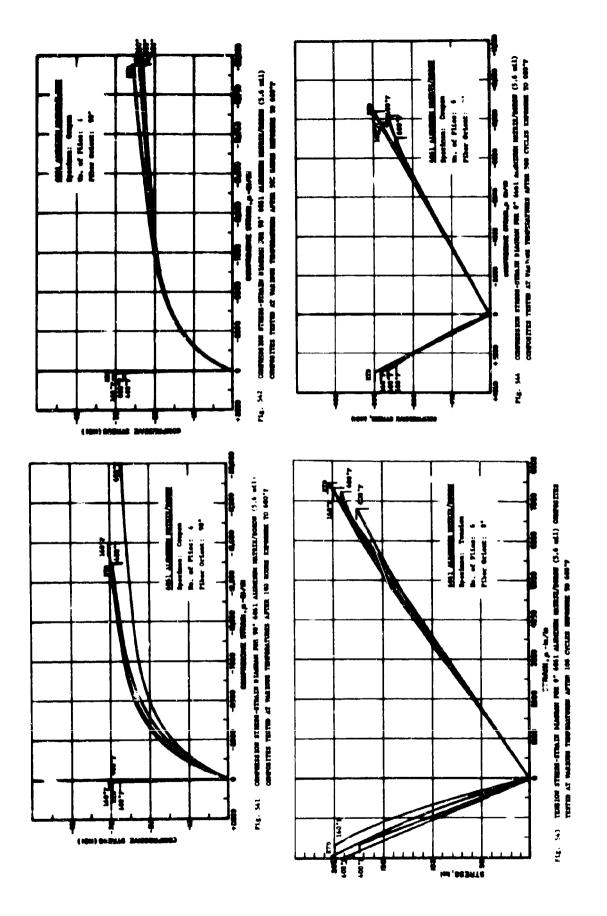
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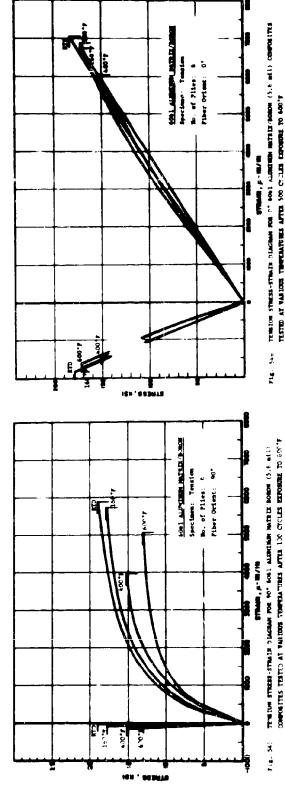


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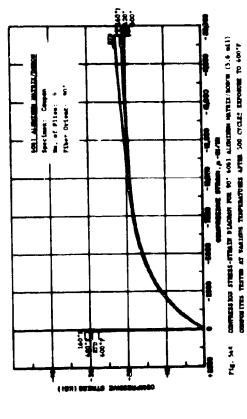
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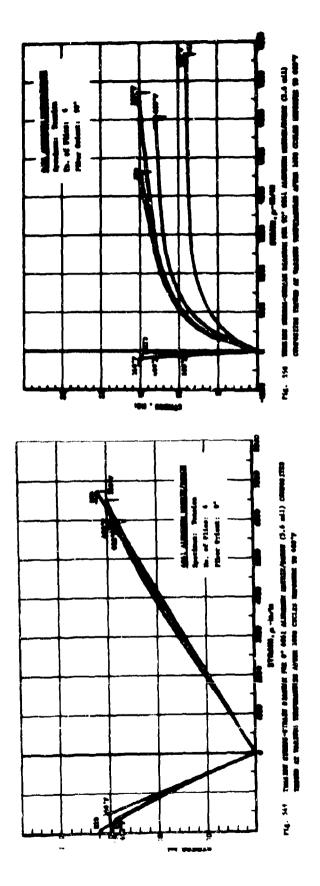


6041 ALMINER MATLE/LORDE Speciamu: Tematom No. of Plies: 6 Fiber Orient: 90*

E . 091 <u>ل</u> 80,7 2.004



THEFOR STREET-STALE DIAGRAM FOR 90' 6651 ALMOSTE METRIC/SECTION (5.6 ed.).
CORPOSITES TEATED AT VALIDES PT-PETALVERS AFTER 348 CITLES ENGINEE TO 640'7. Fig. 54



TAST NXII PROPERTIES SUPPRINCE - LANGER VARIETY FORCES

Ppeceta Rumber	Thickness (Fifes) (In.)	26 16 (In.)	Ortentation	R-Ratio	lest Temp. (*F)	Stress Level (? ^c ult) (ksi)	(bas)	Cycles to Fatlure (cycles)	Opeles Applied Mithese Failure (cycles)	Residual Strength (kel)	3
1929P-86	9	0.043		1.0	RTD	73	150	28.000		,	
1929P-8;	r	0.04.	U	1.0	RTD	7.1	145	. ,			Ismediate Pailure
1929P-88	•	0. (£2.	ί,	1.0	Ę	68.5	140	900		•	
1929P-89	9	0.0		1.0	£	99	135	28,000		•	
192 9P-9 0	\$	0.(43		0.1	E	63.5	130	102,000	•	•	
1929P-91	ı	0.042	ů	0.1	1,091	27.5	150	1,000		•	
1929P-92	ر	6.043		0.1	160 F	7.5	140	•	•	•	Immediate Pailure
1929P-93	Ų	9.043	ij	0.1	16035	6 7	2	161,000	•	•	
1929P-94	9	0.063	L	1.0	1.09T	69.5	135		•	٠	Immediate Felluce
36-4626?	vo	0.043	·O	1.0	1.091	£.5	125	•	•	•	bestiete Patluer
96-46261	4	0.044	Ċ	0.1	J 007	69.5	125	34. 080		•	
1929P-97	•	0.042	c	1.0	4,007	3	115	314,000	•		
19297-98	4	9.0	ů. C	0.1	4.007	75	135	142,000	•	•	
1929P-99	·	0.0		0.1	1,004	8 0.5	145	184,000		•	
19298-100	•	0.043	ò	0.1	4.00*	2	91	1,000	•	•	
1929P-101	ų	3	ت	0.1	1.009	8	0*1	42,000	•	•	
1929P-102	9	 6.0	•0	0.1	J. 009	\$.5	160	~ 8	•	•	
1929P-103	ΨO	C. 68.3	•0	0.1	4.009	*	14.5 24.5	2°80	•	•	
1929P-104	•	0.043	•0	0.1	9	72	2	137,000	•	•	
19299-105	us	0.042	°	1.0	1.009	Z.	21	374,000	•	•	
1928-1	ø	0.045	٥	-1	6	**		2,492,000	•	•	
1928-2	vo	770.0	•0	7	£	ž		2°.	•		
1926-3	φ	0.0	0	-	6	63.5*	R	9.00	•	•	
1928-4		0.045	•	7	E	*.3		200		•	
1928-5	٠	9,0	,	-	£	**	B	9.2	•	•	

* Prom Tenaile Ultimate Stress

Specimen	Thic (P14-s	Thickness (Plies) (In.)	Orientation	R-Matio	Test Tesp. (*)	Stress Let :1 (I ⁶ ult) (sf)	Le (1 (1 et)	Opeles to Failure (eyeles)	Opeles Applicated atthese Patlace (opeles)	Street (tent)	3
1928-6 1928-7 1928-8 1928-9 1928-10	ကာကာလာ က	0.045 0.045 0.043 0.043		7777	160°F 160°F 160°F 160°F	* 2.62 * 2.62 * 4.54 * 5.54	1+1+1+1+1 100 1100 1100 1100 1100 1100	3,000 14,000 73,000 2,956,000	2.0 × 10°	• • • • •	
1928-11 1928-12 1928-13 1928-14	ന ക ക ക ക	0.0.06 0.045 0.045 0.045	00000	तन नन	£.007 £.007 £.007	55.5 61.5 52.5 62.5 64.5 64.5 64.5 64.5	+100 +1100 +1110 +1115	89,000 2,000 29,000 116,000	• • • • •		
1926~16 1928~17 1928~18 1928~19	കാനതു ക	0.000 000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.	00000	77777	4.009 4.009 4.009	55.55 55.55 55.55 55.55 55.55	********	2888 8888 8888			matter Patient
1928-21 1926-22 1926-23 1926-24 1916-25	6	0.000 000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.000 0.		0 000	66 66 6	******	130		2.013 m 195 2.4 m 186	33	Desertation Publishers
1928-26 1928-27 1928-28 1928-29 1928-30	*	9.9.9.9	••••	2222	11111	ingrs ingrs	800 100 100 100 100 100 100 100 100 100		2.433 x 10 ⁶ 2.433 x 10 ⁶	37	Burdiche Patiene

Party Careel la Wiedones Stones

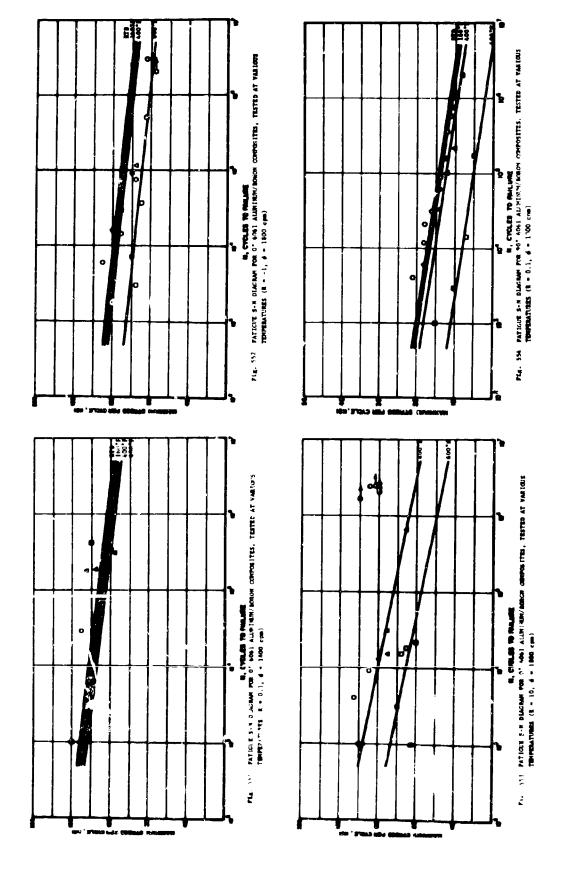
TABLE ANTI LATIGUE PROPERTUS SUPERMI NOST ALLYMBET WATRLE FORENT (T. eff.) COMPOSITIS -

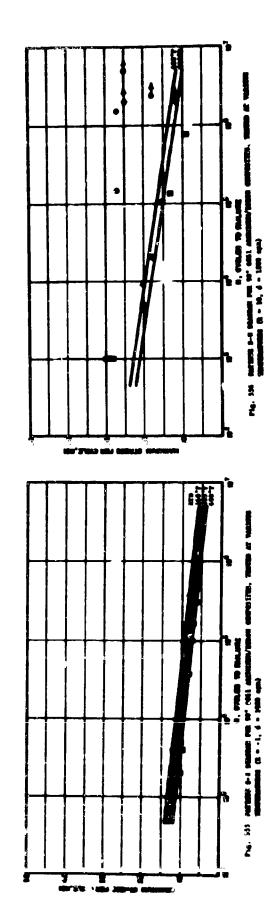
Specimen	en Thickness r (Pites) (In.)	as Trientation In.)		Test Tap. (*F)	Stress Level	cy ves to to Failure (eycles)	Applied without Failure (cycles)	Reciberal Strength (tot)	•
1 - 20, 51			<u>.</u>	3.007					
	9		٠,	1000				•	
10 - 100 C C C	:	÷	91	3			•	•	
19262 - 33	:	-	1:	4004			•	•	
10,083-34		٠	0.5	3 :00 7	110 110		•	•	
1926F-35	o o		21	00% د ا	35.5 - 115	31,000		•	
19788 - 36	9	D 99	0	,009			•	•	
19765-37			2.5	1.009			•	•	
1076s 10	- C	ر م	2	1.009			•	•	
197A8-39			: 31	4,004			•	•	
1928H-40	970.0		01	1.009	97	21,000	•		
42 - 40 20	 (.	ָּטֶל. ניי	-					•	
72.00	. r		• • •	Ē				•	
1,200-1	i c		1.0	00	91 19		•		
25.100.00			0.1	O LA			•	•	
J8-30F61	6 0.044	-	0.1	£	42.5 10	136,000	•		
18.308.81	\frac{1}{2}		0.1	1.091			•	•	
19308 - 82	ا د د	0.5	0.1	1,091	17	900.4	•	•	
1930B · 83	20.00		1.0	1.091			•	•	
77-80(6)	9		0.1	160 5	_		•	•	
1930C-85	6,00.0		0.1	1.091			•	•	
10.000	2	• 98	0.1	4.007			•	•	
1930C-87	9 40		1.0	1.007			•	•	
10.70C-10			- C	1.007			•	•	
100-10	90.0	.06	0.1	1.007	49.5 12	100,000	•	•	
10.70.4	2		0,1	1.007		~	•	•	

48) NATE FIELD PROPERTY NO 6061 ALIMININ'NS ATTENDED NO 17 OFFINE OFFINE TO 18 OFFI

Specimen Number	se nicka:	n 38 In,)	Orientation	R-Ratio	Test (F)	Strict Vel Cout) kst)	cos ro allure (cycles)	Cycles Applicd without Failure (cycles)	Residual Strength (ksi)	S
16-30C-51	1£)	; ()	0.5	6.1	4:004	2	1, no	•	•	
.930C-92	. ψ	7	3	r	1:000	<u> </u>	00°	•	•	
1930C-93	٠	ν - -	36	0.1	4,004		175,000			
1930C-94	ų	3 -0 7	Š	0.1	₹.004	33.5	17,000	•	•	
1530c-95	ţ	e.	يُن	0	J. 309	29 E	1	•	ı	Failed Under Static 14.
1928C1	£.	5.40	°Ú6		RTo	42.5* +10	14,000	1	•	
1928C-42		0.13	0 1/10		α Ι	15.5* + 6	1,153,000	•	•	
.928C-43	w	670	. 06	1.	(L 1)	90 + 	1+4,000		•	
1928C-44	ŗ	0.43	٠, ٢	-1	KTD	1+1	26,000	•	•	
1928c-45	٥	04.3	06ء	• • •	KT D	30 + + 7	100,000	•	•	
19286-46	Ω	(? ()	6د,	7	1.091		25,000	•	•	
1928C-47	æ	m 101	. 36	-1	1,041	£.5* ±12	8.	•	•	
1928C-48	ų.	∴ 043	n ()		1,041		72,000	٠	•	
1928n-19	¥	0.043	3	-1	160		481,000	•	•	
1928D-50	œ	3.044	٠,	-1	160°F		302,000			
.928D-51	·L·	2,043	. 5		3.007		376,000	•	•	
1928D-57	14.	043	.06	-1	400°F			•		
19282-53	4,	0,043	³ 06	-1	4.007	37 * +19		•	•	
1928D-54	ı	0.043	ء 06		4.007			•	•	
1928E-55	9	0.043	°06	- '	4.007			•	•	
19285-56	•	0.043	• 06	-	4.009	01+ × 85	5,000	•	•	
1928E-57	···	043	°06	-1	£ 009	6 + + 67	30,000	•	•	
1926E-56	9	0.042	:06	-1	£009	38 + + 86	37,000	•	•	
1928E-59	9	3.043	.06	-1	4.009	33.5* + 7	164,000	,		
1928E-60	90	0.042	。0 6	-1	600°F	29 * + 6	457,000	•	•	

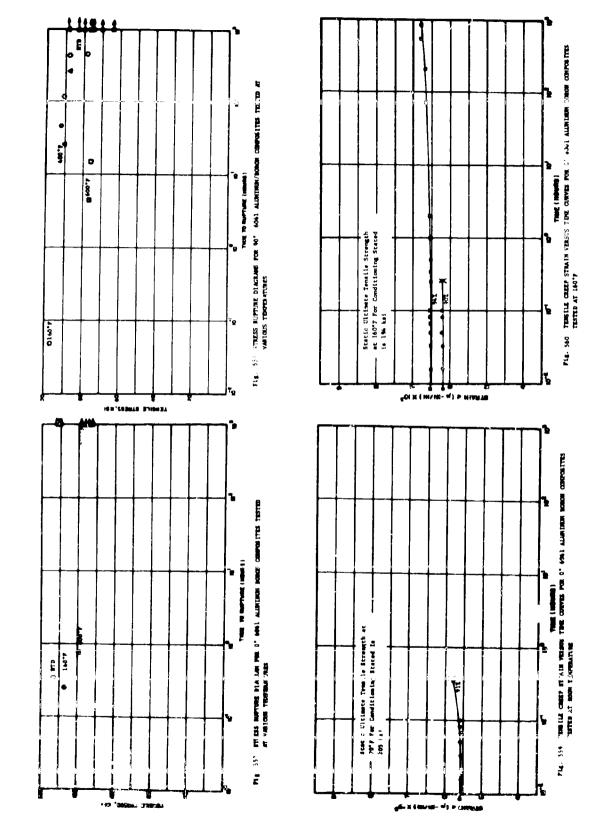
			!! !!	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	11		:	• • • • • • • • • • • • • • • • • • • •	1		
Specimen.	1916 1116		;		da 38 (7)	Stress Livel	Live	voles to Failure	DATE OF STATE OF STAT	Residuel Strength	
I CON		1	'T161,5110n	. P¥-X	(1)	(% ult) (ket)	(ket)	(c)cles)	(cycles)	(ket)	Committee
928E-61		; 1 -		rs es		Š	31.	,			
.928F-82	۵	ال-0′0	:	10	Ē				901.507 6	11.4	Tamenteca Fallure
19265-63	,	رة ج	ਤੰ	10	CL2	2			4 885×106		
19227-64	•	5.9	,	10	C.	70.5	į.	600		;	
19285-45		3.05		10	E	63.5	٠,4	1,511,000	٠		
:928F-66	ŧ	5	á	<u>ب</u>	160.5	7 79	6	9	ı		
3-585-67	÷	0.0	ئ.	207	1600 F			27.			
.928F	æ	0.043	ş	10	1.041	68.5	. 2.	000			Specimen has shooms!
1928(1-63	¢	3	ς,	01	1,041	7.0	-28	2	•	•	VERIETI O'TH WIRTH
J2-136(-6)	æ	0.063	ç	10	160°F	22	-25	;	2.007×106	25.7	
, almos a	£	0.063	·,	10	4.007	52.5	-25	•	•	•	Immediace Patlure
[2+C+,+	ı	÷.8	ş	10	1,007	98	8 7-	20,000	•		
14287-13	۵	O, 0	, ŏ	01	10×10	31.6	57	226,000	•		
	4	790.0		20	1 00%	42	-20	9,000	•		
# T T T	ت	0 04.3		91	J.007	25	-15	. •	2.0x106	20.8	
5 36-51	۲	c. 08.	ş	01	4.004	48.5	- 25	107,000	,	•	
142BC 77	ىد	J.04.1	Ī	10	1.009	64.5	-70	2,000	•	•	
1928M-78	4	- 1 2.	Ī	10	£.009	32	97-	. •			Ismediane Patlure
192 8H- 79	•	0.0	Š	01	1.009	42	ä	127,000	•	•	
1928H-#0	ø	0 041	S	10	1.009	29	•	763,000	•		



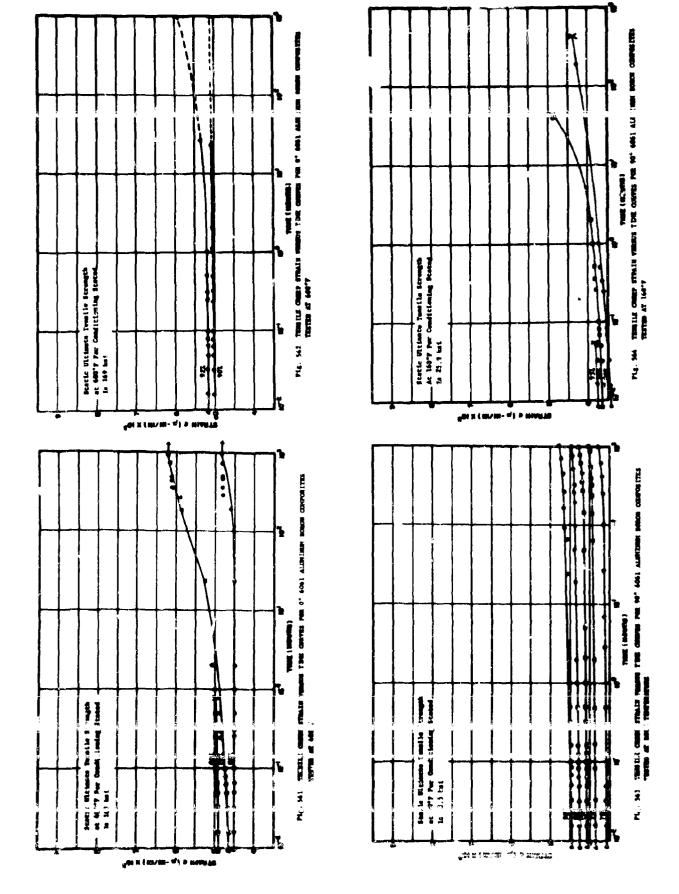


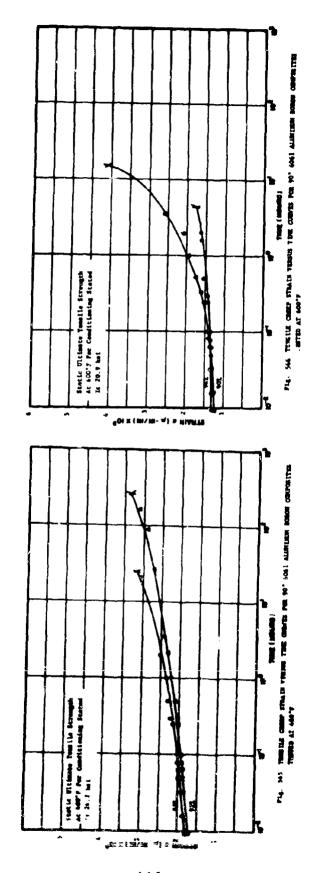
							Time	Applied	
Specimen	14th (P. 14th	'hichness (P ies; (In.)	Orfentation	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	Stress (T ^C ult	Stress Level (1 ^c ult) (ks1)	Estlure (Hours)	Failure (Hours)	Comment
0		170.0	-	£.	7. 61	188.	•	•	Broke during loading
01-4-1	ů	0, 143	ر	CLu	Ç.	185.6	,	ı	Broke during loading
1-1-1-10>		2	9	Ή	נ	196.5	Ę.	•	0
1-1-d-1-10-		÷.	ؿ	e T	Uj	164.3		•	Broke during loading
p10		TO .	ن	, EE	3	164.5	•	0001	Lust gauge during loading
	ı	. 1	O	1,091	i,	9.75	57.		
42-P-	•	;	ڻ	4,091	•	ξα3	•	1000	
61.7.000	r	1	,0	160 F	ę,	180.2			Broke during loading
142-1-11	1	•	ė	160 F	ý. 5	184.3		•	Broke during loading
1620k-115	ι	 	, J	1.00°F	۴3	130.4	•	1000	
-1 -4-2-		ر م	ບ	3 307	8	nc.	,	1000	
1-2-1-317	υ	3.0	. 0	1.00°	χ U	158	.75	•	
41:- 7.71	ſ	() ()	ن	4-00+	0.8	111		1000	
611-02	•,	7, (24.5	ن	±,005	Z	151		•	Broke C. Ing loading
±	U	9.064	ړ	4.00÷	66	551	,	•	Broke during loading
1.24P-121	£	0.045	O	600°F	90	152	,	1000	
19 7. 122	D	1. (O.)	.0	4,009	97	156	,	1000	
1,248-124	ပ	0.045	ņ÷	₹ 009	86 8	106	•	ı	Broke during loading
1927P-124	Þ	0.045	°O	و00 د	96	162	1	•	Broke during loading
1929P-125	•	7.0	•0	F 009	96	159		1000	Lost sause during loading

		1	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	Cost Atomisms (S. mit.) confession	7		; ; ; ;		
# 4	Thickners (Files) (In.)	म् स्माः (स्माः)	071.0		Stres level	(F21)	Tine to Falture (Hours)	line Applied without Fallure (Hours)	Comment
76.0861		ا ا	ø		ž	š. š.	•	1000	
0.97		ż	3	ET.	٠,	17.4	•	0001	
3.0	ı	٠.	.06	a Light	ŭ	1.97		1011	
16.11.00	J	ξ.	2	Ė	Ç	20.0		? 00 1	
1030-105-1		1 C J	.00	: E	ş	1.1-	r	1008	
1.006.1			Ş	1691	22	4	451.7	,	
1930-102		0,000	S.	140 1	3	23.3	45.7		
- 1:3	.,	<u>.</u>	3	1.091	Ç	14.1	.03	•	
20.		c ·	. 06	4.097	S)	21.7	116.2	•	
. 10,	ı	1.1.6	S	1.001	s;	22.0	426.3	1	
.930:103	4	9-1-	Ų÷	d. 007	£3	22.5	25	•	
۔ ان		ô k .	.06	400 F	7	22.7	•	•	Broke during Loading
5	e e		هن.	±.00*	96	21.8	258	c	•
			, Dé	Ji. 00.	*	23.2		•	Broke during Loading
1930-110		.; 3 .∍	° 06	£.007	92	22.2	•	1000	
111 060		9-11 û	\$	1.009	8	18.6	14.8	•	
-		5.	.06	J. 009	92	19.2	•	•	Broke during Loading
1930-113		1.1.52	• \$	1.009	92	19.2		•	Broke during Londing
114	9	3	&	₹.009	z	19.6	4.4	•	
115		.051	\$	4,009	92	10.2	•	1000	



 $\psi = \prod_{i \in I}$





APPENDIX V

DATA SUMMARY FOR 6A1-4V-TITANIUM/BORSIC COMPOSITES

IIT RESEARCH ENSTITUTE

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APPENDIX V

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6	Fig. 594 Creep Strain Versus Time Curves	470

TABLE XI IV STATIC PROPERTIES SUMMARY 6A1-4V-TITANIUM/BorSiC COMPOSITES

(rientation	May I age to	Prior Conditioning	Test Temp. (*F)	E (pst x 10 ⁶)	, (in/in)	oule (kst)	*ult (#-fa./in.)
0.	Tention	None	RI	38.8	0.20	171	4,260
ن.	Tencion	None	700	37.7	0.22	170	057'7
0,	Tention	None	009	37.3	0.23	145	3,960
, 0	Tention	None	900	37.9		138	077'7
.06	Teri ion	None	ĸ	28.2	97 0	85.1	8, 310
, C 6	Tention	None	007	23.1	0.15	53.9	5,050
۵۰ <u>5</u>	Tention	None	600	22.4	0.17	0.72	3,630
↓0 6	Tention	None	80 0	24.2	ı	47.8	4,610
ູ້ບ	Compression	N. SM.	Þ	36.8	0.25	069	19, 500
.0	Compress.on	None	00%	35.8	0.23	198	19,170
•0	Comp Tess	None	900	35.1	0.23	629	18, 360
• 0	Cuar ress	None	900	7. 7.	0.23	577	20,000
.06	Comp resulton	None	Ħ	29.6	0.17	506	11,480
•06	Cong ress lon	None	00 7	29.7	0.17	201	11,700
.06	Compresson	None	009	28.7	0.16	186	13,060
.06	Comp ress !.on	None	800	27.8	0.13	195	14,030
°	Flexirel	None	Ŀ	•	ı	218	•
•0	Flex arel	Mone	007	•	•	211	•
, o	Flexical	Kone	909	•	1	192	•
•	i	;	000			(

TABLE XXIV S.A.TIC PREPIRITIES STEERN 6A1-4V-111AN, "TB, FTS GO 4POSITES

Orientecien	Type Lond	Prior Conditioning	Test Temp. (*F)	E (pai × lt ²)	(in ⁷ in)	gult (kai)	*ule (#-£8./£8.)
	Tension	800°F/1000 Hrs	KT	ا الله	0.22	991	4, 290
• 0	I en g i on	8CU*F/1090 Hrs	907		0.24	156	7, 350
۲,	I en si con	660°F'1000 Hr.	004	35.3	0.23	136	3,780
•	Tens (on	800°F/1000 Hrs	008	35.7	•	113	3,280
°0.	de la la	800°F/100 Hrs	F	27.3	0.18	8 .2	3,740
÷0÷	1675177	800°F/100 Hrs	007	25.7	0 21	6 09	4,310
•04	Tension	\$00*7/100 Mrs	009	22.9	0.24	7 99	4,410
.06	Tens i on	800°F/100 Hrs	9 00	23.2	•	62.8	2,650
, O+,	Tension	800*F/500 Hrs	Ħ	23.6	61.0	59.5	3,090
•00	Tension	800*F/500 Hrs	904	23.4	0.21	6 0.8	3,860
, O	Ten # Lon	\$00°F/500 Hrs	009	5.12	0.23	£ .2	3,7:0
. 0,	Tension	\$00°F/500 Mrs	90	13.7	ı	39.8	. 250
,0 <i>1</i>	Tene i oc.	800"F/1000 Mrs	t	25.5	0.29	51 C	2,140
• US	Tenston	\$00°F/1000 Mrs	904	25 5	0.29	\$0.0	2,630
•0+	Teneton	800°F/1000 Mars	909	23.3	0.20	7.00	1,940
.03	Tens Lon	800°F/1000 Mrs	00	15.9	•	25.8	3,160
• 0	Compression	800°F/100 Mrs	달	37.7	0.22	653	20,730
•0	Compression	800"F/100 Hrs	004	35 7	0.23	%	12,660
•0	Compress ton	\$00°F/100 Hrs	009	31.3	0.20	556	24, 590
ċ	Commercial	800°F/100 Mrs	004	31.9	0.22	535	16, 790

Õ

Çı

l

No Material Left

^e ult (μ-fa./fa.)	18,070	17,610	16,910	16,000	10, 320	10,000	8,400	9,360	10, 390	11,230	10,270	10,930	4,290	3,970	3,490	3, 200	4,790	4,120	3,860	3,950
ult (ksi)	630	565	527	200	205	20%	185	190	212	198	190	189	160	149	125	128	174	148	139	146
, (1a/tn)	0.23	6.23	0.20	0.21	0.16	91.0	0.19	0.19	0 19	0 16	0.19	0 18	0.23	0.18	0.19	•	0.22	0.20	0.24	•
psi x 10 ^f	34.6	8. ¥	33.6	32.9	ब . इ.स.	27.5	27.5	27.4	24 h	28.1	28 2	25.1	38.2	38.7	38.4	39.6	38.0	7.07	1.15	39.1
4 (a.	<u>.</u>	007	6 00	800	r,	707	909	008	1,7	00,	900	800	RT	00,	069	900	臣	007	209	800
Petor Linditioning	SH078/500 HES	800 T/500 Hrs	800°F/500 Hrs	800 F/500 Hrs	800 F/100 Hrs	800 £/100 Hr.	S007F/100 Hrs	800 F7100 Hrs	800 F 7500 Hrs	800°F/500 1:rs	800°F/500 Hrs	800°F'500 Hrs	800°F/100 Cy.	800°F/100 Cy.	800°F/100 Cv.	800'F/109 C).	800°F/500 Cy.	800°F/500 Cy	800°F/500 Cy.	800°F/500 Cy.
Isal Leal	Compare es fon	Compression	ريت 16 د و د الله	Le. pres. for	nores son	Wind - Alleman	UO - 6 - BY HELVY	62 - 1 - 5	C. Marie Maria	Camprie - 1 m	Combatter States	படர்க்கோர்வ	Jensi. A	Tension	Tens ic	Tensio	Tension	Tension	Tention	Tension
rientation	,0	· ·	ပ	, û	aş.	: ق	O.	• C C	ر او	£05	£0.5	٠,	c.	7	٥٠	ő	<u>.</u>	°°	•0	•0

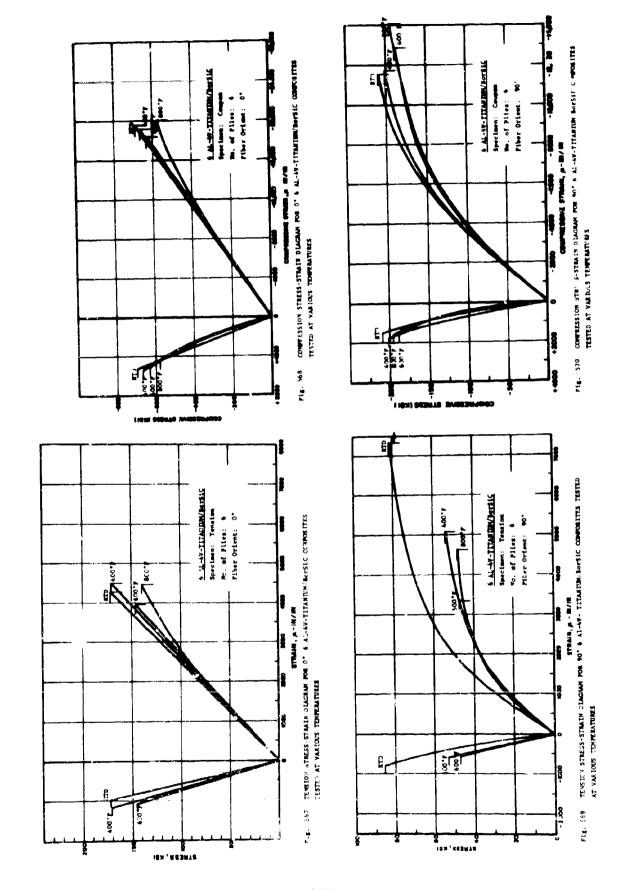
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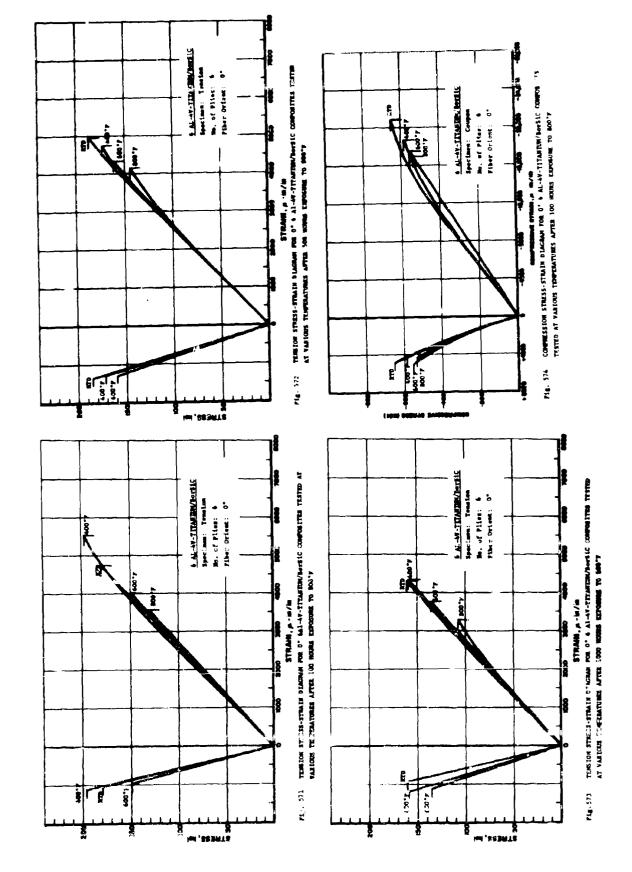
6LE XXIII FILE PROPERTEE SCHAARS

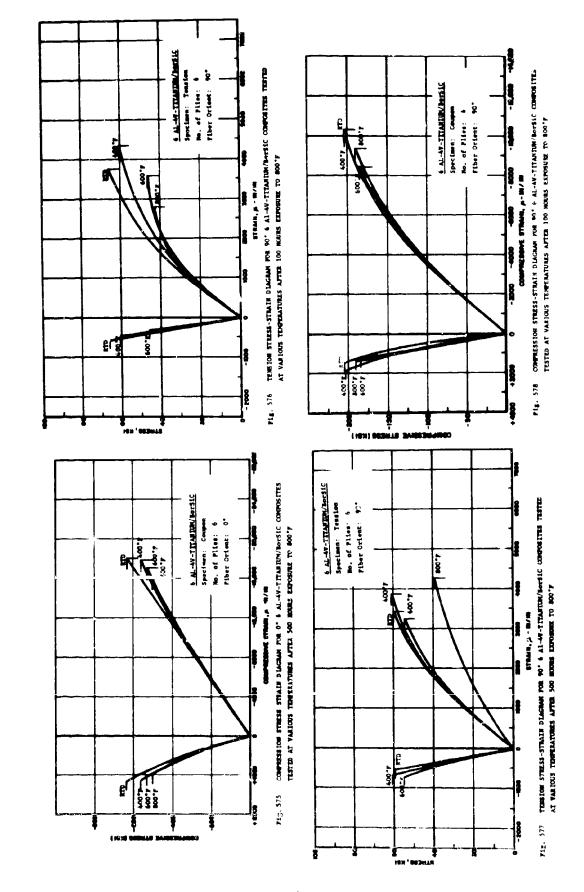
Orientation	Type Load	Prior Conditioning	rest lemp.	(ps1 x 10 ⁶)	(11/11)	ult (kef)	^e ult (u-fn./fn.)
-0	Iension	800°F/1000 CN	RT	1.6.	0.31	148	070'7
, O	Tension	e F/1000 Cv.	U0+	38.5	0.21	141	3,910
0,	:ension	800°E/1000 CV	00+	40.4	0.33	123	3,410
÷0	1. 18 co 1	800°F/1000 Cv.	HOU	32.2	,	9.96	4,330
≥06	Tensin	800°F/100 Cy	Rī	7.72	0.19	9.87	3,260
÷06	Tenspor	890°E/100 Fe.	00+	26.2	0.22	35.0	2,620
್ಥಿರಿಕ	Tension	800°F. LUO CH.	90.	24.7	0.25	26.6	1,470
,06	C. William	800°F/100 Cv.	800	19.6	,	29 1.	3,880
.06	Tension	800°F/560 Cv.	ĸŧ	26.7	97.0	50.4	2,530
. 06	Tension	80C°F/500 Cy.	700	21 6	91.0	38.6	5, 300
.06	Jension	800°F/500 Cy.	009	20.1	0.16	30.4	2,370
្តប្	Tension	300°F/500 Cy.	9.00	20.2	•	26.5	3,090
ون د	Tension	800°F/1000 Cy.	RŢ	22.9	0.21	9.07	2,940
€06	Sen si on	800°F/1000 Cy.	4 90	7.12	0.20	36.1	3,380
≥0 6	Tension	800 F/1000 CV.	009	22.5	0.22	30.4	2,350
06،	.ension	800°F/1000 Cy.	800	21.7	•	27.0	3, 380
٠0	Compression	800°F/500 Cy.	RI				
.0	Compression	800°F/500 Cy.	700				
0.	Compression	800 °F/300 C3.	009				
٥		800'E/500 Cv.	800				

IABLE XXIV STATIC PROPERTIES COMPACTOR SALESTITIANTINGS OF TO CONTRACTORS

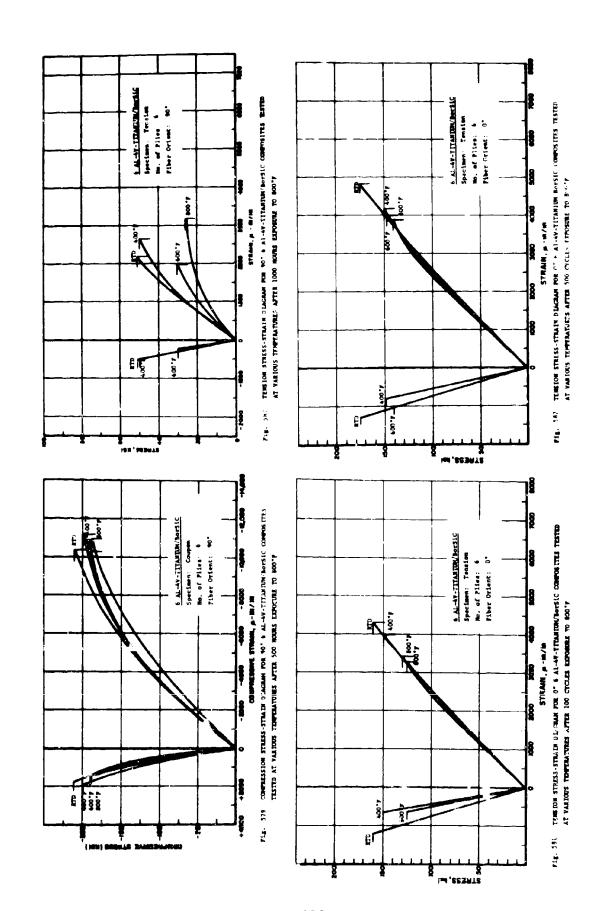
".lt (in./in.)	957 11	027 71	10,290	11,030
³ ult (ksf)	203	197	182	182
(in/in)	6.19	0 15	0.20	0.19
01 8 ,54	2 3	2, " 5	28 €	27.7
Test Temp. (°F)	RT	700	969	800
Prior Conditioning	800°F/500 Cy.	800°F/500 Cy.	800°F/500 Cy.	800°F/500 Cv.
Type Load	Compression	Compress: n	Compression	Compression
Orlentation	۰ 06	a06	್ತಿ06	.06

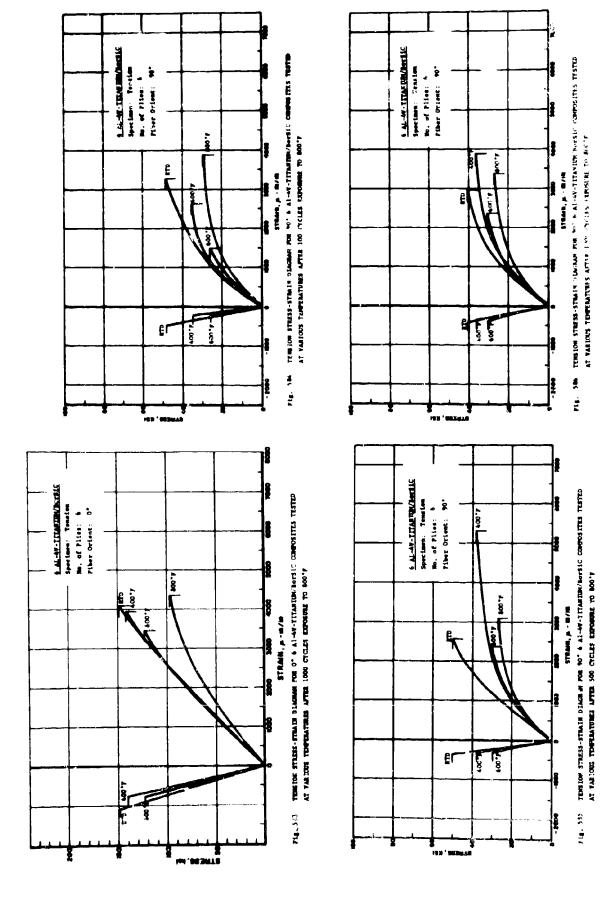






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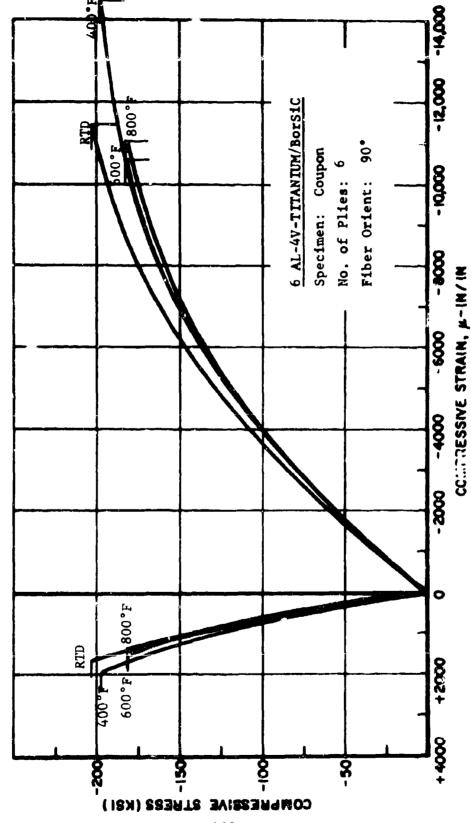




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COMPRESSION STRESS-STRAIN DIAGRAM FOR 90° 6 AL-4V-TITANIUM/BorSic COMPOSITES TESTED AT VARIOUS TEMPERATURES AFTER 500 CYCLES EXPOSURE TO 800°F Fig. 587

は、「一般のでは、「他のでは、「なって、これをはない。」というない。 「他のでは、「他のでは、「他のでは、「他のでは、」」というない。 「はいいない」というない。 「はいいない」というない。 「はいいない これがられる これ

TATE OF THE PROPERTIES SUPPLY OF TANKING ROPES OF TANKING ROPES OF TANKING ROPES OF TANKING THE TANKING TANKIN

Comment																									
Ress ^d us) Strength (kei)	•			•	1	•		•	•	•	•	•		•	1	ı	•	•		180	•	•		•	•
Cycles Applied without Failure (cycles)	•	•	٠	•	•	I	•		•	•	•	•		•	ı	•	•	•	,	7.33 x 10°	•	•		•	
Cycles to Fallure (cycles)	34,000	39,000	12,000	000	76,000	000 71	2,000	1,000	000.61	30,000	000.81	29,000		1,000	76,000	14,000	41,000	000,19	211,000	•	32,000	17,000	9.00	2,000	78. e11
.eve 1 (ks1)	175	ů,	G *	120	113	123	<u>ي. ۲</u>	120	115	ž.	43	105	130	130	60	45	8 5	75	9	2	9	+62	1,00	2 ,5	2,
Stress Lavel	7.3	76	25	70.5	(7.5	73.5	76.5	70.5	£	6.5	65.5	72.5	83	89.5	58.5	69	61.5	54.5	43.5	36	474	55.5*	7 7	* 1	L ,7
Tanp.	, TŒ	KT.J	СĽУ	eT.	K II	1,00	. 3€	3.007	J. 00.	4 007	4,000	FOO F	3.009	J. 009	4 du9	3,008	¥.009	€00°F	£009	3,00g	Ę		£	2	011
h-Rat I.v	٥. ١	0.1	0.1	···c	ů.:	. :	-	٠.1	0	ن.:	0.1	0.1	1.0	٠.٠	0.1	0	0.1	0.1	0.1	0.1	-1		7	.	7-
Orientation	2	-	•	,	۲.	¢.	•	•				r,	19	Ö	0	. 0	ی	0,	٥٠	Û	•0	.0	.0	• •	•0
ess (In.	ر. برور	ن ا ا	0.047	14:1.0	0.6-	6.04s	\$ 70°	1300	0	5+3.0	0.05.	0.050	0.051	0.051	0.050	0.048	6.6	. 6 .0	0.048	0.043	0,043	0,043	0.042	0.038	0.043
Thickness (Plies) (In.	t	4-	Φ	·	vc.		w				9	φ				vc	ن، ر				40			9	ø
Specimen	SN26-18-1	SN26-16-4	SN26-18-7	SN26-1C-2	SN26-1C-5	SN26 - 10 - 8	SN26-10-3	SN26-10-6	SN26-1D-9	57.26-23-1	SN26-23-4	SN26-25-7	SN26-2C-2	SN26-2C-5	SN26-2C-8	F-02-9283	SN26-20-6	SN26-22-9	SN26-38-1	SN26-3B-4	SX26-5B-1	SN26-58-2	SN26-5B-3	SN26-58-4	SN26-5C-1

TABLE XXV. FATIGUE PROPERTIES SUPPARY
641-4V TITANIUM BURSIC
15. mil) COMPOSITES BASELINF DATA

,	Thie	Thickness			Test			Cycles	Cycles Applied	Residue	
Member	(11166)	(Flies) (In.)	Orientation	R-Xat i	Tog. (*)	(T'ult) (kai)	(jes)	Fatlure (cycles)	Failure	Strength (ks1)	Š
SN26-5C-2	ų	3	ذ	-			ä				
EN25-50-3	40	3	·c							•	
SN26-5C-4	ع	1-1-0 0	· c	-	. L	: :	; :	903.47	•		
SNZ6-50-1	ء د	. 1 . 1		7.	- •					٠	
SN26-5D-2	•	9r) 0	. C.		(S) 4	<u>.</u>	£ =	200 120 120 120 120 120 120 120 120 120	1 +	• 1	
1								•)	
5.426-50-3	، ث	0.044	(5	-	¥ 1604	:	, a,	000	,		
1-89-1-58-I	۰	53.0	်	-1	4(1)4	; -	ÿ0.	.00	•	•	
5.827-68-2	vo	ິດ5ດ	c		1, 1414	: 2	<u>.</u>	18,000	•	•	
SN27-6B-3	υ	0.051	ű		Ŏ.	\$ US	<u>0</u>	9	•		
SE27-6C-1	٥	0.051	¢,		1, 009	٠,٠	غَ ا	90,00	•	· ,	
SH27-6C-2	Ð	0.053	ů	-	1:OX	1 200					
SN27-6C-3	vo	0.0	O		1,00						
SN27-6D-1	ψ	0.05	0	• -	#_[11]#						
SN27-6D-2	90	0.05	ۍ د	-	ROO F	14.00	Tent of Services				
SX27-6D-3	•0	0.053	9,	1-	8001°F	1086	Tests were not				
SN25-58-1	9	0.043	,0	10	, IX	-1	ě,	•	2 51 2 106		
SN25-58-2	•	0.043	້ ວ	2	£	. C .	9.1-	1 016 000		Ì,	
SN25-58-3	vo	0.044	•0	01	£		- 105		, , , , , , ,	, <u>1</u>	
S:/25-5c-1	φ	0.045	. 0	10	£		.173	344,000		} .	
S325-5C-2	9	9,046	0 د	10	£		-185	1,677,000	•	•	
5/25-50-3	9	0.046	•0	10	3.007	28	-185	U00 967	•	•	
SN25-50-1	•	0.044	•0	10	4,007	30	-200	000		•	
5725-50-2	•	0.045	ó	61	1.005		-220	000	•	•	
SH25-50-3	•	0.045	0،	01	4.007	33	-210		•	•	Immediate Bathur.
One Spec.	c. Singe										

Specimen Number	Thickness (Plies) (In.)	ness (In.)	Grientation	R-Ratio	Test Temp. (*F)	her se Livel	Live:	Cycles to Failure (cycles)	Cyclis Applied without Failure (cycles)	Residual Strength (ksi)	Сошности
5427-32-1	4	F 5 J	J	01	3 18)9	11	-180	-	•	•	
SN27-38-2	, ,0	0.0	•	21	4.003	58	-160	2.000		•	
SN27-3A-3	5	850.0		: 9	1000	1	071-	759,030	•		
SN27-34-4	9	0.016	5	10	4,009	9,	-150	• •	•	•	Immediate Failure
SN27-3A-5	9	0.049	i.e.	10	a (00) a	57	-145	32,000	•	•	
SN27-32-6	a	0.030	U	10	800 F	- Test	s were no	ב בתני			
SN27-3A-7	40	650.0	U	10	SOU F	- Test	S were no	The second of the second			
SN27-3A-8	J.	0.047	i,	10	800 F	- Test	- Tests were not	י שתבי ז			
SN27-34-9	þ	0.048	6	10	800 · F	- Test	s were no				
SN27-3A-10	۵	340.0		10	800 °F	· Test	S Were Du				
SN26-24-22	¢	0,040	Ú÷.	0.1	RT	70.3	09	,	,		immediate Pail tre
SN26-35-21	ų	0,046		0.1	RTD	53	57	000	•		
SN26-3A-22	۵	9,0.0	, ?,	0.1	RTD	35	30	8,000	•	•	
SN26-4A-21	9	370.0	, CT	0.1	RTD	23.5	20	34,000			
SN26-4A-22	æ	0.048	<u>.</u> ن	6.1	RTD	12	<u> </u>	•	2.15 × 10°	51.2	
SN27-2A-1	9	0,043	96	0,1	4,00%	93	20	3,000	,	•	
SN27-2A-2	9	0.044	06)	1.0	4.00%	74	07	000°s	•	•	
SN27-2A-3	9	0.644	, 06	0.1	4.004	56	30	27,000	•	•	
SN27-2A-4	9	770 0	36 و	0.1	4.00 ÷	46.5	25	206,000			
SN27-2A-5	•	950.0	"ე6	0.1	£,007	33	20	₹ 000	•	•	
SN27-2A-6	9	0,047	, ú 6	0.1	4.009	*	45	•	•		Immediate Failure
SH27-2A-7	9	2750	<u>، ن</u> 6	0.1	600°F	42.5	20	23,000	•	•	
SN27-24-8	9	0.047	, 0 6	0.1	£009	79	30	000.4	٠	•	
SN27-24-9	9	873.0	°06	0.1	¥,009	32	21	67,000			
SN27-14-10	9	0.047	့ပ 6	0.1	₽.009	21	10	134,000	•	•	

TABLE XXV FATTURE PROPERTIES SCHMANY
6A1-4V TITANTIV BOESS
(5.7 of 1) COSPOSTIES RASELLNI DATA

Specimen Nurber	Thickness (Flies) (In	Thickness (Fifes) (In.)	Orientation	N-Ratio	Test Temp. (°F)	Ctriss Live!	uve! (ket)	Cycles Ed Ed Saflure (cycles)	Cycles Applied without Failure (cycles)	Residual Strength (ks.1)	Comment
SN27-2A-11 SN27-2A-12 SN27-2A-13 SN27-2A-14 SN27-2A-14	ተ መጠመግ	0.00 0.00 0.00 0.00 0.00 0.00	, , , , , , , , , , , , , , , , , , ,	555 55	3.008 3.008 3.008	·	95883	13,001 2,644 7,600 -	1 1 (1 1		Failed under Stat Load
SN26-5A-21 SN26-5A-2 SN26-5A-3 SN26-5A-4 SN26-5A-4	99999	0.042 0.043 0.043	900 666 1066 1066 1066	त्त्रात्	KTTC KTTC KTTC KTTC KTCC	35 × 17 × 17 × 17 × 17 × 17 × 17 × 17 × 1	## # # # \$\sum_c \dagger \dag	2,000	3.0 × 10 ⁶	•	lamediate Prilure Immediate Prilure Failed in Northing Immediate Prilure
SN26-5A-6 SN26-5A-7 SN26-5A-8 SN26-5A-9 SN26-5A-10	ወ ውጥ ወ ወ	0.045 0.045 0.043 0.043	, , , , , , , , , , , , , , , , , , ,	7777	# 0007 # 0007 # 0007 # 0007	76.05 55.55 46.05	+1+1+1 ++1 5 % % % 0 1 % % % 0	21,000 1,000 5,000	- - 2.675 x 10 ⁶	- - -112	Immediate Prilure
SN26-5A-11 SN26-5A-12 SN26-5A-13 SN26-5A-14 SN26-5A-15	ଦନ୍ଦ୍ର	0.00 0.042 0.042 0.043 0.043	0 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	44444	3,009 4,009 4,009 4,009	53.5; 42.5; 32.4	+1+1+1+1 20 10 10 10 10	1,000 3,000 356,000 53,000		, , , , ,	
SN26-5A-16 SN26-5A-17 SN26-5A-18 SN26-5A-19 SN26-5A-20	மைம்மை	0.043 0.044 0.044 0.043		7777	800 ° F 800 ° F 800 ° F 800 ° F	TITEST TEST TEST TEST TEST TEST TEST TES	Tests were not run Tests were not run Tests were not run Tests were not run Tests were not run	Tests were not run - Tests were not run			

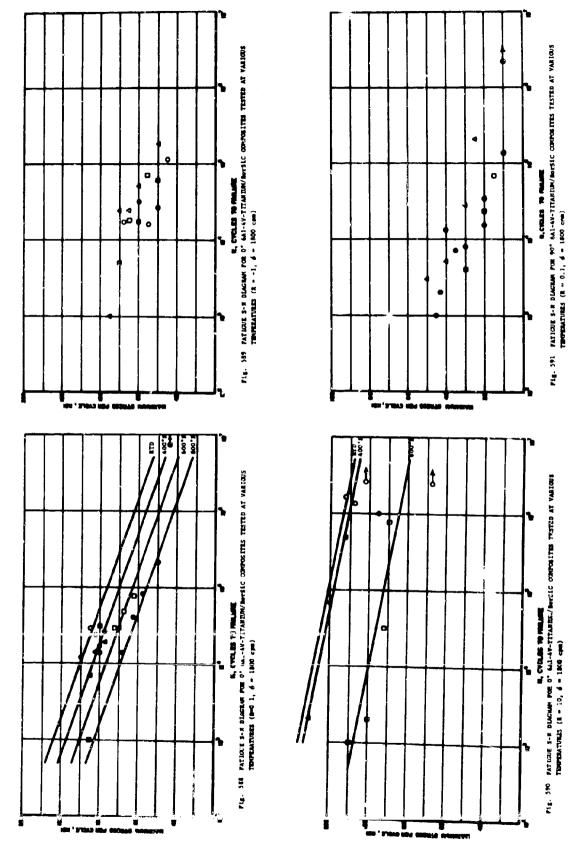
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FAILT THANKE 6 PS10

Co∎€nt				Immediate Failure Immediate Failure	
Residual Strength (ksf)	231		130		
Cycl. 8 Applied without Failure (cycles)	10.3 × 10 ⁶		3.016 x 10 ⁶		
Cycles to Eatlure (cycles)	137,000	1,000	2,000 4,000 110,000 215,000	45,000 2,000 - 557,000	C Tun (C
Stress Level	-100	-130 - 95	-100 -120 -110 -105	-100 -115 -105 -110	were not were not were not were not were not were not
Stress Level	87, 8,6	62 45.5	49.5 60 55 52.5 57.5	53 61.5 56 58.	Teats Tests Tests
Test Temp (字)	RTE RTI)	gg g	4,007 4,007 4,007	4,009 4,009 4,309 4,009	800°F 800°F 800°F 800°F
R-Ratio	10 10	01	01 10 10 01 01	10 10 10 10	10 10 10 10
Orientation	6 . 6 9 6 5	- 06 - 06	, ° ; ; ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° °	, 06 06 06	, 06 06 06 06
nesa (In.)	0.032	0.053	0.053 0.053 0.054 0.054	0.052 0.051 0.052 0.052	0.052 0.053 0.053 0.052
Thickness (Plies) (In.)	மை ம	o o o	ကားသတ္ ကာ	2 00000	\$ \$
Specimen Wasser	SN27-6A-21 SN27-6A-2 SN27-6A-3	SN27-6A-4 SN27-6A-5	SN27-64-6 SN27-64-7 SN27-64-8 SN27-64-9	SN27-6A-11 SN27-6A-12 SN27-64-13 SN27-64-14 SN27-6A-14	SN27-6A-16 SN27-6A-17 SN27-6A-18 SN27-6A-19 SN27-6A-19

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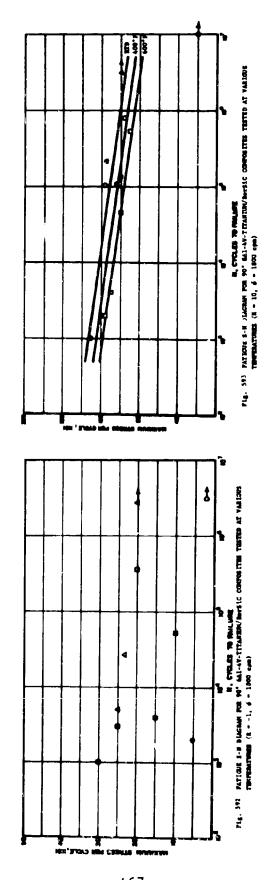
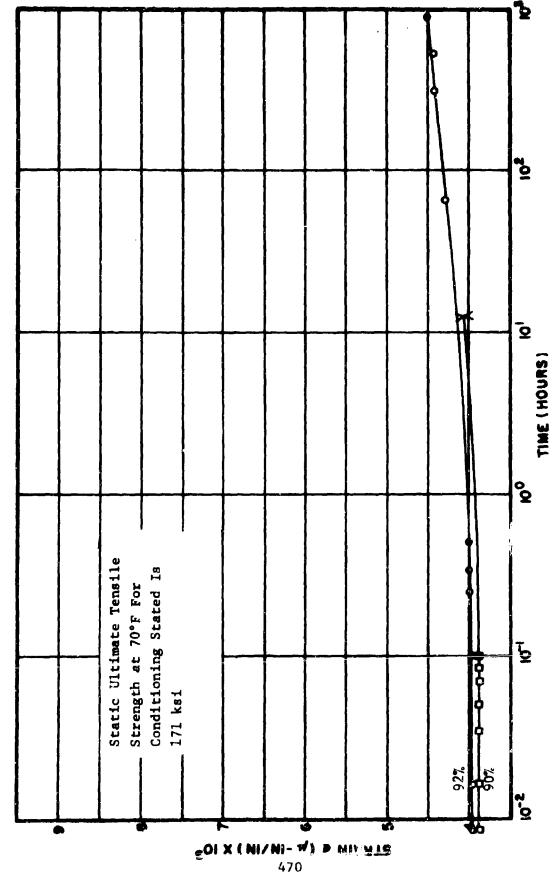


TABLE DAY: CREEP AND STRESS RUPTURE PROGRETES - PORCHY 6A1-4V TITANIUM BOYSIC (5.7 · i); COMPOSITES

							Time	Time	
Specimen Number	Thd (Plie	Thickness (Plies) (In.)	Orientation	Test Temp. (*F)	Stress (Tult	Stress Level (Tult) (ksi)	to Fallure (hours)	without Failure (hours)	Comment
SW26-38-7	9	0.048	.0	2	25	147.5		 	Broke during leading
SH26-3C-3	• •	0.051	, 0	₽	8	141.3	•	1000	
SN26-3C-5	9	0.051	.0	RTD	93	146.0		1000	Lost gage @ 3 min.
SN26-3C-8	ψ		ູ້ກ	RTD					broke during fabrication
SN26-33-3	·Li	0.053	Û	RTD	92	144.4	•	1000	
3N26-3D-6	Ψ	6-0.0	0,	₫.007	85	144.5	•	,	Broke during leading
SNZ6-3D-9	Ψ	0.049	.0	4,00¢	79	134.3	•	•	Broke during lending
SN2C-48-1	¥	0,048	0 ۽	4.007	06	153.0	11.3		Broke in Middle
SN26-4P .4	'n	0.04%	0 ،	4.007	88	149.6	910.		Broke in Middle
SN26-48-7	ψ	0.049	0,	₹.007	82	139.0	•		Broke during leading
\$X26-4C-2	9	0.052	0.	4.009	85	121,8	•	ı	Broke during lending
SW26-4C-5	9	0.053	_. 0	4.009	3 2	122.9	050	•	
SN26-4C-8	9	0.052	.0	₫,009	83	118.6		•	Broke after 2 hours
5,426-40-3	ø	0.052	0,	£.009	9	128.7	•		Broke during lending
SN26-40-6	w	0.053	0,	9,009			•	•	Broke during lesding
6-07-92NS	yo	0.052	, O	800 °F	980	108.8	•	•	de rie
SN28-44-3	vo	0.054	,0	₹.002	*	130,5	•	•	Broke during it ading
SN28-4A-4	vo	0.054	,0	800 °F	9	122.4	•	•	1
SN28-4A-5	w	0.054	٥.	3.000 F	ż	127.8	•	•	į
SN28-64-6	40	0.056	•0	4.008	85	115.6	. 167		

TABLE XXVI (REEP IV SUR SS CUIDA PROGRAM CONDINCT PROGRAM)

Specimen Number	Thic (Pite	Thiciness (Plies) (In.)	Orientation	Test Temp. (*f)	Stress (*Outr	Stress Level ("Oult) (MB1)	Time to Failure (hours)	Time Applied Without Feilure (hours)	Comment
SN25-5A-1	ص	0.047	06	GT.	8	58	,	 	Broke during loading
N25-5A-2	ıΩ	840.0	: 06	RTD	89	63.8	•	•	Broke during loading
SR25-5A-3	φ	6.0	. 06	CF3	ž	6.09	•	•	Broke during loading
3N25-5A-4	ø	0.047	, C o	RTD	17	7.62	•	1000	,
SN25-5A-5	φ	0.048	,06	RTD	20	36.2	•	1001	
N25-5A-6	9	0.048	. 06	J. 007	85	45.8	,	•	Broke during loading
SN25-5A-7	φ	0.047	- 06	4.007	92	6.04	•	•	Broke during loading
N25-5A-8	v		- 06	4.007					Broke during Eubrication
N.25-5A-9	Ð	0.049	.06	3.007	78	42.0	•	•	Broke during loading
SN25-5A-10	φ	970.0	. 06	4.00¢	9	43.1	.033	•	Brcke
N25-5A-11	9	0.041	∘ 06	₫. 009	8	42.3	ı	١	Broke during loading
N25-5A-12	ø	0.041	₌0 6	4.009	.	39.9	•	•	Broke during loading
N75-54-13	9	0.041	: 06	£ 009	80	37.ń	1.0	•	
SN25-5A-14	ů.	0.042	. 06	€000°F	83	39.0	•	•	Broke during loading
SN25-5A-15	w	0.042	. 06	4.009	7.8	36.6	.033	ı	
SN25-54-16	v	0.050	, 06	3.008	88	42.0	•	•	Broke during loading
N25-5A-17	છ	0.050	: 0 6	800°F	8	38.2	•	•	Broke during loading
3N25-5A-18	9	0,050	.06	9.008	78	37.2	•	1	Broke furing loading
SN25-5A-19	v	.051	.06	3.009	75	ر م	•	ı	Broke during loading
C175 5-54-20	v	0.00	•06	#-008	z	40.4	•	•	(Ran 1.8 bours)



TENSILE CREEP STRAIN VERSUS TIME CURVES FOR 0° 6A1-4V-TITANIUM BORSIC COMPOSITES TESTED AT ROOM TEMPERATURE Fig. 5.94

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